

# An electron acceptor molecule in a nanomesh: F<sub>4</sub>TCNQ on *h*-BN/Rh(111)

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## ABSTRACT

The adsorption of molecules on surfaces affects the surface dipole and thus changes in the work function may be expected. The effect in change of work function is particularly strong if charge between substrate and adsorbate is involved. Here we report the deposition of a strong electron acceptor molecule, tetrafluorotetracyanoquinodimethane C<sub>12</sub>F<sub>4</sub>N<sub>4</sub> (F<sub>4</sub>TCNQ) on a monolayer of hexagonal boron nitride nanomesh (*h*-BN on Rh(111)). The work function of the F<sub>4</sub>TCNQ/*h*-BN/Rh system increases upon increasing molecular coverage. The magnitude of the effect indicates electron transfer from the substrate to the F<sub>4</sub>TCNQ molecules. Density functional theory calculations confirm the work function shift and predict doubly-charged F<sub>4</sub>TCNQ<sup>2-</sup> in the nanomesh pores, where the *h*-BN is closest to the Rh substrate, and to have the largest binding energy there. The preferred adsorption in the pores is conjectured from a series of ultraviolet photoelectron spectroscopy data, where the  $\sigma$  bands in the pores are first attenuated. Scanning tunneling microscopy measurements indicate that F<sub>4</sub>TCNQ molecules on the nanomesh are mobile at room temperature, as “hopping” between neighboring pores is observed.

## 1. Introduction

Electron transfer is an essential process that governs many elementary processes in physics and chemistry. For atomic or molecular adsorbates on metallic surfaces simple arguments predict on whether charge is transferred from the surface to the molecule or vice versa. The relevant quantities are the work function of the surface and on the other hand the ionisation potential or electron affinity of the atom or molecule: If the work function is larger than the ionisation potential, an electron is transferred to the metal substrate and the adsorbate charges positively. This effect is well known and e.g. used for surface ionisation of alkali metal atoms [1]. If the electron affinity is larger than the work function, an electron is transferred to the adsorbate from metal substrate [2]. In the case of surface ionisation it was pointed out by Gurney [3] that the quantum mechanical broadening of the adsorbate orbitals lead to the situation where the charge transfer is not complete and thus the electron tunnels back and forth between adsorbate and substrate. Also, the distance dependent interaction between substrate and adsorbate leads to a variation of the ionisation potential and the electron

affinity, and the charge transfer itself changes the work function of the surface locally.

The charge transfer due to polarisation may be inferred from the induced change in the surface dipole. If the adsorbate density is known, the induced dipole may be determined from the Helmholtz equation that relates a change of the work function  $\Delta\Phi$  with the areal density  $n$  or coverage of the dipoles induced by the adsorbed species:

$$\Delta\Phi = -e \frac{p n}{\epsilon_0} \quad (1)$$

where  $e$  is the elementary charge,  $p$  is the induced dipole parallel to the surface normal, and  $\epsilon_0$  is the vacuum permittivity. From Eq. (1) we see that the work function decreases if charge is transferred to the surface and oppositely increases if charge is transferred to the molecule.

If the metal surface is covered with a single layer graphite (graphene, *g*) [4,5] or hexagonal boron nitride (*h*-BN) [6], the reactivity of the system changes. In particular, the charge transfer time across the layer increases, and non-adiabatic effects may be observed [7].

The high electron affinity of the tetrafluorotetracyanoquinodimethane C<sub>12</sub>F<sub>4</sub>N<sub>4</sub> (F<sub>4</sub>TCNQ; see the chemical

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structure in Fig. 1a) has made it to be a frequently employed electron acceptor molecule, with which anion formation on surfaces may be studied. Previously, Barja et al. reported self assembly of TCNQ ( $C_{12}H_4N_4$ ) and  $F_4TCNQ$  ( $C_{12}F_4N_4$ ) on the graphene/iridium system [8]. At low temperatures different adsorption geometries lead to different electronic properties, depending on where the molecules were located in the  $g/Ir(111)$  unit cell [9].

The electron affinity of  $F_4TCNQ$  is 5.2 eV [10] and exceeds the work function of the  $h$ -BN/Rh(111) nanomesh [11] by 1.05 eV. Therefore we expect electron transfer to the molecule and possibly the formation of an open-shell electronic system. This would open perspectives for spintronics as it was put forward for the TCNQ/ $g/Ru(0001)$  where long range magnetic order was found [12].

Here we report about  $F_4TCNQ$  on the  $h$ -BN nanomesh on Rh(111). We have investigated the system with photoemission spectroscopy and scanning tunnelling microscopy (STM) on the same preparations. Density functional theory (DFT) calculations complement the picture and come up with an unexpected prediction of structures with high adsorption energy and transfer of more than one electron onto the molecule.

The substrate in this study is the  $h$ -BN nanomesh on Rh(111) [13]. It features a corrugated single layer of boron nitride [14], where  $13 \times 13$  BN units form a coincidence lattice on  $12 \times 12$  Rh units with the nearest neighbour distances (Fig. 1b). The  $12 \times 12$  unit cell is constituted by 2 nm pores, where the nitrogen atoms are close to on-top Rh configuration, and where nitrogen lone pair bonding occurs [15]. These pores are separated by “wire” regions where the interaction of the  $h$ -BN with the substrate is weak. This pore-wire structure imposes strong lateral electrical fields (dipole rings) that are due to variations of the local electrostatic potential and that lead to the self assembly of molecules in the pores [11]. The electrostatic landscape is such that negative charge is attracted to the pores, and accordingly we expect  $F_4TCNQ$  to first occupy the pores.

## 2. Results and discussion

Fig. 1b displays STM images of the pristine  $h$ -BN nanomesh on Rh(111) substrate with a superstructure lattice constant of 3.2 nm (shown in the inset of Fig. 1b). In Fig. 1c a STM image of the  $h$ -BN/Rh(111) nanomesh after a small dose of  $F_4TCNQ$  with less than 1 molecule per nanomesh unit cell is shown, which corresponds to the data point marked with green square in Fig. 2a. “Bright” protrusions and “dark” depressions are imaged with the periodicity of the nanomesh. This imaging condition is obtained on a regular base after scanning the surface for some time at room temperature. The picture resembles the STM image of  $F_4TCNQ$  on  $g/Ir(111)$ , where the protrusions have been assigned to the molecules [8]. The ratio between dark and bright pores and comparison to the coverage as determined via x-ray photoemission spectroscopy (XPS) and the work function shift of 80 meV that were measured on the same sample rather favors a picture where the dark pores correspond to pores that are occupied with a molecule.

Fig. 2 shows the work function of the  $F_4TCNQ/h$ -BN/Rh(111) system as obtained from the width of He I $\alpha$  excited normal emission ultraviolet photoemission spectra (UPS) as a function of molecular coverage. The coverage  $\Theta$ , given in units of molecules per nanomesh unit cell (mpu), has been determined by Mg K $\alpha$  excited XPS intensity-ratios between the F 1s, the N 1s and the C 1s peaks and the corresponding atomic photoemission cross sections, as shown at a coverage of 7.8 mpu in Fig. 2b–d. Accordingly we find a coverage of one  $F_4TCNQ$  mpu at an atomic F:N ratio of 4:173. Furthermore we note that the splitting in the C 1s emission indicates the existence of chemically different carbon species in the molecule.

As expected for an electron transfer to the molecules the work function increases with coverage. Fitting the data in Fig. 2a to the function  $\Phi = \Phi_0 + \Delta\Phi(1 - \exp(-\Theta/\Theta_0))$  we obtain the red line in Fig. 2a and parameters  $\Phi_0 = 4.2$  eV,  $\Delta\Phi = 2.07$  eV, and  $\Theta_0 = 6.4$  mpu.

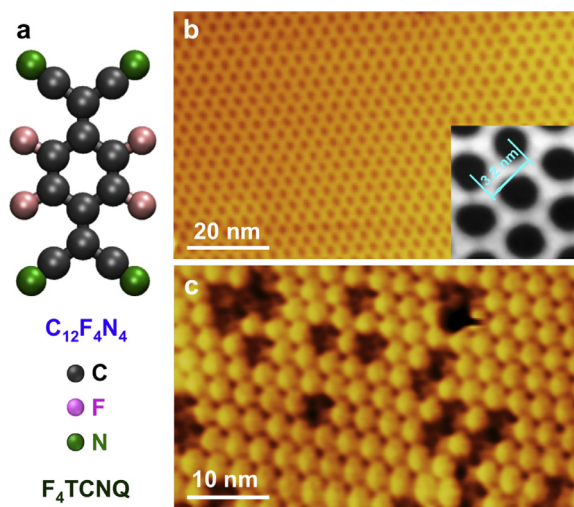


Fig. 1.  $F_4TCNQ$  on  $h$ -BN/Rh(111) surface. (a) Ball and stick model of  $F_4TCNQ$  molecule ( $C_{12}F_4N_4$ ). (b) and (c) Scanning tunnelling microscopy (STM) images at room temperature of pristine  $h$ -BN nanomesh on Rh(111).  $U_t = -1.20$  V,  $I_t = 0.50$  nA (b) and low coverage of  $F_4TCNQ$  molecules on  $h$ -BN/Rh(111) surface.  $U_t = 1.0$  V,  $I_t = 0.3$  nA (c). The right-bottom inset in (b) shows the supercell of  $h$ -BN nanomesh with a lattice constant of 3.2 nm. The wires appear bright and pores dark.

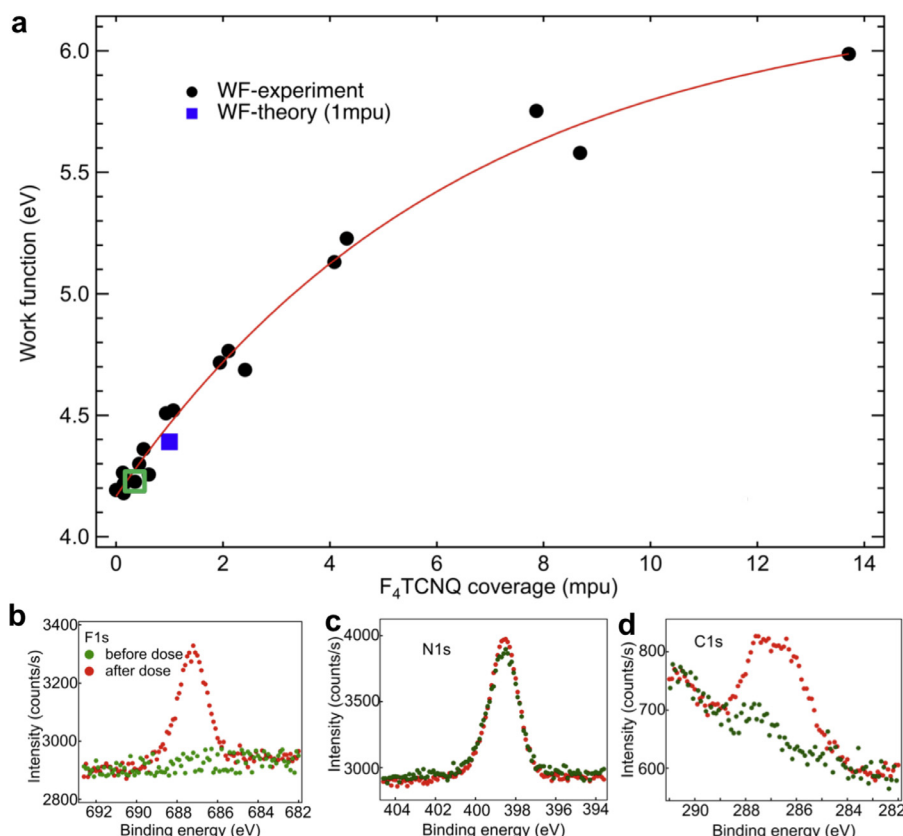
From this we see that at coverages below 2 mpu the work function increases linearly. The work function shift is much more pronounced than in the  $F_4TCNQ/g/SiC$  system where Chen et al. found an increase of 0.7 eV [16]. Given the fact that the work function of  $g/SiC$  is similar to that of  $h$ -BN/Rh(111) this is an indication that the binding of  $F_4TCNQ$  is different to  $h$ -BN on a metal from  $g$  on a semiconductor. For 1 mpu we obtain a work function shift of 0.3 eV at room temperature, and in cooling the sample to 185 K the work function further increases by 50 meV. The temperature dependent work function shift may be due to a higher occupancy in the pores. These values can be compared to the result from the DFT calculations, where we obtain a value  $\Delta\Phi_{DFT, 1 \text{ mpu}}$  of 0.19 eV, as indicated with the blue square in Fig. 2a. We consider this as very good agreement with experimental results and as a confirmation that at low coverage  $F_4TCNQ$  anions are formed on the  $h$ -BN nanomesh.

Fig. 3 shows normal emission He I $\alpha$  UPS excited valence band photoemission data of  $F_4TCNQ$  on  $h$ -BN/Rh(111) as a function of the coverage  $\Theta$ . In Fig. 3a a series of UPS spectra at different  $F_4TCNQ$  coverages from 0 to 4.3 mpu are shown. The spectrum at  $\Theta = 0$  mpu is recorded after annealing the clean nanomesh.

Afterwards  $F_4TCNQ$  has been evaporated successively and measured with XPS and UPS after each evaporation. The spectra are dominated by the well-known valence band pattern of  $h$ -BN/Rh(111) with the Rh 4d valence band peak at 0.88 eV binding energy, and the split  $h$ -BN  $\sigma$ -band [14]:  $\sigma_\alpha$  that accounts for the electronic structure in the nanomesh wires and  $\sigma_\beta$  for that in the nanomesh pores [17].

Like it was first shown for naphthalocyanine on  $h$ -BN nanomesh [14] and later for Xe on  $h$ -BN nanomesh [11], adsorbates attenuate the photoemission signals from the substrate, and since the signal from the pore,  $\sigma_\beta$  is attenuated before the signal from the wires  $\sigma_\alpha$ , a higher adsorption energy in the pores was concluded for naphthalocyanine and for 12 Xe atoms in the pores. Here we observed the same trend: With increasing molecular coverage the  $\sigma_\beta$  intensity decreases before it does the same in the  $\sigma_\alpha$  intensity. In Fig. 3b the spectra of 0.4 mpu and 1.0 mpu are compared, and the difference between the two (1.0 mpu–0.4 mpu) is displayed after scaling by a factor of 5. The above-mentioned is confirmed, in particular the attenuation effect is also seen for the Rh peaks of the substrate.

The assignment of molecular orbitals is not possible at these low coverages, photon energy and emission angle. We observe, however, a



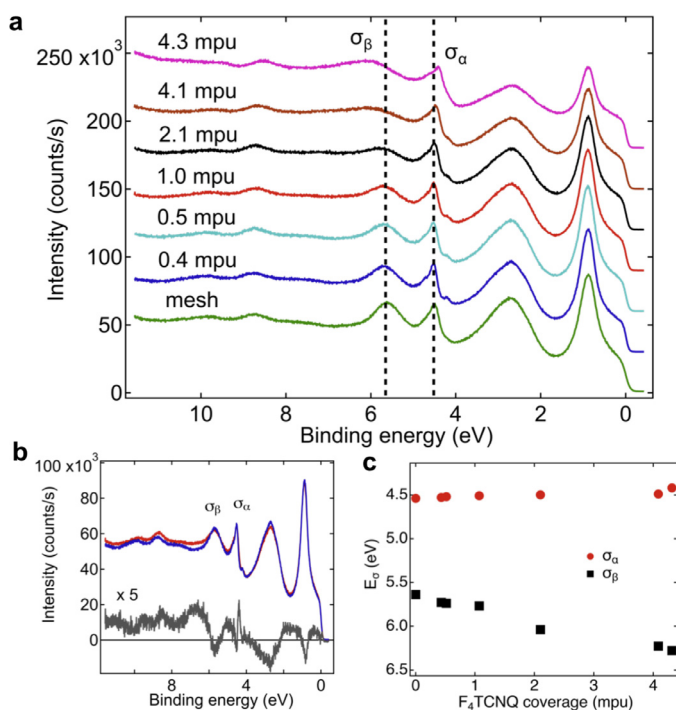
**Fig. 2.** Work function and XPS of  $F_4TCNQ/h-BN/Rh(111)$ . (a) Work function increase with increasing coverage of  $F_4TCNQ$  molecules. The black circles represent experimental data points at different  $F_4TCNQ$  molecular coverages, while the blue square indicates the calculated work function at  $F_4TCNQ$  coverage of 1 molecule per unit cell (mpu). The green open square corresponds to the STM data shown in Fig. 1c. The red line is the fit of an exponential function as described in the text. (b)–(d)  $Mg K\alpha$  XPS ( $\hbar\omega = 1253.6$  eV) spectra of  $F 1s$  (b),  $N 1s$  (c) and  $C 1s$  (d) before (green) and after (red) dosing  $F_4TCNQ$  on  $h-BN$  nanomesh (coverage of 7.8 mpu). After adsorbing on  $h-BN$  surface,  $F_4TCNQ$  shows distinct feature on  $F 1s$  and  $C 1s$  core levels. (For interpretation of the references to color in this figure legend, the reader is referred to the web version of this article.)

shift in binding energy for the boron nitride related  $\sigma$  bands. The sharpest feature in the difference curve lies at the steep onset of the  $\sigma_\alpha$  band at 4.39 eV binding energy and it is accompanied by a sharp negative peak at 4.53 eV. This indicates a  $\sigma_\alpha$  band shift to lower binding energy with increasing coverage and is remarkable since the majority of the molecules must sit in the pores, as implied by the attenuation of the  $\sigma_\beta$  intensity. The spectral weight near the broader  $\sigma_\beta$  peak appears to shift towards higher binding energies with increasing coverage. This may be related to the decrease  $\sigma_\beta$  intensity and the increase of  $F_4TCNQ$  derived molecular orbitals at around 7 eV binding energy. In Fig. 3c the binding energy positions of the  $\sigma_\alpha$  band and the local maximum near the  $\sigma_\beta$  band are shown as a function of the  $F_4TCNQ$  coverage. For flat  $h-BN$  the sigma bands are known to align with the vacuum level and to lead to the “physisorption model” [6], where on the  $Rh(111)$  substrate the  $\sigma_\alpha$  band has the same offset of about 9 eV from the vacuum level like on Ni, Pd and Pt. Therefore it is expected that with increasing work function the  $\sigma_\alpha$  energy rises. The fact that the rise of 14 meV per mpu does not correspond to the 300 meV per mpu indicates again that the vast majority of the  $F_4TCNQ$  molecules are not adsorbed on the wires. The shift in  $\sigma_\beta$  band of  $-161$  meV per mpu may not be explained with the  $h-BN$  “physisorption model” of Nagashima et al. [6]. It rather indicates that the  $F_4TCNQ^-$  molecules influence the boron and the nitrogen atoms in the pore, and the above-mentioned photoemission intensity redistribution between  $\sigma_\alpha$  and  $F_4TCNQ$  molecular orbitals that may impose as well an apparent shift.

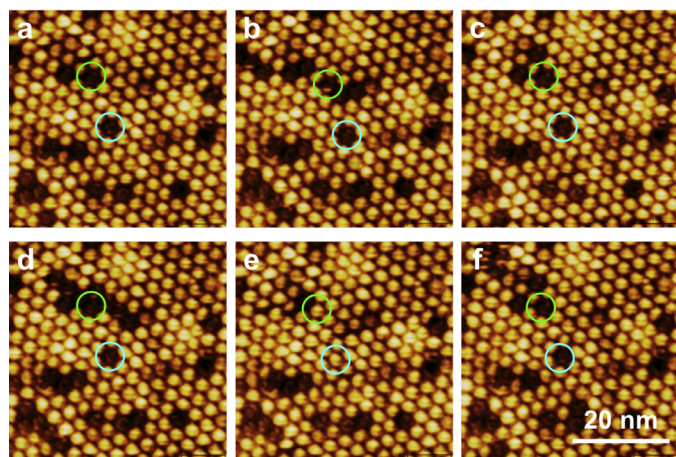
Fig. 4 shows STM images of a sample with low coverage measured with variable temperature (VT)-STM system. As mentioned above, low coverage of  $F_4TCNQ/h-BN/Rh(111)$  displays “bright” and “dark” features on the  $h-BN$  surface at room temperature, which are confirmed by two different instruments, i.e., electron spectroscopy for chemical analysis (ESCA) Park Scientific (Fig. 1c) and VT-STM (Fig. 4). Fig. 4a–f illustrate a sequence of continuous scanning of the same area with same scanning conditions, where the time intervals between the images is 125 s. The green and blue circles in each image indicate the same

positions individually in order to guide the eyes. The 6 images evidently demonstrate  $F_4TCNQ$  “hopping” on the  $h-BN$  surface at room temperature. For instance, the “dark” depressions marked with green and blue circles in Fig. 4d appear to be “bright” protrusions in the Fig. 4e, while the same positions show the “dark” features again in the subsequent Fig. 4f. In order to demonstrate more “hopping” events and in a longer time interval, we made a movie with 40 continuous STM images scanning with same condition on the same area in the Supporting Information. This “hopping” indicates high mobility of  $F_4TCNQ$  molecules on  $h-BN$  surface, which is similar to the case of  $F_4TCNQ$  on  $g/Ir$ , where the  $F_4TCNQ$  molecules were reported to “rotate” even at 77 K [8].

To better understand the experimental results, we carried out DFT calculations. Because the molecular orbitals (for example the highest occupied molecular orbital, HOMO) preserve their character of the free molecule to a large extent, in the following we refer to those orbitals as in the neutral gas-phase molecule, even when the initially lowest unoccupied molecular orbital (LUMO) gains electrons, to simplify the discussion. The DFT results confirm that the  $F_4TCNQ$  molecules like to occupy the pore of the  $h-BN$  nanomesh. The theory predicts the work function change and the charge transfer upon adsorption of one mpu. In Fig. 5 the two lowest energy structures in the nanomesh pores are depicted. They are significantly distinct to each other, in binding energy, charge transfer, work function change and coordination to the beneath  $h-BN$ . In reference to the charge transfer the two structures are labeled as “1–” and “2–”. Intriguingly, the work function change of “2–” is smaller than that of the “1–” structure, which must be related to the onset of covalent chemisorptive bonds and gain of exchange energy in the LUMO in “2–”, which apparently superseeds the Coulomb repulsion in the LUMO. On the other hand “1–” is more physisorptive with ionic and van der Waals character and an open LUMO shell. The work function change of 0.19 eV per mpu of the lowest energy “2–” structure fits well to the experimental result of 0.3 eV. From the experiments we find no direct arguments on the charge state of  $F_4TCNQ$  on  $h-BN$  nanomesh. Rather, the high DFT binding energy difference between



**Fig. 3.** He I $\alpha$  normal emission ultraviolet photoemission spectroscopy (UPS) data of F<sub>4</sub>TCNQ/h-BN/Rh(111). (a) A series of UPS spectra of F<sub>4</sub>TCNQ/h-BN/Rh(111) at different coverages of F<sub>4</sub>TCNQ, showing that the  $\sigma_\alpha$  and  $\sigma_\beta$  bands shift upon increasing of F<sub>4</sub>TCNQ molecules. The two black dashed lines are a guide to the eyes. (b) Comparison of UPS spectra of 0.4 molecule per unit cell (mpu) (blue) and 1.0 mpu (red). The grey line is the difference between these two spectra (1.0 mpu-0.4 mpu). In order to better demonstrate the difference, the zero line has been added, and the data are scaled by a factor of 5. Clearly, the  $\sigma_\alpha$  and  $\sigma_\beta$  bands are affected by adding more F<sub>4</sub>TCNQ molecules, where the  $\sigma_\beta$  band decreases significantly. (c) Shift of peak positions of  $\sigma_\alpha$  and  $\sigma_\beta$  bands with the increase of F<sub>4</sub>TCNQ molecules. The  $\sigma_\alpha$  band shifts towards lower binding energy, while the peak in the  $\sigma_\beta$  band-region shifts to higher binding energy. (For interpretation of the references to color in this figure legend, the reader is referred to the web version of this article.)



**Fig. 4.** F<sub>4</sub>TCNQ “hopping” at low coverage below 1 molecule per unit cell. (a)–(f) A series of sequential STM images of F<sub>4</sub>TCNQ on h-BN/Rh(111) at low coverage of the same area with the same scanning conditions show the F<sub>4</sub>TCNQ molecules “hopping” (appearing and disappearing) on the surface at room temperature. The green and blue circles represent the same positions in every image to guide the eyes. The time interval between two subsequent frames is 125 s. The contrast between dark and bright protrusions, as derived from the histogram of the z-values of panel (d), is 2.6 Å.  $U_t = -1.0$  V,  $I_t = 0.005$  nA. (For interpretation of the references to color in this figure legend, the reader is referred to the web version of this article.)

**Table 1**

Calculated work functions  $\Phi$  in 5 different structures. The binding energies  $E_{\text{bind}}$  in 4 different molecular conformations and the Bader charges on the molecule  $q_{\text{Bader, molecule}}$  in the 3 single molecule structures.

System	$E_{\text{bind}}$ (eV)	$\Phi$ (eV)	$q_{\text{Bader, molecule}}$ (e)
clean	–	4.20	–
1-	3.07	4.61	–1.34
2-	4.15	4.39	–1.89
wire	1.78	4.88	–0.40
dimer (pore)	3.64	4.88	

molecules in the pore and on the wire (see Table 1), would not suggest much diffusion as we observe it at room temperature.

We have to speculate on how the detailed hopping mechanism as we observe it (see Fig. 4) operates: Possibly the STM imaging process triggers hopping events. In Fig. 5c and f the orbital overlap  $\mathcal{O}$ , or the projection of the Kohn–Sham electronic states of the adsorption system on the molecular orbitals of a single molecule (but in the geometry as adsorbed), is shown for “1–” and “2–”. The orbital overlap indicates at which energies the molecular orbitals reside when adsorbed, and if there are large changes, for example due to hybridisation. There are clearly defined states, in particular the HOMO-1, HOMO and LUMO, and the states above the LUMO are separated, like for the molecule in the gas phase. The gas phase eigenvalues are indicated with solid circles: Black for the occupied orbitals, and red for unoccupied ones. The energy alignment between gas phase and adsorbate system is arbitrary, and done to fit the HOMO and LUMO reasonably well. In the structure “F<sub>4</sub>TCNQ<sup>2–</sup>”, where the Bader charge of the molecule is close to  $-2e$ , the two fold degenerate, i.e., spin-degenerate, LUMO of the gas-phase molecule is clearly below the Fermi energy, thus agreeing with the charge state “2–”. Instead the charge state of “F<sub>4</sub>TCNQ<sup>1–</sup>” seems to pin at the Fermi level, indicating half filling.

In Table 1 we also show the binding energy for a F<sub>4</sub>TCNQ<sup>2–</sup> dimer. It is significantly lower than that of two molecules in two different pores. From this we expect single pore occupancy to be the lowest energy structure at low coverages.

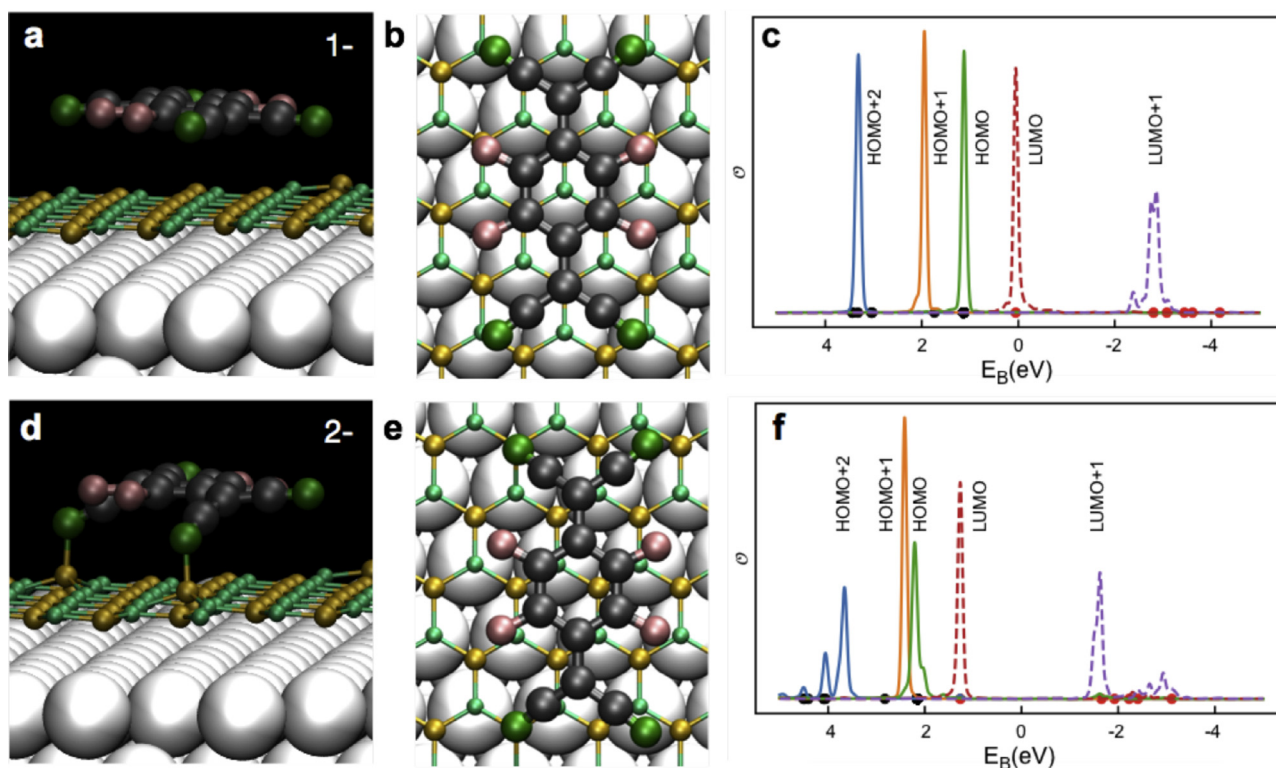
### 3. Conclusions

In conclusion, combining controlled experimental photoemission and scanning tunneling microscopy measurements with density functional theory calculations, we demonstrate electron transfer from h-BN/Rh(111) to F<sub>4</sub>TCNQ. As negatively charged molecules (electron acceptors) coalesce on the surface, the work function of the system increases. Upon charging, the  $\sigma_\alpha$  and  $\sigma_\beta$  bands of h-BN nanomesh shift towards lower and higher binding energy, respectively. The molecules occupy preferentially the pores of the h-BN/Rh(111) nanomesh, though display mobility and molecular “hopping” on the h-BN surface at room temperature. DFT results indicate that the lowest energy structure is an anionic “2–” species. Our work paves the way to tune the electronic and structural properties of two-dimensional materials by using adsorptions of organic molecules.

### 4. Methods

#### 4.1. Experimental

The experiments were performed in two ultrahigh-vacuum (UHV) systems with base pressure of  $1 \times 10^{-10}$  mbar. One is a variable-temperature STM (Omicron VT-STM), and the other is a user-modified Vacuum Generators ESCALAB 220 with a Mg K $\alpha$  lab source at an energy of  $\hbar\omega = 1253.6$  eV, with a monochromatized He I $\alpha$  ( $\hbar\omega = 21.2$  eV) source equipped with a room-temperature STM (Park Scientific) that allows photoemission and STM measurements on the same sample [18,19]. The work function has been determined from the width of the



**Fig. 5.** DFT results of two different  $F_4TCNQ$  structures in the  $h$ -BN/Rh(111) nanomesh pores. (a)–(c) The “1–” structure with one electron transfer. (a) Perspective view showing a floating molecule, (b) top view: The molecule is centered on top of the  $h$ -BN nitrogen atom with on top of Rh coordination and (c) the orbital overlap  $\phi$  with the LUMO pinned at the Fermi level. (d)–(f) The “2–” structure with two electrons transferred, that has a 26% higher binding energy than “1–”. (d) Perspective view, showing the covalent bonding of two cyano (CN) groups to two boron atoms which get pulled off the  $h$ -BN layer. (e) Top view: The molecule is shifted away from the pore center such that two CN-B bonds become operational and (f) the orbital overlap  $\phi$ , where the LUMO lies below the Fermi level. Note that the distinct distortion of the molecules imposes a non-rigid shift of the molecular orbital energies. The solid circles in (c) and (f) are the gas phase eigenvalues. Colour code for atomic models: Boron orange, carbon black, fluorine pink, nitrogen green, rhodium whitish. (For interpretation of the references to color in this figure legend, the reader is referred to the web version of this article.)

He I $\alpha$  photoemission spectrum recorded at a sample bias of  $-9$  V. The STM measurements were carried out with electrochemically etched tungsten tips. All STM images were taken in constant-current mode at room temperature. At negative tunneling voltages  $U_t$  electrons are tunneling from the substrate to the tip. The  $h$ -BN/Rh(111) samples were produced with the standard recipe [13]. Particularly purified  $F_4TCNQ$  molecules were evaporated on  $h$ -BN/Rh(111) substrates kept at room temperature using a Knudsen cell at 365 K. Then the samples were transferred to the analysis chambers for the photoemission or the STM measurements.

#### 4.2. Theory

Calculations were performed using the Kohn–Sham DFT formalism within the Gaussian plane wave (GPW) method [20] as implemented in the QuickStep module [21] in the CP2K program package [22]. The exchange-correlation term was approximated with the rB86-vdW-DF2 exchange-correlation functional [23] that explicitly includes the van der Waals interactions. In the GPW scheme we used the expansion of the molecular orbitals with Gaussian basis functions [24], and the electron density is expressed with a plane wave basis set up to 700 Ry with a relative cut-off of 70 Ry. Dual-space pseudopotentials [25] were used to describe the interaction of valence electrons with atomic cores. We sampled the first Brillouin zone at  $\Gamma$  point only. The theoretical lattice constant of Rh of 3.8147 Å was employed in the calculations. The substrate was modelled with four layers of Rh(111), of which the two lowest were kept fixed at their bulk positions during the relaxation.

#### Conflict of interest

The authors declare no competing financial interest.

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#### Supplementary material

Supplementary material associated with this article can be found, in the online version, at doi:10.1016/j.susc.2018.04.026.

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