

Accepted Manuscript

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PII: S0925-8388(15)01123-8

DOI: <http://dx.doi.org/10.1016/j.jallcom.2015.04.108>

Reference: JALCOM 33990

To appear in: *Journal of Alloys and Compounds*

Received Date: 16 February 2015

Revised Date: 15 April 2015

Accepted Date: 17 April 2015

Please cite this article as: M.E. Camilo, E.d.O. Silva, L.R.P. Kassab, J.A.M. Garcia, C.B. de Araújo, White light generation controlled by changing the concentration of silver nanoparticles hosted by $\text{Ho}^{3+}/\text{Tm}^{3+}/\text{Yb}^{3+}$ doped $\text{GeO}_2 - \text{PbO}$ glasses, *Journal of Alloys and Compounds* (2015), doi: <http://dx.doi.org/10.1016/j.jallcom.2015.04.108>

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White light generation controlled by changing the concentration of silver nanoparticles hosted by Ho^{3+} / Tm^{3+} / Yb^{3+} doped $\text{GeO}_2 - \text{PbO}$ glasses

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Abstract

Frequency upconversion (UC) experiments were performed with $\text{GeO}_2\text{-PbO}$ glasses, containing silver nanoparticles (NPs), doped with holmium (Ho^{3+}), thulium (Tm^{3+}) and ytterbium (Yb^{3+}) ions. The samples were excited using a continuous-wave diode laser operating at 980 nm. The UC intensities and the colors of the samples were controlled by changing the concentrations of the rare-earth ions (REI) and the silver NPs nucleated inside the samples. The colors observed spanned the visible range from the red-yellow to the blue.

For a particular combination of silver NPs and REI concentrations it was obtained white light generation corresponding to the CIE coordinates $X=0.33$ and $Y=0.33$.

Keywords: frequency upconversion; rare-earth ions; energy transfer; metallic nanoparticles; silver nanoparticles; germanate glass; white light generation.

1. Introduction

Frequency upconversion (UC) processes in organic and inorganic materials have been largely exploited in the past decades because of their potential to convert light from the infrared to the visible [1-7]. In particular the investigation of glasses doped with rare-earth ions (REI) for possible photonic applications is still attracting large interest in the present days

Photoluminescence (PL) is the main technique used to investigate UC processes. When the three primary colors (red-green-blue) are obtained with adjustable intensities, the color of the luminescent sample can be controlled and for a specific choice of parameters white light generation (WLG) can be obtained. Large number of experiments in this area was reported with glasses doped simultaneously with holmium (Ho^{3+}), thulium (Tm^{3+}), and ytterbium (Yb^{3+}) ions. For instance, in experiments where Yb^{3+} ions are resonantly excited, energy transfer (ET) from Yb^{3+} to Ho^{3+} and from Yb^{3+} to Tm^{3+} are responsible for the frequency upconverted PL. Green light at 545 nm and red emission at 658 nm are produced by the excited Ho^{3+} ions while blue emission at 477 nm is emitted by the excited Tm^{3+} ions. The CIE (Commission Internationale de l'Éclairage) chromaticity diagram is used to characterize the samples' color. In the studies towards WLG the materials parameters are

choose in order to obtain the CIE coordinates as near as possible from the white point ($X=0.33$; $Y=0.33$). There are many reports that present reasonable results using different materials doped with $\text{Ho}^{3+}/\text{Tm}^{3+}/\text{Yb}^{3+}$, such as aluminum-germanate glasses [8], oxyfluoride nanoglass [9], SrMoO_4 nanocrystals [10], tellurite glasses [11,12] and lead-germanate glasses [13]. However to reach the white point is challenging.

In another research direction to obtain UC with large efficiency, several authors exploited the enhanced PL of REI in the presence of metallic nanoparticles (NPs). Then, the PL enhancement is due to the influence of two effects: the increase of the local field on the REI located in the proximity of the NPs and the ET from excited NPs to the REI. This procedure to obtain larger PL efficiency was demonstrated for various glass compositions doped with different REI and containing silver or gold NPs [13-19]

In particular, germanate glasses with REI and metallic NPs are good candidates for photonic applications due to their wide transmission window, from the visible to mid-infrared region, good mechanical strength, large chemical durability, low cut-off phonon energy ($\sim 800 \text{ cm}^{-1}$) and high refractive index (~ 2) [14]. UC processes have been studied in germanate glasses doped with Eu^{3+} [20,21], Tm^{3+} [22], Er^{3+} [23], Er/Yb^{3+} [24], $\text{Tm}^{3+}/\text{Yb}^{3+}$ [25], $\text{Er}^{3+}/\text{Tm}^{3+}/\text{Yb}^{3+}$ [26] and $\text{Tm}^{3+}/\text{Ho}^{3+}/\text{Yb}^{3+}$ [13] and Nd^{3+} [27]. In all references, except in ref. [13], the samples contained silver or gold NPs fabricated using the melt-quenching method followed by appropriated heat-treatment of the samples.

In the present work, we analyze the influence of silver NPs on the colors emitted by $\text{GeO}_2\text{-PbO}$ glasses doped with Ho^{3+} , Tm^{3+} , and Yb^{3+} excited at 980 nm to extend the results of [13]. It is shown for the first time that the presence of silver NPs allows observation of WLW in samples whose REI concentrations, in the absence of NPs, could not enable WLW. Also, it is shown that changing the amount of silver NPs inside the $\text{GeO}_2\text{-PbO}$ glass

samples doped with fixed concentrations of the rare-earth ions, allows the control of the samples' color.

2. Experimental details

The samples investigated were fabricated using the melt-quenching method with the starting composition: 40GeO₂-60PbO. The doping species were Ho₂O₃ (0.9 and 0.65 wt. %), Tm₂O₃ (0.60 wt.%), Yb₂O₃ (3.2 wt.%) and AgNO₃ (1.0 wt.% and 2.5 wt.%). Two different sets of samples were prepared with the same Tm₂O₃ and Yb₂O₃ concentrations but with different concentrations of Ho₂O₃ and AgNO₃; the first set includes samples without NPs studied in our previous report [13] and also samples having 1.0 wt.% of AgNO₃. The second set have samples prepared with 0.6 wt.% of Ho₂O₃ and 2.5 wt.% of AgNO₃, but with the same concentrations of Tm₂O₃ and Yb₂O₃ as the other set of samples.

The reagents were melted in an alumina crucible for 1 h at 1200 °C, quenched in a preheated brass mold, annealed at 420 °C during 1 h (to avoid internal stress) and cooled to room temperature inside the furnace. After this procedure, the samples were polished and cut in order to have parallel faces with 2.0 mm thickness. The samples prepared with AgNO₃ were submitted to heat-treatment (HT) during 12 h and 24 h at 420°C to thermally reduce the Ag⁺ ions to Ag⁰, enabling the nucleation of silver NPs.

Although the study of the mechanisms of silver NPs formation is beyond the scope of the present article, we recall that the AgNO₃ compound added to the starting glass constituents easily decompose at the temperatures used to prepare the samples by the

reaction $2\text{AgNO}_3 \rightarrow \text{Ag}_2\text{O} + 2\text{NO}_2 + \frac{1}{2}\text{O}_2$. On the other hand, Ag₂O decomposes to metallic

Ag^0 and to O_2 through $\text{Ag}_2\text{O} \rightarrow 2\text{Ag} + \frac{1}{2}\text{O}_2$ [28, 29]. Also individual silver ions produced during melting may reduce to Ag^0 during the HT due to electrons extracted from atoms that are intrinsic to the glass, namely the nonbridging oxygens.

The samples containing Ag NPs, used in the experiments, were labeled as: A (0.9% Ho^{3+} ; 0.6% Tm^{3+} ; 3.2% Yb^{3+} ; 1.0% AgNO_3 ; HT: 12h); B (0.9% Ho^{3+} ; 0.6% Tm^{3+} ; 3.2% Yb^{3+} ; 1.0% AgNO_3 ; HT: 24h); C (0.65% Ho^{3+} ; 0.6% Tm^{3+} ; 3.2% Yb^{3+} ; 2.5% AgNO_3 ; HT: 24h); D (0.65% Ho^{3+} ; 0.6% Tm^{3+} ; 3.2% Yb^{3+} ; 2.5% AgNO_3 ; HT: 12h)

A 200 kV transmission electron microscope (TEM) was used to determine the sizes and shapes of the silver NPs. A commercial spectrophotometer was used to measure the optical absorption spectra from 400 to 2000 nm. The PL measurements were made by exciting the samples using a 980 nm continuous-wave diode laser (maximum power on the samples: 200 mW). A spectrometer attached to a photomultiplier and a computer was used to analyze the PL signals.

3. Results and discussion

The linear absorption spectra of the samples were similar to the ones presented in [13], except for the amplitudes of the absorption peaks that depend on the REI and NPs concentrations. Figure 1 shows the absorbance spectra for two samples doped with Tm^{3+} , Ho^{3+} , Yb^{3+} ions (with and without metallic NPs). The absorption transitions associated with the Tm^{3+} ions correspond to the bands centered at 450 nm (${}^3\text{H}_6 \rightarrow {}^1\text{G}_4$), 680 nm (${}^3\text{H}_6 \rightarrow {}^3\text{F}_{2,3}$) and 800 nm (${}^3\text{H}_6 \rightarrow {}^3\text{H}_4$). The Ho^{3+} transitions are associated to the bands at 450 (${}^5\text{I}_8 \rightarrow {}^5\text{F}_1 + {}^5\text{G}_6$), 480 (${}^5\text{I}_8 \rightarrow {}^5\text{F}_2 + {}^5\text{F}_3$), 530 (${}^5\text{I}_8 \rightarrow {}^5\text{F}_4 + {}^5\text{S}_2$), 640 (${}^5\text{I}_8 \rightarrow {}^5\text{F}_5$) and 930 nm (${}^5\text{I}_8$

→ 5I_5). The absorption band at 980 nm corresponds to the Yb^{3+} ions ($^2F_{7/2} \rightarrow ^2F_{5/2}$). The absorption spectra did not show the localized surface plasmons (LSP) band because the volume fraction occupied by the NPs is small. However the TEM images obtained for all heat-treated samples show silver NPs with average diameter of ≈ 7 nm but filling a small volume fraction of the samples. Energy dispersive spectroscopy (EDS) data (not shown) corroborated the Ag composition of the NPs. The LSP band in the range from 400 to 500 nm was clearly observed in GeO_2 -PbO glass doped with $Er^{3+}/Tm^{3+}/Yb^{3+}$ because 4.0 wt.% of $AgNO_3$ was used and consequently larger amount of Ag NPs was nucleated [26]. The background observed in the whole spectral region is due to the rough polishing of the sample's surfaces.

Figures 2 and 3 present the PL spectra of the samples excited at 980 nm. The blue light at 477 nm is due to the emission from Tm^{3+} ions that corresponds to the transition $^1G_4 \rightarrow ^3H_6$. The green light at 545 nm is emitted by the Ho^{3+} ions, due to the transition ($^5F_4; ^5S_2$) → 5I_8 . The red light at 660 nm has contributions of both REI and the transitions responsible for this emission are $^1G_4 \rightarrow ^3F_4$ from Tm^{3+} and $^5F_5 \rightarrow ^5I_8$ from Ho^{3+} .

It can be observed for the samples prepared with 1.0 wt.% of $AgNO_3$ (samples A and B) that the amplitudes of all transitions are increased with respect to the sample without silver NPs. An enhancement of about 50% for the blue and red emissions and 113% for the green PL was obtained in sample B. On the other hand, for the samples prepared with 2.5 wt.% of $AgNO_3$ (samples C and D), a smaller PL increase is observed for the blue and red emissions while the green emission was partially quenched. This behavior is attributed to the closer proximity between the REI and the NPs in the sample with larger amount of NPs.

In order to identify the UC pathways the PL intensity, I_{UC} , was measured as a function of the laser power, P , for each UC emission. In the absence of saturation, it is expected that $I_{UC} \propto P^N$ and the slope determined in the log-log plot of I_{UC} versus P indicates the number of photons, N , that participate in the UC process. Fig 5 indicates the values of $N = 2.33$ (corresponding to 477 nm), 2.00 (at 545 nm), and 1.54 (at 665 nm) for sample B. We recall that the slopes measured for the samples prepared without silver NPs [13] were 2.47 (477 nm), 1.84 (545 nm) and 1.52 (665 nm). The UC pathways proposed in the present case are the same proposed in [13] as illustrated in Fig.4. Due to the large Yb^{3+} ions concentration and due to the larger cross-section of the transition ${}^2\text{F}_{7/2} \rightarrow {}^2\text{F}_{5/2}$ in comparison with the cross-sections of all transitions of the Ho^{3+} and Tm^{3+} ions, we attribute the efficient UC process to the ET from the excited Yb^{3+} ions to the Ho^{3+} and Tm^{3+} ions. The deviation from the expected slopes ($N=2$ and 3) is attributed to saturation of the Yb^{3+} absorption transition. The results obtained for the samples A, C and D also show slopes smaller than 2 and 3, as in sample B.

Fig 6 presents the CIE chromaticity diagram with coordinates corresponding to the emission spectra presented in Fig 2. The calculated coordinates for the samples A and B are (0.34; 0.44) and (0.32; 0.42), respectively. We recall that the coordinates of the optimized sample produced without NPs (0.31; 0.35) were already published in [13]. It can be seen that the coordinates for the samples produced with silver NPs are far from the white point (0.33; 0.33). This effect is due to the fact that the PL transition (${}^5\text{S}_2; {}^5\text{F}_4 \rightarrow {}^5\text{I}_8$) is more affected than the other transitions by the presence of the silver NPs as evidenced by the 113 % enhancement observed for the green intensity.

A different behavior was observed for the samples C and D. For these samples the CIE coordinates presented in Fig. 7 are (0.32; 0.33), (0.33; 0.33) for samples with silver NPs and (0.32; 0.37) for the sample without NPs. These results are correlated with Fig 2 which shows that the blue and red emissions were enhanced by the nucleation of silver NPs whereas the green emission was quenched. The quenching of the green emission modifies the PL spectrum in a way that WLG was obtained.

4. Conclusion

This is the first report that shows the possibility of using silver NPs for controlling the color of germanate glass samples keeping fixed the rare-earth ions concentrations. The samples were prepared by the conventional melt-quenching method followed by heat-treatment. The samples' colors changed by selecting different heat-treatment time in order to change the amount of silver nanoparticles inside the samples. A remarkable result is the white light generation obtained with samples containing REI concentrations that were not adequate for this purpose if no silver NPs were present.

Acknowledgements

We acknowledge financial support by the Brazilian agencies Conselho Nacional de Desenvolvimento Científico e Tecnológico (CNPq) and Fundação de Amparo a Ciência e Tecnologia do Estado de Pernambuco (FACEPE). The work was performed in the framework of the National Institute of Photonics (INCT de Fotônica). We acknowledge the Nanotechnology National Laboratory (LNNano) for HR-TEM measurements

References

1. R. Vadrucchi, C. Weder, Y. C. Simon, *J. Mater. Chem. C* 2 (2014) 2837-2841.
2. Y. C. Simon, C. Weder, *J. Mater. Chem.* 22 (2012) 20817-20830.
3. A. Monguzzi, F. Meinardi, *J. Phys. Chem. A* 118 (2014) 1439-1442.
4. M Haase, H. Scäfer, *Angew. Chem. Int. Ed.* 50 (2011) 5808-5829.
5. F. Auzel, *Chem. Rev.* 104 (2004) 139-174.
6. W. Xu, X. Min, X. Chen, Y. Zhu, P. Zhou, S. Cui, S. Xu, L. Tao, H. Song, *Sci. Reports* 4 (2014) 5087.
7. J. W. Wang, A. T. Peter, *J. Am. Chem. Soc.* 132 (2010) 947-949.
8. H. Gong, D. Yang, X. Zhao, E. Y. B. Pun, H. Lin, *Opt. Mater.* 32 (2010) 554-559.
9. X. Liu, B. Chen, E. Y. B. Pun, H. Lin, C. Liu, J. Heo, *Mater. Lett.* 61 (2013) 3751-3754.
10. D. Li, Y. Wang, X. Zhang, G. Shi, G. Liu, Y. Song, *J. Alloys Comp.* 550 (2013) 509-513.
11. L. X. Jun, H. Gong, B. J. Shen, H. Lin, E. Y. B. Pun, *J. Appl. Phys.* 105 (2009) 106109.
12. N. Q. Wang, X. Zhao, C. M. Li, E. Y. B. Pun, H. Lin, *J. Lumin.* 130 (2010) 1044-1047.
13. M. E. Camilo, E. de O. Silva, T. A. A. de Assumpção, L. R. P. Kassab, C. B. de Araújo, *J. Appl. Phys.* 114 (2013) 163515.
14. L. R. P. Kassab, C. B. de Araújo, *Germanate and tellurite glasses for photonic applications*, in: *Photonics Research Developments*. V. P. Nilsson (Ed.), Nova Science Publishers Inc., New York, 2008 (Chapter 13) ISBN: 978-1-60436-720-5.
15. T. Som, B. Karmakar, *J. Opt. Soc. Am. B* 26 (2009) B21-B27.

16. S. K. Singh, N. K. Giri, D. K. Rai, S. B. Rai, *Solid State Sci.* 12 (2010) 1480-1483.
17. V. A. G. Rivera, S. P. A. Osorio, D. Manzani, Y. Messaddeq, L. A. O. Nunes, E. Marega, *Opt. Mater* 33 (2011) 888-892.
18. T. T. Sun, J. Lin, H. Y. Wei, Z. B. Feng, *J. Inorg. Mater.* 26 (2011) 1215-1220.
19. T. Som, B. Karmakar, *J. Alloys Comp.* 509 (2011) 4999-5007.
20. L. R. P. Kassab, D. S. da Silva, R. A. Pinto, C. B. de Araújo, *Appl. Phys. Lett.* 94 (2009) 101912.
21. L. R. P. Kassab, D. S. da Silva, C. B. de Araújo, *J. Appl. Phys.* 107 (2010) 113506.
22. T. A. A. de Assumpção, L. R. P. Kassab, A. S. L. Gomes, C. B. de Araújo, N. U. Wetter, *Appl. Phys. B* 103 (2011) 165-169.
23. D. M. da Silva, L. R. P. Kassab, S. R. Luthi, C. B. de Araújo, A. S. L. Gomes, M. J. V. Bell, *Appl. Phys. Lett.* 90 (2007) 081913.
24. L. R. P. Kassab, F. A. Bonfim, J. R. Martinelli, N. U. Wetter, J. J. Neto, C. B. de Araújo, *Appl. Phys. B.* 94 (2009) 239-242.
25. T. A. A. de Assumpção, D. M. da Silva, L. R. P. Kassab, and C. B. de Araújo, *J. Appl. Phys.* 106 (2009) 063522.
26. M. E. Camilo, T. A. A. Assumpção, D. M. da Silva, D. S. da Silva, L. R. P. Kassab, C. B. de Araújo, *J. Appl. Phys.* 113 (2013) 153507.
27. D. S. da Silva, T. A. A. de Assumpção, L. R. P. Kassab, C. B. de Araújo, *J. Alloys Comp.* 586 (2014) S 516-S 519.
28. Y. Chiu, U. Rambabu, M. -H. Hsu, H. -P. D. Shieh, and C. -Y. Chen, *J. Appl. Phys.* 94, 1996 (2003).
29. A. Kipke and H. Hofmeister, *Mater. Chem. Phys.* 111, 254 (2008).

Figure Captions

1. (color online) Absorption spectra of the sample prepared with 2.5 wt.% of AgNO_3 . The red (black) line corresponds to the sample with Ag NPs (without NPs). The inset show images of the silver NPs nucleated in the sample. Sample thickness: 3mm.
- 2.(color online) Luminescence spectra of the $\text{Ho}^{3+}/\text{Tm}^{3+}/\text{Yb}^{3+}$ doped PbO-GeO_2 glass prepared with 1.0 wt.% of AgNO_3 . The three curves correspond to different heat-treatment times (0, 12 and 24 hours). Excitation: 200 mW at 980 nm.
- 3.(color online) Luminescence spectra of the $\text{Ho}^{3+}/\text{Tm}^{3+}/\text{Yb}^{3+}$ doped PbO-GeO_2 glass prepared with 2.5 wt % of AgNO_3 . The three curves correspond to different heat-treatment times (0, 12 and 24 hours). Excitation: 200 mW at 980 nm.
- 4.(color online) Simplified energy level diagram of the Ho^{3+} , Tm^{3+} , and Yb^{3+} ions. The radiative transitions are indicated by solid lines. The energy transfer from Yb^{3+} to Tm^{3+} and Ho^{3+} are indicated by brown dotted lines, and the phonon emissions are represented by black dashed lines.
- 5.(color online) Dependence of the upconversion intensity versus the laser power for the sample prepared with 1.0 wt.% of AgNO_3 (sample B) that was heat treated during 24 h.
- 6.(color online) CIE chromaticity diagram for the samples prepared with 1.0 wt % of AgNO_3 (samples A and B); the results for the samples without silver NPs are also shown.
- 7.(color online) CIE chromaticity diagram for the samples prepared with 2.5 wt % of AgNO_3 (samples C and D); the results for the samples without silver NPs are also shown.

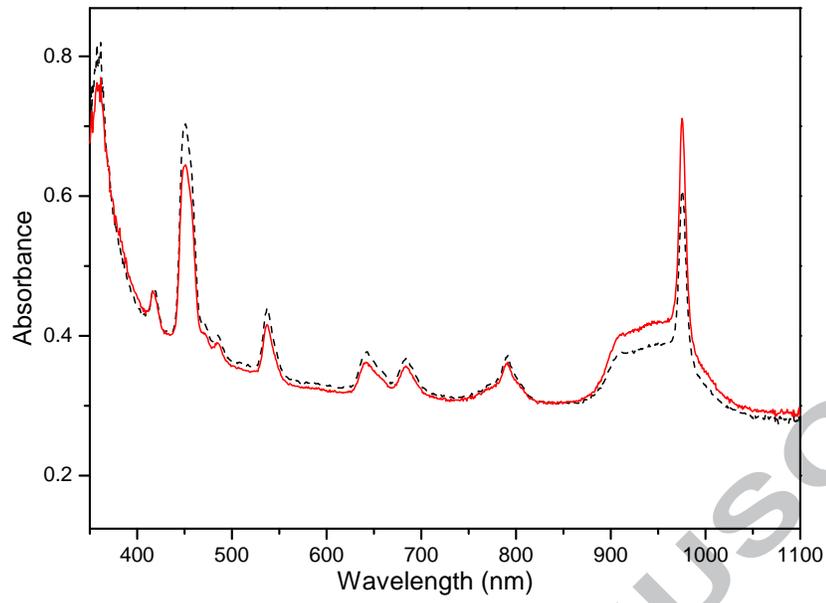


Figure 1 Camilo et al.

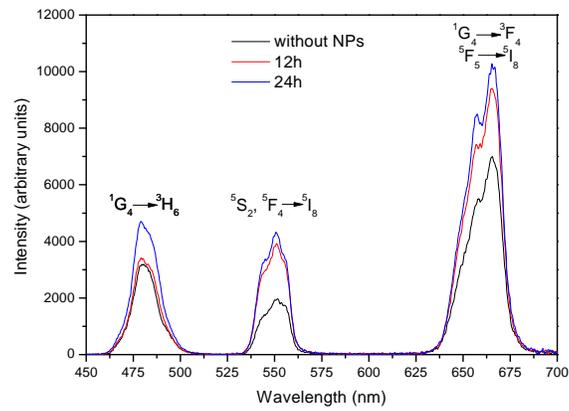


Figure 2 Camilo et al.

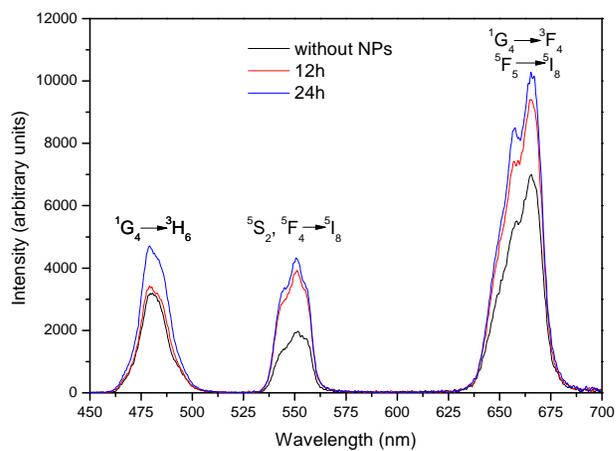


Figure 3 Camilo et al.

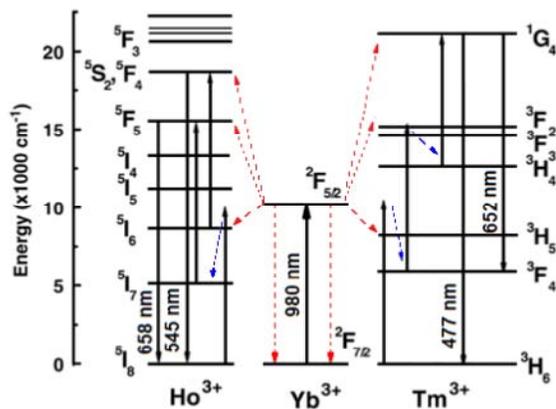


Figure 4 Camilo et al.

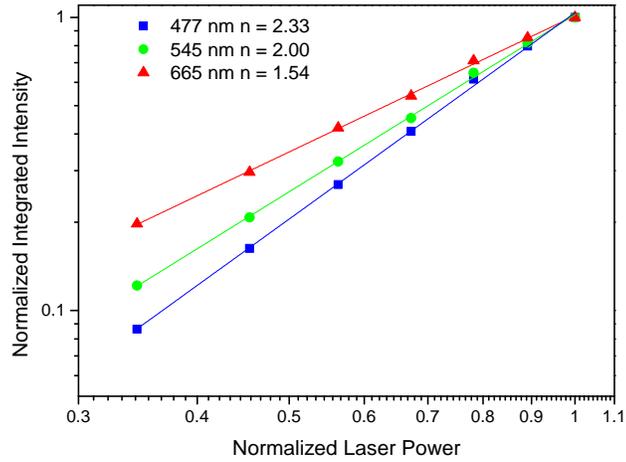


Figure 5 Camilo et al.

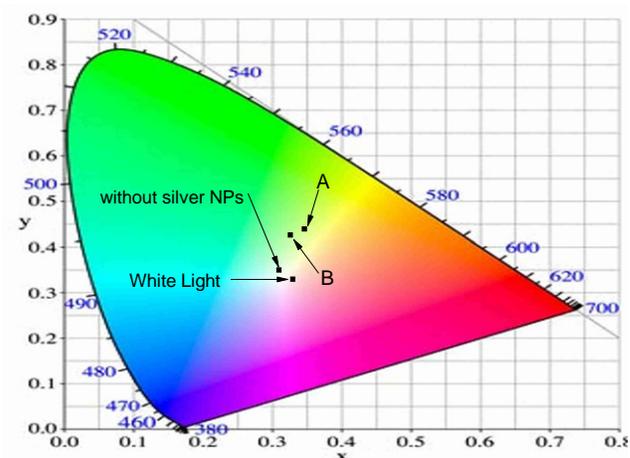


Figure 6 Camilo et al.

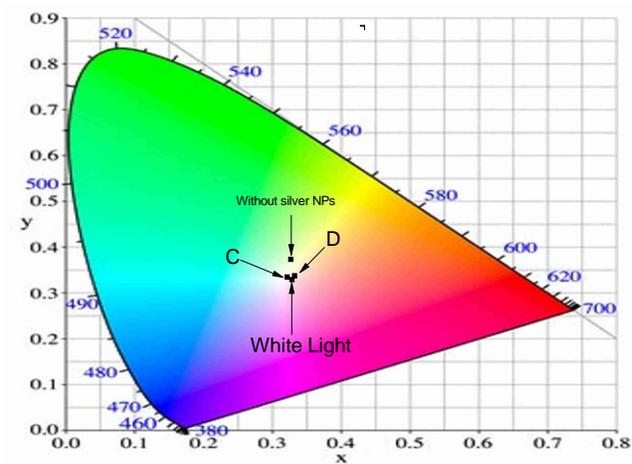
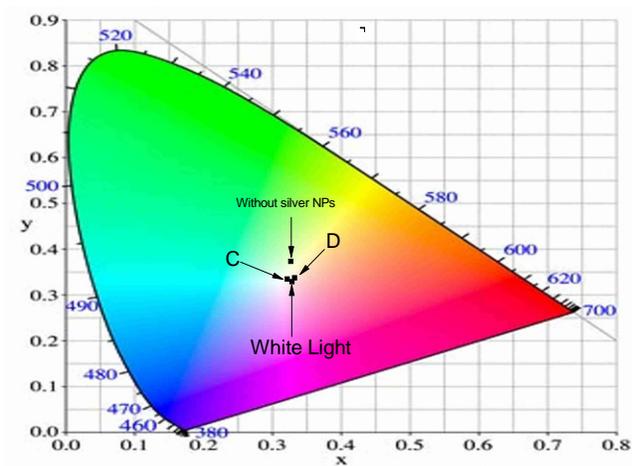


Figure 7 Camilo et al.

Graphical



Highlights

Successful fabrication of $\text{Ho}^{3+}\text{-Tm}^{3+}\text{-Yb}^{3+}$ doped lead-germanate glasses containing silver nanoparticles.

Efficient color control of $\text{Ho}^{3+}\text{-Tm}^{3+}\text{-Yb}^{3+}$ doped glass by changing the concentration of silver nanoparticles.

Surface plasmon enhanced multicolor upconversion emission in $\text{Ho}^{3+}\text{-Tm}^{3+}\text{-Yb}^{3+}$ doped lead-germanate glass.

White light generation due to the presence of silver nanoparticles