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Enhanced acetone sensing properties of titanium dioxide nanoparticles with a sub-ppm detection limit

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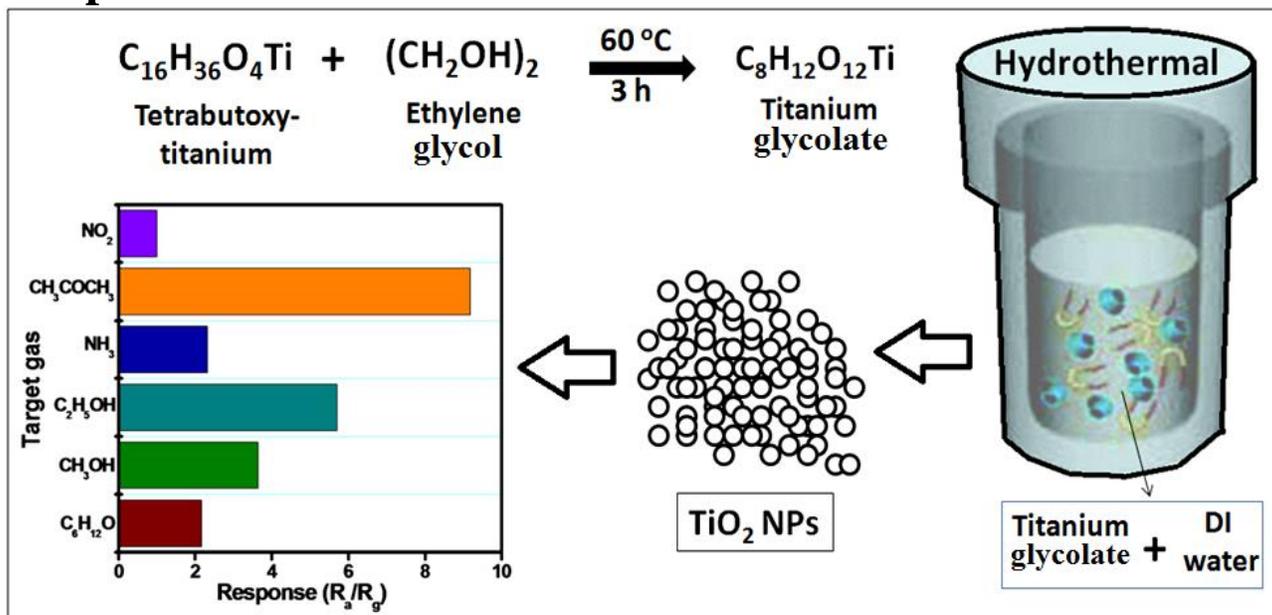
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Graphical abstract



Highlights

- Simple hydrothermal synthesis strategy of TiO₂ NPs.
- Structural analysis confirms the formation of pure anatase phase of TiO₂.
- TiO₂ sensor, composed of NPs, is highly sensitive and selective towards CH₃COCH₃ @270 °C.
- TiO₂ sensor demonstrates the enhanced response of 15.24 to 1000 ppm exposure of CH₃COCH₃ @270 °C along with rapid response and recovery times.
- TiO₂ sensor is able to detect as low as 500 ppb concentration of CH₃COCH₃.

Abstract

In the present study, a simple hydrothermal approach has been successfully applied for a large scale synthesis of anatase titanium dioxide nanoparticles (TiO₂ NPs) using titanium glycolate precursors and is utilized for the fabrication of low-cost high performance acetone (CH₃COCH₃) gas sensors after corroborating the crystallinity, phase-purity, and surface morphology investigations. Several randomly distributed TiO₂ aggregates, composed of NPs, are noticed from morphology analysis. Chemiresistive properties of as-fabricated TiO₂ sensors attempted towards host of oxidizing and reducing gases, reveal a superior selectivity to CH₃COCH₃ with a maximum response of 15.24 (1000 ppm) @270 °C compared to other target gases. One of the key features of as-fabricated TiO₂ sensor is the lowest detection limit of 500 ppb to CH₃COCH₃ with rapid response and recovery times, signifying commercial potential of the developed sensor materials. The effect of operating temperature along with various concentrations of CH₃COCH₃ on the gas sensing properties of TiO₂ sensor has thoroughly been investigated and reported. Finally, the interaction mechanism between the CH₃COCH₃ molecules

and the TiO₂ NPs sensor was elaborated in depth for a thorough understanding sensor performance experimentally and supposedly.

Keywords: Metal oxide; Hydrothermal synthesis; TiO₂ NPs; Chemiresistive properties; Acetone sensor

1. Introduction

In recent years, chemiresistive sensors based on semiconducting metal oxide nanostructures are an extensively investigated class of gas sensors for developing efficient chemically stable and mechanically robust gas sensor devices for human safety and environmental monitoring applications due to their flexibility in production and low cost [1-13]. So far, the commercial traditional gas sensing devices have rapidly expanded, but they still failed to meet the demand of low-cost, high sensitivity, low detection limit, short response/recovery time, and lower temperature operation in combination. It is important to develop such kind of gas sensing materials, which detect the colorless and harmful gases at a very low concentration level as these gases are toxic in nature and have direct impact on human health and eco-system. Higher sensitivity and lower detection limit of a gas sensor material may sense a smaller quantity of leaked gas, which eventually helps in avoiding serious accidents. Thus, to meet this demand, development of different materials using a variety of synthesis routes, *viz.* chemical and physical syntheses, are important to meet the challenges in the field of gas sensors. Till date, several semiconducting metal oxides such as tungsten oxide (WO₃) [1, 2, 4], tin dioxide (SnO₂) [3], zinc oxide (ZnO) [5, 6], cobalt oxide (Co₃O₄) [7], iron oxide (Fe₂O₃) [8, 9], titanium dioxide (TiO₂) [10, 11], nickel oxide (NiO) [12], and indium oxide (In₂O₃) [13] etc., have been reported as successful gas sensing materials for detecting a variety of toxic and inflammable gases with different detecting level capacities. Among known semiconducting metal oxides, nanostructured TiO₂ is one of the largely investigated materials for pigments in coating, gas sensor, self-cleaning glass, energy storage, photovoltaic, and photocatalysis applications because of its low-cost, high band gap energy, excellent chemical stability, and environmental friendliness [14-18]. A myriad of efforts have been undertaken successfully to synthesize TiO₂ nanostructures in different morphologies using hydrothermal [19], electrospinning [20], flame spray pyrolysis [21], sol-gel [22], anodization [23], matrix-assisted pulsed laser deposition [24], and screen printing [25] etc.,

followed the gas sensing application in literature. Amongst several known methods, hydrothermal synthesis is an easy and highly efficient chemical process used for obtaining nanostructures of metal oxides and chalcogenides at an ambient temperature. It is also well-known green approach because of its closed system conditions throughout the synthesis [26]. Moreover, hydrothermal process has quite a few advantages over other chemical methods in terms of economy, low-temperature, additives and catalyst-free, ability to scale up synthesis with high purity product, and environmental cordiality etc., [8, 19, 26]. Acetone, a colorless and clear liquid, is one of the general volatile organic compounds with the chemical formula CH_3COCH_3 . Long-term inhalation of CH_3COCH_3 at high levels can cause fatigue, eye-irritation, headache, as well as nerve and kidney damage, as found by World Health Organization [27-29]. Furthermore, CH_3COCH_3 , produced through hepatocytes by means of decarboxylation of excessacetyl-coenzyme A, is one of the commonly considered biomarkers for type-I as well as type-II diabetes [27, 28]. Thus, the development of low-cost high performance gas sensor systems, which can detect CH_3COCH_3 at low concentration levels, is crucial for the early diagnosis of diabetes in humans and process-control industry related applications.

In the present study, we have successfully synthesized anatase TiO_2 nanoparticles (NPs) using a titanium glycolate precursor through a simple hydrothermal route, which avoids the procedure of annealing treatment at elevated temperature, and which further were applied for the chemiresistive properties towards various target gases as well as operating temperatures. Initially, the as-synthesized TiO_2 NPs were characterized for confirming the structure and morphology by various means. Gas sensing properties demonstrate that the sensor based on TiO_2 NPs has a high sensitivity and selectivity towards acetone at an operating temperature of 270°C . One of the key features of developed sensor is that it can detect acetone in parts per billion (ppb) levels with rapid response and recovery times.

2. Experimental

2.1 Materials

All chemicals [Tetrabutoxytitanium ($\text{C}_{16}\text{H}_{36}\text{O}_4\text{Ti}$), ethylene glycol [$(\text{CH}_2\text{OH})_2$], acetone [$(\text{CH}_3)_2\text{CO}$], and $\text{CH}_3\text{CH}_2\text{OH}$] in the present study are of analytical grade and were used as received. Distilled water was used during all the experiment.

2.2 Synthesis of TiO_2 NPs

A low-temperature hydrothermal route was employed for the preparation of TiO₂ NPs using titanium glycolate as a source of Ti. Initially, the titanium glycolate precursor was produced using the procedure described in elsewhere with minor alterations [30]. In a typical procedure, 2.33 g of C₁₆H₃₆O₄Ti was slowly added in 60 ml (CH₂OH)₂ and the desired mixture was magnetically stirred in a water bath @60 °C for 3 h. After the completion of reaction, the obtained clear solution was transferred into the (CH₃)₂CO and distilled water mixture. The prepared suspension was then magnetically stirred for 2 h without any heat treatment. Subsequently, the obtained solid product was collected with the help of centrifugation and washed several times with distilled water and CH₃CH₂OH successively and dried @60 °C. Afterwards, the as-synthesized titanium glycolate precursor (0.5 gm) was dispersed in 30 ml of distilled water and the desired mixture was transferred to 50 ml capacity Teflon-lined autoclave. The hydrothermal reaction was carried out at 180 °C for 6 h and finally, synthesized TiO₂ product was collected by centrifugation followed by washing with distilled water and dried at 60 °C for 2 h. The schematic of formation of TiO₂ NPs using hydrothermal is shown in Fig. 1(a). Structural information of the as-synthesized TiO₂ product was investigated using X-ray diffraction spectroscopy (XRD, Model: Rigaku X-ray diffractometer with Cu-K_α source ($\lambda=0.15418$ nm)), Fourier infrared transform spectroscopy (FTIR, Model: Perkin-Elmer100 spectrophotometer), and X-ray photoelectron spectroscopy (XPS, Model: VG, Multilab 2000, Thermo VG, Scientific, UK). Morphology of TiO₂ was confirmed from digital images taken on scanning electron microscopy (SEM, Model: JEOL-6300F). Specific surface area and pore-size distribution of as-synthesized TiO₂ product were analyzed using Brunauer-Emmett-Teller (BET, Model: Belsorp II, BEL Japan Inc.) and Barrett–Joyner–Halenda (BJH) techniques, respectively.

2.3 Sensor fabrication and measurement

As-synthesized TiO₂ product was dispersed in distilled water in order to form a slurry/paste, which then was covered on a ceramic tube (4 mm length, external and internal diameter 1.2 mm and 0.8 mm, respectively), with the help of a paint brush, and thereafter used as sensing material. Both the ends of ceramic tube consist of two gold (Au) electrodes (0.5 mm wide each) for contacts, which are connected to platinum wires for signal collection (as shown in Fig. 5(b)). A Ni-Cr thermocouple was inserted inside the ceramic tube to control the operating temperature of sensor, which was managed by altering the heating current passed throughout the Ni-Cr coil. The gas sensing properties, in terms of change in electrical resistance, of as-

fabricated TiO₂ sensor were measured with the help of a static-test sensing system under laboratory conditions. The sensor was placed inside the airtight chamber (Volume: 20 L) and the certain amount of target gases inside the chamber was attained with the help of a micro-syringe. The recovery of sensors was achieved by allowing a fresh air inside the sensing chamber. The different kind of target gases such as CH₃COCH₃, ethanol (C₂H₅OH), methanol (CH₃OH), nitrogen dioxide (NO₂), ammonia (NH₃), and cyclohexanol (C₆H₁₂O) were tested. The response (S) of the sensor was calculated using following relation:

$$\text{Response (S)} = R_a/R_g \quad (1)$$

where R_a and R_g are the resistances of the sensor in presence of air and gas, correspondingly.

3. Results and discussion

3.1 Structural elucidation and morphological evolution

A typical XRD pattern of as-synthesized TiO₂ NPs together with the Miller indices of the crystallographic planes corresponding to all the different diffraction peaks is shown in Fig. 1(b). The XRD pattern of TiO₂ reveals the diffraction peaks located at 2θ of 25.28°, 37.80°, 48.04°, 53.89°, 62.68°, 68.76°, 70.30°, 75.10°, and 82.65°, belonging to (101), (004), (200), (105), (204), (116), (220), (215), and (224) Bragg's scattering planes, respectively. All observed diffraction peaks, without any additional impurity peaks, and their equivalent peak positions are analogous to the JCPDS Card #21-1272 of pure tetragonal anatase phase of TiO₂. The observed diffraction peaks are widened type, which is caused by the smaller size of the as-synthesized TiO₂. The average crystallite size in (101) direction is ≈ 14 nm, calculated from Scherrer relation ($D = 0.9\lambda/\beta \cdot \cos\theta$, where D is the crystallite size, λ is the wavelength of X-ray, β is the full width at half maximum of (101) plane, and θ is the diffraction angle). The Raman spectrum of as-synthesized TiO₂ product is shown in Fig. 1(c), which obviously suggesting four active Raman peaks at 144, 397, 514, and 638 cm⁻¹. The highly intense peak observed at 144 cm⁻¹ together with the peak located at 638 cm⁻¹ is due to the E_g mode of anatase TiO₂. In addition to this, the peaks positioned at 397 and 514 cm⁻¹ are due to B_{1g} and A_{1g} vibration modes, respectively. The peaks at 144 and 397 cm⁻¹ confirm an involvement of the O-Ti-O bending vibration. However, the peaks at 514 and 638 cm⁻¹ are of Ti-O stretching vibration. All observed Raman modes are entirely analogous to previously reported anatase TiO₂ [30, 31].

The wide range XPS spectrum of TiO₂ is shown in Fig. 2(a), suggesting the presence of Ti and O peaks. The core-level spectrum of Ti 2p, shown in Fig. 2(b), which demonstrates two

broad peaks belonging to Ti 2p_{3/2} and Ti 2p_{1/2} doublets situated at binding energies of 458.67 and 464.53 eV, respectively. The Ti 2p_{3/2} and Ti 2p_{1/2} doublet with the binding energy difference of 5.86 eV is the main characteristic of Ti⁺⁴ valance state of anatase TiO₂ in conjunction with octahedral co-ordination with O [32, 33]. The O1s core level spectrum of TiO₂, shown in Fig. 2(c), confirms major O1s peaks located at 530.1 and 531.93 eV. These peaks are assigned to the lattice oxygen (Ti-O-Ti) and hydroxyl (-OH) groups of TiO₂, continually, whereas the binding energy peak positioned at 533.57 eV is the characteristic of C=O [32-35]. Thus, the XPS study corroborates the formation of phase pure anatase TiO₂.

Surface morphology, playing an important role in gas sensing applications, of as-synthesized TiO₂ product was analyzed by SEM technique and presented in Fig. 3(a). The SEM surface image demonstrates that the closely bounded agglomerations of NPs. Random distribution of TiO₂ aggregates, composed of nanoparticles, with numerous pores stuck between them are clearly seen from the SEM micrograph. Such kind of surface morphology is preferred for gas sensing applications, since the circulation of target gas molecules is very fast through the porous structure compared to flat/dense surfaces, thereby, a higher performance is anticipated [36]. The EDX spectroscopy (Fig. 3(b)) was used to study the elemental composition present inside the synthesized TiO₂ product, which clearly suggesting that the synthesized product is composed of titanium (Ti) and oxygen (O). The information regarding the atomic percentages of Ti and O elements are given in the inset-table of Fig. 3(b). EDX mapping images of TiO₂ shown in Fig. 3(c) highlight the existence of O and Ti in the final product with appropriate proportions. Fig. 4(a) shows the nitrogen (N₂) adsorption/desorption isotherm of TiO₂ NPs obtained by hydrothermal method, suggesting a typical IV-type isotherm behavior along with H₃-type hysteresis loop [5, 7]. As-synthesized TiO₂ NPs exhibit a specific surface area of 25.2169 m²/g and ~25 nm average pore-diameter (calculated using BJH pore size distribution curve, Fig. 4(b)), confirming an existence of the mesoporous character.

3. 2 Gas sensing properties

Gas sensing properties of the as-fabricated TiO₂ NPs sensor, whose on-field photograph is shown in Fig. 5(a), were studied towards NO₂, CH₃COCH₃, C₂H₅OH, CH₃OH, NH₃, and C₆H₁₂O gases at different operating temperatures. The plot of response as a function of different target gases (500 ppm each), shown in Fig. 5(c), is suggesting that the TiO₂ NPs sensor displays maximum selectivity to CH₃COCH₃ gas in comparison to other test gases @270 °C. Therefore,

CH₃COCH₃ was considered as the selective target gas throughout the entire gas sensing study. The obtained response values for NO₂, CH₃COCH₃, C₂H₅OH, CH₃OH, NH₃, and C₆H₁₂O gases are 1.2, 9.19, 5.71, 3.64, 2.32, and 2.17, respectively. The observed TiO₂ response value of 9.19 to 500 ppm CH₃COCH₃ is ~5 times higher than that of the fifth ranking C₆H₁₂O gas. Selectivity coefficient (K) of CH₃COCH₃ over the other target gases was calculated using following relation [7]:

$$K=R_{\text{CH}_3\text{COCH}_3}/R_X \quad (2)$$

where R_{CH₃COCH₃} and R_X are the responses of TiO₂ NPs sensor in presence of CH₃COCH₃ and other target gases (X). The calculated ‘K’ values for NO₂, C₂H₅OH, CH₃OH, NH₃, and C₆H₁₂O are 7.65, 1.59, 2.52, 3.96, and 4.23, respectively. Higher ‘K’ value suggested that the as-fabricated sensor has a superior ability to discriminate the target gas (here CH₃COCH₃) between the mixture of gases. It is well-known that the operating temperature significantly influences the gas sensing properties of sensors based on semiconducting metal oxides [36-40]. Thus, the effect of operating temperature on the gas sensing performances of TiO₂ sensors was thoroughly studied and obtained results are displayed in Fig. 5(d). For the response study of TiO₂ NPs sensor, with respect to operating temperature, a fixed 500 ppm concentration of CH₃COCH₃ was used. The TiO₂ sensor starts responding to CH₃COCH₃ at the operating temperature of 250 °C and attains the maximum response value at 270 °C. The response value of TiO₂ sensors above 270 °C is decreased with increasing temperature. A maximum response value of 9.19 at 270 °C upon 500 ppm exposure of CH₃COCH₃ is obtained; as a result 270 °C was used as an optimized sensing temperature for TiO₂ NPs sensor in further gas sensing studies. TiO₂ is a well known catalyst for CH₃COCH₃. At its surface acetone undergoes oxidation to CO₂ and H₂O in the presence of adsorbed oxygen [41]. The adsorbed oxygen molecules help to improve the response while the moisture present on the sensor surface hinders the CH₃COCH₃ oxidation. Gas sensing response, being a surface phenomenon, strongly depends on the adsorption and desorption kinetics and usually exhibits volcanic shape temperature dependence (see Fig. 5(d)). Here, the sensor response first increases with temperature, attains a maximum and then decreases with further rise in temperature. This behavior is attributed to the temperature dependence of adsorption and desorption rates and can be explained by considering the surface reactions and gas diffusion kinetics [42, 43]. Initially, with temperature, the adsorption increases while at higher temperatures desorption of gases or molecular oxygen from the sensor surface is more

favorable. Besides, at higher temperatures the low utilization rate of the sensing material is owing to low gas adsorption ability of the gas molecule causes poor response [44]. At equilibrium, a maximum response towards CH_3COCH_3 is owing to suitable activation energy and enhanced reaction rate [44, 45]. At this temperature, the high selectivity is attributed to the higher adsorption activities of the TiO_2 surfaces for CH_3COCH_3 as compared to that of other investigated gases [46, 47]. Thus, based on the selectivity and temperature dependent response study it is concluded that the TiO_2 NPs sensor of anatase phase is highly sensitive and selective to CH_3COCH_3 gas in comparison to other target gases @270 °C. Thereafter, the effect of CH_3COCH_3 concentration (0.5-500 ppm) on TiO_2 sensors was studied thoroughly @270 °C and reported herein.

The dynamic resistance plot of TiO_2 NPs sensor upon 0.5-500 ppm levels exposure of CH_3COCH_3 at the operating temperature of 270 °C is shown in Fig. 6(a). One of the key features of fabricated TiO_2 sensor is the lowest detection limit of CH_3COCH_3 . TiO_2 NPs sensor is capable to detect as low as 0.5 ppm (500 ppb) of CH_3COCH_3 with response of 1.18. The variation of TiO_2 response as a function of CH_3COCH_3 concentration is shown in Fig. 6(b), demonstrating the increase in response value with CH_3COCH_3 concentration. A maximum 15.24 response of 15.24 is evidenced to 1000 ppm exposure of CH_3COCH_3 . Low response at lower concentrations of CH_3COCH_3 attributes to the lower surface coverage of CH_3COCH_3 gas molecules as a result of lower surface interactions, which limits the sensor response. Conversely, a higher concentration of CH_3COCH_3 covers increasingly surface of TiO_2 , which eventually enhances the surface interactions, consequently, increases the response value.

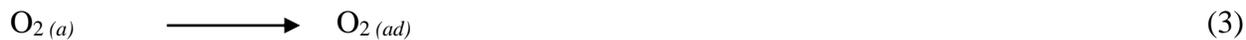
The reversibility study of as-fabricated TiO_2 NPs sensor upon random concentration of CH_3COCH_3 is shown in Fig. 6(c), suggesting the sensor shows higher reversibility and reproducibility towards CH_3COCH_3 with identical sensor response values and response kinetics. The observed response values to 10, 100, 50, and 200 ppm CH_3COCH_3 are 1.67, 4.25, 3.19, and 6, respectively. From Fig. 6(c) it is clear that, the as-fabricated sensor exhibits an excellent reversibility upon successive low and high levels exposure of CH_3COCH_3 . The response and recovery times, considered as key parameters of the chemiresistive-type of sensors, were measured and defined as the time taken by sensor to attain 90% of its maximum resistance change in case of adsorption and desorption, correspondingly [48]. Ideal gas sensors supposed to have a rapid response and recovery times. The response-recovery plot of TiO_2 sensor upon 500

ppm exposure of CH_3COCH_3 @270 °C is shown in Fig. 7(a), demonstrating a rapid response time of 10 s and recovery time of 9 s. Precise reproducible feature along with nearly square shaped response have clearly been evidenced from the response-recovery plot (Fig. 7(a)) [49]. It is observed that, initially, the TiO_2 sensor has responded rapidly to CH_3COCH_3 and quickly has attained a nearly stable state. Afterwards, the sensor is reached to its baseline level, when exposed to fresh air. From the viewpoint of practical application of sensors, not only higher responses and rapid response-recovery times but also reproducibility and stability in response are considered as essential parameters too. Hence, reproducibility, and stability of TiO_2 NPs sensor to CH_3COCH_3 were studied systematically and reported herein. The response repeatability study towards consecutive five cycles of 500 ppm CH_3COCH_3 @270 °C was carried out and the obtained results are displayed in Fig. 7(b). Almost the same response value of ~9.19 is achieved upon successive exposure of CH_3COCH_3 , demonstrating the excellent reproducibility in response of TiO_2 sensor toward CH_3COCH_3 . The plot of stability study of TiO_2 sensor, for the period of 10 days, towards fixed a 500 ppm concentration of CH_3COCH_3 @270 °C is shown in Fig. 7(c). The stability study demonstrates that the TiO_2 NPs sensor had retained its response value superbly. The observed gas sensing properties such as response, selectivity, response-recovery times, reproducibility, and stability of the TiO_2 sensor makes it suitable candidate for selective and sub-ppm detection of CH_3COCH_3 in variety of applications. We have compared the CH_3COCH_3 sensing performance of hydrothermally grown TiO_2 NPs with those of reports available in literature for TiO_2 simply and the comparison is tabulated in table 1. From the table, it can be clearly noticed that the as-fabricated TiO_2 NPs sensor is demonstrating excellent sensing properties, in terms of lowest detection limit, rapid response and recovery times, and lower operating temperature, than that of results quoted in literature.

3. 3 Gas sensing mechanism

It is well-known that the gas sensing mechanism of semiconducting metal oxides depends upon the change in their electrical resistance signatures, which primarily is caused by means of the adsorption/desorption of target gas molecules on the top of sensor surface, upon the interaction of different target gases. As the TiO_2 sensor is exposed to fresh air, oxygen molecules from air adsorb on the TiO_2 surfaces, by capturing electrons from the conduction band of TiO_2 resulting in the formation of an electron depletion layer (as shown in Fig. 8) leading to the higher resistance, which forms the adsorbed oxygen species (such as O^- , O_2^- , and O^{2-}). Adsorbed

oxygen has temperature dependent character [5]. It has O_2^- nature below 100 °C, O^- in between 100-300 °C, and O^{2-} above 300 °C. In the present case, the sensor temperature is 270 °C thus, the O^- species are dominants. The adsorption on the sensor surface takes place as per following reactions (wherein 'a' stands for air, 'ad' stands for adsorption, and 'g' stands for gas):



Herein, TiO_2 is n-type semiconductor with electrons (e^-) as majority carriers. The thermal activation barrier between adsorbed acetone and oxygen to form the acetone-oxygen complex is 10 kJ/mol [45]. The first principle calculations have revealed that the adsorption energy for acetone on TiO_2 surface is negative indicating the process to be exothermic and spontaneous [46]. Hence, when reducing CH_3COCH_3 gas molecules come in contact with the TiO_2 sensor, the previously adsorbed oxygen species readily interact with the organic gas molecules by releasing the trapped e^- back to the conduction band of TiO_2 . Thereby, the conductivity of TiO_2 increases due to the e^- donating nature of CH_3COCH_3 and consequently, the resistance decreases. Decrease in resistance of TiO_2 leads to the reduction in the size of electron depletion layer as shown in the schematic of Fig. 8. Thus, it is prevalent that the adsorbed oxygen molecules on the sensor surface play a dominating role in the governing sensing mechanism. The possible reaction mechanism is as follows [20, 49]:



4. Conclusions

A simple and rapid hydrothermal chemical strategy has been successfully applied for the synthesis of anatase TiO_2 NPs and their chemiresistive activity has been envisaged towards various target gases. Gas sensing results demonstrate that the TiO_2 NPs-based sensor has high selectivity to CH_3COCH_3 and capable to detect its concentration up to ppb level limit at an operating temperature of 270 °C. Our gas sensing results suggest that the as-fabricated TiO_2 sensors is a fascinating candidate for the development of low-cost high performance CH_3COCH_3 -based sensors, in view of its excellent selectivity, lower detection limit, rapid

response/recovery times, and outstanding dynamic properties along with the high response in stability.

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Figure captions

Fig. 1(a) Synthesis scheme of TiO₂ NPs, (b) XRD, and (c) Raman spectrums of as-synthesized anatase TiO₂.

Fig. 2(a) Wide range, (b) Ti 2p, and (c) O1s XPS spectrums of TiO₂.

Fig. 3(a) SEM and (b) EDX spectrum, and (c, d, e) EDX elemental mapping images of TiO₂.

Fig. 4(a) N₂ adsorption-desorption isotherm, and (b) BJH pore size distribution plot of as-synthesized TiO₂.

Fig. 5(a) Photograph of as-fabricated TiO₂ sensor, (b) Schematic view of fabrication of sensor, (c) Selectivity study of TiO₂ sensor towards various target gases at fixed 500 ppm concentration @270 °C, and (d) Temperature dependent CH₃COCH₃ response values of TiO₂ sensor.

Fig. 6(a) Dynamic resistance plot of TiO₂ sensor towards various concentrations of CH₃COCH₃ @270 °C, (b) Plot of response as a function of CH₃COCH₃ concentrations @270 °C, (c) Reversibility study of TiO₂ sensor upon random concentration of CH₃COCH₃ @270 °C.

Fig. 7(a) Response-recovery plot of TiO₂ sensor upon 500 ppm exposure of CH₃COCH₃, (b) Reproducibility study of TiO₂ sensor towards successive exposure of CH₃COCH₃, and (c) Stability study of TiO₂ sensor upon fixed 500 ppm exposure of CH₃COCH₃ @270 °C.

Fig. 8 Schematic of gas sensing mechanism of TiO₂ sensor in presence of air and CH₃COCH₃ atmosphere, respectively.

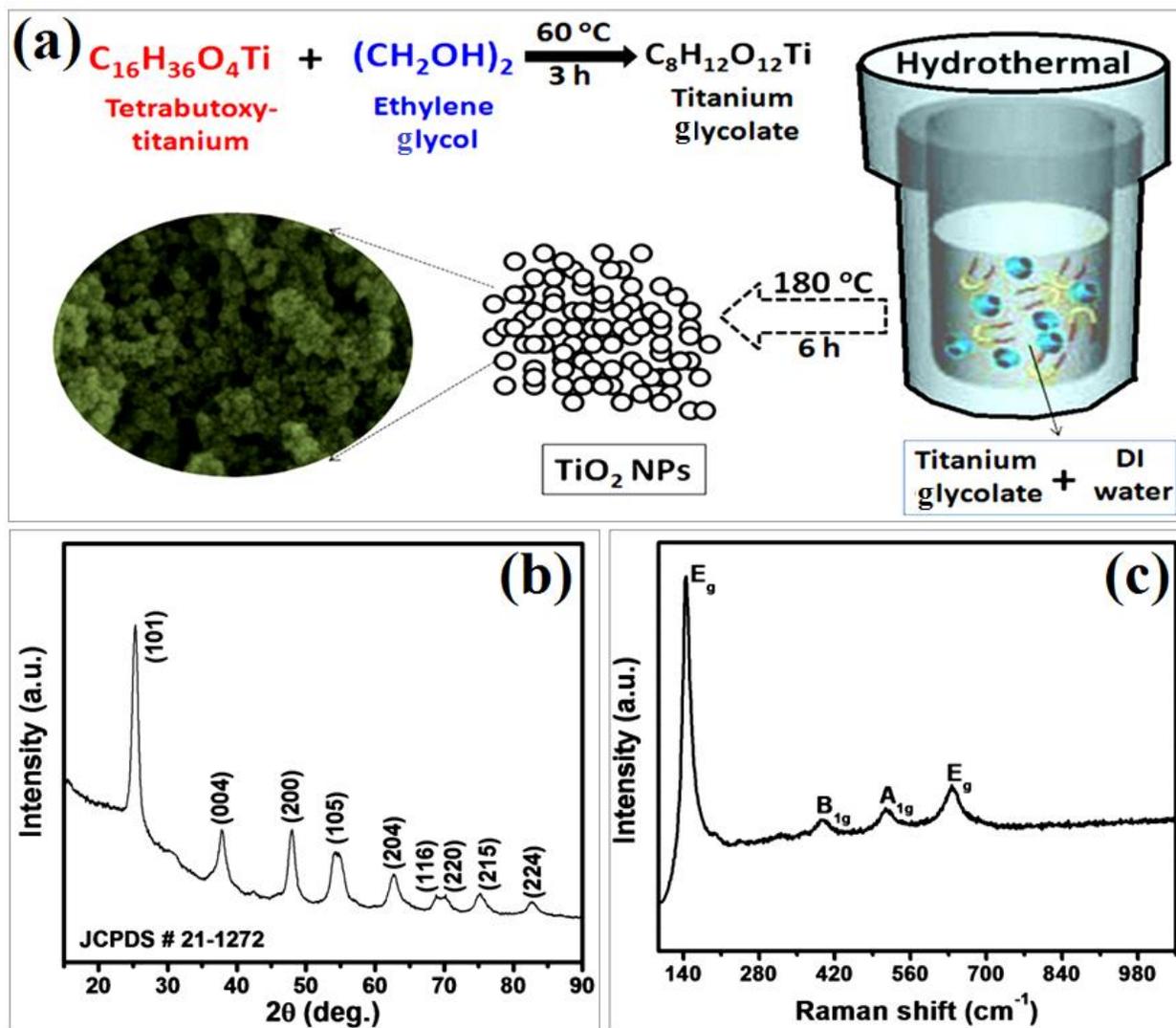


Fig. 1

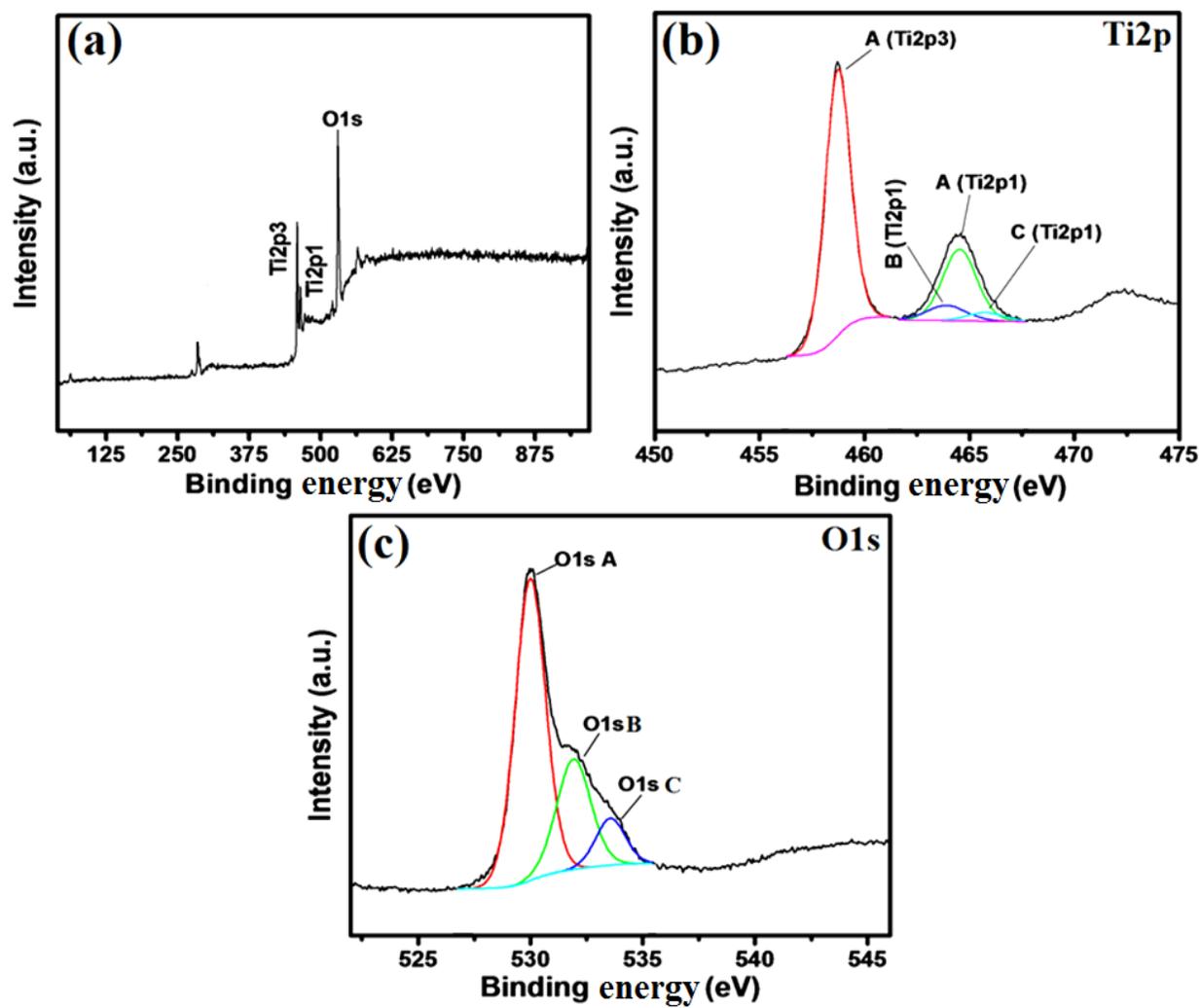


Fig. 2

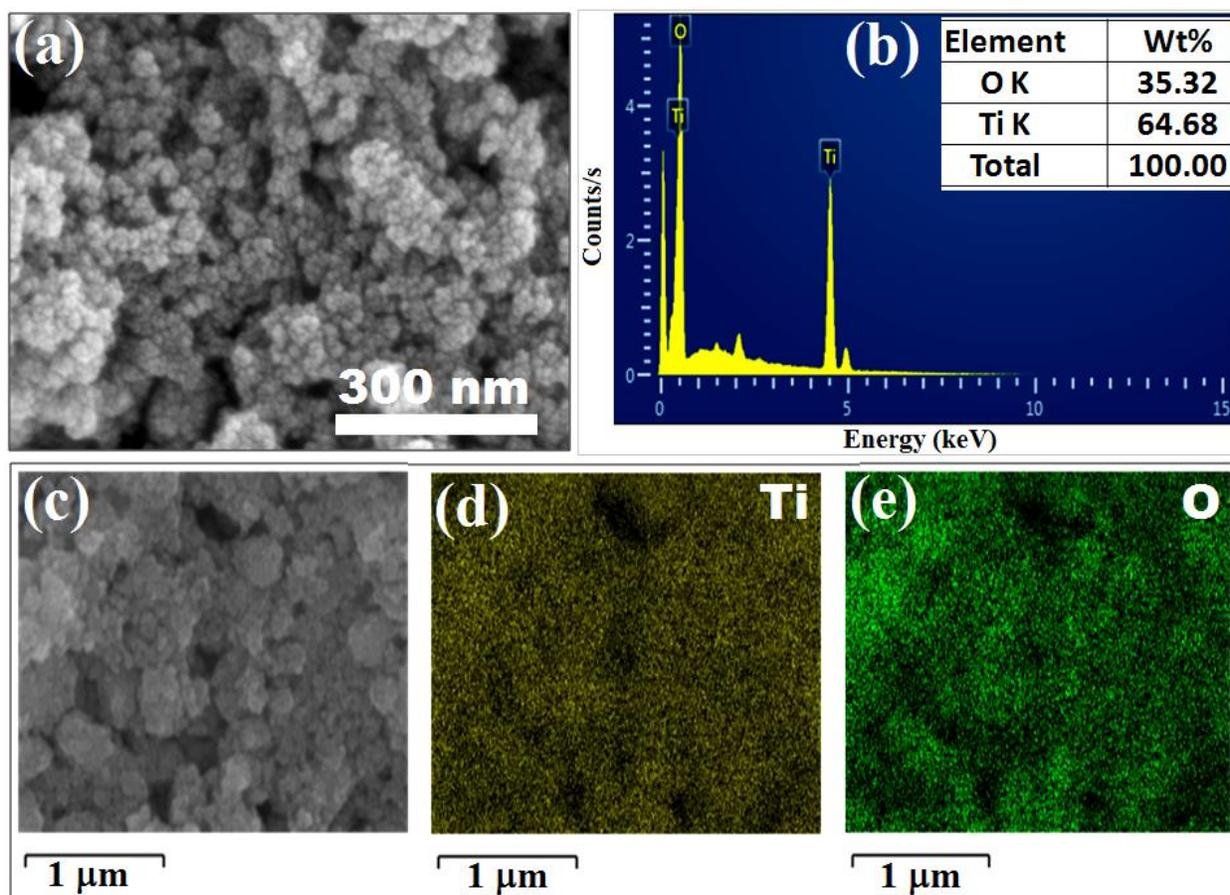


Fig. 3

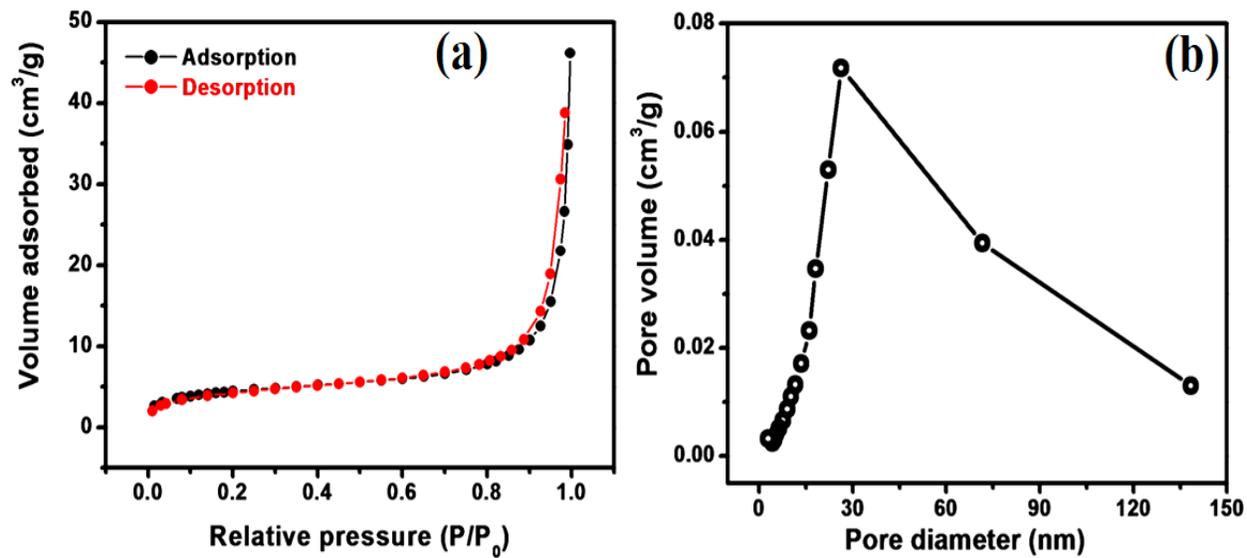


Fig. 4

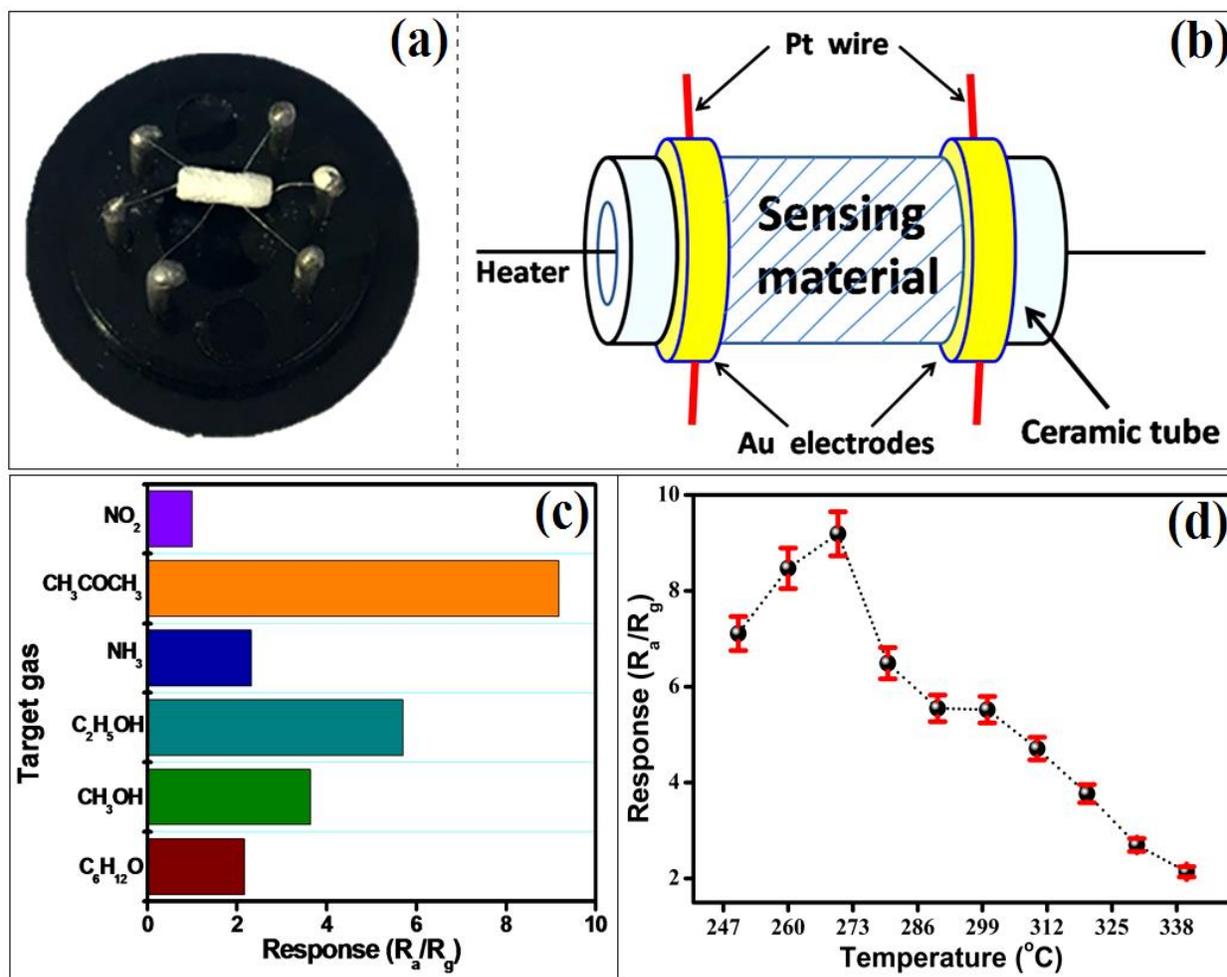


Fig. 5

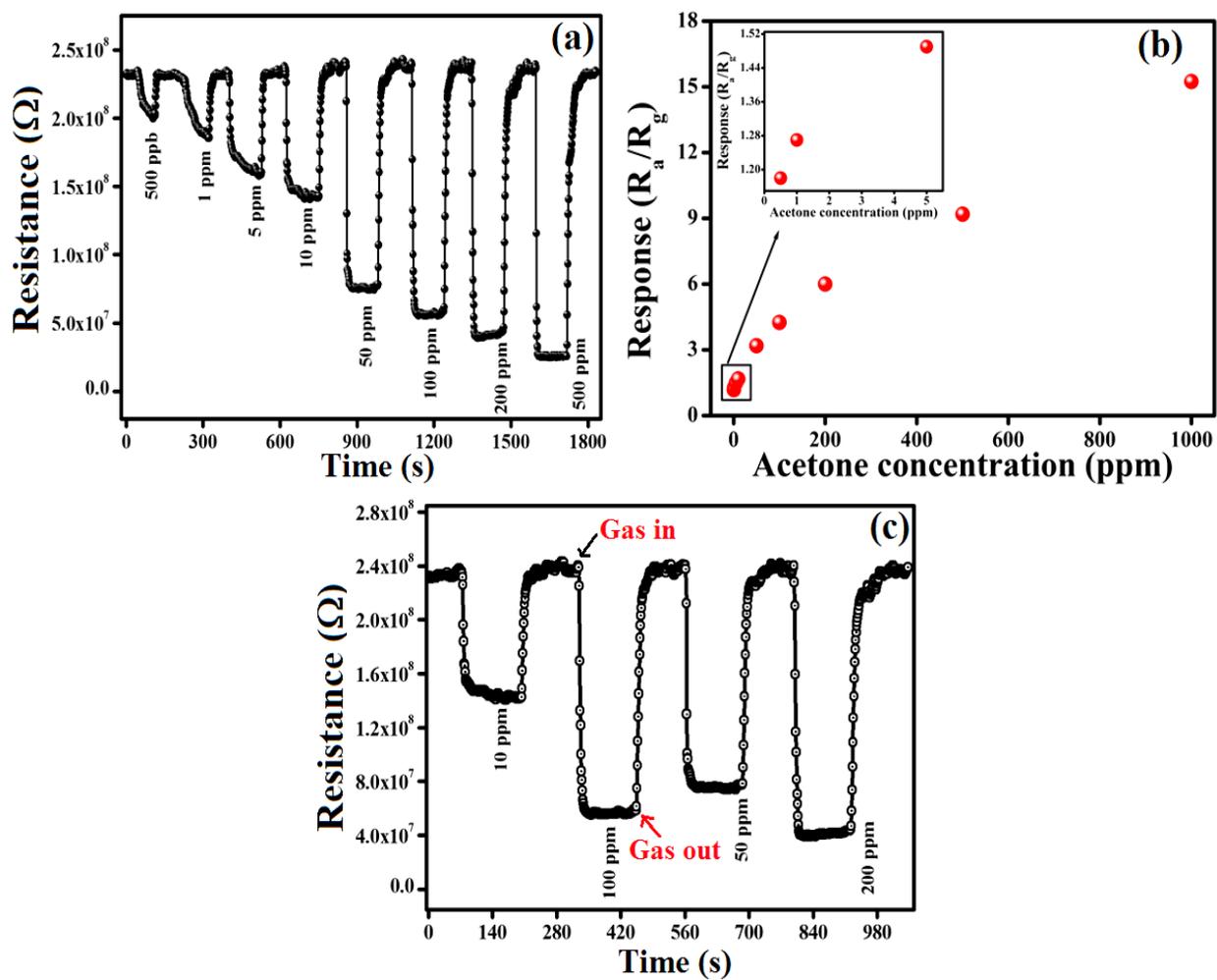


Fig. 6

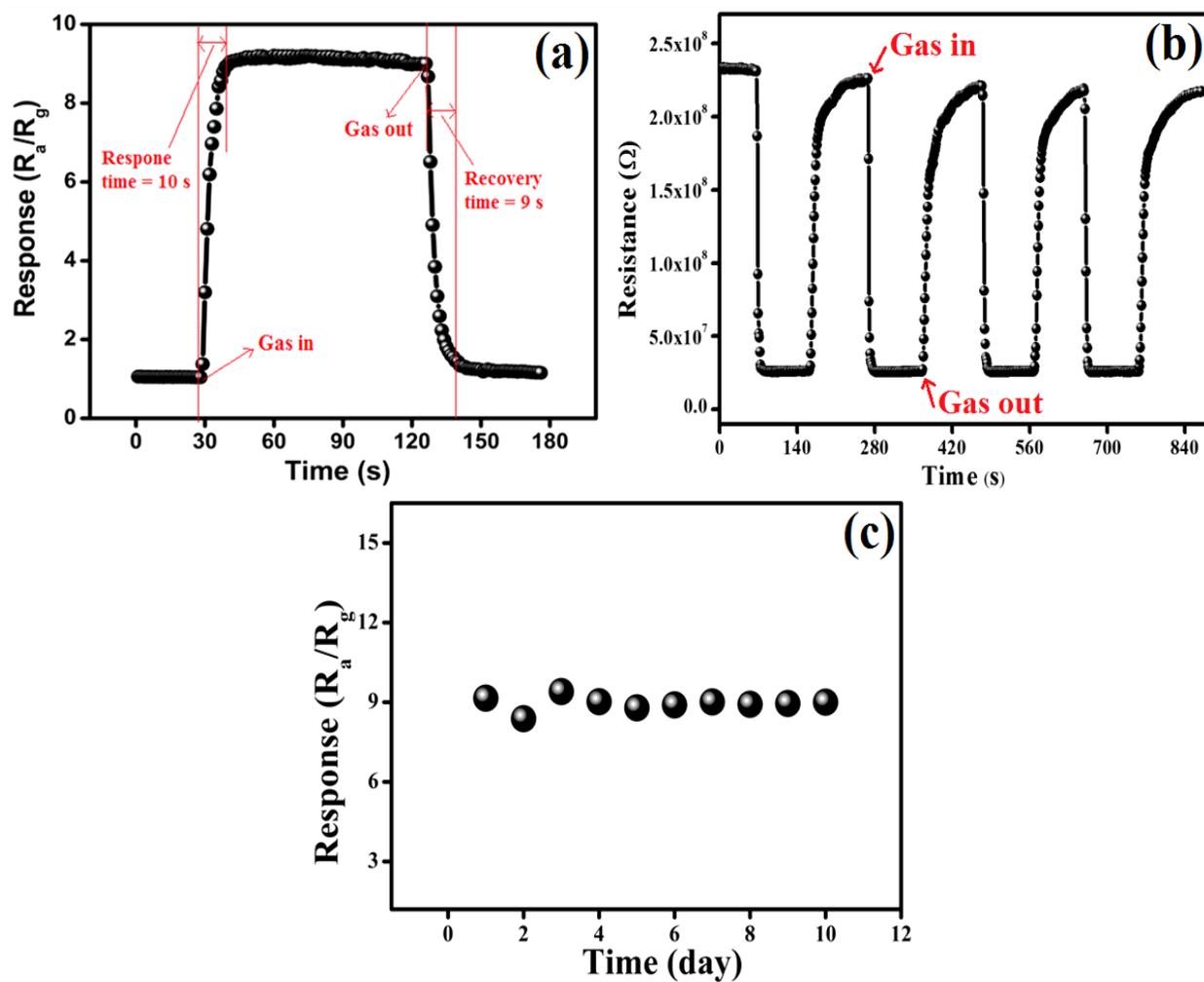


Fig. 7

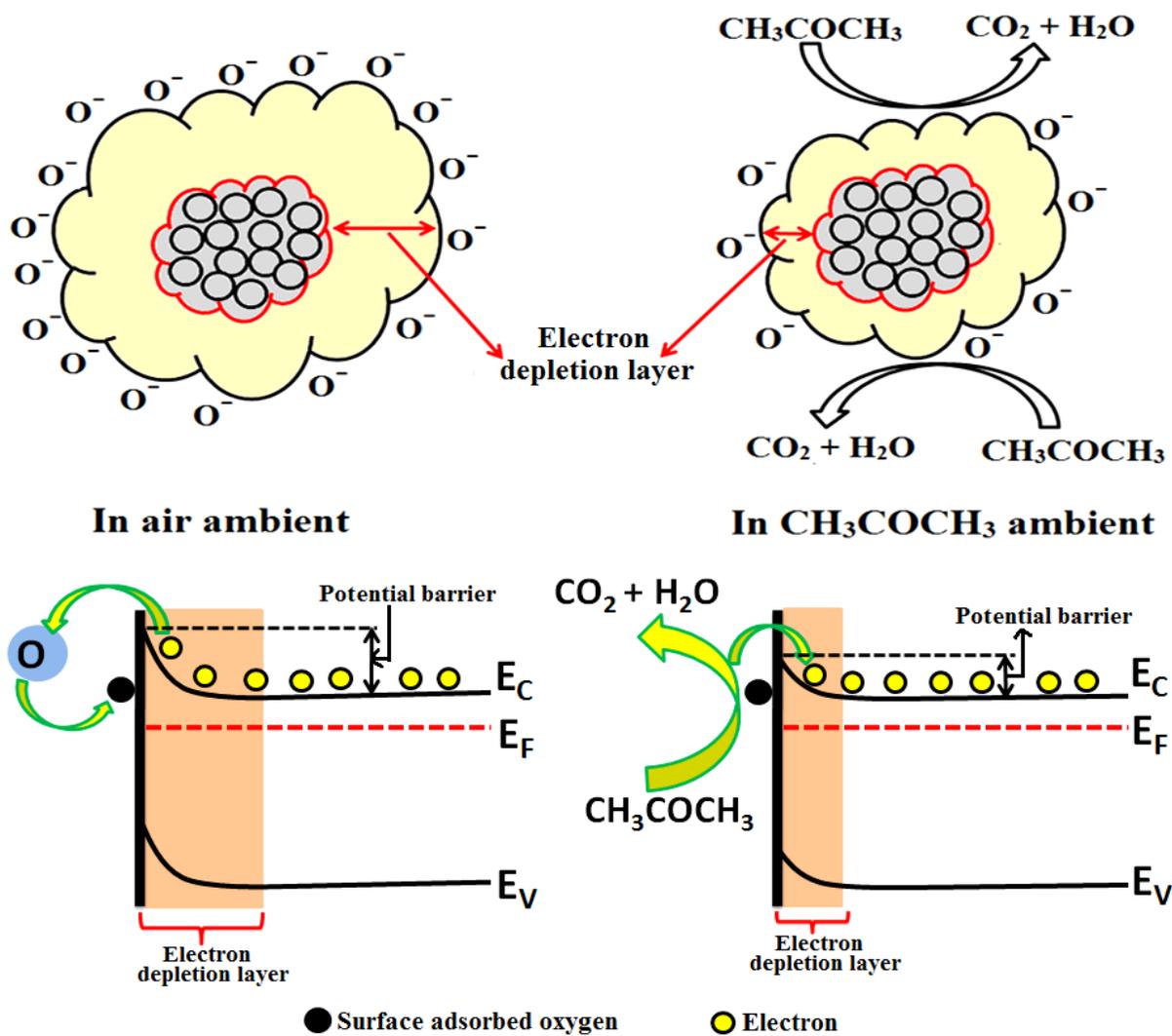


Fig. 8

Table 1: Comparison of gas sensing parameters of TiO₂ NPs sensor with other published CH₃COCH₃ sensors.

Table 1.

Material	Synthesis route	Sensor tempt.	Res (R _a /R _g)	Acetone (ppm)	Res-t	Rec-t	Lowest detection limit	Ref.
TiO ₂ nanorods	Electrospinning	500 °C	~21	500 ppm	Several seconds	Several seconds	1 ppm	20
TiO ₂ NPs	Flame spray pyrolysis	400 °C	~6	400 ppm	Several min	Several min	25 ppm	21
TiO ₂ hollow microspheres	one-pot hydrothermal	320 °C	6.9	100 ppm	Several seconds	Several seconds	10 ppm	46
Ag-TiO ₂ NPs	Sol-gel	270 °C	8.9	1 ppm	29 s	2100 s	1 ppm	22
TiO ₂ nanoflowers	Hydrothermal route	60 °C	2.99	700 ppm	10 s	45 s	1 ppm	50
TiO ₂ Nanotubes	Anodization	150 °C	2.9	500 ppm	100 s	180 s	10 ppm	23
Nanoporous TiO ₂	Hydrothermal	370 °C	25.97	500 ppm	13 s	8 s	20 ppm	44
TiO ₂ NPs	Matrix assisted pulsed laser deposition	400 °C	6	100 ppm	240 s	-	20 ppm	24
Mesoporous TiO ₂	Sol-gel	400 °C	53	400 ppm	-	-	100 ppm	47
TiO ₂ NPs	Screen printing	393 °C	1.51	1.5 ppm	50 s	50 s	1.5 ppm	25
TiO ₂ NPs	Matrix assisted pulsed laser deposition	350 °C	3.04	20 ppm	4 min	14 min	20 ppm	51
TiO ₂ spherical collides	Sol-gel method	350 °C	2.3	1000 ppm	150 s	350 s	1000 ppm	52
TiO₂ NPs	Hydrothermal	270 °C	9.19	500 ppm	10 s	9 s	500 ppb	This work

Res = Response, Res-t = Response time, Rec-t = Recovery time.