

CHARACTERIZATIONS AND ELECTRON PARAMAGNETIC RESONANCE (EPR) QUASICRYSTALS IN AN ICOSAHEDRAL PHASE

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ABSTRACT

This article deals with a study on the icosahedral phase quasicrystal of $Al_{63}Cu_{25}Fe_{12}$ produced by casting in plasma induction furnace under an argon atmosphere to guarantee it's homogenously. But found in phase transformations from one phase to another, the structures formed by the quasicrystalline alloy having ordered icosahedral phase, leads to low electrical conductivity; but considering the location of weak electron-electron interactions. For microstructural characterization was made using the Following techniques; X-ray Diffraction, Scanning Electron Microscopy, Energy Dispersive Spectroscopy of, DSC Thermal Analysis and Spectroscopy Electron Paramagnetic Resonance (EPR), where the spectrum Obtained experimental EPR presents an amount of resonance $g = 2,036$. EPR spectrum provide a detailed view of magnetic equivalent sites to those of paramagnetic transition metal compositional variation in the alloy quasicrystalline and transition temperature effects on the magnetic spins in the sample ordering.

Keywords: EPR Spectrum; Quasicrystal; Microstructural Characterization ; Icosahedral Phase.

1. INTRODUCTION

The quasicrystals are solid atomic structures with periodic and quasi crystallographic symmetry forbidden, since the physical and chemical properties are different from conventional crystalline solids, by presenting a long-range periodicity ¹. The quasicrystalline alloy AlCuFe system and other families of quasicrystal have attracted considerable interest because of good physical and chemical properties, mechanical properties and tribological will know: a low electrical and thermal conductivity, good corrosion resistance and oxidation, high hardness and low friction coefficient ². Investigative awakening in quasicrystal material comes from the thermodynamic stability, non-toxicity and low production cost of quasicrystalline alloy and great potential in various area of health, industry and technological field. Some important applications of this material under study are: Some important applications of this material under study are: photonics, electronic, magnetic, biomaterials and good hydrogen storers to be used in catalytic reactions.

The power system quasicrystalline Al-Cu-Fe has compositions in the range of 58-70% Al, 20-28% Fe and 10-14% Cu with, allowing the thermodynamic stability in quasicrystalline icosahedral phase. The transformation from the solid phase is due quasicrystalline (β crystalline, amorphous and icosahedral and present intermetallic). But it is emphasized the transition from one phase to another, there is presence of amorphous gamma this process occurs through nucleation from a single crystal of $\gamma-Al_2O_3$ as the core of a layer of amorphous alumina present in quasicrystal under normal temperature and pressure conditions. The preparation of the alloy Al-Cu-Fe, takes place through a thermal aluminum reaction as the reaction (1), this overheating is fusion or even the rapid solidification ³. They are used iron powders, copper and aluminum which are added to reduce the adiabatic combustion temperature and thereby avoid vaporization of the other components.



Aluminum, copper and iron oxide powders were weighed according to the stoichiometry of equation (2) for the production of quasicrystalline materials Al-Cu-Fe with a nominal composition of $Al_{63}Cu_{25}Fe_{12}$,



The icosahedral phase quasicrystalline present in the Al-Cu-Fe alloy is formed by peritectic solidification of crystalline and amorphous phases of high temperature react with a liquid phase ⁴.

In view of the formation of quasicrystalline phase from other phases occurred in icosahedral structure, the intermetallic compounds are related to the AlFe AlFeCu ⁵. The β -AlFe structure is a driving force in the formation

of quasicrystalline phase AlFeCu. Therefore, these structures share similar properties and can say both are naturally brittle. Therefore, it is expected that the AlFeCu mechanism suffers from being fragile intermetallic system similar to AlFe ⁶.

When the quasicrystal is subjected to the heat treatment of 8 hours, no decrease in oxidation of the alloy, being able to obtain and observe the icosahedral structure as shown in Figure. 1.

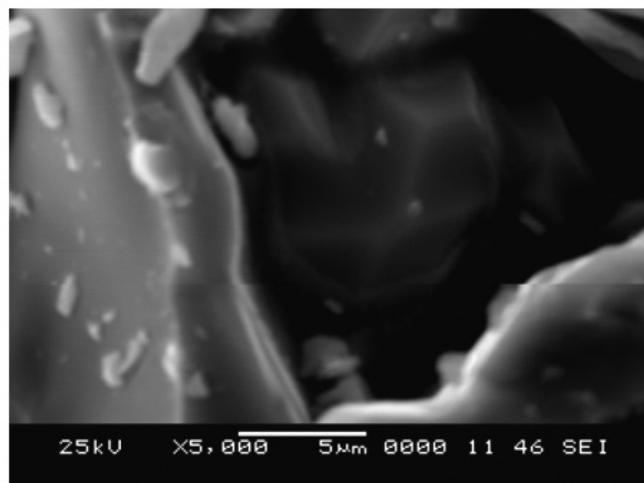


Figure 1. SEM image of the surface quasicrystalline alloy $Al_{63}Cu_{25}Fe_{12}$ in icosahedral phase.

Conduction electrons is an important study in quasicrystals because a physical understanding of the electronic density of states. Therefore, another interesting aspect of that emerges from the location of the unpaired electron spins of the electrons present in quasicrystals in its magnetic moment contained in the transition metals (Mn, Ni, Co, Nb, and Fe) and rare earth metal atoms.

The interaction between the magnetic exchanges unpaired d f moments leads to magnetic ordering phenomena carrying a magnetic ordering of their spins or antiferromagnetic type at low temperatures ⁷.

However, to date the quasicrystalline alloy (Al-Mn-Si) and (Al-Mn-B-Pd) exhibited ferromagnetism by which were investigated by electron paramagnetic resonance, verified the existence of ferromagnetic resonance that is reported in ferromagnetic phase. This work has the purpose to study some techniques of

characterization and magnetic effects through the use of electron paramagnetic resonance in the icosahedral phase of quasicrystal Al-Cu-Fe.

2. THEORY

2.1 EPR Spectroscopic Study of $Al_{63}Cu_{25}Fe_{12}$ Quasicrystal in Non-magnetic Phase

The electrical resistivity of quasicrystals is several orders of magnitude, but in the transition metals is higher. Typical lower resistivity at room temperature are in the range $\rho \approx 10^2 - 10^3 \mu\Omega\text{cm}$ for quasicrystals decagonal and icosahedral with $\rho \approx 10^3 - 10^6 \mu\Omega\text{cm}$, however, compared to the resistivity of aluminum metal which $\rho_{(77\text{K})} = 0,3 \mu\Omega\text{cm}$ ⁸. The resistivity of the quasicrystals and the remaining orders of magnitude lower than that seen in the insulation so that the quasicrystals can be generally considered as weak metals or semimetals.

Paramagnetic Resonance Spectroscopy metal gives rise to two reasons: The first is the fact that the intensity of the paramagnetic resonance spectroscopy line is proportional to the static spin susceptibility, the conduction electrons, it is of a type Pauli, their order of magnitude can be given by⁹.

$$\chi_p = \frac{(g\mu_B)^2}{k_B T_F} \quad (3)$$

An electron of an atom in the periodic quasi network then has a total angular momentum which is the sum of orbital angular momentum and orbital spin and spin in accordance with equation (4) below¹⁰:

$$\vec{J} = \vec{L} + \vec{S} \quad (4)$$

The relation between the orbital and spin magnetic moments and their angular momentum is different, i.e. that the total magnetic moment, in general, is not parallel to the total angular momentum. Here μ_B the same relation holds for an atom as a whole, where an electron from an atom then has a total angular momentum. However the components and μ_z maintaining a proportional relationship which is expressed as:

$$J_z = \hbar m_j \quad (5)$$

In this case j is the sum of the angular momentum of all its electrons and the quantum number can assume both integer values as semi-integer, where in unit steps and, is the Fermi temperature of the order of 10^4K according to equation (5).

$$T_F = \frac{E_F}{k_B} \quad (6)$$

This susceptibility remains four orders of magnitude smaller than, the type of Curie paramagnetic susceptibility of localized electrons at temperatures of liquid He to a comparable density of electron spins (since electrons are not located in a magnetically ordered state). The second reason for the weakness of the paramagnetic resonance spectroscopy signal is the surface that mediate the effect of conduction electrons with their.

Spins, where the microwave field penetrates the metal sample only to the radius of the layer of depth order given by,

$$\delta = \sqrt{\frac{\rho}{\mu_0 \pi \nu}} \quad (7), \text{ with } \mu_0 \text{ is the magnetic permeability in vacuum.}$$

The layer effect dramatically reduces the number of excited electrons which contribute to electron paramagnetic resonance signal. In the X-band electron paramagnetic resonance with the resonance frequency of free electrons $g_f \approx 2$ in $\nu = 9,6\text{GHz}$ a typical resistivity $\rho \approx 2.10^3 \mu\Omega\text{cm}$ quasicrystal in a sample which results in a depth of layer $\delta = 23 \mu\text{m}$ ¹¹. For electron paramagnetic resonance up field held in $\nu = 109\text{GHz}$, the depth of the layer reduces the $\delta = 7 \mu\text{m}$. The microwave field can thus excite a small amount of conduction electrons with the surface layer of a single sample. The factor g of conduction electrons in metals is generally not very different from the value of free electrons $g_f \approx 2$, the reason is that the exchange interaction between the conduction electrons fast moving average is zero and therefore unable to change the resonance frequency.

The small difference between actual values and the free electrons is of the order $g - g_f \leq 10^{-2}$ normally found in metal; it is attributed to the spin-orbit coupling, which causes the electron energy dependent upon the relative

orientation between its two angular momentum, orbital and spins¹². A related situation should be applied to quasicrystals as well. The average conduction electrons should be centered again towards the value of free electrons for the same reason. The spin-orbit coupling, however, deserves more attention in this study.

2.2 Electron Paramagnetic Resonance in Quasicrystals Magnetic

Some quasicrystals icosahedral systems such as; Al-Pd-Re, Al-Cu-Ni, Al-Cu-Ga, Al-Ba-Mo, Al-Cu-Mg, Al-Mn-Pd, Al-Co-Cu and Al-Pd-Ce are perfectly diamagnetic, others may contain paramagnetic centers in addition to itinerant magnetic moments of the conduction electrons. Examples thereof are transition metals containing icosahedral families Al-Pd-Mn, Al-V-Cu, Al-Fe-Ni, Pt-Al-Fe, Al-Co-Ni, Fe-Cr-Al and Cu-Al-Fe, where the Cr atoms, Co, Cu, Ni, Pt, Mn and Fe have unpaired electrons in their energy sublevel "d"¹³. In this case two types of electronic resonances can be observed in both the conduction electrons in the electron spin location in quasi periodic work.

The quasicrystalline alloy containing iron atom Hamiltonian of disturbed crystalline electric field is usually much greater than the spin-orbit coupling should result in small displacements g of localized electron resonances. There is an important difference between the spin-orbit coupling induced displacements of the conduction electrons and localized moments. An electron located is moving in a spherical electrostatic potential with its electrons and the effect of the crystalline electric field meets the electronic orbital angular momentum, whereas it does not affect much the network own potential at the atomic level. The resulting displacement is therefore very small in quasi periodic quasicrystal work.

A conduction electron on the other hand, moves in the crystal a total potential. For regular metals this potential has the periodicity of the structure. In quasicrystals the periodicity of the structure does not exist, so that significant variations in potential sites without translational periodicity. This presents a distribution displacements g which is not necessarily small. The conduction s electrons and localized spins are coupled by an exchange interaction of form¹⁴,

$$H_{ex} = \sum_{i=1}^n \int d^3 \vec{R} J_{sd} (\vec{R}_i - \vec{R}) \vec{S}_{di} (\vec{R}_i) \cdot \vec{S}_F (\vec{R}_i - \vec{R}) \quad (7)$$

Here J_{sd} is the exchange coupling constant, $\vec{S}_F (\vec{R}_i - \vec{R})$ is the driving density of free electrons at one point \vec{R} , $\vec{S}_{di} (\vec{R}_i)$ it is the total spin of the d electron of the transition metal atom located at point \vec{R}_i work and the sum is over all atoms of the magnetic structure of quasicrystalline superlattice. This interaction introduces a broadening and a shift of the resonance. However, as the two resonant frequencies are almost equal, the resonance of conduction electrons and localized spins are aligned at the same time have to meet the natural frequencies of the coupled spins.

The resonance frequency becomes a function of the frequencies of the two subsystems, the exchange parameter J_{sd} and the network spin relaxation damping of the two systems. Application of Spin Resonance spectroscopy quasicrystal the sample occurred at the end of a resonant cavity. The quantity measured is the power absorbed by the sample from change when the external field is varied.

In this experiment the reflection was conducted in a band X (irradiation frequency 9.6 GHz, corresponding to the absorption center field 3400 G for $g \approx 2$ through a sample powder in quasicrystal $Al_{63}Cu_{25}Fe_{12}$ icosahedral phase. At room temperature until the liquid He, the spin resonance signal electrons can be detected just below 60 K in a relatively narrow line width, in the range of appeared $\Delta H_{pp} = (4,5-5)\text{G}$ according to Figure 2.

Its significance factor g varied in a small range to a low temperature of 4K mode was observed in the experiment done $g = (2,006-2,00063)$ whereas the intensity of the line by the reverse enhanced cooling of the Curie temperature $1/T$ (the Curie law of a paramagnetic material), as disclosed in Figure 2b). The signal obtained left the day of electron paramagnetic resonance experience confirmed the origin of the sample quasicrystal; this was achieved by removed from resonance. But, it is also more likely to originate from impurities that are in the sample, another form of extrinsic quasicrystalline structure.

The quasicrystal in high-field transmission electron spin resonance spectrometer operated in the frequency of microwave 109,270 GHz (corresponding to the absorption center of the field $B = 3,890\text{ T}$ to $g_f \approx 2$). The camp was swept in a range 0-6 T, where we can see that the magnetic field operates at up to 14 T fields. The transmitted signal is detected by a thermal bolometer. In this high-field transmission experiment of electron spin resonance was clearly detected quasicrystal $Al_{63}Cu_{25}Fe_{12}$ measurements were

carried out at a temperature below. The spectra obtained in this experiment did not appear clear or pure dispersion, but have a slight mixture of absorption. A weak signal with a narrow spectral width is observed near the position of free electrons in $B = 3,8684$ T, which is obviously the same seen in X band in Figure 2. In Figure 3 obtained by electron paramagnetic resonance shows the g factor for the icosahedral phase in quasicrystal Al-Cu-Fe where this shift is because of conduction electrons originating from the iron, because there was a slight variation for a low temperature 4 K, however, the intensity of the line increases by cooling the inverse of the Curie temperature $1/T$.

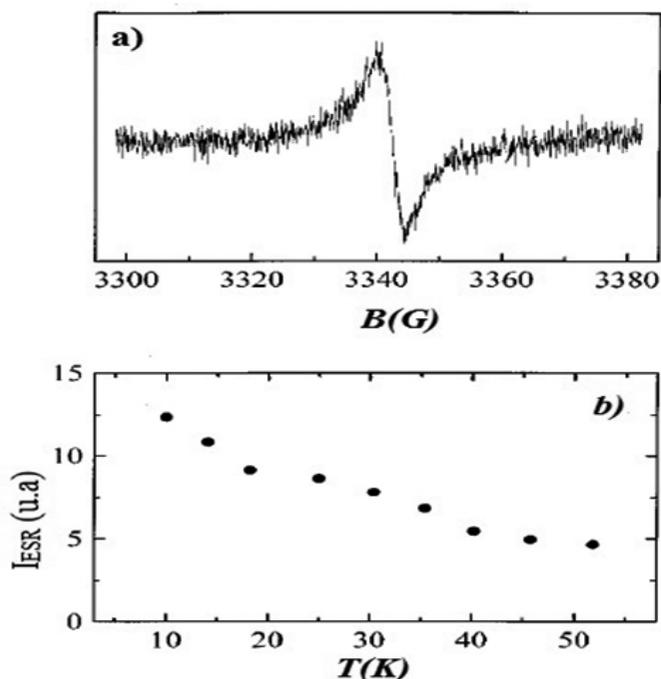


Figure 2. Electron paramagnetic resonance signal representation of the icosahedral quasicrystal $Al_{63}Cu_{25}Fe_{12}$ the 25 K. 2 (b) the intensity of the paramagnetic resonance signal of electrons. (Corresponding static susceptibility of electrons in the paramagnetic resonance) as a function of temperature.

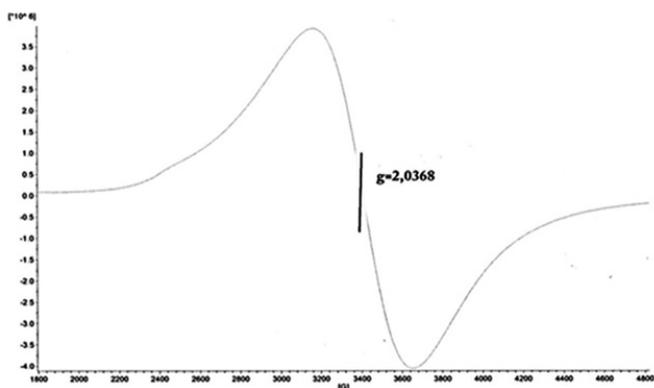


Figure 3. Electron spin resonance signal in quasicrystal $Al_{63}Cu_{25}Fe_{12}$ through reflection of electrons in the X-band.

The behavior of the icosahedral phase quasicrystal AlCuFe by some important magnetic properties by electron paramagnetic resonance, and the influence of the magnetic field and the Curie temperature factor $1/T$, these factors magnetism quasicrystalline the alloy is crucial for applications in some industrial sectors. The structures formed by quasicrystalline alloys have an icosahedral phase order, which provides low electrical conductivity, but is due to the location of weak electron-electron interactions. The microstructure effect on quasicrystal is observed by a high polarization field, to analyze the heterogeneous spin caused by defects, dislocations and fluctuations of random

magneto crystalline.

The $Al_{63}Cu_{25}Fe_{12}$ quasicrystal has localized magnetic moments on iron atoms (Fe) since only 10^{-4} Fe atoms have centered moments. The resistivity of the icosahedral quasicrystal Al-Fe-Cu and Al-Pd-Mn are almost the same, so that a similar concentration of conduction electrons must exist in both quasicrystals. Furthermore, both quasicrystal families are structurally almost identical.

3. MATERIALS AND METHODS

From the pre-established stoichiometry $Al_{62}Cu_{25}Fe_{12}$ were heavy aluminum powders (99.97%), iron (99.9%) and copper (99.9%) in an analytical balance, the Gehaka brand, AG-200. The alloy was obtained by melting the pure elements of air. The induction furnace was used with Argon atmosphere controlled at 5.0 with the objective of obtaining a good homogenization of the quasicrystalline phase.

We used a high-frequency generator (40 kVA) POLITRON of manufacture. Each element that makes up the quasicrystalline alloy contains 10g; this measurement procedure was performed on a Shimadzu balance. The training method was through solidification in a cold hearth furnace to generate a heterogeneous alloy, being a common procedure a mixture of quasicrystalline phase to the crystalline phase.

To be a proportional increase of the quasicrystalline phase in the sample, it is necessary to heat treat, so that will favor the peritectic transformation of phases. This was accomplished using heat treatment in a resistance furnace Nabertherm, where the samples are maintained during the period of 8h and 24 h at a temperature of 1123 K. The diffraction of X-ray (XRD) was used to monitor the progress of the phases and stability of samples during casting. It was used for both a diffraction SIEMENS D5000 and is used CuK radiation whose wavelength is $\lambda = 1,5406 \text{ \AA}$. To examine the morphology of the powders being quasicrystalline using a LEO Scanning Electron Microscope, Model 1430, OXFORD coupled to a probe. The samples after casting and were placed in catalytic tests dispersion in isopropyl alcohol solution. The energy dispersive spectroscopy (EDS) in the sample binds quasicrystalline $Al_{62}Cu_{25}Fe_{12}$ micrograph was obtained at an energy dispersive analysis after 8 hours of heat treatment coupled with the SEM sample which has previously been metallized with gold (average thickness 12nm). The thermal properties of alloys were analyzed by Differential Scanning Calorimeter (DSC) at $10 \text{ }^\circ\text{C}\cdot\text{min}^{-1}$ heating rate under N_2 flow.

4. RESULTS AND DISCUSSION

4.1 X-ray Diffraction

The diffraction spectra of X-rays with the sample $Al_{62}Cu_{25}Fe_{12}$ stoichiometry is shown in Figure 4 respectively for the thermal treatment of 24 hours at 1123 K. This diffraction peak can identify the phases; the icosahedral quasicrystalline phase (I-phase) and tetragonal phases (θ - $AlCu_3$, θ - $CuAl_2$ and β - Al_3Cu) and intermetallic (ω - Al_7Cu_2Fe , λ - $Al_{13}Fe_4$, λ - Al_3Fe_2 , ω - Al_7Cu_2Fe and ϕ - $Al_{10}Cu_{10}Fe_2$).

These results also suggest that the β phase is directly formed from the liquid alloy. Moreover, the β phase transforms below 873 K to form the λ and θ stages, which are solid solutions induced by the solubility of Cu and Fe. The intensity of the peaks corresponding to the i-phase layer is higher than the peaks are specifically related to the β -phase.

However, $Al_{62}Cu_{25}Fe_{12}$ in the alloy melt, i-phase coexisting with a small amount of AlFe(Cu) solid solution¹⁵. The composition of $Al_{62}Cu_{25}Fe_{12}$ has phases in balance with other crystalline phases such as β - Al_2Fe_3 , γ - $Al_{13}Fe_4$, γ - Al_3Fe_2 , θ - $AlCu$, ω - Al_7Cu_2Fe and ϕ - $Al_{10}Cu_{10}Fe_2$. Among these phases ω crystalline shows a great similarity with the icosahedral ψ phase. The coordination of Fe atoms is very similar in face-centered cubic structure with Cu being passivity with Al_2O_3 . From the solidification path, the ϕ phase seems to be formed at equilibrium conditions with high thermodynamic stability, by a peritectic reaction from a crystalline and amorphous intermetallic phase due to phase transitions.

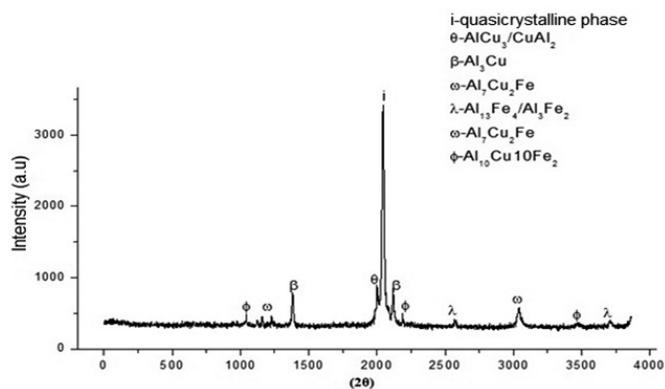


Figure 4. X-ray diffract gram (XRD) of the thermally treated sample $Al_{62}Cu_{25}Fe_{12}$ time of 24 h.

4.2 Scanning Electron Microscopy and EDS

Figure 5 and 6 are respectively represent the results obtained from samples of quasicrystals by Scanning Electron Microscopy and EDS spectrum post after being subjected to heat treatment with 8 hours.

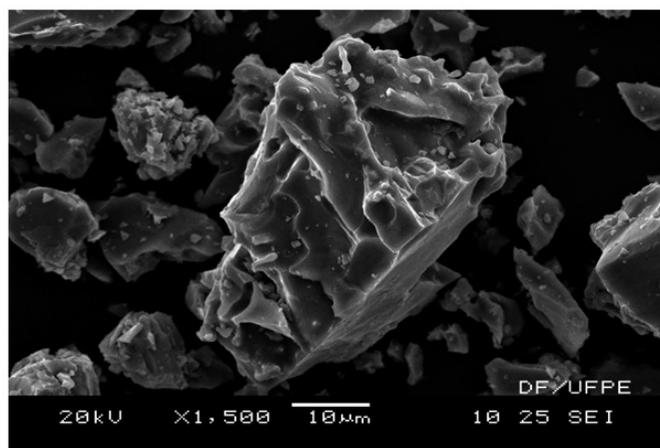


Figure 5. SEM showing the formation of intermetallic compound of $Al_{62}Cu_{25}Fe_{12}$ alloy after heat treatment of 8 hours.

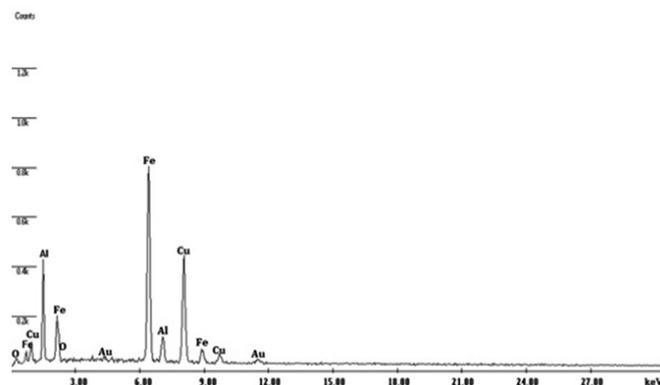


Figure 6. EDS spectrum of quasicrystalline alloy $Al_{62}Cu_{25}Fe_{12}$ showing the formation of intermetallic with irregular polyhedral.

The analysis Scanning Electron Microscopy (SEM) and Energy Dispersive Spectroscopy (EDS) were used for sample quasicrystal and produced Figures. 5 and 6 with the heat treatment of 8 hours. The image that the Scanning Electron Microscopy (SEM) and Energy Dispersive Spectroscopy spectrum

(EDS), view the microstructure of quasicrystalline alloy with surface irregular crystals and geometric uniformity.

The phases present already been processed (β -crystalline, and intermetallic icosahedral), there are show the strong presence of Fe and Cu conduction electrons which are protected by the thin layer of aluminum oxide which enables the peritectic reaction between phases¹⁶. In the spectrum of elemental analysis EDS, there is a greater prevalence of aluminum than the other elements (copper and iron) that makes up the quasicrystalline alloy. The existence of $\gamma-Al_2O_3$ favors spinel formation on Al_2O_3 oxidation in the presence of Cu or CuO. However, the possibility of formation of other oxides such as spine $CuFe_xAl_{2-x}O_4$, are essential reactions and catalytic composites of Fe_3O_4 and $CuAl_2O_4$ are complex to form a thin film, the passive quasicrystalline structure of the alloy.

4.3 Thermal Analysis DSC

Figure 7 shows the thermal behavior during non-isothermal heating made by Differential Scanning Calorimeter (DSC) conducted at a heating rate of $10^\circ C/MIM-1$ for quasicrystalline alloy powder during the heating $Al_{62}Cu_{25}Fe_{12}$ 400-1473K, exothermic events occur are clearly defined and distinguished in a low temperature range of 773 K, however, relates to the formation of $\phi-AlFe_3$ and- $\omega-Al_7Cu_2Fe$ phases, which represents a fusion event or dissolution of the phases (amorphous intermetallic) corresponds to merging with predominance of copper and iron. A more complex sequence of thermal events is observed above 773 K. The increased heating rate causes a moderate change (~ 293 K) of the phase transition temperatures towards higher values, otherwise the following thermal events does not change. The $\omega-Al_7Cu_2Fe$ phase is the main precursor of ϕ -phase ($i-Al_{62}Cu_{25}Fe_{12}$)¹⁷.

As the homogenization of the alloy composition is gradually completed, $\omega-Al_7Cu_2Fe$ makes if main constituent. The enthalpy of the minimum first peak is smaller than the other two peaks, indicating that the highest concentration of phases of $\omega-Al_7Cu_2Fe$ type and ($i-Al_{62}Cu_{25}Fe_{12}$). This result indicates the need for a detailed analysis of the microstructural evolution and the formation and stability of constituent phases during non-isothermal heat treatments, because this enables rich oxide layer that is essential for catalytic purposes.

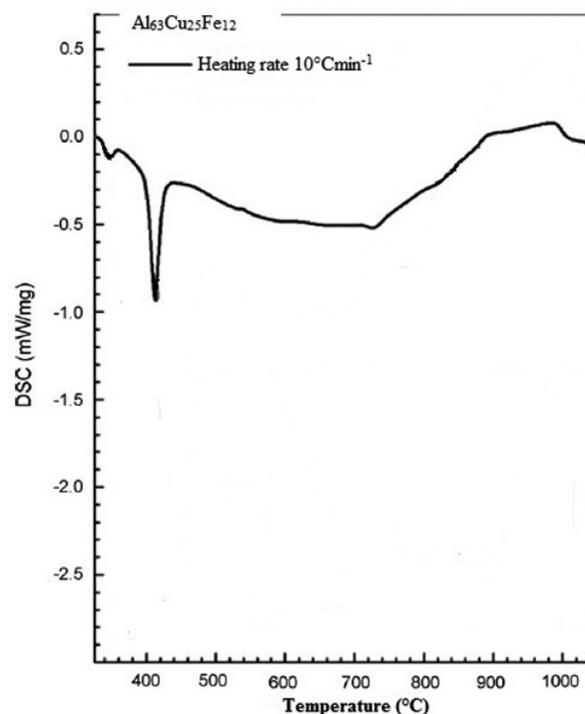


Figure 7. DSC curve of quasicrystal $Al_{62}Cu_{25}Fe_{12}$.

CONCLUSIONS

- The main conclusions of the study are as follows:
 - The icosahedral phase becomes stable above 948 K and the crystalline phase a solid solution which controls the formation of the icosahedral phase due to its surface having rotational symmetry that are important to describe the mobility of conduction electrons at the surface;
 - The nuclear magnetic resonance spectroscopy in quasicrystal is able to detect signs of both itinerant conduction electrons and localized magnetic moments of atoms can provide important information about the electronic properties of quasicrystalline materials;
 - The use of a conventional X-band in the electron paramagnetic resonance in the conventional apparatus which employs downside scanning a relatively narrow field of 0-1 T;
 - The sharp widening of the spectrum icosahedral quasicrystal $Al_{63}Cu_{25}Fe_{12}$ is doubly investigated. The first is the distribution of local electric fields due to the absence of periodicity in quasicrystal reticulum and the second is the exchange interaction energy sublevels in $s-d$ between the conduction electrons and magnetic moments localized d sublevel in Fe ;
 - An abnormal temperature dependence of the electron in the line width in the electron paramagnetic resonance, which narrow 20-5 K by a factor of 3,7 which can originate from the change in the exchange interaction strength in the sublevels $s-d$ due to conduct screening electrons of the locations of its magnetic moment in sublevel d Fe atom.

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REFERENCES

1. JAMSHIDI, A.L.C.L.; NASCIMENTO, L.; RODBARI, R.J.; BARBOSA, G.F.; MACHADO, F.L.A.; PACHECO, J.G.A.; BARBOSA, C.M.B.M. Use Alloy Quasicrystalline $Al_{62.2}Cu_{25.3}Fe_{12.5}$ for Steam Reforming of Methanol. *J. Chem. Eng. Process. Technol.* 5:187, pp.2-6, 2014.
2. OLSSON, S.; FREDRIK, E.; BIRCH, J.; HULTMAN, L. Formation of α -approximant and quasicrystalline Al-Cu-Fe thin films. *Thin Solid Films* 526, pp. 74-80, 2012.
3. LI, L.; BI, Q.; YANG, J.; FU, L.; WANG, L.; WANG, S.; LIU, W. Large-scale synthesis of Al-Cu-Fe submicron quasicrystals. *Scripta Materialia* 59, pp. 587-590, 2008.
4. SRIVASTAVA, V.C.; HUTTUNEN-SAAKIVIRTA, E.; CUI, C.; UHLENWINKEL, V.; SCHULZ, A.; MUKHOPADHYAY, N.K. Bulk synthesis by spray forming of Al-Cu-Fe and Al-Cu-Fe-Sn alloys containing a quasicrystalline phase. *Journal of Alloys and Compounds* 597, pp.258-268, 2014.
5. WIDJAJA, E.J.; MARKS, L.D. Microstructural evolution in Al-Cu-Fe quasicrystalline thin films. *Thin Solid Films* 441, p.63-71, 2003.
6. NASCIMENTO, L.; AGOSTINHO, L.C.L.; CAVALCANTI, F.B. Grouping Model in Fermi Surface Applied to Quasicrystals. *Revista Colombiana de Materiales* N 3: pp.55-62, 2012.
7. AZHAZHAA, V.; KHADZHAYB, G.; MALIKHIN, S.; MERISOVB, B.; PUGACHOV, A. The electrical resistivity of Ti-Zr-Ni quasicrystals in the interval 1.3-300 K. *Physics Letters A* 319, pp.539-543, 2003.
8. ROY, M. Formation and magnetic properties of mechanically alloyed $Al_{65}Cu_{20}Fe_{15}$ quasicrystal. *Journal of Magnetism and Magnetic Materials* 302, pp.52-55, 2006.
9. TRSIC, M.; PINTO, M.F.S. *Química Quântica: Fundamentos e Aplicações*, Editora Manole LTDA., Barueri-SP, 2009.
10. MISRA, S.K.; KAHRIZI, M.; SINGH, K.; VENKATRAMANI, N.; BAHADUR, D. EPR, X-ray, and magnetization studies of metastable alloys Ti_2Fe , $Al_{40}Cu_{10}Mn_{25}Ge_{25}$ and $Al_{65}Cu_{20}Fe_{15}$ as prepared by mechanical alloying. *Journal Magnetic Materials* 150, pp.430-436., 1995.
11. GOMILŠEKA, J.P.; ARČONB, I.; KODREB, A.; DOLINŠEKB, J. EXAF study of the Fe local environment in icosahedral AlCuFe and its relation to magnetism of quasicrystals. *Solid State Communications* 123, pp.527-530, 2012.
12. MARION, J.B.; THORNTON, S.T. *Classical Dynamics of Particles and Systems* - 5ª. Ed., Saunders College Publishing / 2004.
13. AGOSTINHO, L.C.L. *Estudo da Aplicabilidade dos Quasicristais AlCuFe em Reações Catalíticas na Oxidação do Metanol*, Dissertação (Mestrado em Ciências de Materiais). Universidade Federal da Paraíba, João Pessoa-Paraíba, 2009.
14. SAKURAI, J. J. *Modern Quantum Mechanics*, Revised Edition, Addison-Wesley, 1994.
15. SALES, M.; MERSTALLINGER, A.; USTINOV, A.I.; POLISHCHUK, S.S.; MELNICHENKO, T.V. Effect of the addition of crystalline β -phase in Al-Cu-Fe quasicrystalline coatings on their tribological properties. *Surface & Coatings Technology* 201, pp.6206-6211, 2007.
16. SVIRIDOVA, T.A. SHEVCHUKOV, A.P. SHELEKHOV, E.V.; DIAKONOV, D.L.; GORSHENKOV, M.V. KALOSHKIN, S.D. TCHERDYNTSE, V.V. LI, Z.; FAN, G. Nano quasicrystalline Phase in Mechanically Alloyed and Heat-Treated $Al_{73}Cu_{11}Cr_{16}$. *Acta Physica Polonica A* 126, pp.599-602, 2014.
17. S. YIN, Z. XIE, Q. BIAN, B. HE, Z. PAN, Z. SUN, Z. WEI, L. Qian and S. Wei: *J. Alloys Compd.*, 455, pp.314-321, 2008.