



# Highly sensitive detection of H<sub>2</sub>O adsorbed on Si(111)7 × 7 and Si(100)2 × 1 surfaces by means of slow highly charged Xe ions

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## ABSTRACT

The proton yields from Si(111)7 × 7 and Si(100)2 × 1 reconstructed surfaces irradiated with slow ( $v < 0.25v_{\text{Bohr}}$ ) highly charged Xe ions are obtained. H<sub>2</sub>O molecules adsorbed over time on the surfaces at room temperature, under ultra-high vacuum can be detected as an increase of the proton yield. For the Si(100)2 × 1 surface, the proton yield with HCl-irradiation time is discussed based on temporal variation of the H<sub>2</sub>O coverage. The proton desorption efficiency with Xe<sup>50+</sup> is more than about ten times and twice as compared with Xe<sup>29+</sup> and Xe<sup>44+</sup>, respectively. For the Si(111)7 × 7 and Si(100)2 × 1 surfaces, the proton yield in each time increases with the charge states  $q$  to the power of 6 and 4, respectively, and is changed with time with the power laws held on.

## 1. Introduction

A highly charged ion (HCI) has a large potential energy,  $E_p$ , which is the summation of the ionization energies obtained by removing electrons from a neutral atom. Being slower than the electrons in a solid ( $< 10^6$  m/s), when a HCI approaches a solid surface, electrons move from the surface to the HCI within a few femtoseconds, which is shorter than the time period during which the HCI interacts with the surface. The HCI then releases most of its  $E_p$  resulting in the emission of secondary electrons, ions, neutrals, and photons caused by capturing electrons from the surface. The emission mechanism of secondary ions and neutrals, which is known as potential sputtering, differs from that due to kinetic sputtering [1–3]. With HCI irradiation, protons are emitted as typical secondary ions when hydrogen species (e.g., H<sub>2</sub>, H<sub>2</sub>O, or other hydride molecules) have been adsorbed either physically or chemically as impurities on the material. Proton desorption is caused by Coulomb repulsion between a proton and an adjacent ion after giving out two electrons from a chemical bond of hydrogen (pair-wise potential sputtering (PWPS) [4]). The proton yield increases with the charge state,  $q$ , to the power of 3–6 [5–8].

Proton desorption depends on surface chemical property. For

example, in the case of gold surfaces covered with a surfactant monolayer (ML), the proton yield from carboxyl group of mercaptoundecanoic acid (HS(CH<sub>2</sub>)<sub>10</sub>COOH) is larger than that from methyl group of dodecanethiol (HS(CH<sub>2</sub>)<sub>11</sub>CH<sub>3</sub>). A hydrogen atom in the carboxyl group is easily desorbed as compared with that in the methyl group. In the process of HCI capturing two bonding electrons from a chemical bond, the hydrogen atom is more easily ionized as a proton from OH bond than from CH bond, because the electronegativity of oxygen is higher than that of carbon [9]. A HCI captures electrons into its outer shells. In case of the fluorine adsorbed on Si(100)2 × 1 surface, the angular distribution of F<sup>+</sup> ion has been observed. The desorbed F<sup>+</sup> ion image indicates the bond direction and angle between F and Si on the surface. Above the surface, a HCI captures electrons from a chemical bond of F–Si into its high Rydberg states, and desorbs F without destroying surface structure before colliding with the surface [10].

In this study, we focus on clean reconstructed Si surfaces that contain almost no hydrogen. The structure of a reconstructed Si surface after being cleaned at 1500 K differs from that of the bulk. In a Si(100) surface, a Si atom on the surface has two dangling bonds and approaches an adjacent Si atom, where the two atoms bond to form a 2 × 1 Si dimer [11]. The remaining single dangling bonds of the Si atoms remain on the

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surface and are exposed by the residual gas, under ultra-high vacuum (UHV). Dissociative adsorption of  $\text{H}_2\text{O}$ , which is one of the main constituents of the residual gas, occurs on the surface [12]. The  $\text{H}_2\text{O}$  coverage depending on the flux of  $\text{H}_2\text{O}$  molecules changes constantly until saturation and have been obtained using temperature-programmed desorption (TPD) [13]. In the case of a Si(111) surface, the structure of a Si(111) $7 \times 7$  surface, called the dimer adatom stacking-fault model, is reconstructed to from the top layer to the third layer of the surface [14]. As there are 19 dangling bonds in the  $7 \times 7$  unit cell, dissociative adsorption of  $\text{H}_2\text{O}$  occurs on the surface similar to that of a Si(100) $2 \times 1$  surface [12].

In this paper, we show that the high  $q$ -HCIs (up to  $q=50$ ) usefully act as the ultra-sensitive probe for the investigation into the initial stage of the hydrogen-adsorption process, which is a development from previous work [15].

## 2. Experimental details

HCIs were produced at the University of Electro-Communications in an electron-beam ion trap (EBIT) known as the Tokyo EBIT [16], where they were transported to a collision chamber using a beamline [17]. The HCIs had a kinetic energy  $E_k = 3.5q$  keV owing to the acceleration voltages 3 kV and  $-0.5$  kV that were applied to the drift tube in the EBIT and to a sample, respectively. The collision chamber was equipped with an instrument for time-of-flight secondary ion mass spectrometry (TOF-SIMS) and a low-energy electron diffraction (LEED) apparatus, where the base pressure was  $4.0 \times 10^{-8}$  Pa. The partial pressure of the residual gas was obtained using a quadrupole mass spectrometer, the main constituents of the residual gas at the base pressure were estimated as 35%  $\text{H}_2$ , 31%  $\text{H}_2\text{O}$ , 25% CO and negligible  $\text{CO}_2$ .

The TOF-SIMS setup has been described elsewhere [8]. Secondary ions were detected with a micro-channel plate (MCP) detector with a resistive anode, i.e., a position sensitive detector (PSD). It was confirmed that secondary ions were collected within the active area of the PSD with cylindrical electrostatic lenses; trajectories of proton were simulated using SIMION 3D software [18], and every trajectory ended in the active area for the protons having less than 25 eV. Protons are dominantly emitted within a few tens eV due to potential sputtering [6]. Therefore, the errors of collection efficiency of secondary ions in this TOF-SIMS instrument is  $98.1 \pm 0.5\%$  negligible compared with others. Three meshes were attached to the TOF tube and lenses. The transmission efficiency of secondary ions passing through the meshes was  $48.5 \pm 0.5\%$ , as estimated from the open-area ratio of the mesh. The detection efficiency of the MCP for 1.5-keV  $\text{H}^+$  was assumed to be  $60.5 \pm 5\%$ , after Gao et al. [19]. Considering these factors, the absolute efficiency was determined to be  $\eta = 28.8 \pm 1.8\%$ . The secondary ion counts and yields measured in this work were corrected by dividing by  $\eta$ .

Si(111) and Si(100) samples were cut into  $5 \text{ mm} \times 20 \text{ mm}$  size from 0.5-mm-thick Si(111) and Si(100) n-type wafers with resistivities of 10 and 1  $\Omega\text{cm}$ , respectively. The samples were cleaned ultrasonically in (i) acetone for 10 min and (ii) wash fluid (Semico Clean 23; Furuuchi Chemical Co., Ltd., Japan) for 10 min and then rinsed in ultra-pure water ( $> 18 \text{ M}\Omega\text{cm}$ ). The samples were heated in situ at 750 K to degas them and their holders under UHV until the pressure had stabilized near the base pressure for at least 10 h. The samples were flashed repeatedly at 1500 K by resistive heating and were cooled slowly, at rates of less than 2 K/s, from 1200 K to room temperature [20]. The Si(111) $7 \times 7$  and Si(100) $2 \times 1$  structures were confirmed from their LEED patterns, and no contamination was detected within the sensitivity of Auger electron spectroscopy (AES).

Most of the Si(111) $7 \times 7$  and Si(100) $2 \times 1$  samples were irradiated with  $\text{Xe}^{q+}$  ( $q=29, 38, 44, 47$  and  $50$ ); however, one of the Si(100) $2 \times 1$  samples was not irradiated with  $\text{Xe}^{47+}$ . The intensity of the Xe HCI beam through a 1-mm-diameter aperture was less than 1700/s. The surface normal of the samples was oriented so that the angle of incidence of the

Xe HCI beam was  $60^\circ$ . The filament of the ionization gauge was switched off to avoid generation of atomic hydrogen whose sticking probability is higher than that of  $\text{H}_2$ . The samples were flashed at 1500 K for a few seconds before being irradiated with Xe HCIs. The TOF-SIMS measurement was recorded immediately after sample flashing; this was done to obtain the proton yield that was proportional to the amount of  $\text{H}_2\text{O}$  adsorbed on the surface, which in turn increased with time. In this paper, time indicates HCI-irradiation time from a started record of TOF-SIMS measurement cycle.

## 3. Results and discussion

Fig. 1 shows the TOF spectra from Si(111) $7 \times 7$  surfaces irradiated with Xe HCIs at 2.5 min after sample flashing. The peaks are identified as photons,  $\text{H}^+$ ,  $\text{H}_2^+$  and  $\text{Si}^{n+}$  ( $n \leq 4$ ). The intensity of  $\text{H}^+$  from the Si(111) $7 \times 7$  surface irradiated with  $\text{Xe}^{50+}$  was one order of magnitude smaller than that from a hydrogen-terminated Si(111) $1 \times 1$  surface in previous work [8]. The peaks of  $\text{Si}^{n+}$  are obviously shaped with the charge state  $q$ . The each yield of  $\text{Si}^{n+}$  increases with the charge state  $q$  to the power of 2, and are affected by potential sputtering, since the yields of  $\text{H}^+$ ,  $\text{Si}^+$  and  $\text{Si}^{2+}$  from a hydrogen-terminated Si(111) $1 \times 1$  surface is almost independent of kinetic energy of HCIs in the energy range of 119–257 keV [21].  $\text{Si}^{n+}$  ions produced by HCI capturing electrons acquires desorption momentum of Si atom by Coulomb repulsion force between  $\text{Si}^{n+}$  ions. Fig. 2 shows the TOF spectra from a Si(111) $7 \times 7$  surface irradiated with  $\text{Xe}^{50+}$  at 20 and 140 min after sample flashing. At the later time, the intensity of  $\text{H}^+$  had increased drastically and those of  $\text{O}^+$  and  $\text{O}^{2+}$  had increased only slightly, which is clear evidence for the dissociative adsorption of  $\text{H}_2\text{O}$ .

Fig. 3 (a) and (b) show the proton yields  $Y_{\text{H}^+}^{(111)}$  and  $Y_{\text{H}^+}^{(100)}$  from Si(111) $7 \times 7$  and Si(100) $2 \times 1$  surfaces obtained in 20 and 45 min, respectively, after sample flashing. Each value of  $Y_{\text{H}^+}^{(111)}$  and  $Y_{\text{H}^+}^{(100)}$  is a 5-min average, which increased linearly with time after sample flashing, and increased with the charge state  $q$ . The intersections of the proton yields with the horizontal-axis are  $-20$  to  $-28$  and  $-5$  to  $-10$  min, respectively. In Fig. 3(b),  $\text{H}_2\text{O}$  adsorption on the surface from 800 K to room temperature occurs for several min. In Fig. 3(a), it is considered that adsorbed  $\text{H}_2\text{O}$  have slightly been remaining on the surface after sample flashing because of insufficient flashing time. The increase in the amount of  $\text{H}_2\text{O}$  adsorbed on the surfaces could be detected as increases of  $Y_{\text{H}^+}^{(111)}$  and  $Y_{\text{H}^+}^{(100)}$ . Increases per min of proton yield, namely  $dY_{\text{H}^+}^{(111)}/dt$  and  $dY_{\text{H}^+}^{(100)}/dt$ , are given in Table 1.

Fig. 4 (a) and (b) show charge state dependence of proton yield from Si(111) $7 \times 7$  and Si(100) $2 \times 1$  surfaces in each time after sample flashing. The proton yields in Fig. 4(a) and (b) are translated from those in Fig. 3(a) and (b), respectively. For Si(111) $7 \times 7$  and Si(100) $2 \times 1$  surfaces, the proton yield in each time increases with the charge state  $q$  to the power of 6 and 4, respectively. The power law are applied to the

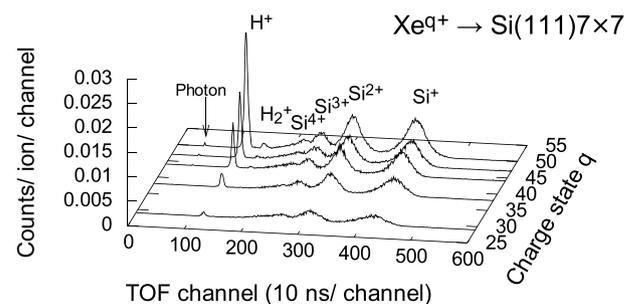
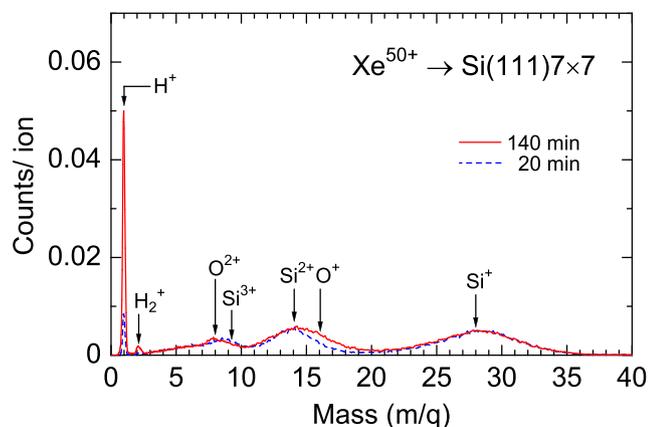


Fig. 1. Time-of-flight spectra from Si(111) $7 \times 7$  surfaces irradiated with 3.5q-keV  $\text{Xe}^{q+}$  ( $q=29, 38, 44, 47$  and  $50$ ) at 2.5 min after sample flashing.

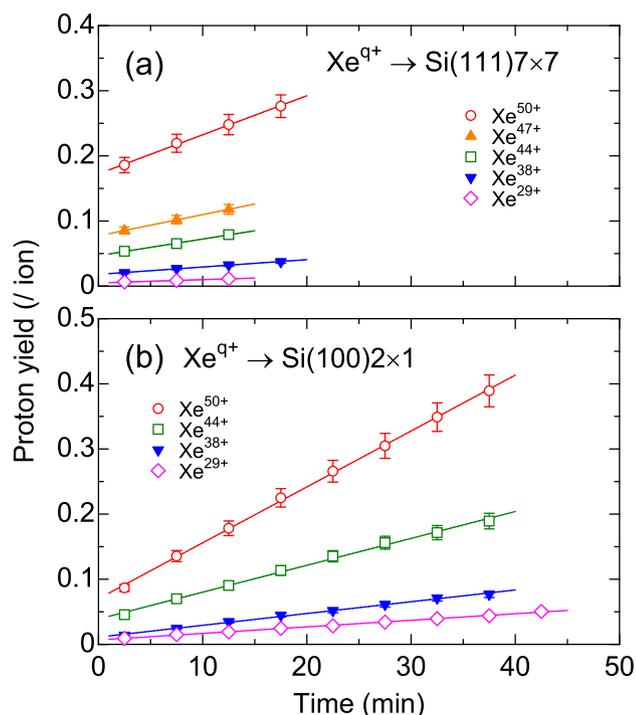


**Fig. 2.** Mass spectra from a Si(111)7 × 7 surface irradiated with 175-keV Xe<sup>50+</sup>. The dashed (blue) and solid (red) line represent the mass spectra at 20 and 140 min, respectively, after sample flashing.

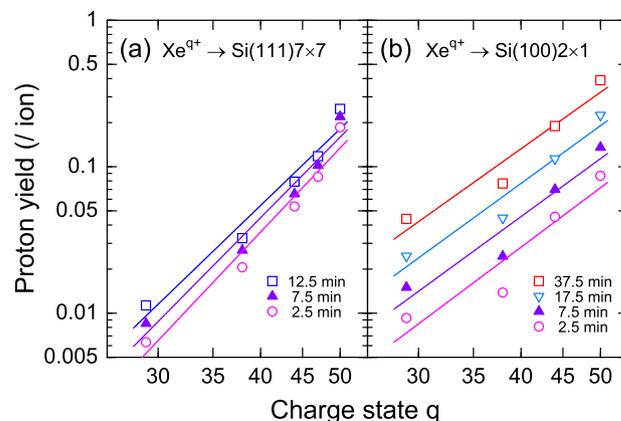
**Table 1**

For Si(111)7 × 7 and Si(100)2 × 1 surfaces, increases per min of proton yield, and ratios of differential of proton yield to that of H<sub>2</sub>O coverage.

$q$	$dY_{\text{H}^+}^{(111)}/dt \times 10^{-3}$ (/ion min)	$dY_{\text{H}^+}^{(100)}/dt \times 10^{-3}$ (/ion min)	$dY_{\text{H}^+}^{(100)}/d\theta_{\text{H}_2\text{O}}^{(100)}$ (/ion ML)
50	$6.0 \pm 0.4$	$8.6 \pm 0.5$	$2.8 \pm 0.6$
44	$2.6 \pm 0.2$	$4.1 \pm 0.3$	$1.3 \pm 0.3$
38	$1.1 \pm 0.1$	$1.8 \pm 0.1$	$0.58 \pm 0.12$
29	$0.50 \pm 0.03$	$1.0 \pm 0.1$	$0.32 \pm 0.07$



**Fig. 3.** Proton yields from (a) Si(111)7 × 7 surfaces and (b) Si(100)2 × 1 surfaces irradiated with Xe HCIs versus time after sample flashing. The open circles (red), solid triangles (orange), open squares (green), solid upside-down triangles (blue) and open diamonds (pink) represent the proton yields for Xe<sup>50+</sup>, Xe<sup>47+</sup>, Xe<sup>44+</sup>, Xe<sup>38+</sup> and Xe<sup>29+</sup>, respectively.



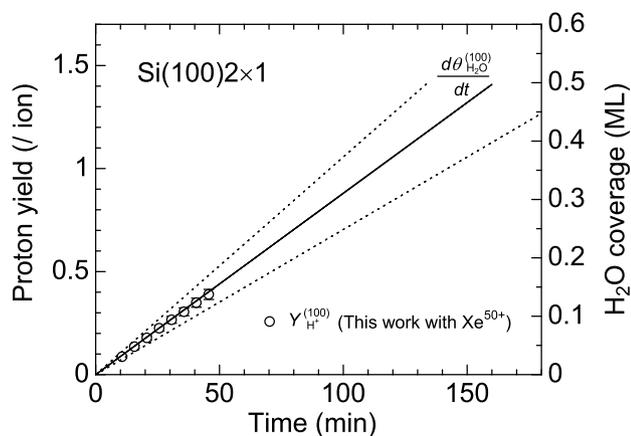
**Fig. 4.** Charge state dependence of proton yield from (a) Si(111)7 × 7 surfaces and (b) Si(100)2 × 1 surfaces in each time after sample flashing. Proton yields of (a) and (b) are translated from those in Fig. 3(a) and (b), respectively. For each value, the error is within ±6–9%. The open circles (pink), solid triangles (purple), open squares (blue), open upside-down triangles (light blue) and open squares (red) represent the proton yields at 2.5, 7.5, 12.5, 17.5 and 37.5 min, respectively, after sample flashing.

PWPS model predicted in lower charge states ( $q \leq 10$ ) [4]. With the power law  $q^6$  and  $q^4$  held on, the proton yields are changed with time. The power law depends on not time but the charge state  $q$ .

In the case of a Si(100)2 × 1 surface, we deduce that H<sub>2</sub>O is preferentially adsorbed on the surface for the following two reasons. (1) The sticking probability of H<sub>2</sub>O on the surface at room temperature is unity, which is sufficiently higher than those of the other residual-gas components H<sub>2</sub>, CO, and CO<sub>2</sub> [22,23]. (2) The impinging rate of H<sub>2</sub>O is estimated to be  $P/\sqrt{2\pi mk_B T} \simeq (3.6 \pm 0.7) \times 10^{14}/\text{m}^2\text{s}$ , where  $P$  estimated as the partial pressure of H<sub>2</sub>O at the base pressure was  $(1.0 \pm 0.2) \times 10^{-8}$  Pa,  $m$  is the H<sub>2</sub>O mass,  $k_B$  is the Boltzmann constant, and  $T$  is 300 K (room temperature), and is large enough compared with that of incident Xe HCIs, which is estimated as being less than  $2.2 \times 10^9/\text{m}^2\text{s}$ . This indicates that Xe HCIs do not prevent H<sub>2</sub>O from adsorption on the surface. In TOF-SIMS measurement, it must be emphasized that surface damage in the case of HCl irradiation is extremely little as compared with that in the case of singly charged ion of which the intensity, e.g., a few nA, is seven orders of magnitude larger than that of HCl. The H<sub>2</sub>O coverage,  $\theta_{\text{H}_2\text{O}}^{(100)}$ , is ideally half a ML in saturation, where the surface is fully terminated by –H and –OH. In this HCl-SIMS experiment, we define the proton desorption efficiency to be the ratio of  $dY_{\text{H}^+}^{(100)}/dt$  to the H<sub>2</sub>O adsorption rate, namely  $d\theta_{\text{H}_2\text{O}}^{(100)}/dt$ . The proton desorption efficiency needs to consider as the average efficiency of proton desorption from Si–OH and Si–H. Fig. 5 shows the  $Y_{\text{H}^+}^{(100)}$  obtained in this work with Xe<sup>50+</sup> (open circles). The intersection of the proton yields with the horizontal-axis in Fig. 3(b) is corrected to 0 min. The  $Y_{\text{H}^+}^{(100)}$  is compared with the H<sub>2</sub>O adsorption rate (solid line) calculated from the sticking probability of H<sub>2</sub>O at unity and the partial pressure of H<sub>2</sub>O. The H<sub>2</sub>O adsorption rate is estimated to be  $(3.1 \pm 0.6) \times 10^{-3}$  ML/min. The proton desorption efficiency, namely  $dY_{\text{H}^+}^{(100)}/d\theta_{\text{H}_2\text{O}}^{(100)}$ , was calculated as given in table 1. These results show a remarkable increase of the proton desorption efficiency with the charge state  $q$ . For example, the proton desorption efficiency with Xe<sup>50+</sup> is more than about ten times and twice as compared with Xe<sup>29+</sup> and Xe<sup>44+</sup>, respectively.

#### 4. Conclusion

With Xe HCl irradiation, the proton yields increase with increasing



**Fig. 5.** The proton yields (open circles) from the Si(100)2 × 1 surface irradiated with Xe<sup>50+</sup> versus time. The intersection of the proton yields with the horizontal-axis in Fig. 3(b) is corrected to 0 min. The solid and dotted lines represent the H<sub>2</sub>O coverage, which is calculated from the sticking probability of H<sub>2</sub>O at unity and the partial pressure of H<sub>2</sub>O at  $(1.0 \pm 0.2) \times 10^{-8}$  Pa, and minimum and maximum errors of the H<sub>2</sub>O coverage.

the amount of H<sub>2</sub>O adsorbed on Si(111)7 × 7 and Si(100)2 × 1 surfaces over time after sample flashing. In particular, the proton desorption efficiency as derived from  $dY_{H^+}^{(100)}/d\theta_{H_2O}^{(100)}$  for the Si(100)2 × 1 surface with Xe<sup>50+</sup> is  $3.5 \pm 1.1$  /ion ML, and is more than about ten times and twice as compared with Xe<sup>29+</sup> and Xe<sup>44+</sup>, respectively. For the Si(111)7 × 7 and Si(100)2 × 1 surfaces, the proton yield in each time increases with the charge state  $q$  to the power of 6 and 4, respectively, and is changed with time with the power laws held on. The power law depend on not time but the charge state  $q$ . The proton desorption efficiency indicates the power law  $q^6$  and  $q^4$ , respectively. A high  $q$ -HCI is a highly sensitive probe for hydrogen on various materials.

#### CRedit authorship contribution statement

**Satoshi Takahashi:** Investigation, Writing - original draft. **Masahide Tona:** Investigation, Writing - review & editing. **Nobuyuki Nakamura:** Conceptualization, Writing - review & editing. **Chikashi Yamada:** Conceptualization, Writing - review & editing. **Makoto Sakurai:** Conceptualization, Writing - review & editing. **Shunsuke Ohtani:** Supervision.

#### Declaration of Competing Interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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