

# Influence of nanographene platelets (NGP) incorporation on Fe<sub>3</sub>O<sub>4</sub> nanoparticles as materials additives for enhancement thermal properties stearic acid

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**Abstract.** In this work, Fe<sub>3</sub>O<sub>4</sub> nanoparticles, and Fe<sub>3</sub>O<sub>4</sub>/NGP composite were used as material additive for enhancement thermal properties of stearic acid (SA). The both material additive were synthesized using sol-gel method. Phase change material (PCM) composites SA-Fe<sub>3</sub>O<sub>4</sub> and Sa-Fe<sub>3</sub>O<sub>4</sub>/NGP mixtures were made through the dispersion technique with three different weight % ratio of material additives into stearic acid: 1 wt.%, 3 wt.%, and 5 wt.%. X-Ray Diffractometer (XRD) and Fourier Transform Infrared (FTIR) spectroscopy were used to investigate the structural properties. Magnetic properties also measured by vibrating sample magnetometer (VSM) to see influence of NGP in PCM composites. Differential Scanning Calorimetry (DSC) and Thermogravimetric Analysis (TGA) were used in order to analyse the thermal properties of the samples. The results show an enhancement of the latent heat, thermal stability as well as specific heat by the presence of material additives in SA. Compare to SA-Fe<sub>3</sub>O<sub>4</sub>, SA-Fe<sub>3</sub>O<sub>4</sub>/NGP show better improvement in enhancement of thermal performance of SA. The improvement by about 41.2 % in specific heat and 21.2% in latent heat.

## 1. Introduction

Thermal energy storage (TEM) has attracted more attention due to its heating and cooling application for waste heat recovery, solar thermal energy utilization and thermal management of electronics [1-3]. The variety of phase change material (PCM) especially the organic phase change material has been widely studied and has become the key technologies in the thermal energy storage application. Stearic acid which the common preferred organic phase change material is typically used due to their nontoxic, high energy storage density, thermal stability, and inconspicuous volume change [4-6]. Unfortunately, most of the organic phase change material including the stearic acid have a very low thermal conductivity which can limit their application. Therefore, some various methods are proposed in order to overcome this problem and one of the most promising method is by dispersing the nanoparticle into the PCM itself [7-9].

Adapting the idea of nanofluid, the nanomaterial utilization into the PCM is considered as a promising solution to enhance the thermal ability of the PCM in order to absorb and release heat [10]. A recent researchs show that the addition of the high thermal properties metal oxides such as Fe<sub>3</sub>O<sub>4</sub>, TiO<sub>2</sub> into the PCM has produced an enhancement in both latent heat and thermal conductivity [11-13]. Not only metal oxide, graphene with nano-sized material also shows a tremendous effect to the thermal



properties of the PCM. For an example a research by Harish et.al [14] shows that the addition of the single-walled carbon nanohorn into the PCM has produced an increment into the thermal conductivity as well as its thermal latent heat. But unfortunately, there are only a few researches that investigate the addition of metal-oxide and graphene combination into the PCM. Therefore, in this research, Sa/Fe<sub>3</sub>O<sub>4</sub> and Sa/Fe<sub>3</sub>O<sub>4</sub>/NGP are dispersed into the stearic acid and both of structural and thermal properties were observed.

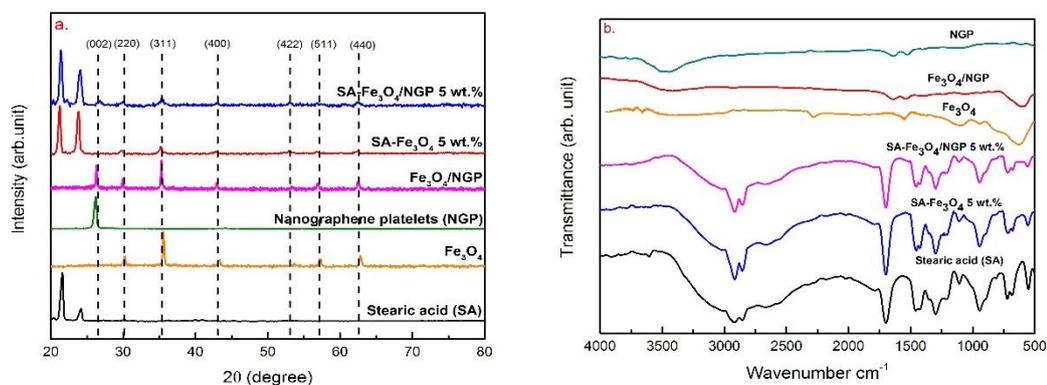
## 2. Experimental

In this work, Iron (II) sulfate heptahydrate (FeSO<sub>4</sub> · 7H<sub>2</sub>O, 99), sodium hydroxide (NaOH), nanoraphene platelets (NGP), and Stearic acid (Sa) as a phase change material are used without further purification. Fe<sub>3</sub>O<sub>4</sub> nanoparticle prepared by using the same method as our previous study [15] while the Fe<sub>3</sub>O<sub>4</sub>/NGP with 10% NGP is prepared by sol-gel method. In brief, the NGP is dissolved in 80 ml of distilled water and 40 ml ethanol by using the ultrasonic treatment. The resulted 10% NGP solution is mixed into the 2 gram of Fe<sub>3</sub>O<sub>4</sub> and are stirred for 2 h. The homogeneous suspension then is heated in the 120°C for 3h or 3 h to affect the deposition on the graphene sheets. The resulting nanocomposite is isolated by centrifugation and dried in vacuum at 70 °C for 12 h. The PCM mixtures of Sa/Fe<sub>3</sub>O<sub>4</sub> and Sa/ Fe<sub>3</sub>O<sub>4</sub>/NGP composites are prepared by dispersing the additives materials in the Sa with ultrasonic water bath for 2 hours. The ratio of Fe<sub>3</sub>O<sub>4</sub> nanoparticle and Fe<sub>3</sub>O<sub>4</sub>/NGP nanocomposite were varied by 1, 3, and 5 wt% with the stearic acid.

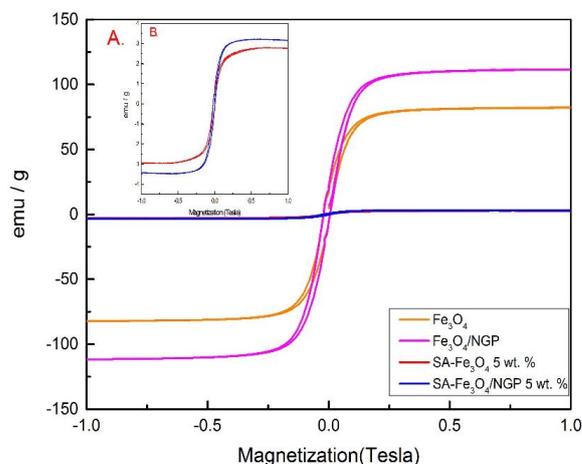
The structural properties and chemical structures of all the samples are measured by the X-Ray Diffraction (XRD) and Fourier Transform Infrared (FTIR) respectively while the magnetic behaviors of the sample are measured by the vibrating sample magnetometer (VSM) measurement. For the thermal properties, Differential Scanning Calorimetry (DSC) is used in order to know the phase change behavior and specific heat capacity of the samples while the Thermo-gravimetric Analysis (TGA) is used to observe the degradation process of the samples.

## 3. Results and discussion

The XRD spectrum of Sa, Fe<sub>3</sub>O<sub>4</sub> and NGP are plotted in Figure 1(a) along with the spectrum of Sa/Fe<sub>3</sub>O<sub>4</sub> and Sa/Fe<sub>3</sub>O<sub>4</sub>/NGP with 5 wt% ratio as a representation of the Sa mixtures. The XRD pattern of the Sa shows the presence of two high peaks below the 2θ = 25°. These peaks are due to Sa crystallization peak. Another pattern from the XRD spectra of Fe<sub>3</sub>O<sub>4</sub> confirms the inverse cubic spinel phase form at 2θ = 30.2°, 35.5°, 53.4°, and 57.0° which are attributed to the (220), (311), (511), and (440) crystalline planes. The strong diffraction pattern is also shown at 2θ = 26° for the (200) crystalline plane of NGP. The peaks of the Fe<sub>3</sub>O<sub>4</sub> as well as the peaks of Sa can be obviously seen in the XRD spectra of Sa/Fe<sub>3</sub>O<sub>4</sub> and Sa/Fe<sub>3</sub>O<sub>4</sub>/NGP with the additional NGP peak in the spectra. The XRD results indicate that all of the samples are consisted of the desired materials.



**Figure 1.** (a) XRD patterns and (b) FTIR spectrum of SA, Fe<sub>3</sub>O<sub>4</sub>, NGP, Fe<sub>3</sub>O<sub>4</sub>/NGP SA-Fe<sub>3</sub>O<sub>4</sub> wt. % and SA-Fe<sub>3</sub>O<sub>4</sub>/NGP 5 wt.%.



**Figure 2.** VSM Spectra of (a)  $\text{Fe}_3\text{O}_4$ ,  $\text{Fe}_3\text{O}_4/\text{NGP}$ ,  $\text{SA-Fe}_3\text{O}_4$  and  $\text{SA-Fe}_3\text{O}_4/\text{NGP}$  and (b)  $\text{SA-Fe}_3\text{O}_4$  and  $\text{SA-Fe}_3\text{O}_4/\text{NGP}$

**Table.1** Magnetic Saturation of  $\text{Fe}_3\text{O}_4$ ,  $\text{Fe}_3\text{O}_4/\text{NGP}$ ,  $\text{SA-Fe}_3\text{O}_4$  and  $\text{Fe}_3\text{O}_4/\text{NGP}$

Sample	M-S (emu/g)
$\text{Fe}_3\text{O}_4$	82
$\text{Fe}_3\text{O}_4/\text{NGP}$	112
$\text{SA-Fe}_3\text{O}_4$ 5 wt. %	2.76
$\text{SA-Fe}_3\text{O}_4/\text{NGP}$ 5 wt. %	3.16

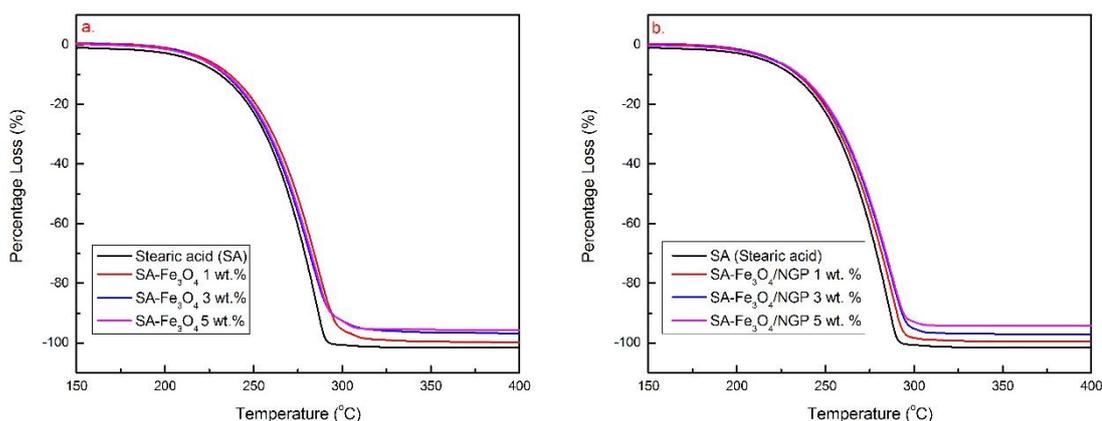
Figure 1(b) shows the FTIR transmittance pattern of Sa,  $\text{Sa-Fe}_3\text{O}_4$ , and  $\text{Sa-Fe}_3\text{O}_4/\text{NGP}$ . The transmittance pattern of Sa shows the strong absorption bands at  $2917\text{ cm}^{-1}$ ,  $2849\text{ cm}^{-1}$  and  $1703\text{ cm}^{-1}$  which are attributed to the stretching vibration of  $-\text{CH}_3$ ,  $-\text{CH}_2$  and  $\text{C}=\text{O}$  molecules respectively while another absorption bands at  $1464\text{ cm}^{-1}$ ,  $1296\text{ cm}^{-1}$ ,  $934\text{ cm}^{-1}$  and  $720\text{ cm}^{-1}$  correspond to in plane, out of plane and in plane swinging of OH molecules [16]. There are no transmittance pattern of the  $\text{Fe}_3\text{O}_4$  and NGP that can be seen in the  $\text{Sa-Fe}_3\text{O}_4$ , and  $\text{Sa-Fe}_3\text{O}_4/\text{NGP}$  due to the small amount of the additives materials. However, the similar transmittance pattern of the Sa in  $\text{Sa/Fe}_3\text{O}_4$  and  $\text{Sa/Fe}_3\text{O}_4/\text{NGP}$  spectra indicate that there are no chemical pattern takes place between the PCM and the additives.

The magnetic hysteresis curve of  $\text{Sa/Fe}_3\text{O}_4$  and  $\text{Sa/Fe}_3\text{O}_4/\text{NGP}$  alongside with  $\text{Fe}_3\text{O}_4$  and  $\text{Fe}_3\text{O}_4/\text{NGP}$  as the additive materials are plotted in Figure 2. The magnetic saturation values are also listed in the Table 1. The magnetic behaviour of  $\text{Fe}_3\text{O}_4$  is increased with the presence of the NGP particles. The same tendency is also observed for  $\text{Sa/Fe}_3\text{O}_4$  and  $\text{Sa/Fe}_3\text{O}_4/\text{NGP}$  with the smaller magnetic saturation due to the small amount of the  $\text{Fe}_3\text{O}_4$  nanoparticle and  $\text{Fe}_3\text{O}_4/\text{NGP}$  nanocomposite in the Sa.

For the thermal degradation process, the TGA measurements results of the  $\text{Sa/Fe}_3\text{O}_4$  (Figure 3(a)) and  $\text{Sa/Fe}_3\text{O}_4/\text{NGP}$  (Figure 3(b)) show that the Sa begins to decompose at about  $200^\circ\text{C}$ . However, the slightly change in temperature decomposition are found in the  $\text{Sa/Fe}_3\text{O}_4$  samples and are more distinguishable in the  $\text{Sa/Fe}_3\text{O}_4/\text{NGP}$  samples. Higher temperature decomposition process can provides the higher thermal stability on the PCM which is very important to the thermal energy storage. The TGA measurement also shows the information about the residual weight composition of the sample after the decomposition process. The percentage loss of Sa in Table 2 corresponds to the composition of the samples  $\text{Sa/Fe}_3\text{O}_4$  and  $\text{Sa/Fe}_3\text{O}_4/\text{NGP}$ . DSC curve of the  $\text{Sa/Fe}_3\text{O}_4$  and  $\text{Sa/Fe}_3\text{O}_4/\text{NGP}$  with 1, 3, and 5wt% ratios are shown in Fig. 4a and Fig. 4b respectively in order to know the melting and solidifying process of the Sa composites as the PCM. The melting and solidifying points of the samples from the DSC curves are tabulated in Table 3 along with the latent heat values. The presence of the  $\text{Fe}_3\text{O}_4$  and  $\text{Fe}_3\text{O}_4/\text{NGP}$  do not cause any favourable effect on the melting and solidification

point. The Sa completely melt at temperature 70.5 °C while its solidifying temperature is 66 °C. The slightly change occurs in the melting and solidifying temperature for the Sa/Fe<sub>3</sub>O<sub>4</sub> and Sa/Fe<sub>3</sub>O<sub>4</sub>/NGP. Both latent heat of melting and solidifying are higher for Sa/Fe<sub>3</sub>O<sub>4</sub> and Sa/Fe<sub>3</sub>O<sub>4</sub>/NGP than that of Sa. The enhancement of the latent heat can be explained with molecular attraction between Sa and the nanocomposite [17-18]. The influence of the addition of Fe<sub>3</sub>O<sub>4</sub> and Fe<sub>3</sub>O<sub>4</sub>/NGP on enhancing the value of latent heat showed the enhancement ability of the Sa composites to absorb and release heat with greater energy capacity.

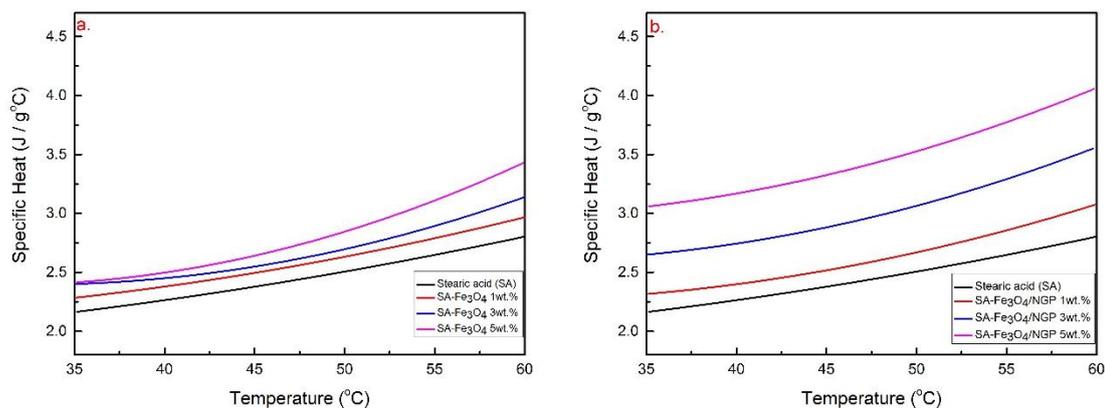
The specific heat values of the samples are also calculated from the DSC results and are plotted in



**Figure 3.** TGA (a) SA-Fe<sub>3</sub>O<sub>4</sub> 5 wt. % various wt. % and (b) SA-Fe<sub>3</sub>O<sub>4</sub>/NGP various wt. %

**Table 2.** TGA result of SA- Fe<sub>3</sub>O<sub>4</sub> 5 wt. % various wt. % and SA- Fe<sub>3</sub>O<sub>4</sub>/NGP various wt. %

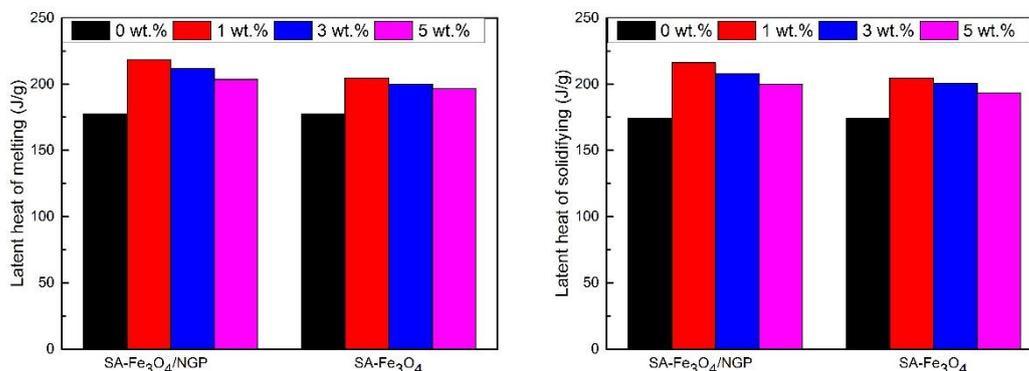
Sample	Onset Temp	Percentage Loss
Stearic acid (SA)	251.6°C	100%
SA-Fe <sub>3</sub> O <sub>4</sub> 1 wt. %	251.4°C	99.89%
SA-Fe <sub>3</sub> O <sub>4</sub> 3 wt. %	250.2°C	96.41%
SA-Fe <sub>3</sub> O <sub>4</sub> 5 wt. %	251.6°C	94.49%
SA-Fe <sub>3</sub> O <sub>4</sub> /NGP 1 wt. %	253.5°C	99.19
SA-Fe <sub>3</sub> O <sub>4</sub> /NGP 3 wt. %	253.6°C	96.99
SA-Fe <sub>3</sub> O <sub>4</sub> /NGP 5 wt. %	254.2°C	94.17



**Figure 4.** Specific heat of solid polynomial fitting curve (a) SA-Fe<sub>3</sub>O<sub>4</sub> various wt. % and (b) SA-Fe<sub>3</sub>O<sub>4</sub>/10% NGP various wt. %

**Table 3.** DSC result of SA-Fe<sub>3</sub>O<sub>4</sub> 5 wt. % various wt. % and SA-Fe<sub>3</sub>O<sub>4</sub>/10% NGP various wt. %

Sample	Solid-liquid			Liquid-solid		
	T <sub>onset</sub>	T <sub>peak</sub>	T <sub>endset</sub>	T <sub>onset</sub>	T <sub>peak</sub>	T <sub>endset</sub>
Stearic acid (SA)	67.5	70.5	76	67.2	66.6	63.1
SA-Fe <sub>3</sub> O <sub>4</sub> 1 wt. %	67.53	70.48	76.1	66.26	66.72	62.81
SA-Fe <sub>3</sub> O <sub>4</sub> 3 wt. %	67.52	70.16	75.8	65.88	66.83	62.74
SA-Fe <sub>3</sub> O <sub>4</sub> 5 wt. %	67.4	70.4	76.1	64.6	66.6	62.2
SA-Fe <sub>3</sub> O <sub>4</sub> /10% NGP 1 wt. %	67.78	70.57	76.2	67.84	66.73	63.16
SA-Fe <sub>3</sub> O <sub>4</sub> /10% NGP 3 wt. %	67.61	70.43	76.1	67.89	66.7	63.24
SA-Fe <sub>3</sub> O <sub>4</sub> /10% NGP 5 wt. %	67.63	70.42	75.9	67.89	66.8	63.45

**Figure 5.** Latent heat melting proses and solidifying proses.

the Figure 4(a) and Figure 4(b) for the Sa/Fe<sub>3</sub>O<sub>4</sub> and Sa/Fe<sub>3</sub>O<sub>4</sub>/NGP respectively. The result showed that Fe<sub>3</sub>O<sub>4</sub> nanoparticles increase the specific heat value of Sa and further increase for Fe<sub>3</sub>O<sub>4</sub>/NGP. The value of specific heat also increase as with the addition of weight percent of the Fe<sub>3</sub>O<sub>4</sub> added to either with or without NGP. The enhancement of specific heat indicates that the addition of the Fe<sub>3</sub>O<sub>4</sub> and Fe<sub>3</sub>O<sub>4</sub>/NGP in the Sa also enhance the ability of the Sa in the sensible heat condition.

The comparison of latent heat storage between pure SA and PCM composites and also data of characterization are tabulated in Table 3. In charging process both PCM composites in onset, peak, and endset temperature do not significantly change from pure SA. On the other hand in discharging process, PCM composites shows difference behaviour in onset temperature from pure SA. In SA-Fe<sub>3</sub>O<sub>4</sub> discharging curve shows that onset temperature become more smaller with increasing Fe<sub>3</sub>O<sub>4</sub> wt.%. While in SA-Fe<sub>3</sub>O<sub>4</sub>/NGP discharging process curve shows a small increases in onset temperature. It is shows that small amount of NGP in Fe<sub>3</sub>O<sub>4</sub>/NGP can maintain the onset temperature of pure stearic acid. In Figure 5 plotted latent heat storage charging and discharging process of PCM composites. From Figure 6 shows that latent heat storage of samples increases with content of material additives various wt. %. Maximum enhancement of latent heat PCM composites occurs in content material additives at 1 wt.%. The enhancement of latent heat of SA-Fe<sub>3</sub>O<sub>4</sub>/NGP shows higher improvement than SA-Fe<sub>3</sub>O<sub>4</sub>. Latent heat of SA-Fe<sub>3</sub>O<sub>4</sub>/NGP improved by about 23% while SA-Fe<sub>3</sub>O<sub>4</sub> improved by 15%. The enhancement of latent heat storage PCM composite is due to molecular interaction between molecules SA and material additives. Molecular interactions between molecules and material additives is physical interactions that are affected by the large surface area of material additives. The larger surface area of material additives produces greater interaction molecular with molecules SA [18-20]. From the result of XRD spectra Fe<sub>3</sub>O<sub>4</sub>/NGP have smaller crystallite size than Fe<sub>3</sub>O<sub>4</sub> that can be lead to increased surface area. Therefore the enhancement of latent heat SA with addition Fe<sub>3</sub>O<sub>4</sub>/NGP have better improvement than Fe<sub>3</sub>O<sub>4</sub>.

#### 4. Conclusion

Fe<sub>3</sub>O<sub>4</sub> nanoparticles and Fe<sub>3</sub>O<sub>4</sub>/NGP nanocomposites were synthesized using sol gel method and used as material additives into PCM composites various wt. % prepared by dispersion using sonication method. Material additives used as enhancement latent heat storage of stearic acid. Based on XRD synthesized PCM composite shows that crystal structure SA and material additives remain unchanged. Thermal stability of SA-Fe<sub>3</sub>O<sub>4</sub>/NGP slightly increase than SA because the incorporation of NGP in nanocomposites. Enhancement of specific heat and latent heat storage were confirmed via DSC curve show that both of Fe<sub>3</sub>O<sub>4</sub> and Fe<sub>3</sub>O<sub>4</sub>/NGP increased thermal performance of SA. In comparison material additives for percentage of enhancement specific heat and latent heat storage of SA, Fe<sub>3</sub>O<sub>4</sub>/NGP have higher improvement than Fe<sub>3</sub>O<sub>4</sub> because of small content NGP in nanocomposites.

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