

NANO EXPRESS

Open Access



# Effect of Composition, Interface, and Deposition Sequence on Electrical Properties of Nanolayered Ta<sub>2</sub>O<sub>5</sub>-Al<sub>2</sub>O<sub>3</sub> Films Grown on Silicon by Atomic Layer Deposition

Junpeng Li, Jianzhuo Wu, Junqing Liu\*  and Jiaming Sun\*

## Abstracts

Nanolayered Ta<sub>2</sub>O<sub>5</sub>-Al<sub>2</sub>O<sub>3</sub> composite films were grown on *n*-type silicon by atomic layer deposition (ALD) within the overlapped ALD window of 220–270 °C. Moreover, post-annealing treatment was carried out to eliminate defects and improve film quality. Nanolayered Ta<sub>2</sub>O<sub>5</sub>-Al<sub>2</sub>O<sub>3</sub> composite films remain amorphous after 700 °C annealing. The effects of composition, interface, and deposition sequence on electrical properties of Ta<sub>2</sub>O<sub>5</sub>-Al<sub>2</sub>O<sub>3</sub> composite films were investigated in detail utilizing MIS devices. The results demonstrate that the formation of Ta<sub>2</sub>O<sub>5</sub>-Al<sub>2</sub>O<sub>3</sub> composite films by mixing Al<sub>2</sub>O<sub>3</sub> into Ta<sub>2</sub>O<sub>5</sub> can decrease the leakage current effectively, but it leads to the decrease of the dielectric constant and the enhancement of the hysteresis effect. The interfaces in composite films are not conducive to prevent the leakage current. The deposition sequence of Si/(Al<sub>2</sub>O<sub>3</sub>/Ta<sub>2</sub>O<sub>5</sub>)<sub>*n*</sub>, Al<sub>2</sub>O<sub>3</sub> as the first covering layer, reduces the leakage current and the hysteresis effect effectively. Therefore, the electrical properties of Ta<sub>2</sub>O<sub>5</sub>-Al<sub>2</sub>O<sub>3</sub> composite films could be regulated by adjusting components and structures via ALD to acquire relatively great dielectric constants and acceptable leakage currents.

**Keywords:** Ta<sub>2</sub>O<sub>5</sub>, Nanolayered films, Electrical property, Atomic layer deposition, Post-annealing

## Background

With the shrinking of the sizes, the limitations of silicon oxide (SiO<sub>2</sub>) gate dielectric for ultra large-scale integration (ULSI) devices have been reached, hence developing new gate dielectrics for next generation of microelectronic devices has become an urgent task in semiconductor industry [1]. It is required that the leakage current of new gate dielectrics has to be lower than that of the conventional SiO<sub>2</sub> under the same equivalent oxide thickness. Therefore, various high-*k* dielectric materials have been recommended to replace SiO<sub>2</sub> [2, 3].

Recently, alternative metal oxide films have been extensively investigated such as Ta<sub>2</sub>O<sub>5</sub>, Al<sub>2</sub>O<sub>3</sub>, ZrO<sub>2</sub>, HfO<sub>2</sub>, Nb<sub>2</sub>O<sub>5</sub>, and TiO<sub>2</sub>. Among them, tantalum pentoxide

(Ta<sub>2</sub>O<sub>5</sub>) has been considered as one of the most promising candidates to replace SiO<sub>2</sub> due to its relatively high dielectric constant of about 20~60 [4–8]. However, Ta<sub>2</sub>O<sub>5</sub> has noticeable high-field conductivity and cannot prevent carriers leakage due to its small band gap of 4.4 eV, which means this metal oxide cannot be independently used as a dielectric film. Hence, it is necessary to introduce an excellent insulating material to block leakage current [9]. Al<sub>2</sub>O<sub>3</sub> is one of the most investigated materials with large band gap (8.7 eV) and high breakdown electric field [10–13]. To optimize the electrical property of Ta<sub>2</sub>O<sub>5</sub> as gate dielectric, ultrathin Al<sub>2</sub>O<sub>3</sub> can be mixed into Ta<sub>2</sub>O<sub>5</sub> thin films for its current-blocking capability [14–16]. This composite structure is believed to provide a high dielectric constant and an acceptable leakage current by controlling the composition and structure [17–23].

\* Correspondence: [junqingliu@nankai.edu.cn](mailto:junqingliu@nankai.edu.cn); [jmsun@nankai.edu.cn](mailto:jmsun@nankai.edu.cn)  
Research Center for Photonics and Electronics Materials, School of Materials Science and Engineering and National Institute for Advanced Materials, Nankai University, Tongyan Road 38, Tianjin 300350, China

As for film deposition methods, atomic layer deposition (ALD) based on saturated self-limiting surface reactions has become an important film deposition technique in the semiconductor industry. It exhibits many advantages over other deposition routes, such as precise thickness control at atomic layer level, high uniformity over a large area, excellent conformity in many complex nanostructures, and controllable film structure and composition [24–28]. Min-Kyu et al. [29] reported the film deposition of  $\text{Ta}_2\text{O}_5$  via thermal and ozone ( $\text{O}_3$ ) ALD using pentaethoxytantalum as Ta precursor. Hyuncho et al. [30] reported the growth of the  $\text{ZrO}_2/\text{Ta}_2\text{O}_5$  multi-laminate films by ALD and the relation between their dielectric and chemical properties. Partida-Manzanera et al. [4] reported  $(\text{Ta}_2\text{O}_5)_x(\text{Al}_2\text{O}_3)_{1-x}$  thin films deposited by ALD using pentakis(dimethylamino)tantalum as Ta precursor and DI water as oxidizer, and the effects of tantalum doping and annealing on dielectric performance. Nevertheless, the effect of composition, interface, and the deposition sequence in composite thin films on electrical properties of  $\text{Ta}_2\text{O}_5\text{-Al}_2\text{O}_3$  film deposited by ALD still need to be further illustrated.

In this work, we deposited nanolayered  $\text{Ta}_2\text{O}_5\text{-Al}_2\text{O}_3$  composite thin films on *n*-type silicon wafers by ALD technology using pentakis(dimethylamino)tantalum (PDMATa) and trimethylaluminum (TAM) as metal precursors, as well as  $\text{O}_3$  as an oxidizer. Moreover, post-annealing treatments were carried out to eliminate defects and improve film quality [31]. The electrical properties of films were studied utilizing the MIS device with  $\text{Ta}_2\text{O}_5\text{-Al}_2\text{O}_3$  as dielectric layer [32]. The effects of film composition, interface, and the deposition sequence on electrical properties of film were investigated in detail by capacitance-voltage and current-voltage measurement.

## Methods

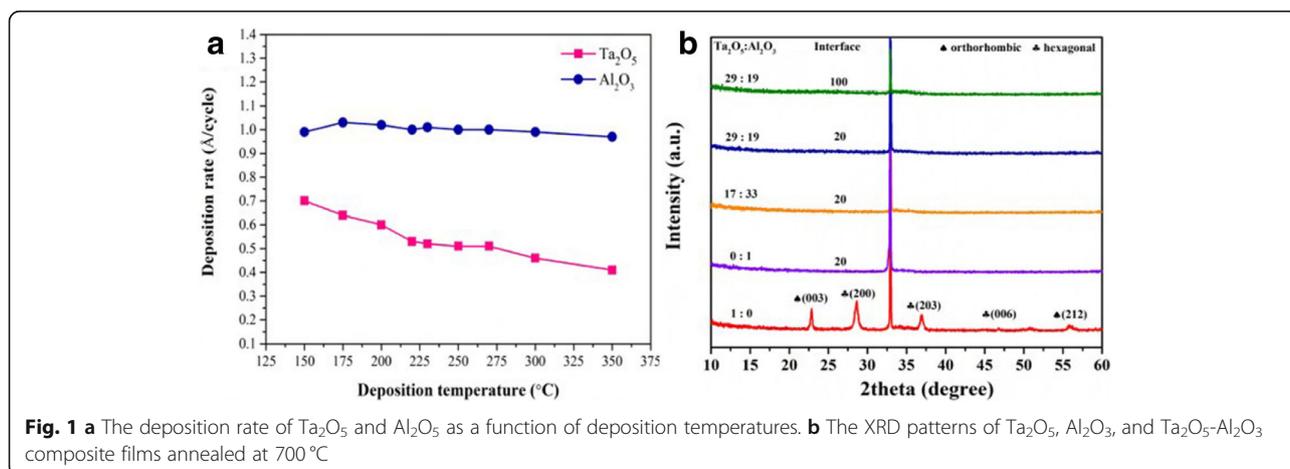
Nanolayered  $\text{Ta}_2\text{O}_5\text{-Al}_2\text{O}_3$  composite films were grown onto oriented *n*-type silicon wafers using an ALD reactor (MNT Ltd.). Trimethylaluminum was held at room

temperature and pentakis(dimethylamino)tantalum was heated to 80 °C. Ozone as an oxidant was generated from oxygen (99.999% purity) by an ozone generator (Newland Ltd.). High purity nitrogen gas (99.999%) was used as the carrying and purging gas. Moreover, the temperature of the reactor chamber and the delivery lines was remained at 230 °C and 120 °C, respectively. All the samples were annealed at 700 °C for 2 h under nitrogen ambient. The Al electrodes on both sides of the samples were deposited by physical vapor deposition. The samples were annealed at 250 °C for 0.5 h to assure reliable ohmic contacts. The samples with varying ratios and varying interface number were prepared by controlling the ALD cycles or sub-layer thickness of  $\text{Ta}_2\text{O}_5$  and  $\text{Al}_2\text{O}_3$ .

The thicknesses and refractive indexes of all samples were measured by an ellipsometer. The crystal structure of the  $\text{Ta}_2\text{O}_5\text{-Al}_2\text{O}_3$  films was characterized by glancing angle X-ray diffraction (GAXRD) with Cu  $K\alpha$  radiation. Current-voltage (*I-V*) measurements were carried out by a Keithley 2410 1100 V source measurement unit (Keithley Instruments Inc.) and capacitance-voltage (*C-V*) measurements were carried out by TH2828S LCR meter (Tonghui Electronics). All the measurements were completed at room temperature.

## Results and Discussion

Figure 1a shows the change of deposition rate as a function of deposition temperature. There is an overlap for ALD temperature windows of  $\text{Ta}_2\text{O}_5$  and  $\text{Al}_2\text{O}_3$ . Therefore,  $\text{Ta}_2\text{O}_5\text{-Al}_2\text{O}_3$  composite films can be deposited within the temperature range of 220–270 °C, in which it is controllable to grow uniform and high-quality dielectric films by ALD manner. Moreover, the deposition rates of  $\text{Ta}_2\text{O}_5$  and  $\text{Al}_2\text{O}_3$  are constant 0.52 Å/cycle and 1.01 Å cycle in ALD temperature windows, respectively. The deposition rates can be used to design the thickness and component contents of the composite film. Annealing treatment is regarded as a necessary process to eliminate



**Fig. 1** **a** The deposition rate of  $\text{Ta}_2\text{O}_5$  and  $\text{Al}_2\text{O}_3$  as a function of deposition temperatures. **b** The XRD patterns of  $\text{Ta}_2\text{O}_5$ ,  $\text{Al}_2\text{O}_3$ , and  $\text{Ta}_2\text{O}_5\text{-Al}_2\text{O}_3$  composite films annealed at 700 °C

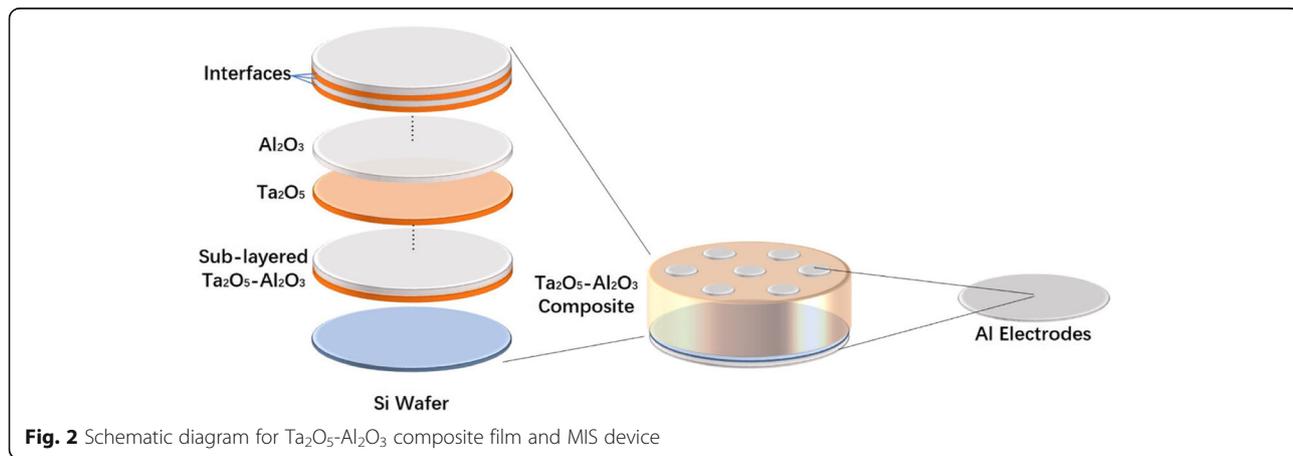
**Table 1** The experimental design for studying the effects of composition, interface, and deposition sequence on electrical properties

	ALD cycles			Composition (Ta <sub>2</sub> O <sub>5</sub> :Al <sub>2</sub> O <sub>3</sub> )	Interfaces (in film)	Deposition sequence (first layer)
	Ta <sub>2</sub> O <sub>5</sub>	Al <sub>2</sub> O <sub>3</sub>	Major cycle			
I	86	0	10	1:0	0	Ta <sub>2</sub> O <sub>5</sub>
	72	12	10	38:12	20	Ta <sub>2</sub> O <sub>5</sub>
	55	19	10	29:19	20	Ta <sub>2</sub> O <sub>5</sub>
	50	23	10	27:23	20	Ta <sub>2</sub> O <sub>5</sub>
	44	26	10	23:26	20	Ta <sub>2</sub> O <sub>5</sub>
	32	33	10	17:33	20	Ta <sub>2</sub> O <sub>5</sub>
	0	54	10	0:1	0	Al <sub>2</sub> O <sub>3</sub>
II	11	4	50	29:19	100	Al <sub>2</sub> O <sub>3</sub>
	23	8	25	29:19	50	Al <sub>2</sub> O <sub>3</sub>
	43	15	13	29:19	26	Al <sub>2</sub> O <sub>3</sub>
	55	19	10	29:19	20	Al <sub>2</sub> O <sub>3</sub>
	63	22	9	29:19	18	Al <sub>2</sub> O <sub>3</sub>
	80	28	7	29:19	14	Al <sub>2</sub> O <sub>3</sub>
III	72	12	10	38:12	20	Al <sub>2</sub> O <sub>3</sub>
	72	12	10	38:12	20	Ta <sub>2</sub> O <sub>5</sub>

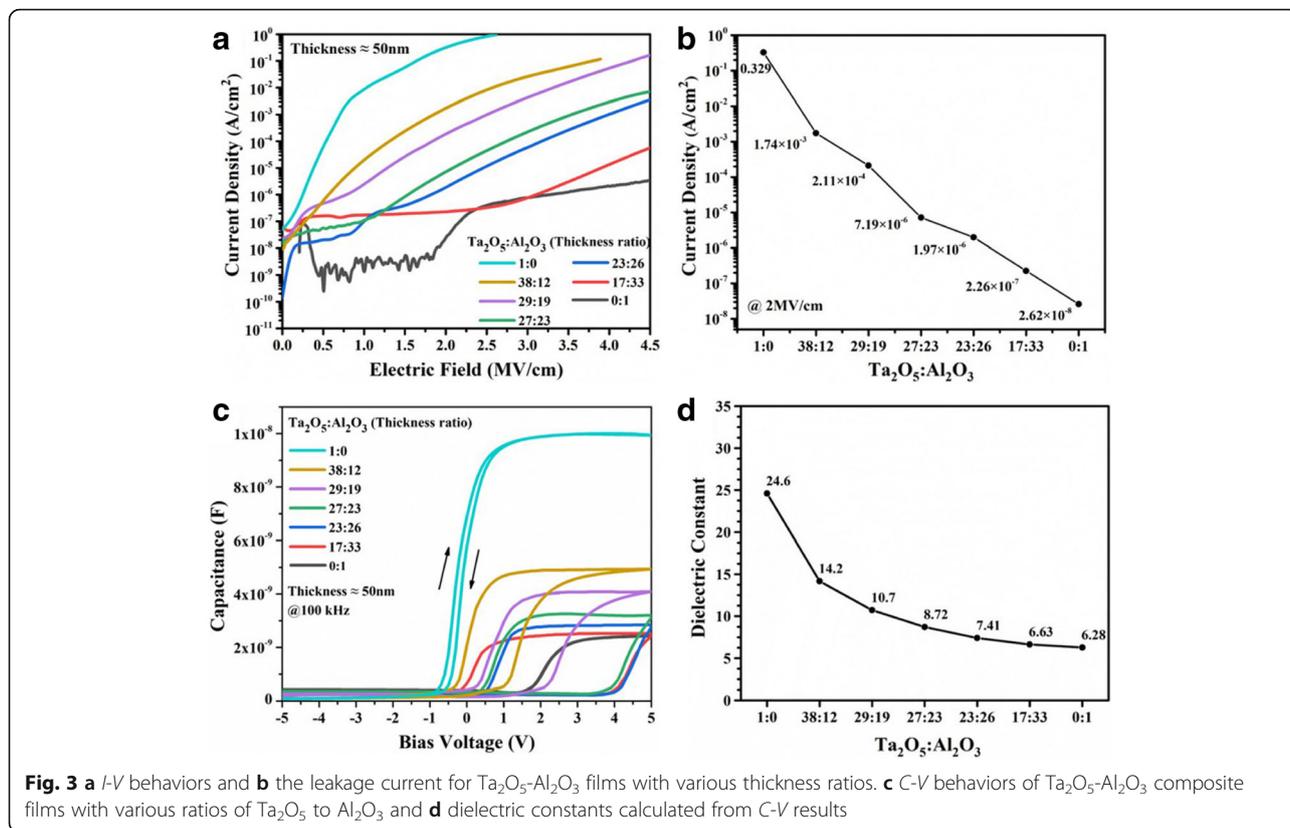
defects and improve film quality [33]. Figure 1b shows the GAXRD patterns of Ta<sub>2</sub>O<sub>5</sub>, Al<sub>2</sub>O<sub>3</sub>, and Ta<sub>2</sub>O<sub>5</sub>-Al<sub>2</sub>O<sub>3</sub> films annealed at 700 °C. Pure Al<sub>2</sub>O<sub>3</sub> film remained amorphous state after 700 °C annealing. In the pattern of Ta<sub>2</sub>O<sub>5</sub>, the strong peaks at 22.8° and 56.8° are indexed to the orthorhombic Ta<sub>2</sub>O<sub>5</sub> (PDF Card 25-0922), and the peaks at 28.5°, 36.9°, and 46.8° are indexed to the hexagonal Ta<sub>2</sub>O<sub>5</sub> (PDF Card 18-1304). However, no diffraction peak was detected in the pattern of Ta<sub>2</sub>O<sub>5</sub>-Al<sub>2</sub>O<sub>3</sub> composite films with various composition and interfaces. One possible explanation is that crystallization is inhibited in the ultrathin Ta<sub>2</sub>O<sub>5</sub> sub-layers. The other is that amorphous Al<sub>2</sub>O<sub>3</sub> mixed in the composite film increases the crystallization temperature of Ta<sub>2</sub>O<sub>5</sub> film.

Three series of experiments, as shown in Table 1, were carried out to investigate the effects of component ratio, the number of interface, and deposition sequence on electrical properties. The nanolayered Ta<sub>2</sub>O<sub>5</sub>-Al<sub>2</sub>O<sub>3</sub> composite films have a periodic structure consisted of several sub-layered Ta<sub>2</sub>O<sub>5</sub>-Al<sub>2</sub>O<sub>3</sub>. The electrical properties of composite films were studied utilizing the metal-insulator-semiconductor (MIS) devices, as shown in Fig. 2.

To study the effect of the component ratio in composite films on the electrical properties, in experiment I, the thickness ratios of Ta<sub>2</sub>O<sub>5</sub> to Al<sub>2</sub>O<sub>3</sub> in films varied from 1:0 to 0:1. Figure 3a shows that the curves of current density versus electric field intensity. For pure Al<sub>2</sub>O<sub>3</sub> film, it is difficult to inject current due to its strong insulativity. For pure Ta<sub>2</sub>O<sub>5</sub>, it shows obvious leakage current and low breakdown field strength. In Fig. 3b, the current density of pure Ta<sub>2</sub>O<sub>5</sub> (Ta<sub>2</sub>O<sub>5</sub>:Al<sub>2</sub>O<sub>3</sub> = 1:0) film at 2 MV/cm is 0.329 A/cm<sup>2</sup> due to high-field conductivity and abundant grain boundary as the leakage paths [34]. Then, the current density decreases correspondingly with decreasing the thickness ratios of Ta<sub>2</sub>O<sub>5</sub> to Al<sub>2</sub>O<sub>3</sub> from 1:0 to 0:1, and it finally declines down to 2.62 × 10<sup>-8</sup> A/cm<sup>2</sup>. The results demonstrate that the mixing Al<sub>2</sub>O<sub>3</sub> into Ta<sub>2</sub>O<sub>5</sub> thin film can decrease the leakage current effectively. One reason is Al<sub>2</sub>O<sub>3</sub> with wide band gap has strong insulativity and can act as a barrier layer to prevent leakage current. The other is that the amorphous phase of composite film blocks leakage current path. To calculate the dielectric constants of Ta<sub>2</sub>O<sub>5</sub>-Al<sub>2</sub>O<sub>3</sub> composite films, the C-V measurement was carried out at 100 kHz at a ramp rate of 100 mV/s, as shown in Fig. 3c. A low capacitance state is a depletion region in the negative voltage range and a high capacitance state is an accumulation region in the positive voltage range for MIS capacitors. The capacitances decrease with reducing the thickness ratio of Ta<sub>2</sub>O<sub>5</sub> to Al<sub>2</sub>O<sub>3</sub>. Moreover, the C-V data of Ta<sub>2</sub>O<sub>5</sub>-Al<sub>2</sub>O<sub>3</sub> composite films display significant flat band shifts to more positive voltages and additionally significant hysteresis with

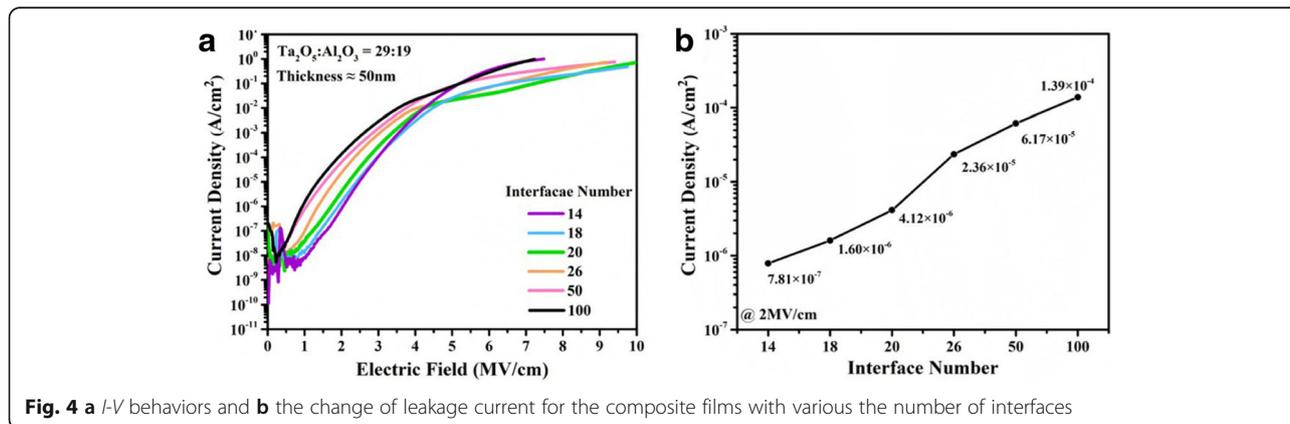


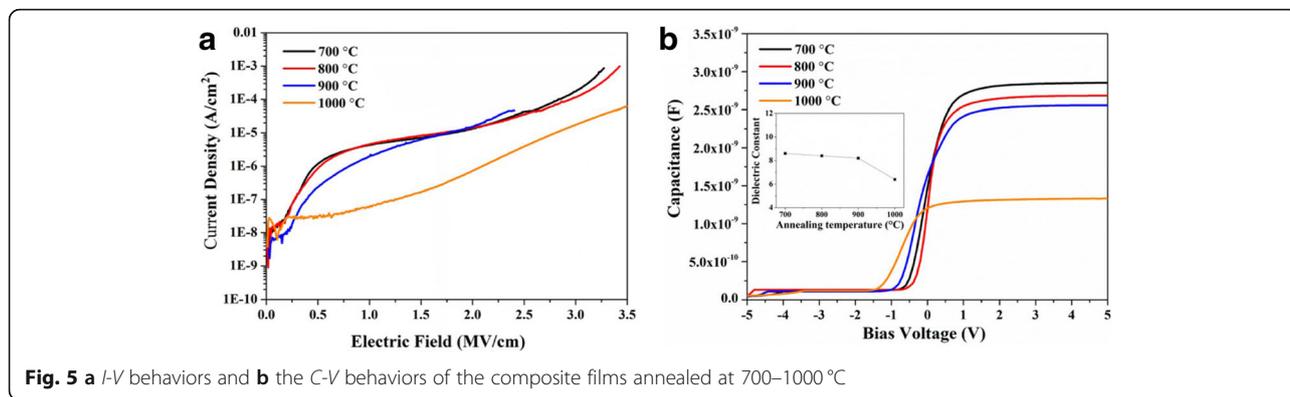
**Fig. 2** Schematic diagram for Ta<sub>2</sub>O<sub>5</sub>-Al<sub>2</sub>O<sub>3</sub> composite film and MIS device



increasing  $\text{Al}_2\text{O}_3$  content ratios. The positive shifts of flat band voltage can be attributed to the negative charges from trapping of electrons as well as fixed charges at the interface or in the film. Hysteresis effect in *C-V* measurements is normally attributed to charge trapping in the oxide or at the interface, mobile charge, and remnant polarization [35]. In Fig. 3d, the dielectric constant of pure  $\text{Ta}_2\text{O}_5$  ( $\text{Ta}_2\text{O}_5:\text{Al}_2\text{O}_3 = 0:1$ ) and pure  $\text{Al}_2\text{O}_3$  film was calculated at 24.6 and 6.28, respectively. For  $\text{Ta}_2\text{O}_5\text{-Al}_2\text{O}_3$  composite films, as is expected, the dielectric constants decrease continuously with the increase of  $\text{Al}_2\text{O}_3$  content correspondingly.

To explore the effect of interface in composite films on the electrical properties, in experiment II, the number of the interfaces varied from 14 to 100. Figure 4a shows the leakage current behaviors of  $\text{Ta}_2\text{O}_5\text{-Al}_2\text{O}_3$  composite films with various number of interfaces. It can be found that the interface has smaller effects on leakage current compared to the film component. In Fig. 4b, the current density of  $\text{Ta}_2\text{O}_5\text{-Al}_2\text{O}_3$  composite films is  $7.81 \times 10^{-7} \text{ A/cm}^2$  when the number of interfaces is 14, and then it increases continuously with increasing the number of interfaces from 14 to 100 at the electric field of 2 MV/cm. These results demonstrate





that interfaces in Ta<sub>2</sub>O<sub>5</sub>-Al<sub>2</sub>O<sub>3</sub> films are not conducive to prevent the leakage current. These defects trend to generate at interfaces due to the different ionic radius and valence states for Ta<sup>5+</sup> and Al<sup>3+</sup>. Moreover, more interfaces mean thinner Ta<sub>2</sub>O<sub>5</sub>-Al<sub>2</sub>O<sub>3</sub> sub-layers in fixed-thickness film. The interface defect density will increase with the reduction of film thickness [36], which may cause an increase of leakage current. In addition, the effect of SiO<sub>2</sub> interface on the electrical properties of the nanolayered film is relatively minor after 700 °C annealing under N<sub>2</sub> ambient. Before ALD processes, the native oxide has been removed by an HF last cleaning step immediately before the deposition. The HF step gives rise to a hydrogen-passivated surface, which becomes the initial state for the ALD process. After the film deposition, the samples were annealed at 700 °C under N<sub>2</sub> ambient. The inert gas can prevent the oxidation of Si and the further growth of SiO<sub>2</sub> interface. Moreover, the Al<sub>2</sub>O<sub>3</sub> films are not permeable for oxygen diffusion [37]. Al<sub>2</sub>O<sub>3</sub> as a barrier layer in nanolayered Ta<sub>2</sub>O<sub>5</sub>-Al<sub>2</sub>O<sub>3</sub> film can suppress oxygen diffusion toward the interface between Si and nanolayered film. Therefore, the effect of the SiO<sub>2</sub> interface on the electrical properties of the nanolayered film is limited below 900 °C annealing. However, the SiO<sub>2</sub> interface has an effect on the electrical properties of nanolayered

Ta<sub>2</sub>O<sub>5</sub>-Al<sub>2</sub>O<sub>3</sub> film when the annealing temperature is above 1000 °C. As shown in Fig. 5, the reduction of leakage current and dielectric constant can be attributed to the growth of the SiO<sub>2</sub> interface during the annealing processes. The effect of the deposition sequence on the electrical properties was compared in experiment III. The deposition sequence of composite films on silicon was first Ta<sub>2</sub>O<sub>5</sub> and then Al<sub>2</sub>O<sub>3</sub>, which was defined as Si/(Ta<sub>2</sub>O<sub>5</sub>/Al<sub>2</sub>O<sub>3</sub>)<sub>n</sub>. Otherwise, it was defined as Si/(Al<sub>2</sub>O<sub>3</sub>/Ta<sub>2</sub>O<sub>5</sub>)<sub>n</sub>. Figure 6a, b depicts the leakage current behaviors and the curves of C-V. The current density of Si/(Ta<sub>2</sub>O<sub>5</sub>/Al<sub>2</sub>O<sub>3</sub>) film is higher than that of Si/(Al<sub>2</sub>O<sub>3</sub>/Ta<sub>2</sub>O<sub>5</sub>) film at the electric field of 4 MV/cm, and the breakdown field of Si/(Ta<sub>2</sub>O<sub>5</sub>/Al<sub>2</sub>O<sub>3</sub>) film is obviously weaker than that of Si/(Al<sub>2</sub>O<sub>3</sub>/Ta<sub>2</sub>O<sub>5</sub>) film. In addition, the hysteresis of the C-V curve for Si/(Ta<sub>2</sub>O<sub>5</sub>/Al<sub>2</sub>O<sub>3</sub>) film is obviously greater. It is reported that Al<sub>2</sub>O<sub>3</sub> thin film has a low interface trap density [38, 39] and can improve interfacial properties [22]. It can be seen that there are lesser defects at the Si/Al<sub>2</sub>O<sub>3</sub> interface compared to the Si/Ta<sub>2</sub>O<sub>5</sub> interface. Moreover, the Al<sub>2</sub>O<sub>3</sub> films are not permeable for oxygen diffusion. It can act as a barrier layer to cover Si in order to prevent the diffusion of oxygen in film toward Si/Al<sub>2</sub>O<sub>3</sub> interface.

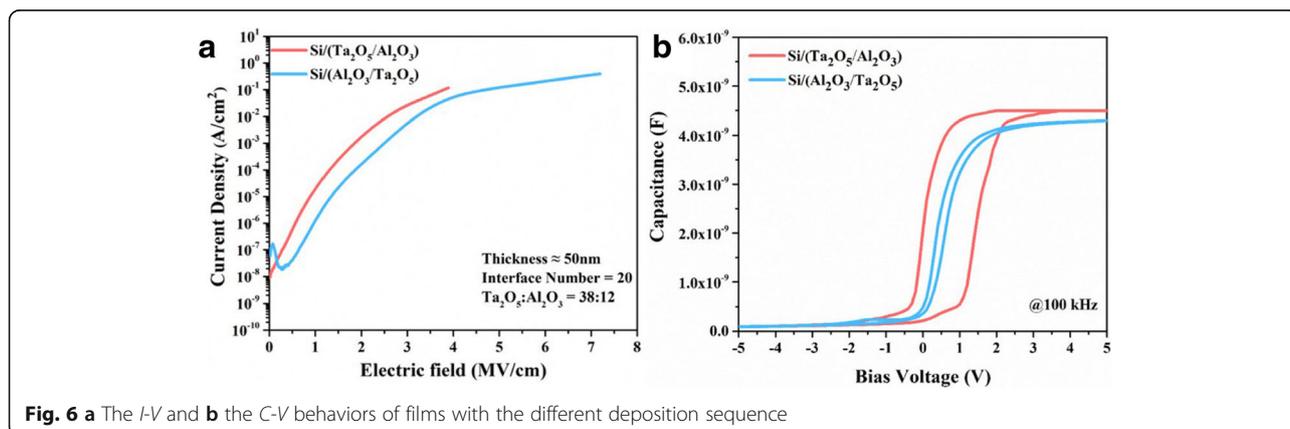


Fig. 6 a The I-V and b the C-V behaviors of films with the different deposition sequence

The above results illustrate that film composition, structure, and interface state density act as the key factors to affect the electrical properties. A compromise property was obtained by mixing  $\text{Al}_2\text{O}_3$  into  $\text{Ta}_2\text{O}_5$  film. The increase of film crystallinity can not only increase the dielectric constant, but also increase the leakage current due to abundant grain boundary as a leakage path. Moreover, high interface state density should be avoided for the laminated or doped film on account of the negative influence on leakage current. Therefore, the amorphous dielectric film with high dielectric constant, relatively large band gap energy, and low interface state density may be a promising gate dielectric to replace  $\text{SiO}_2$ . In addition, deposition technology also as a key factor has an important effect on electrical properties of gate dielectric.

### Conclusions

Nanolayered  $\text{Ta}_2\text{O}_5$ - $\text{Al}_2\text{O}_3$  composite films were grown on *n*-type silicon by ALD. The overlapped temperature window for  $\text{Ta}_2\text{O}_5$  and  $\text{Al}_2\text{O}_3$  is 220~270 °C using pentakis(dimethylamino)tantalum as the Ta precursor and  $\text{O}_3$  as the oxidant. Nanolayered  $\text{Ta}_2\text{O}_5$ - $\text{Al}_2\text{O}_3$  composite films remain amorphous after annealing treatment at 700 °C. The formation of  $\text{Ta}_2\text{O}_5$ - $\text{Al}_2\text{O}_3$  composite films by introducing  $\text{Al}_2\text{O}_3$  into  $\text{Ta}_2\text{O}_5$  can decrease the leakage current effectively due to the excellent insulator for amorphous  $\text{Al}_2\text{O}_3$ , but lead to the decrease of the dielectric constant. Moreover, the interfaces in composite films are not conducive to prevent the leakage current. In addition, the deposition sequence of  $\text{Si}/(\text{Al}_2\text{O}_3/\text{Ta}_2\text{O}_5)_n$ ,  $\text{Al}_2\text{O}_3$  as the first covering layer, reduces effectively the leakage current and the hysteresis effect due to its thermostability and barrier effect. Therefore, the electrical properties of  $\text{Ta}_2\text{O}_5$ - $\text{Al}_2\text{O}_3$  composite films could be regulated by adjusting components and structures via ALD to acquire relatively great dielectric constants and acceptable leakage currents.

### Abbreviations

ALD: Atomic layer deposition; C-V: Capacitance-voltage; GAXRD: Glancing angle X-ray diffraction; I-V: Current-voltage;  $\text{O}_3$ : Ozone; PDMA: Pentakis(dimethylamino)tantalum; TAM: Trimethylaluminum; ULS: Ultra large-scale integration

### Acknowledgements

This work is supported by the National Natural Science Foundation of China (no. 61674085).

### Funding

National Natural Science Foundation of China (no. 61674085).

### Availability of Data and Materials

All data are fully available without restriction.

### Authors' Contributions

JL carried out the experiments and measurements. JW was involved in the measurements. JL designed the study and drafted the manuscript. JS supervised the overall study. All authors read and approved the final manuscript.

### Competing Interests

The authors declare that they have no competing interests.

### Publisher's Note

Springer Nature remains neutral with regard to jurisdictional claims in published maps and institutional affiliations.

Received: 31 December 2018 Accepted: 18 February 2019

Published online: 04 March 2019

### References

- Chen W, Ren W, Zhang Y, Liu M, Ye ZG (2015) Preparation and properties of  $\text{ZrO}_2$  and  $\text{TiO}_2$  films and their nanolaminates by atomic layer deposition. *Ceram Int* 41:5278–5282
- Young CD, Heh D, Nadkarni SV, Choi R, Peterson JJ, Barnett J, Lee BH, Bersuker G (2006) Electron trap generation in high-*k* gate stacks by constant voltage stress. *IEEE T Reliab* 6:123–131
- Gusev EP, Narayanan V, Frank MM (2006) Advanced high-*k* dielectric stacks with polySi and metal gates: recent progress and current challenges. *IBM J Res Dev* 50:387–410
- Partida-Manzanera T, Roberts JW, Bhat TN, Zhang Z, Tan HR, Dolman SB, Sedghi N, Tripathy S, Potter RJ (2016) Comparative analysis of the effects of tantalum doping and annealing on atomic layer deposited  $(\text{Ta}_2\text{O}_5)_x(\text{Al}_2\text{O}_3)_{1-x}$  as potential gate dielectrics for GaN/ $\text{Al}_x\text{Ga}_{1-x}\text{N}$ /GaN high electron mobility transistors. *J Appl Phys* 119:1059–1052
- Kolkovsky V, Kurth E, Kunath C (2016) Enhanced dielectric properties of thin  $\text{Ta}_2\text{O}_5$  films grown on 65 nm  $\text{SiO}_2/\text{Si}$ . *Phys Status Solidi* 13:786–789
- Cheng S, Sang L, Liao M, Liu J, Imura M, Li H, Koide Y (2012) Integration of high-dielectric constant  $\text{Ta}_2\text{O}_5$  oxides on diamond for power devices. *Appl Phys Lett* 101:331–359
- Zhang L, Li J, Zhang XW, Jiang XY, Zhang ZL (2010) High-performance ZnO thin film transistors with sputtering  $\text{SiO}_2/\text{Ta}_2\text{O}_5/\text{SiO}_2$  multilayer gate dielectric. *Thin Solid Films* 518:6130–6133
- Kolkovsky V, Lukat K, Kurth E, Kunath C (2015) Reactively sputtered hafnium oxide on silicon dioxide: structural and electrical properties. *Solid State Electron* 106:63–67
- Atanassova E, Georgieva M (2010) High-*k*  $\text{HfO}_2$ - $\text{Ta}_2\text{O}_5$  mixed layers: electrical characteristics and mechanisms of conductivity. *Microelectron Eng* 87:668–676
- Wei D, Edgar JH, Briggs DP, Retterer ST (2014) Atomic layer deposition  $\text{TiO}_2$ - $\text{Al}_2\text{O}_3$  stack: an improved gate dielectric on Ga-polar GaN metal oxide semiconductor capacitors. *J Vac Sci Technol*, B 32:060602–060604
- Chang S, Song YW, Lee S, Sang YL, Ju BK (2008) Efficient suppression of charge trapping in ZnO-based transparent thin film transistors with novel  $\text{Al}_2\text{O}_3/\text{HfO}_2/\text{Al}_2\text{O}_3$  structure. *Appl Phys Lett* 92:113505
- Kukli K, Ritala M, Leskelä M (2001) Development of dielectric properties of niobium oxide, tantalum oxide, and aluminum oxide based nanolayered materials. *J Electrochem Soc* 148:156–162
- Chun BS, Wu HC, Abid M, Chu IC, Serrano-Guisan S, Shvets IV, Choi DS (2010) The effect of deposition power on the electrical properties of Al-doped zinc oxide thin films. *Appl Phys Lett* 97:1245
- Nakamura R, Toda T, Tsukui S, Tane M, Ishimaru M, Suzuki T, Nakajima H (2014) Diffusion of oxygen in amorphous  $\text{Al}_2\text{O}_3$ ,  $\text{Ta}_2\text{O}_5$ , and  $\text{Nb}_2\text{O}_5$ . *J Appl Phys* 116:222904
- Atanassova E, Spassov D, Novkovski N, Paskaleva A (2012) Constant current stress of lightly Al-doped  $\text{Ta}_2\text{O}_5$ . *Mater Sci Semicond Process* 15:98–107
- Spassov D, Atanassova E, Paskaleva A (2011) Lightly Al-doped  $\text{Ta}_2\text{O}_5$ : electrical properties and mechanisms of conductivity. *Microelectron Reliab* 51:2102–2109
- HongHwa LU (2011) Effects of the Ta content on the microstructure and electrical property of reactively sputtered  $\text{Ta}_x\text{Zr}_{1-x}\text{N}$  thin films. *Thin Solid Films* 519:4987–4991
- Zhang H, Solanki R, Roberds B, Bai G, Banerjee I (2000) High permittivity thin film nanolaminates. *J Appl Phys* 87:1921–1924
- Nam M, Kim A, Kang K, Choi E, Kwon SH, Lee SJ, Pyo SG (2016) Characterization of atomic layer deposited  $\text{Al}_2\text{O}_3/\text{HfO}_2$  and  $\text{Ta}_2\text{O}_5/\text{Al}_2\text{O}_3$  combination stacks. *Sci Adv Mater* 8:1958–1962
- Kukli K, Kemell M, Vehkamäki M, Heikkilä MJ, Mizohata K, Kalam K, Ritala M, Leskelä M, Kundrata I, Fröhlich K (2017) Atomic layer deposition and properties of mixed  $\text{Ta}_2\text{O}_5$  and  $\text{ZrO}_2$  films. *AIP Adv* 7:025001

21. Jögi I, Tamm A, Kukli K, Kemell M, Lu J, Sajavaara T, Ritala M, Leskelä M (2010) Investigation of  $ZrO_2$ - $Gd_2O_3$  based high- $k$  materials as capacitor dielectrics. *J Electrochem Soc* 157:G202-10
22. Ding SJ, Zhu C, Li MF, Zhang DW (2005) Atomic-layer-deposited  $Al_2O_3$ - $HfO_2$ - $Al_2O_3$  dielectrics for metal-insulator-metal capacitor applications. *Appl Phys Lett* 87:886
23. Ding SJ, Xu J, Huang Y, Sun QQ, Zhang DW, Li MF (2008) Electrical characteristics and conduction mechanisms of metal-insulator-metal capacitors with nanolaminated  $Al_2O_3$ - $HfO_2$  dielectrics. *Appl Phys Lett* 93:79
24. Lee S, Kim H, Lee J, Yu IH, Lee JH, Hwang C (2014) Effects of  $O_3$  and  $H_2O$  as oxygen sources on the atomic layer deposition of  $HfO_2$  gate dielectrics at different deposition temperatures. *J Mater Chem C* 2:2558-2568
25. Smith SW, McAuliffe KG, Conley JF (2010) Atomic layer deposited high- $k$  nanolaminate capacitors. *Solid State Electron* 54:1076-1082
26. Sang WL, Kwon OS, Hwan Han J, Seong Hwang C (2008) Enhanced electrical properties of  $SrTiO_3$  thin films grown by atomic layer deposition at high temperature for dynamic random access memory applications. *Appl Phys Lett* 92:G127
27. Zhu MW, Gong J, Sun C, Xia JH, Jiang X (2008) Investigation of correlation between the microstructure and electrical properties of sol-gel derived ZnO based thin films. *J Appl Phys* 104:247
28. Jun JH, Choi DJ, Kim KH, Oh KY, Hwang CJ (2014) Effect of structural properties on electrical properties of lanthanum oxide thin film as a gate dielectric. *Jpn J Appl Phys* 42:3519-3522
29. Kim MK, Kim WH, Lee T, Kim H (2013) Growth characteristics and electrical properties of  $Ta_2O_5$  grown by thermal and  $O_3$ -based atomic layer deposition on TiN substrates for metal-insulator-metal capacitor applications. *Thin Solid Films* 542:71-75
30. Cho H, Park KW, Park CH, Cho HJ, Yeom SJ, Hong K, Kwak NJ, Ahn JH (2015) Abnormally enhanced dielectric constant in  $ZrO_2/Ta_2O_5$  multi-laminate structures by metallic Ta formation. *Mater Lett* 154:148-151
31. Hao T, Deng Z, Liu Z, Huang C, Huang J, Hai L, Chong W, Cao Y (2011) Effects of post-annealing on structural, optical and electrical properties of Al-doped ZnO thin films. *Appl Surf Sci* 257:4906-4911
32. Roy Chaudhuri A, Fissel A, Osten HJ (2014) Superior dielectric properties for template assisted grown (100) oriented  $Gd_2O_3$  thin films on Si(100). *Appl Phys Lett* 104:18
33. Wang X, Ishiwara H (2014) Improvement of electrical property of sol-gel-derived lead zirconate titanate thin films by multiple rapid thermal annealing. *Jpn J Appl Phys* 40:7002-7006
34. Nguyen NV, Richter CA, Yong JC, Alers GB, Stirling LA (2000) Effects of high-temperature annealing on the dielectric function of  $Ta_2O_5$  films observed by spectroscopic ellipsometry. *Appl Phys Lett* 77:3012-3014
35. Johnson RS, Hong JG, Lucovsky G (2001) Electron traps at interfaces between Si(100) and noncrystalline  $Al_2O_3$ ,  $Ta_2O_5$ , and  $(Ta_2O_5)_x(Al_2O_3)_{1-x}$  alloys. *J Vac Sci Technol B* 19:1606-1610
36. Saint-Cast P, Heo YH, Billot E, Olwal P, Hofmann M, Rentsch J, Glunz SW, Preu R (2011) Variation of the layer thickness to study the electrical property of PECVD  $Al_2O_3/c$ -Si interface. *Energy Procedia* 8:642-647
37. Lebedev M S, Ayupov B M (2008) Investigation of thin-film nanocomposite materials by monochromatic null ellipsometry. 9th International workshop and tutorials EDM'2008, session I, 30-33
38. Geng GZ, Liu GX, Shan FK, Liu A, Zhang Q, Lee WJ, Shin BC, Wu HZ (2014) Improved performance of InGaZnO thin-film transistors with  $Ta_2O_5/Al_2O_3$  stack deposited using pulsed laser deposition. *Curr Appl Phys* 14:S2-S6
39. Werner F, Cosceev A, Schmidt J (2012) Interface recombination parameters of atomic-layer-deposited  $Al_2O_3$  on crystalline silicon. *J Appl Phys* 111:073710

Submit your manuscript to a SpringerOpen<sup>®</sup> journal and benefit from:

- Convenient online submission
- Rigorous peer review
- Open access: articles freely available online
- High visibility within the field
- Retaining the copyright to your article

---

Submit your next manuscript at ► [springeropen.com](https://www.springeropen.com)

---