

Reaction kinetic analysis of damage rate effects on defect structural evolution in Fe–Cu

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ABSTRACT

In Fe–Cu alloys, Cu precipitates are formed during high-energy particle irradiation. If there exists energetic binding between vacancies and Cu atoms, vacancy clusters (voids) are formed in precipitates at an initial stage of irradiation, separate from voids in the matrix, because of the migration of Cu atoms with vacancies. In this paper, the damage rate dependence on the formation and annihilation of voids in the precipitates and in the matrix is simulated by reaction kinetic analysis. The initial formation of voids at precipitates, the annihilation of them with an increased dosage and new formation of voids in the matrix are simulated, and the results are compared with the experiments. In a high damage rate of 3.3×10^{-7} dpa/s, the formation of voids in Cu precipitates is not significant, but the formation of voids in the matrix is dominant, different from those in a low damage rate of 1.5×10^{-10} dpa/s.

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1. Introduction

For the development of nuclear materials, accelerated irradiation experiments are usually employed to reduce irradiation time. In order to estimate the lifetime of nuclear reactor pressure vessel steels from the irradiation experiment using materials testing reactors, we have to understand the damage rate dependence of defect structural evolution. Clustering of point defects and precipitation are strong damage rate dependence.

As a model system, Fe–0.6wt.% Cu alloy was chosen. Cu atoms are almost insoluble in Fe below 800 K and Cu precipitates are formed in Fe–Cu alloys during thermal aging at high temperatures, as well as upon high-energy particle irradiation [1,2]. The precipitates obstruct dislocation motion and induce embrittlement in Fe-based alloys containing Cu impurities, such as old commercial reactor pressure vessel (RPV) steels. Besides Cu precipitates, defect clusters such as interstitial clusters and vacancy clusters are also formed during irradiation, which are further contributing factors to increasing hardness and decreasing ductility of RPV steels.

Nagai et al. have been reported the formation of voids and Cu precipitates, and concluded that voids were surrounded by Cu precipitates [3]. Xu et al. also detected the coexistence of Cu precipitates and voids [4,5]. In our previous simulations of the Fe–Cu system, we did not consider the energetic binding between Cu atoms and vacancies. Without the binding, the formation of voids in Cu clusters did not occur [6,7]. In this paper, reaction kinetic analysis, using rate equations, was performed to simulate the Cu

precipitation in the Fe–Cu system introducing the binding energy between Cu atoms and vacancies. The damage rate dependence of precipitation and void growth was also discussed.

2. Reaction kinetic analysis

The calculation model is based on the rate theory. The alloy calculated is Fe–0.6wt.% Cu. The model describes the reaction rates among various point defect reactions [8,9]. The following assumptions were made in the calculation:

- (1) Mobile defects are interstitials, vacancies and Cu–vacancy pairs.
- (2) The binding and the thermal dissociation are considered for Cu–vacancy pairs.
- (3) The time dependence of eleven variables is calculated up to 1 dpa: the concentration of interstitials, interstitial clusters (interstitial type dislocation loops), vacancies, vacancy clusters (voids) in the matrix, Cu atoms, Cu–vacancy pairs, Cu–vacancy clusters, the total interstitials in interstitial clusters, the total vacancies in vacancy clusters in the matrix, the total vacancies in void in Cu–vacancy clusters, and the total Cu atoms in Cu–vacancy pairs. The cluster size is taken as an average one.
- (4) Di-interstitials and di-vacancies are set for stable nuclei of clusters [10,11]. Direct formation of clusters in cascades, interstitial clusters (three interstitials) and vacancy clusters (three vacancies), is also taken in the simulation. The formation rate is assumed to be 1.0×10^{-5} of the total point defect production as explained in 4.1.

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Table 1

Values used in simulation. The values with * were determined to fit experimental results as discussed in 4.1. Z denotes the number of reaction sites without $Z_{I,IC}$, $Z_{V,IC}$, and $Z_{VCu,IC}$.

Z^*	$Z_{I,IC}^*$	$Z_{VCu,IC}^*$	$Z_{V,IC}^*$	P_{IC}^*	P_{VC}^*	E_I	E_V	B_{VCu}^*	$B_{V,PC}^*$	$B_{V,Void}^*$	C_S	M_{VCu}^*
10	44	40	40	$10^{-5}P$	$10^{-5}P$	0.15 eV [12]	0.57 eV [13]	0.1 eV	1.0 eV	1.59 eV [14]	10^{-10} [10]	$0.01M_I$

(5) The material temperature is 573 K during irradiation.

The concentrations of interstitials (C_I), vacancies (C_V), Cu atoms in the matrix (C_{Cu}) and Cu–vacancy pairs (C_{VCu}) are expressed as:

$$\frac{dC_I}{dt} = P_I - 2Z_{I,I}M_I C_I^2 - Z_{I,V}(M_I + M_V)C_I C_V - Z_{I,VCu}(M_I + M_{VCu})C_I C_{VCu} - Z_{I,IC}M_I C_I S_{IC} - Z_{I,VC}M_I C_I S_{VC} - Z_{I,PC}M_I C_I S_{PC} - M_I C_I C_S - N_I P_{IC},$$

$$\frac{dC_V}{dt} = P_V + B_{V,Cu}M_V C_{VCu} + B_{V,PC}M_V S_{PC} - 2Z_{V,V}M_V C_V^2 - Z_{I,V}(M_I + M_V)C_I C_V - Z_{V,Cu}M_V C_V C_{Cu} - Z_{V,VCu}(M_V + M_{VCu})C_V C_{VCu} - Z_{V,VC}M_V C_V S_{VC} - Z_{V,IC}M_V C_V S_{IC}$$

$$- Z_{V,PC}M_V C_V S_{PC} - M_V C_V C_S - N_V P_{VC}, \frac{dC_{VCu}}{dt} = Z_{V,Cu}M_V C_V C_{Cu} - B_{V,Cu}M_V C_{VCu} - Z_{I,VCu}(M_I + M_{VCu})C_I C_{VCu} - Z_{V,VCu}(M_V + M_{VCu})C_V C_{VCu} - Z_{VCu,Cu}M_{VCu} C_{VCu} C_{Cu} - 2Z_{VCu,VCu}M_{VCu} C_{VCu}^2 - Z_{VCu,VC}M_{VCu} C_{VCu} S_{VC} - Z_{VCu,IC}M_{VCu} C_{VCu} S_{IC} - Z_{VCu,PC}M_{VCu} C_{VCu} S_{PC} - M_{VCu} C_{VCu} C_S,$$

$$\frac{dC_{Cu}}{dt} = Z_{I,VCu}(M_I + M_{VCu})C_I C_{VCu} + B_{V,Cu}M_V C_{VCu} - Z_{V,Cu}M_V C_V C_{Cu} - Z_{VCu,Cu}M_{VCu} C_{VCu} C_{Cu}.$$

where P is the production rate of interstitials and vacancies, and Z is the number of sites in the spontaneous reaction of each process. M is the mobility of defects and is expressed as $\nu \exp(-\frac{E_M}{kT})$, where ν is an effective frequency associated with vibration of the defects in the direction of the saddle point and taken as $10^{13}/s$. E , k , and T are the migration energy, the Boltzmann constant and temperature, respectively. S is the sink efficiency to mobile defects [10,11]. N is the number of atoms in clusters formed directly in cascades. B is the dissociation probability of vacancies with voids and is expressed as $\exp(-\frac{K}{kT})$, where K is the binding energy. $K_{V,Cu}$, $K_{V,PC}$ and $K_{V,Void}$ are the binding energies between vacancies and Cu atoms, between vacancies and voids in Cu–vacancy clusters, and between vacancies and voids in the matrix, respectively. The subscripts I , V , Cu , VCu , PC , IC , VC , and S denote interstitials, vacancies, Cu atoms, Cu–vacancy pairs, Cu–vacancy clusters, interstitial type dislocation loops, voids, and surfaces, respectively. The surfaces are expressed by sink efficiency C_S . It is almost $(a/L)^2$ at the center of foil specimens in the case of random walk of point defects, where a and h are atomic distance and foil thickness, respectively [10].

The concentrations are fractional units. S is expressed as

$$S_{VC} = (48\pi^2 R_{VC} C_{VC}^2)^{1/3}, \\ S_{IC} = 2(\pi R_{IC} C_{IC})^{1/2}, \\ S_{PC} = (48\pi^2 (R_{PV} + R_{PCu}) C_{PC}^2)^{1/3}.$$

The nucleation rates of interstitial type dislocation loops (concentration, C_{IC}), voids in the matrix (C_{VC}), and Cu–vacancy clusters (C_{PC}) are:

$$\frac{dC_{IC}}{dt} = P_{IC} + Z_{I,I}M_I C_I^2,$$

$$\frac{dC_{VC}}{dt} = P_{VC} + Z_{V,V}M_V C_V^2,$$

$$\frac{dC_{PC}}{dt} = Z_{V,VCu}(M_V + M_{VCu})C_V C_{VCu} + Z_{VCu,VCu}C_{VCu}^2 + Z_{VCu,Cu}M_{VCu} C_{VCu} C_{Cu}.$$

P_{IC} and P_{VC} are the production rates of interstitial type dislocation loops and voids directly from cascades, and determined to be 1.0×10^{-5} of the total point defect production as explained in 4.1.

The total accumulation of interstitials in loops (R_I), that of vacancies in voids (R_V) from the matrix and that of vacancies in voids from Cu–vacancy clusters (R_{PV}) are

$$\frac{dR_{IC}}{dt} = 2Z_{I,I}M_I C_I^2 + Z_{I,IC}M_I C_I S_{IC} + Z_{VCu,IC}M_{VCu} C_{VCu} S_{IC} - Z_{V,IC}M_V C_V S_{IC} + N_I P_{IC},$$

$$\frac{dR_{VC}}{dt} = 2Z_{V,V}M_V C_V^2 + Z_{I,VC}M_V C_V S_{VC} + Z_{VCu,VC}M_{VCu} C_{VCu} S_{VC} - Z_{I,VC}M_I C_I S_{VC} + N_V P_{VC},$$

$$\frac{dR_{PV}}{dt} = 2Z_{V,VCu}(M_V + M_{VCu})C_V C_{VCu} + 2Z_{VCu,VCu}M_{VCu} C_{VCu}^2 + Z_{VCu,Cu}M_{VCu} C_{VCu} C_{Cu} - Z_{VCu,PC}M_{VCu} C_{VCu} S_{PC} - Z_{I,PC}M_I C_I S_{PC} + Z_{V,PC}M_V C_V S_{PC} - B_{V,PC}M_V S_{PC}.$$

The total accumulation of Cu atoms in Cu–vacancy clusters (R_{PCu}) is

$$\frac{dR_{PCu}}{dt} = Z_{V,VCu}(M_V + M_{VCu})C_V C_{VCu} + 2Z_{VCu,VCu}M_{VCu} C_{VCu}^2 + 2Z_{VCu,Cu}M_{VCu} C_{VCu} C_{Cu} - Z_{VCu,PC}M_{VCu} C_{VCu} S_{PC}.$$

The parameters used are listed in Table 1. The values marked by * were adjusted to fit to previous experimental results [5], and an experimental result shown in Fig. 6 as discussed in 4.1.(see Table 2).

3. Results

The change of the concentrations in the point defects, the point defect clusters, Cu atoms, Cu–vacancy pairs, Cu atoms in Cu clusters, and vacancies in Cu–vacancy clusters under a damage rate of 1.5×10^{-8} dpa/s is shown in Fig. 1. The increase of Cu atoms in Cu clusters, the decrease of isolated Cu atoms in the matrix, and the growth of interstitial type dislocation loops and voids in the matrix are seen. Fig. 2 shows the relationship between the concentration of vacancies and Cu atoms in Cu–vacancy clusters as three damage rates. At the initial stage of irradiation, Cu atoms and vacancies increases linearly, and then the vacancy concentration decreases at the dose indicated in each figure. The decrease is the dissociation of vacancies from voids in Cu–vacancy clusters.

The dose dependence of the concentration of vacancies in voids in the matrix and in Cu–vacancy clusters is compared in Fig. 3. We determined the binding energies of vacancies in voids in

Table 2

Irradiation doses in Fig. 6. From 1' to 9' and from a to d were irradiated at 573 K with 1.5×10^{-8} dpa/s and at 563 K with 3.3×10^{-7} dpa/s, respectively.

Symbols	1'	2'	3'	4'	5'	6'	7'	8'	9'	a	b	c	d
Dose (dpa)	4×10^{-6}	1.3×10^{-5}	4×10^{-5}	1.2×10^{-4}	4×10^{-4}	1.2×10^{-3}	3×10^{-3}	6×10^{-3}	1.6×10^{-2}	9.6×10^{-3}	4×10^{-2}	7.4×10^{-2}	0.2

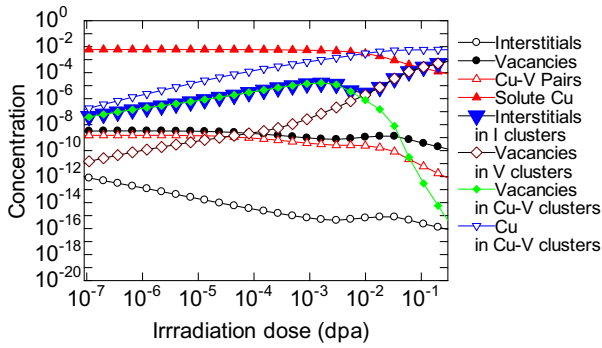


Fig. 1. The irradiation dose dependence of the concentrations of the point defects, the point defect clusters, Cu atoms, Cu-vacancy pairs, Cu atoms in Cu clusters and vacancies in Cu-vacancy clusters under a damage rate of 1.5×10^{-8} dpa/s.

Cu-vacancy clusters and voids in the matrix to be 1.0 eV and 1.59 eV, respectively so as to fit experimental results as discussed in 4.1. Therefore the decrease of vacancies in Cu-vacancy clusters is caused by the low binding energy between vacancies and voids in Cu-vacancy clusters. The total vacancy concentrations in voids in Cu-vacancy clusters and voids in the matrix at three irradiation damage rates are shown in Fig. 4 with dose. The vacancy concentration of high damage rate is higher than that of low damage rate in the range lower than 10^{-1} dpa. This damage rate dependence is

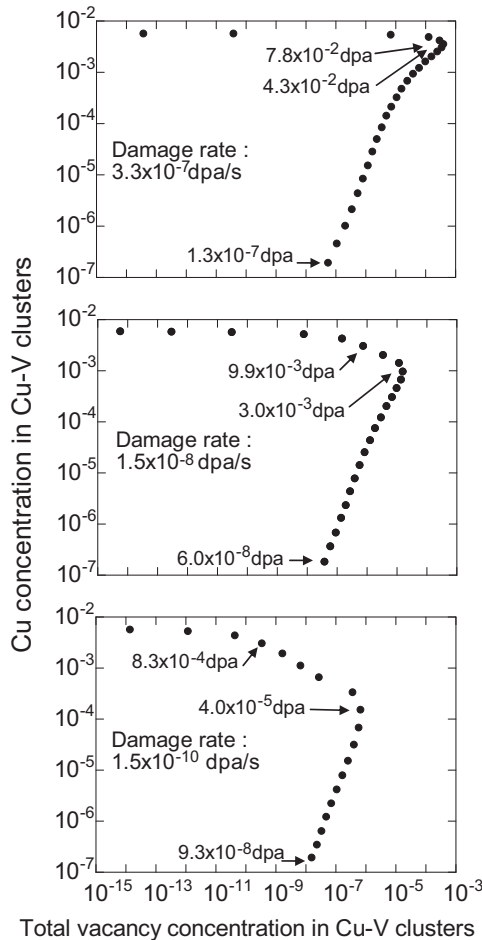


Fig. 2. Damage rate dependence of vacancies and Cu concentrations in Cu-vacancy clusters. The arrows indicate damages at points. 7.8×10^{-2} , 9.9×10^{-3} and 8.3×10^{-4} dpa are doses that half of Cu atoms are in Cu-vacancy clusters.

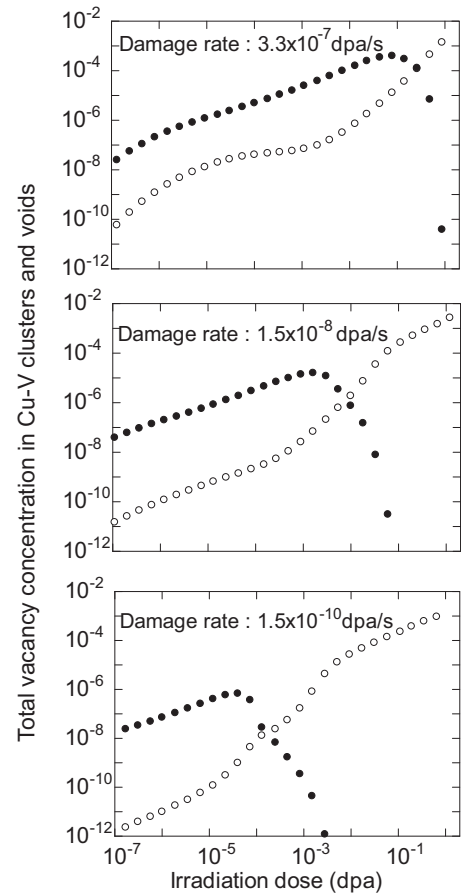


Fig. 3. Vacancy concentrations in voids in Cu-vacancy clusters (solid circles) and in the matrix (open circles).

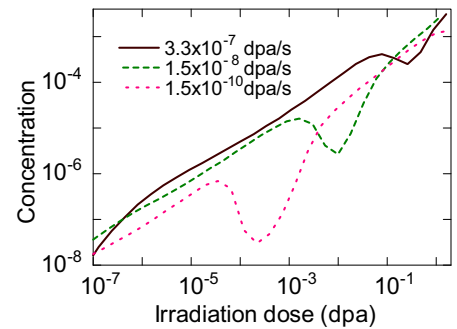


Fig. 4. Irradiation dose dependence of the total vacancy concentrations in voids in Cu-vacancy clusters and in the matrix.

the same as those with no binding energy between vacancies and Cu atoms [6,7]. The slope of the line before and after matrix voids formation is very similar. It can be explained as follows. The time dependence of vacancy accumulation in voids (dR_{VC}/dt) and in vacancy-Cu clusters (dR_{PV}/dt) is functions of many variables such as C_V and C_I . These values are almost constants or smooth functions of time (dpa) as shown in Fig. 1. Therefore dR_{VC}/dt and dR_{PV}/dt are almost constants, and R_{VC} and R_{PV} increase linearly with time (dpa).

Fig. 5 shows the relationship between the total vacancy concentration in voids in Cu-vacancy clusters and in the matrix, and Cu concentration in Cu-vacancy clusters. It can be seen that the relationship between Cu concentration and the total accumulated

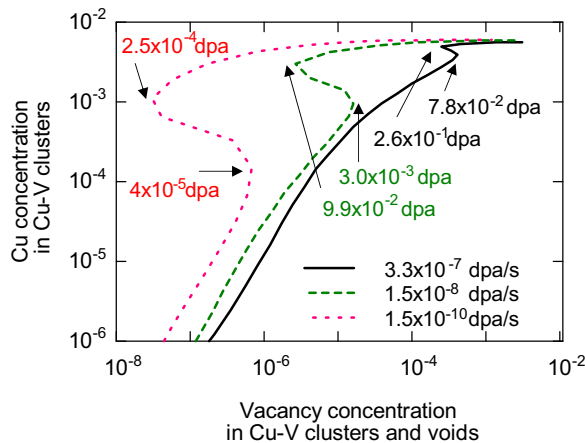


Fig. 5. Dose dependence of the concentration of Cu atoms in Cu-vacancy clusters, and vacancies in voids in Cu-vacancy clusters and in the matrix.

vacancy concentration is almost linear in the high damage rate of 3.3×10^{-7} dpa/s.

4. Discussion

4.1. Comparison with experiments

The authors have reported the formation of Cu precipitates and point defect clusters in Fe–Cu binary model alloys [4,5]. Results of positron annihilation experiments indicated that Cu precipitates were formed in these irradiations with different damage rates. Cu precipitates did not grow monotonously when increasing the irradiation dose. Fig. 6 shows examples of coincidence Doppler broadening measurements of Fe-0.6wt.% Cu irradiated under improved temperature control with an irradiation facility SSS at the Kyoto University Reactor (1.5×10^{-8} dpa/s [16], 1'–9') at 573 K and with Japan Materials Testing Reactor (3.3×10^{-7} dpa/s, a–d) of Japan Atomic Energy Agency at 563 K. The irradiation doses in the figure are listed in Table 1. Two parameters, S and W , are defined as the ratio of the low momentum ($|P_L| < 4 \times 10^{-3} m_0c$) and high momentum ($20 \times 10^{-3} m_0c < |P_L| < 30 \times 10^{-3} m_0c$) regions in the Doppler broadening spectrum, respectively, to the total region [4,5]. S represents the smaller Doppler shift resulting from annihilation of positrons with valence electrons, and the increase in the size or density of vacancies causes S to increase. W comes from annihilation of positrons with core electrons, and the increase in size or density of Cu precipitates increases W . The changes in S and W are not independent. The change in size and density of voids or precipitates can change both S and W , since S and W are defined as the ratio of certain regions in the Doppler broadening spectrum to the total region. For example, in pure Fe, there is no formation of Cu precipitates, but the formation of voids increases S and consequently decreases W as shown in Fig. 6. The decrease of S parameter from 7' (3.0×10^{-3} dpa) in Fig. 6 means the decrease of vacancy concentration in Cu-vacancy clusters. In our simulation as shown in Fig. 2 and Fig. 5, the start of the decrease at 3.0×10^{-3} dpa was replicated by adjusting values marked by * in Table 1. For example, if $B_{V,PC}$ decreased, the start of the decrease of vacancy accumulation (3.0×10^{-3} dpa) became high dose, and if P_{VC} increased, the vacancy accumulation in voids was higher than that in experiments.

4.2. Damage rate dependence of Cu precipitation

The Cu precipitation in Cu-vacancy clusters has strong damage rate dependence. In the simulation of the high damage rate of

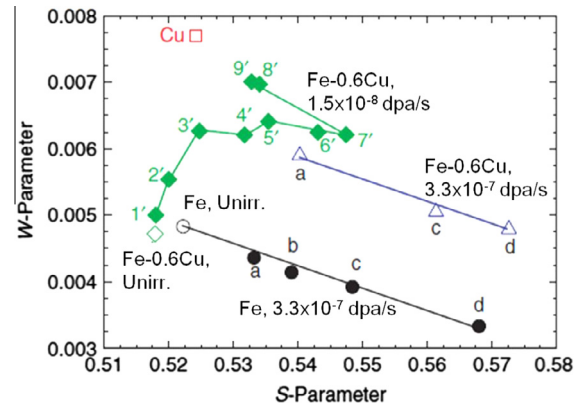


Fig. 6. Coincidence Doppler broadening measurement of Fe-0.6wt.% Cu and Fe irradiated with 1.5×10^{-8} dpa/s at 573 K and with 3.3×10^{-7} dpa/s at 563 K. Solid squares and open triangles are Fe-0.6wt.% Cu, and solid circles are Fe. Unirradiated Cu, Fe and Fe-0.6wt.% Cu are also indicated in the figure. Irradiation doses are in Table 2.

3.3×10^{-7} dpa/s, the precipitation continues to a high dose as shown in Fig. 2. It is caused by the consumption of vacancies by vacancy-interstitial mutual annihilation. The computational result (Fig. 5) shows that the total vacancy concentrations in voids in the matrix and in Cu-vacancy clusters at 7.8×10^{-2} dpa and 2.6×10^{-1} dpa are 4.1×10^{-4} and 2.5×10^{-4} , respectively. The Cu concentrations in Cu-vacancy clusters at 7.8×10^{-2} dpa and 2.6×10^{-1} dpa are 0.06% and 0.19%, respectively. According to the calculation, d (0.2 dpa) of Fe-0.6wt.% Cu with 3.3×10^{-7} dpa/s in Fig. 6 should be at the left side of c (7.4×10^{-2} dpa/s). Therefore the present simulation is not fully replicating the experimental result. Smaller difference of S between c and d in Fe-0.6wt.% Cu than that between c and d in Fe is, however, the evidence of the decrease of S between 7.8×10^{-2} dpa and 2.6×10^{-1} dpa. In the low damage rate of 1.5×10^{-10} dpa/s, the precipitation was accelerated.

4.3. Damage rate dependence of vacancy accumulation

The vacancy accumulation in Cu clusters is caused by the binding between Cu atoms and vacancies. Cu atoms are added to precipitates with the help of vacancies, and vacancies also accumulate at the precipitate forming voids. The binding energy has not been reported yet. Even the volume size factor of Cu in Fe is not determined yet as 17.53% [17], 15% [18] and 0.4% [19]. In order to explain the experimental results of the voids in Cu clusters, however, the binding between Cu atoms and vacancies is inevitable.

There are two types of vacancy accumulation. One is the void formation in Cu clusters. The other is that in the matrix without Cu atoms. The shrinkage of initially formed voids in Cu-vacancy clusters is caused by evaporation of vacancies from Cu-vacancy clusters. The binding energy of 1.0 eV is used to fit the experimental result of 1.5×10^{-8} dpa/s. After the shrinkage of voids in Cu-vacancy clusters, the growth of voids in the matrix occurs. It is caused by the difference of the binding energy between vacancies in voids in Cu-vacancy clusters and those in voids in the matrix. The binding energy of 1.5 eV is used for vacancies in voids in the matrix. The binding energy of vacancies in large voids is the same as the formation energy of vacancies. The formation energies of vacancies in Fe and Cu are 1.59 eV [14] and 1.28 eV [15], respectively. In small Cu-vacancy clusters in Fe, Cu atoms are bcc structure and the formation energy of vacancies must be lower than that in fcc structure. Therefore the binding energy of 1.0 eV used is a reasonable value.

5. Concluding remarks

The damage rate dependence of Cu precipitation and void growth was simulated for Fe–Cu alloy by using reaction kinetic analysis. The results almost replicated our previous coincidence Doppler broadening measurements. The shrinkage of voids in Cu–vacancy clusters was explained by the easy evaporation of vacancies from voids surrounding by Cu atoms. In the low damage rate of 1.5×10^{-10} dpa/s, the precipitation was accelerated, indicating the promotion of the Cu precipitation in nuclear power reactors. Our experiments were upon a model alloy of Fe–Cu and the composition was largely different from that in pressure vessel steels. However, there is a strong possibility of the same precipitation behavior in the steels.

The simulation was a preliminary one. For the next step, the following points should be included in the rate equations. (1) Initial inhomogeneous distribution of Cu atoms, (2) coalescence of Cu precipitate, (3) thermal dissociation of small clusters such as di-vacancies and (4) migration mechanism of Cu–vacancy pairs.

References

- [1] J.T. Buswell, C.A. English, M.G. Hetherington, W.J. Phythian, G.D.W. Smith, G.M. Worrall, Effects of radiation on materials, in: N.H. Packan, R.E. Stoller, A.S. Kumar (Eds.), 14th International Symposium, ASTM STP, vol. II, American Society for Testing and Materials, Philadelphia, PA, 1990, p. 127.
- [2] P.J. Othen, M.L. Jenkins, G.D.W. Smith, W.J. Phythian, *Phil. Mag. Lett.* 64 (1991) 383.
- [3] Y. Nagai, Z. Tang, M. Hassegawa, T. Kanai, M. Saneyasu, *Phys. Rev. B* 63 (2001) 134110.
- [4] Q. Xu, T. Yoshiie, K. Sato, *Phys. Rev. B* 73 (2006) 134115.
- [5] Q. Xu, T. Yoshiie, *Phil. Mag.* 91 (2011) 3716.
- [6] S. Yanagita, Q. Xu, T. Yoshiie, H. Ino, *Mater. Trans.* 43 (2002) 1663.
- [7] S. Yanagita, T. Yoshiie, H. Ino, *J. Jpn. Inst. Metals* 64 (2000) 115.
- [8] T. Yoshiie, Q. Xu, K. Sato, K. Kikuchi, M. Kawai, *J. Nucl. Mater.* 377 (2008) 132.
- [9] T. Yoshiie, T. Ito, H. Iwase, Y. Kaneko, M. Kawai, I. Kishida, S. Kunieda, K. Sato, S. Shimakawa, F. Shimizu, S. Hashimoto, N. Hashimoto, T. Fukahori, Y. Watanabe, Q. Xu, S. Ishino, *Nucl. Inst. Meth. Phys. Res. B* 269 (2011) 1740–1743.
- [10] N. Yoshida, M. Kiritani, *Phys. Soc. Jpn.* 35 (1973) 1418.
- [11] M. Kiritani, H. Takata, Y. Maehara, *Phys. Soc. Jpn.* 38 (1975) 1677.
- [12] H.E. Schaefer, D. Butteweg W. Dander, in: M.T. Robinson, R.W. Young, Jr. (Eds.), *Proc. Int. Conf. on Fundamental Aspects of Radiation Damage in Metals*, Gatlinburg, 1975, (CONF-751006, Natl. Techn. Inf. Service, Springfield, VA, 22161) p. 463.
- [13] S. Takaki, J. Fuss, H. Kugler, U. Dedek, H. Schultz, *Rad. Eff.* 79 (1983) 87.
- [14] K. Furderer, K.P. Doring, M. Gladisch, N. Haas, D. Herlach, J. Major, H.J. Mundinger, J. Rosenkranz, W. Schafer, L. Schimmele, M. Schomlitz, W. Schwarz, A. Seeger, *Mater. Sci. Forum* 15–18 (1987) 125.
- [15] H.E. Schaefer, W. Stuck, F. Banhart, W. Bauer, *Mater. Sci. Forum* 15–18 (1987) 117.
- [16] T. Yoshiie, Y. Hayashi, S. Yanagita, Q. Xu, Y. Satoh, H. Tsujimoto, T. Kozuka, K. Kamae, K. Mishima, S. Shiroya, K. Kobayashi, M. Utsuro, Y. Fujita, *Nucl. Inst. Meth. Phys. Res. A* 498 (2003) 522.
- [17] H.W. King, *J. Mater. Sci.* 1 (1966) 79.
- [18] J.Z. Jiang, U. Gonser, C. Gente, R. Bormann, *Appl. Phys. Lett.* 63 (1993) 1056.
- [19] O. Coren o-Alonso, J. Coren o-Alonso, *Intermetallics* 12 (2004) 117.