

A new superlattice structure in the $\text{Al}_{2.75}\text{Ir}$ and $\text{Al}_{2.63}\text{Rh}$ 1/0 approximants

Kazue Nishimoto¹, Natsumi Yamaguchi², Ryuji Tamura²

¹Institute of Multidisciplinary Research for Advanced Materials (IMRAM), Tohoku University, 2-1-1, Katahira, Sendai-shi, Miyagi, 980-8577, Japan

²Department of Materials Science and Technology, Tokyo University of Science, 6-3-1 Nijuku, Katsushika-ku, Tokyo, 125-8585, Japan

nishi-mt@tagen.tohoku.ac.jp

Abstract. We investigate the superstructure of binary $\text{Al}_{2.75}\text{Ir}$ and $\text{Al}_{2.63}\text{Rh}$ 1/0 approximants by X-ray diffraction method and transmission electron microscopy. $\text{Al}_{2.75}\text{Ir}$ possesses a $2a \times 2a \times 2a$ *C*-centred monoclinic lattice with space group *C2* or *Cm* or *C2/m*. $\text{Al}_{2.63}\text{Rh}$ possesses a $2a \times 2a \times 2a$ face centred lattice with space group *F23* or *Fm $\bar{3}$* .

1. Introduction

Binary $\text{Al}_{2.75}\text{Ir}$ and $\text{Al}_{2.63}\text{Rh}$ are 1/0 approximants of ternary Al-TM quasicrystals. The structure of $\text{Al}_{2.75}\text{Ir}$ and $\text{Al}_{2.63}\text{Rh}$ have been investigated by XRD, and $\text{Al}_{2.75}\text{Ir}$ and $\text{Al}_{2.63}\text{Rh}$ possess a primitive cubic lattice with space group *P23* (the lattice parameter $a = 7.674(1) \text{ \AA}$ and $a = 7.6692(1) \text{ \AA}$, respectively) [1]. Fig.1 shows schematic crystal structure of (a) $\text{Al}_{2.75}\text{Ir}$ and (b) $\text{Al}_{63.3}\text{Cu}_{12.3}\text{Ir}_{24.4}$ phases. In the case of $\text{Al}_{2.75}\text{Ir}$, the primitive cubic lattice is composed by Ir_{12} (green circle) icosahedron containing central Ir atom and 9~10 Al atoms (small yellow circle). It should be noted that the Al sites inside the Ir icosahedron are partially occupied with a pentagon-dodecahedral like distribution and surround central Ir atom. In contrast, the structure of ternary Al-Cu-TM 1/0 approximants are different from the binary Al-TM 1/0 approximants. The ternary Al-Cu-TM 1/0 approximants possess the double *Fm $\bar{3}$* lattice with the lattice constant $=15.3844(2) \text{ \AA}$ ($2a \times 2a \times 2a$) due to the ordered Cu site inside of the TM icosahedron [2, 3]. Therefore, the structure of Al-Cu-TM 1/0 approximant is a $2a \times 2a \times 2a$ FCC (space group *Fm $\bar{3}$*).

In the case of Al-Ir, the order-disorder phase transition of Al inside TM icosahedron is predicted by theoretical calculation using pair potentials fitted to an ab initio database [4]. Moreover, Mihalkovič reported that the ordered phase of $\text{Al}_{2.75}\text{Ir}$ is possibly insulate phase. Similarly, for $\text{Al}_{2.73}\text{Ir}$, very weak superlattice reflections were reported in the selected area electron diffraction patterns (SAED) [5]. Hence, it is necessary to investigate the superlattice structure of binary Al-TM (TM = Ir, Rh) 1/0 approximants.

In the present work, we have investigated the superstructure and the composition dependence of the $\text{Al}_{2.75}\text{Ir}$ and $\text{Al}_{2.63}\text{Rh}$ by the powder X-ray diffraction (XRD) method and transmission electron microscopy (TEM) study to understand the behaviour of atoms inside TM icosahedral clusters.



2. Experimental

Al-TM(TM=Ir, Rh) with composition ranging from 70 to 75 % Al were melted by arc method under an argon atmosphere (Al (99.99%), Rh (99.9%) and Ir (99.9%)). And the samples were annealed at 1373K for about 72 hours or 723K for 3 days in the quartz tube under the argon atmosphere, followed by water quenching. Powder X-ray diffraction experiments were carried out using Cu K α to examine the phase constitution. Additionally, TEM study were performed by JEOL -TKP2 or TOPCON-002b operating at 200kV. The samples for TEM were crushed by agate mortar and dispersed on a copper micro-grid mesh.

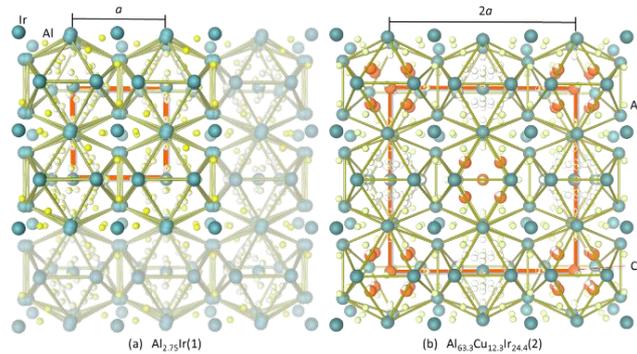


Fig. 1. Crystal structures of (a) $\text{Al}_{2.75}\text{Ir}$ with an $a \times a \times a$ lattice (space group $P23$) [1] and (b) $\text{Al}_{63.3}\text{Cu}_{12.3}\text{Ir}_{24.4}$ with an $2a \times 2a \times 2a$ lattice (space group $Fm\bar{3}$) [2]. The Al sites (small circle) inside the Ir icosahedron are partially occupied with a pentagon-dodecahedral-like distribution.

3. Results and discussions

3.1. XRD

Fig. 2 shows XRD patterns of (a) $\text{Al}_{73}\text{Ir}_{27}$, (b) $\text{Al}_{73.5}\text{Rh}_{26.5}$ and (c) $\text{Al}_{72.5}\text{Rh}_{27.5}$. The most peaks are indexed by an $a \times a \times a$ lattice with reported space group $P23$, but weak superlattice peaks appeared. In Fig. 2, superlattice peaks are indicated by black circles and open circles. For $\text{Al}_{73}\text{Ir}_{27}$, the superlattice peaks can be indexed as $2a \times 2a \times 2a$ lattice. In addition, the peaks parallel to the (100) seem to be split into two peaks, when the $K\alpha_2$ removed. Then, it is possible that the superlattice structure is monoclinic.

On the other hand, for Al-Rh, $\text{Al}_{2.63}\text{Rh}$ phase is observed in the composition range of Al70~74.5%. Indexed as a $2a \times 2a \times 2a$ lattice, the indexes ($h_F k_F l_F$) satisfied the reflection condition ($h_F + k_F, k_F + l_F, l_F + h_F = \text{even}$). It is possible that the superlattice structure possesses FCC lattice with a lattice constant $a_F (= 2a) = 15.340\text{\AA}$. Additionally, superlattice peaks disappeared in the poorer Al compositions below Al72.5%. The composition dependence on the superstructure suggests that the number of Al inside the TM icosahedron contribute to the superstructure.

3.2. TEM

3.2.1 $\text{Al}_{2.63}\text{Rh}$

Figure 3 shows selected area diffraction patterns of $\text{Al}_{72.5}\text{Rh}_{27.5}$ taken along [100], [110] and [111]. We observed the superlattice reflections at $h/2 k/2 l/2$ positions indicated by open triangle in the [110] SAED pattern. In the [111], there are no superlattice reflections, suggesting that the superlattice structure is cubic lattice. All the superlattice reflections can be indexed by considering a $2a \times 2a \times 2a$ lattice ($h_F k_F l_F$), and satisfy the FCC reflection condition ($h_F + k_F, k_F + l_F, l_F + h_F = 2n$). Therefore, the superlattice of $\text{Al}_{2.63}\text{Rh}$ can be explained as $2a \times 2a \times 2a F23$ or $Fm\bar{3}$.

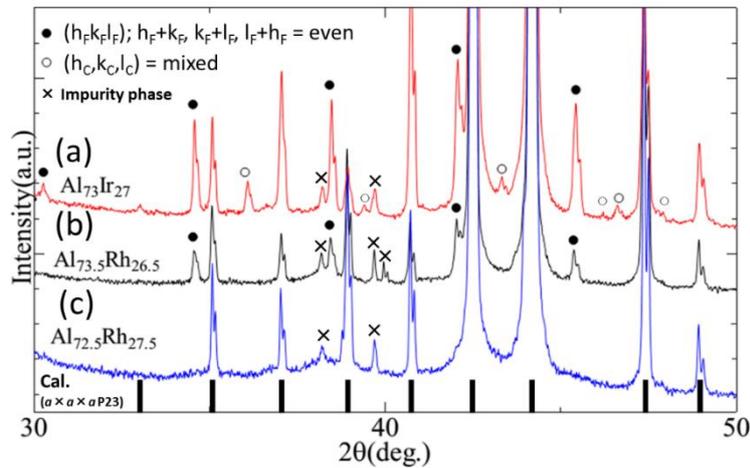


Fig. 2. XRD patterns of (a) $\text{Al}_{73}\text{Ir}_{27}$, (b) $\text{Al}_{73.5}\text{Rh}_{26.5}$ and (c) $\text{Al}_{72.5}\text{Rh}_{27.5}$.

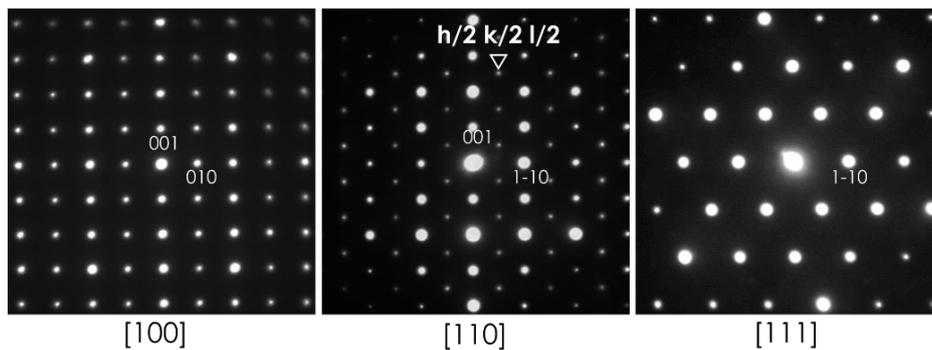


Fig 3. Selected area electron diffraction patterns of Al-Rh taken along [100], [110], [111] and (b) a bright field TEM image.

3.2.2 $\text{Al}_{2.75}\text{Ir}$

Figure 4 presents SAED patterns of $\text{Al}_{73}\text{Ir}_{27}$ taken along [100], [001], [110], [011] and [111]. We observed the superlattice reflections at $h/2 k/2 l/2$, $h/2 k/2 0$ and $0 0 l/2$ positions indicated by open and filled triangles. Thus, all the superlattice reflections can be indexed by considering a $2a \times 2a \times 2a$ lattice ($h_c k_c l_c$). In the case of [111], the superlattice reflections appeared at the point to break the cubic 3-fold symmetry. It indicates that the superlattice structure are not cubic lattice. Considering the XRD result, it is possible that the $\text{Al}_{2.75}\text{Ir}$ possesses a monoclinic lattice. Moreover, all reflections satisfy the C -centred lattice reflection condition ($h_c k_c l_c$; $h_c + k_c = 2n$). Therefore, the superlattice structure of $\text{Al}_{2.75}\text{Ir}$ is $2a \times 2a \times 2a$ $C2$ or Cm or $C2/m$.

Fig 5 shows TEM images of $\text{Al}_{7.25}\text{Ir}$ annealed at 723K for 3 days ((a, b) and (c) a HRTEM image. (b) and (c) were taken along [100]). In Fig. 5(a), dark contrast line indicates domain boundary of ordered phase due to the structural phase transition. The structural phase transition might be same mechanism of order-disorder phase transition of Cd_6M 1/1 approximants at about 100 K [6]. The domain structure is the rectangular shape of about a few 100 nm. Also, in the SAED taken from domains (Fig.5b), there are diffuse streaks along [001]. Additionally, many dark parallel lines generated by stacking faults can be recognized thickly at intervals of a few dozen nm parallel to (001) in the domains (Fig. 5b). It implies that the stacking of ordered clusters tend to easily disarranged to [001]. To obtain the insulate

phase for Al-Ir phase, it is interesting to investigate the electronic state of superlattice structures and lattice defects.

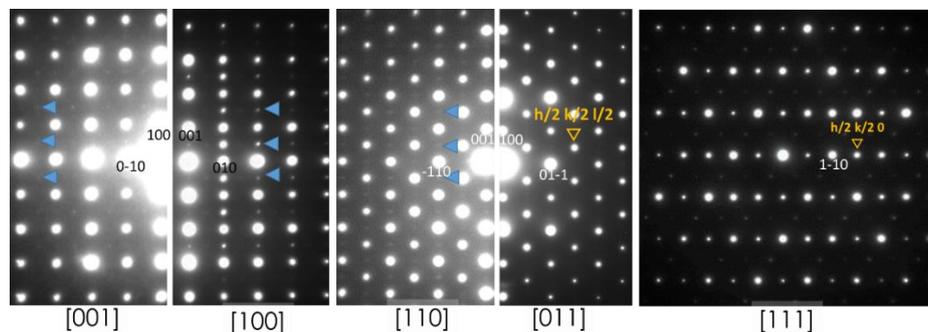


Fig 4. Selected area electron diffraction patterns of $\text{Al}_{7.25}\text{Ir}$ taken along [100], [110], [011] and [111] annealed at 1373K for 72h.

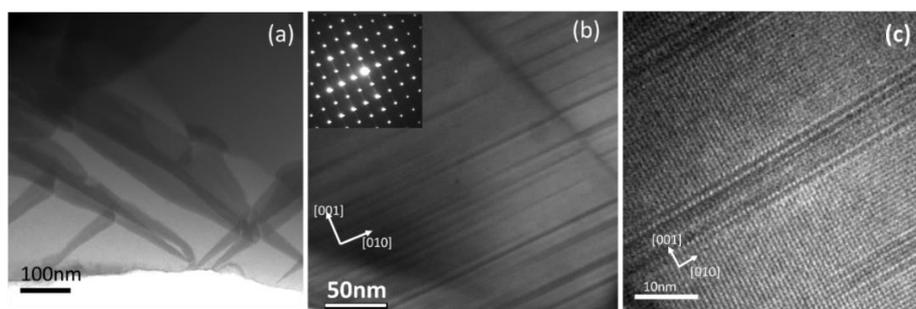


Fig. 5 (a) Bright field TEM image of $\text{Al}_{2.75}\text{Ir}$ annealed at 723k for 3 days. (b) BF and (c) lattice images were taken along [100].

In this work, we reported the new superlattice structure of binary $\text{Al}_{2.75}\text{Ir}$ and $\text{Al}_{2.63}\text{Rh}$ phases, which was effected by number of Al atoms in side TM(TM = Ir, Rh) icosahedron cluster. $\text{Al}_{2.75}\text{Ir}$ possesses a $2a \times 2a \times 2a$ monoclinic lattice with space group $C2$ or Cm or $C2/m$. $\text{Al}_{2.63}\text{Rh}$ possesses a $2a \times 2a \times 2a$ face centred lattice with space group $F23$ or $Fm\bar{3}$. The composition dependence of superlattice reflections suggest that the superlattice structure is determined by the number of ordered Al atoms inside TM icosahedron.

Acknowledgements

TEM observations were performed by JEOL-TKP02 at Prof. Masami Terauchi's laboratory, IMRAM, Tohoku University.

References

- [1] Grin Y, Peters K, Burkhardt U, Gotzmann K and Ellneret. M 1997 *Z. Kristallogr.* **212** 439
- [2] Dshemuchadse J, Kuczera P and Steurer W 2013 *Intermetallics* **23** 337
- [3] Grushko B, Gwóźdz J and Yurechko M 2000 *J. Alloys. Comp.* **305** 219
- [4] Oishi A, Nishimoto K and Tamura R 2009 *Z. Kristallogr.* **224** 115
- [5] Mihalkovič M and Henley C L 2013 *Phys. Rev. B* **88** 064201
- [6] Nishimoto K, Sato T and Tamura R 2013 *J. Phys. Cond. Mat.* **25** 235403