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Topical Report



Investigation of Performance of SCN-1 Pure Glass as Sealant Used in SOFC

W.N Liu
X. Sun
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M.A. Khaleel

March 2010



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Investigation of Performance of SCN-1 Pure Glass Used in SOFC

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March 2010

Prepared for the SECA Core Technology Program
National Energy Technology Laboratory, Pittsburgh, Pennsylvania
under Contract DE-AC05-76RL01830

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Executive Summary

As its name implies, a self-healing glass seal has the potential of restoring its mechanical properties upon being reheated to SOFC stack operating temperature, even when it has experienced some cooling induced damage/cracking at room temperature. Such a self-healing feature is desirable for achieving high seal reliability during thermal cycling. On the other hand, self-healing glass is also characterized by its low mechanical stiffness and high creep rate at the typical operating temperature of SOFCs. Therefore, from a design perspective, it is important to know the long term geometric stability and thermal mechanical behaviors of the self-healing glass under the stack operating conditions. These predictive capabilities will guide the design and optimization of a reliable sealing system that potentially utilizes self-healing glass as well as other ceramic seal components in achieving the ultimate goal of SOFC.

In this report, we focus on predicting the effects of various generic seal design parameters on the stresses in the seal. For this purpose, we take the test cell used in the leakage tests for compliant glass seals conducted at PNNL as our initial modeling geometry. The effect of the ceramic stopper on the geometry stability of the self-healing glass sealants is studied first. Then we explore the effect of various interfaces such as stopper and glass, stopper and PEN, as well as stopper and IC plate, on the geometry stability and reliability of glass during the operating and cooling processes.

Acronyms

CTE coefficient of thermal expansion

FE finite element

NETL National Energy Technology Laboratory

SECA Solid-State Energy Conversion Alliance

SEM scanning electron microscopy

SOFC solid-oxide fuel cell

Acknowledgements

Pacific Northwest National Laboratory is operated for the U.S. Department of Energy by Battelle under Contract DE-AC05-76RL01830. The work summarized in this report was funded as part of the Solid-State Energy Conversion Alliance Core Technology Program by the U.S. Department of Energy's National Energy Technology Laboratory. We would like to acknowledge the technical direction from Travis Shultz and Briggs White. Technical discussions with Drs. Matt Chou and Jeff Stevenson are also gratefully acknowledged.

Contents

Executive Summary	ii
Acronyms	iii
Acknowledgements	iv
1.0 Introduction	8
2.0 Technical Approach and Model Description.....	9
3.0 Results and Discussions on Parametric Study.....	15
4.0 Conclusions	27
5.0 References	29

Figures

Figure 1 Setup of leakage test (provided by Matt Chou).....	9
Figure 2 Geometry of PEN, ceramic stopper and seal in the test cell	9
Figure 3 Numerical model	10
Figure 4 FE mesh used in the simulation.....	11
Figure 5 Temperature dependent Young's modulus [3].....	12
Figure 6 Dimension change of specimen in CTE test [7].....	12
Figure 7 Temperature dependent CTE [6].....	13
Figure 8 Setup of viscosity measurement with TMA. (a) before viscosity measurement; (b) after viscosity measurement.....	14
Figure 9 Creep behavior of the pure glass [6].....	14
Figure 10 Schematic of model of setup without ceramic stoppers	16
Figure 11 Deformed configurations (a) after applying top pressure; (b) creep in 2s	16
Figure 12 Deformation history of the inner side if the glass	17
Figure 13 Deformed glass with distribution of equivalent creep strain after creep. (a) 36s of creep without stopper; (b) 1000s of creep with ceramic stopper	18
Figure 14 History of equivalent creep strain.....	19
Figure 15 Maximum principal stress contour of glass seal at the different time: weak interface between stopper and glass seal	20
Figure 16 Shear stress S12 contour of glass seal at the different time: weak interface between stopper and glass seal	21
Figure 17 Maximum principal stress contour of glass seal at the different: strong interface of stopper and glass seal	21
Figure 18 Shear stress S12 contour of glass seal at the different time: strong interface of stopper and glass seal	22
Figure 19 Shear stress distribution during the shutdown process with strong and weak interfaces of stopper and glass seal	23
Figure 20 Normal stress distribution during the shutdown process with strong and weak interface of stopper and glass	23

Figure 21 No sliding interface of stopper with PEN and IC: interfacial shear stress of glass (a) with PEN; (b) with IC.....	24
Figure 22 No sliding interface of stopper with PEN and IC: interfacial normal stress of glass with (a) PEN; (b) IC	25
Figure 23 Sliding interface of stopper with PEN: interfacial shear stress of glass with (a) PEN; (b) IC	25
Figure 24 Sliding interface of stopper with PEN: interfacial normal stress of glass with (a) PEN; (b) IC	26
Figure 25 Sliding interfaces of stopper with PEN and IC: interfacial shear stress of glass with (a) PEN; (b) IC.....	26
Figure 26 Sliding interfaces of stopper with PEN and IC: interfacial normal stress of glass with (a) PEN; (b) IC	27

1.0 Introduction

Seals are the most critical components in commercializing the planar SOFC technology. They must adequately prevent the leakage of air and fuel, effectively isolate the fuel from the oxidant, and insulate the cell from short circuit. Glass joining is widely accepted in SOFC as a sealing approach since it provides a cost effective and relatively simple method for bonding ceramic and metal parts. However, the softening point of the glass component typically limits the maximum operating temperature to which the joint may be exposed.

As its name implies, a self-healing glass seal has the potential of restoring its mechanical properties upon being reheated to stack operating temperature [1,2], even when it has experienced some cooling induced damage/crack at room temperature. Such a self-healing feature is desirable for achieving high seal reliability during thermal cycling. On the other hand, self-healing glass is also characterized by its low mechanical stiffness and high creep rate at the typical operating temperature of SOFCs. Therefore, from a design perspective, it is important to know the long term geometric stability and thermal mechanical behaviors of the self-healing glass under the stack operating conditions. These predictive capabilities will guide the design and optimization of a reliable sealing system that potentially utilizes self-healing glass as well as other ceramic seal components in achieving the ultimate goal of SOFC.

In this report, we focus on predicting the effects of various generic seal design parameters on the stresses in the seal. For this purpose, we take the test cell used in the leakage tests for compliant glass seals conducted at PNNL as our initial modeling geometry. The effect of the ceramic stopper on the geometry stability of the self-healing glass sealants is studied first. Then we examine the influence of interfacial strength at various interfaces, i.e., stopper/glass, stopper/PEN, as well as stopper/IC plate, on the geometry stability and reliability of glass during the operating and cooling processes.

2.0 Technical Approach and Model Description

Figure 1 and Figure 2 show the test setup and the associated seal geometry of the test cell, respectively.

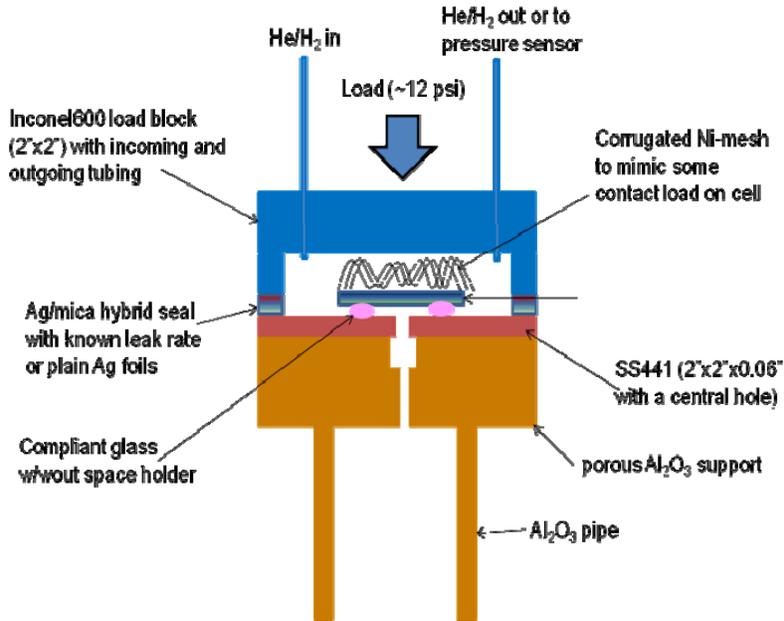


Figure 1 Setup of leakage test (provided by Matt Chou)

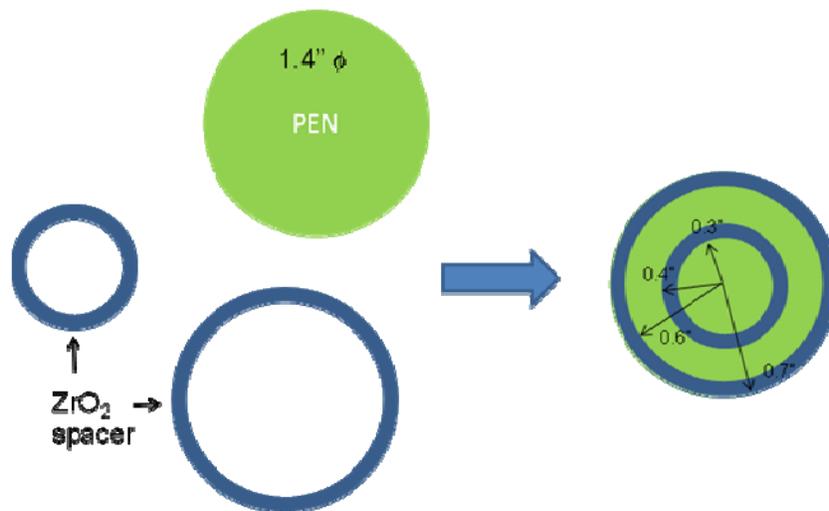


Figure 2 Geometry of PEN, ceramic stopper and seal in the test cell

To simulate the glass behavior in the test, only the core part of the setup in Figure 1 is taken into account, which includes PEN, stopper/glass, and IC plate. Figure 3 shows the computational cross-section of the test cell. Utilizing the symmetrical nature of the test cell geometry, a 2-dimensional axisymmetric model is used in this study. The test cell includes PEN (anode, electrolyte, and cathode), ceramic stoppers, glass seal, and SS 441 interconnect.

The simulation was performed using the commercial finite element (FE) software package MARC. Figure 4 shows the FE mesh used in the simulation with details around the seal, PEN and IC interface area. Two types of sealing systems are considered: 100% glass seals, and glass seals with ceramic stopper rings at both inner and outer radius of the glass seal, see Figure 3.

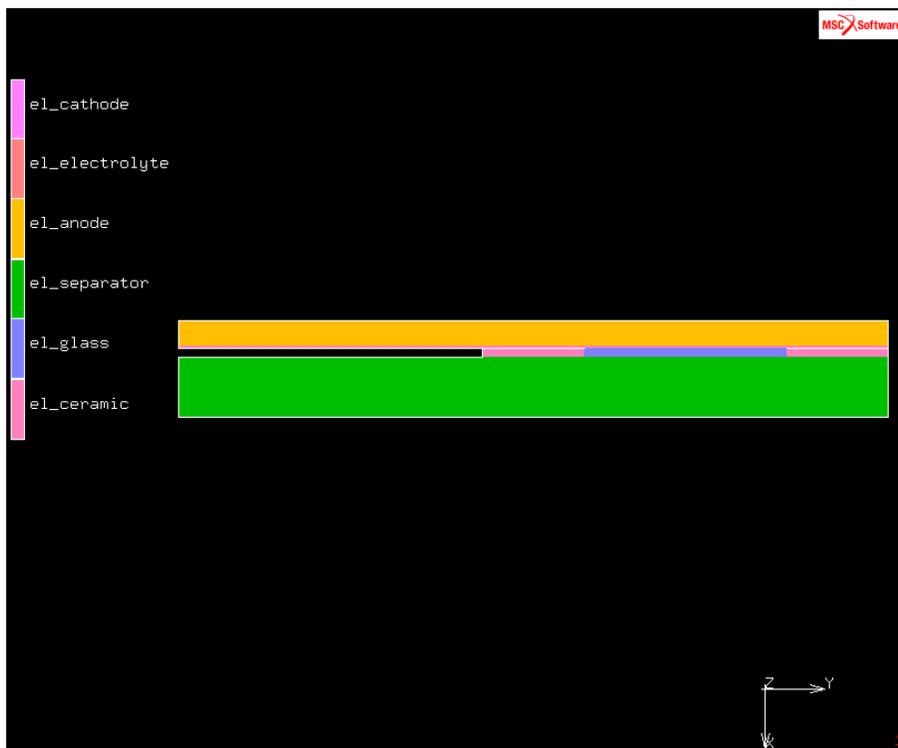


Figure 3 Numerical model

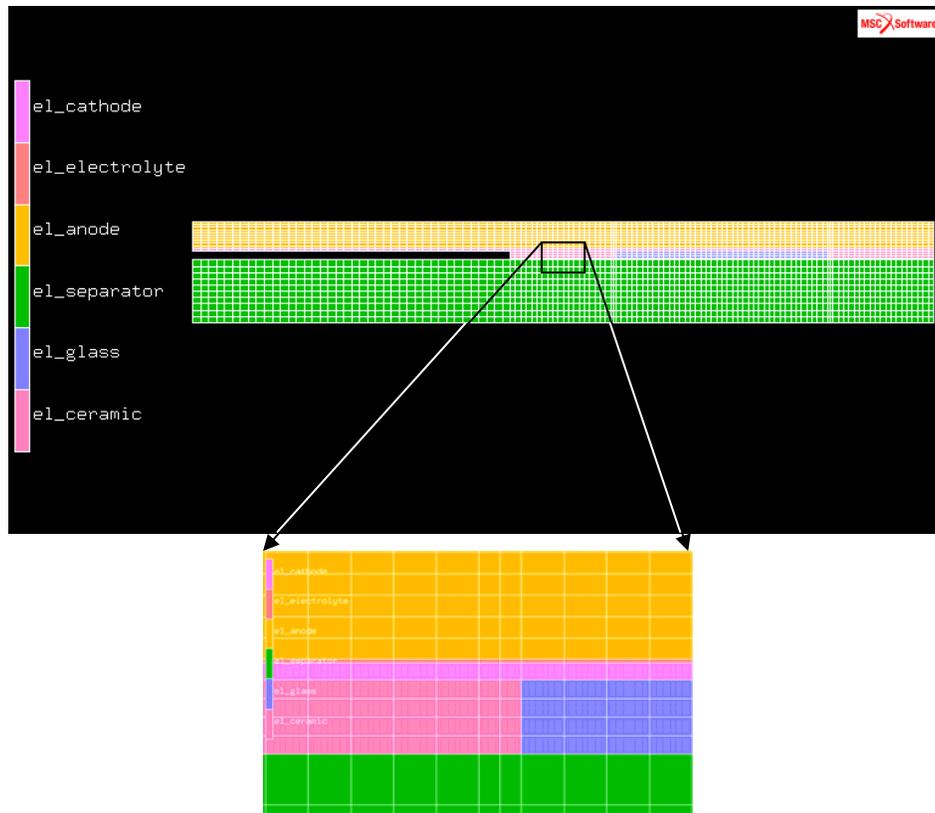


Figure 4 FE mesh used in the simulation

Currently, SCN-1 is used as the pure glass seal in the test cell modeling. Its thermal and mechanical properties are temperature dependent. Figure 5 depicts the temperature dependent Young's modulus, partially measured experimentally by ORNL [3]. It may be seen that the measurement was only conducted up to the glass transition temperature, i.e., T_g of this glass. This is because the testing method is no longer valid when the temperature exceeds T_g . In this analysis, the high temperature modulus is extrapolated in a manner consistent with typical glass behavior [4, 5].

The glass transition temperature T_g of SCN-1 is about 500°C , which is substantially lower than the typical operating temperature of SOFC ($\sim 800^\circ\text{C}$). The ORNL measured CTE for SCN-1 [7] only goes to about 600°C , see Figure 6, so temperature dependent CTE data from the literature [6] was used in the current simulation (see Figure 7) for temperatures greater than 600°C .

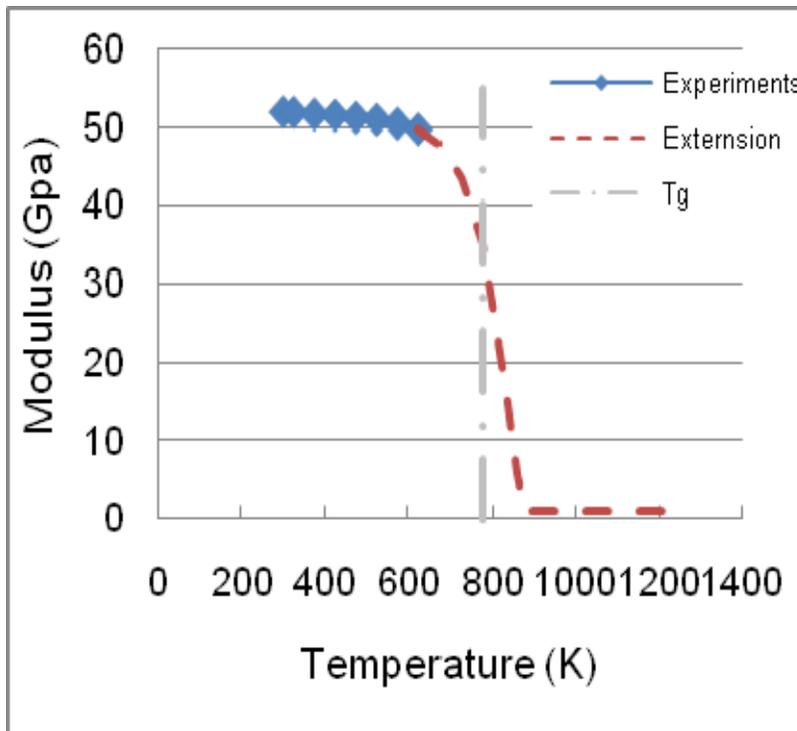


Figure 5 Temperature dependent Young's modulus [3]

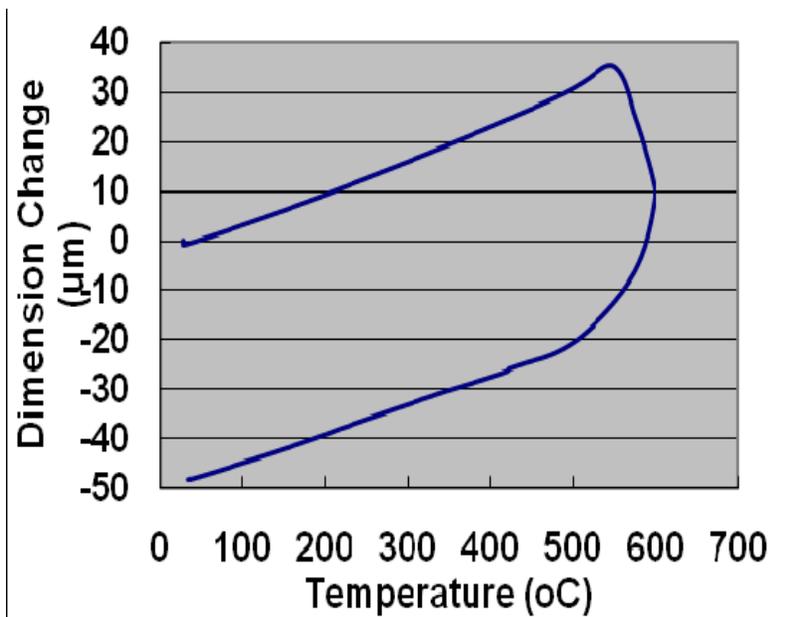


Figure 6 Dimension change of specimen in CTE test [7]

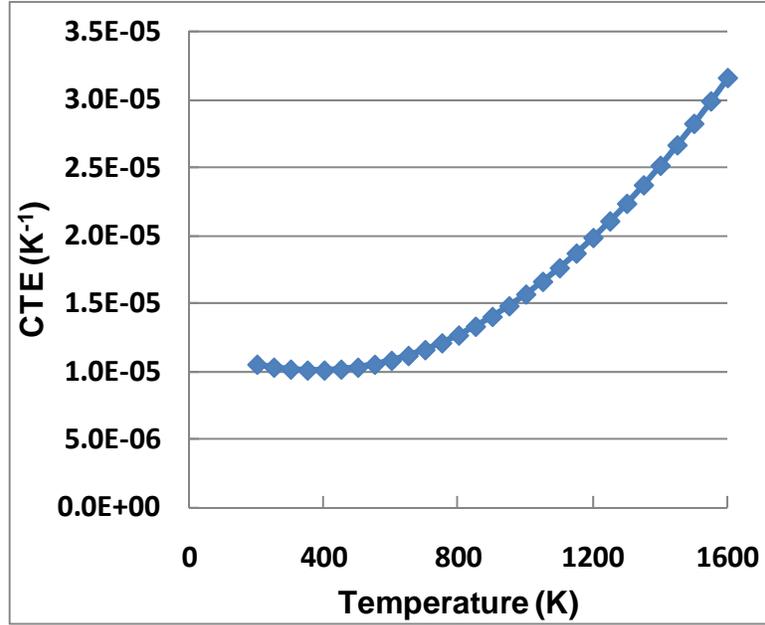


Figure 7 Temperature dependent CTE [6]

At the typical operating temperature of SOFC, creep behavior of the pure glass is unavoidable. Its creep behavior at high temperature was measured at ORNL using the setup shown in Figure 8. The measurements were performed isothermally at temperatures between 600 °C and 850 °C. Three different loads were used in the viscosity measurement. A constant heating rate of 5°C/min was applied under a constant load [8].

Viscous behavior of the pure glass can be evaluated by the linear creep law in equation (1.1) [8]:

$$\dot{\epsilon} = \frac{1}{\eta} \sigma \quad (0.1)$$

where

$$\eta = \eta_0 \exp\left(\frac{Q_v}{RT}\right) \quad (0.2)$$

R is the universal gas constant with a value of 8.314 J/K/mol. Based on the measurement results, the parameters for SNC-1 self-healing glass can be determined as $\eta_0 = 1.397 \times 10^{-9} \text{ Pa} \cdot \text{s}$,

$Q_v = 283.32 \text{ kJ/mol}$ [6]. The viscous parameter η of the self-healing glass is plotted as a function of temperature in Figure 9.

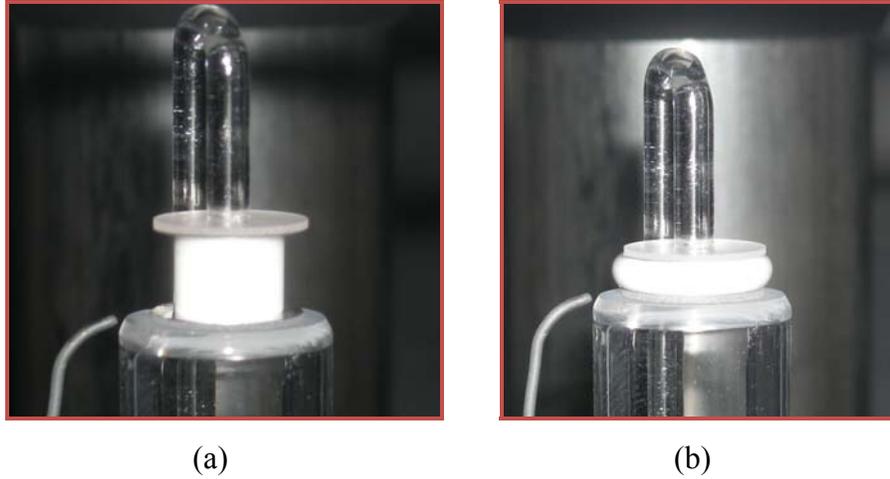


Figure 8 Setup of viscosity measurement with TMA. (a) before viscosity measurement; (b) after viscosity measurement

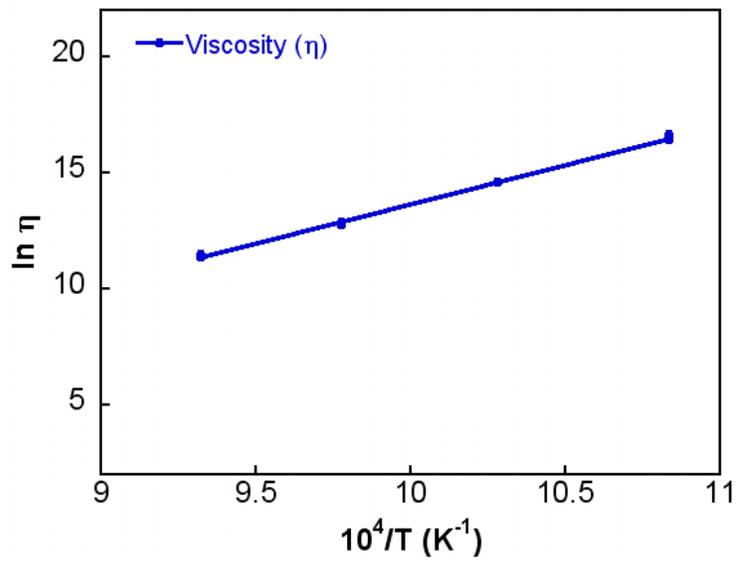


Figure 9 Creep behavior of the pure glass [6]

The initial stress-free temperature is assumed to be the stack assembly temperature of 850 °C. The operating temperature is 800°C. The test cell is given the initial stress-free temperature, then dropped to the operating temperature in a quasi-static analysis step. The CTE mismatch among various components will lead to various degrees of thermal stresses in different components due to the temperature drop. However, at the constant stack operating temperature, the glass seal will creep under the thermal stress, resulting in stress relaxation and redistribution. Upon shutdown of the test cell and cooling down to the room temperature of 25 °C, new thermal stresses will be created due to the CTE mismatch. We will present the glass seal stress and the interfacial stress distributions at different stages of the thermal profile described above to compare the effects of these different sealing systems.

3.0 Results and Discussions on Parametric Study

To explore the performance of the self-healing glass in SOFC stacks, different design and interface conditions were considered. At first, the behavior of the glass was studied with and without ceramic stoppers. Then, the effect of viscosity of the glass at high temperature on the stress in the glass was studied using creep analysis. Subsequently, the interfacial conditions in the ceramic stopper/glass, stopper/PEN, as well stopper/IC plate interfaces were investigated at two extreme conditions: fully bonded and sliding. For the stopper and glass interface, two cases were considered, one with a gap of 0.0635mm between the stopper and glass seal, and one with no gap between them.

Geometry Stability of the Glass Seal

In the first case, pure SCN-1 glass seal was applied without the ceramic stoppers at its inner and outer radius, see Figure 10. A constant pressure of 0.08MPa was applied on the top surface of PEN. The glass was allowed to flow freely along the interfaces with PEN and IC plate. The deformed configurations immediately after the pressure application and after 2 seconds of creep deformation are illustrated in Figure 11, respectively. Immediately after the application of the top pressure, the glass still keeps its original geometry. However, the glass is predicted to flow

quickly with time in creep analysis. The final thickness of the glass is almost invisible. Figure 12 shows the history of deformation of left side, i.e., inner radius, of the glass. The in-plane deformation increases almost linearly, and the thickness of the glass decreases in the same linear manner. That means that the glass will quickly flow out between the PEN and IC plate if no physical containment is established at the PEN and IC.

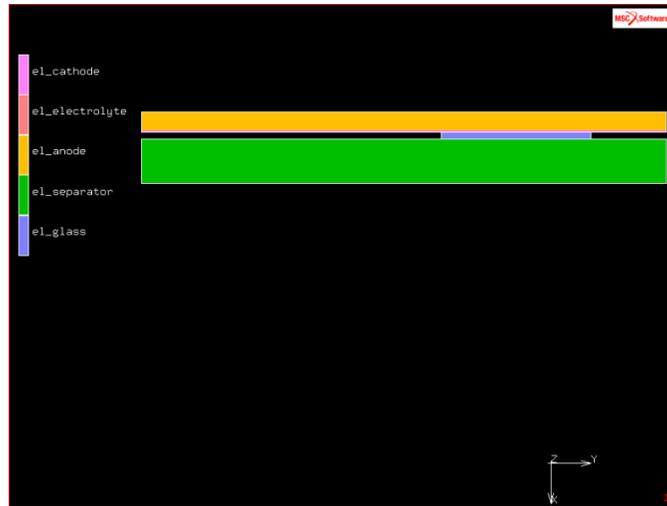


Figure 10 Schematic of model of setup without ceramic stoppers

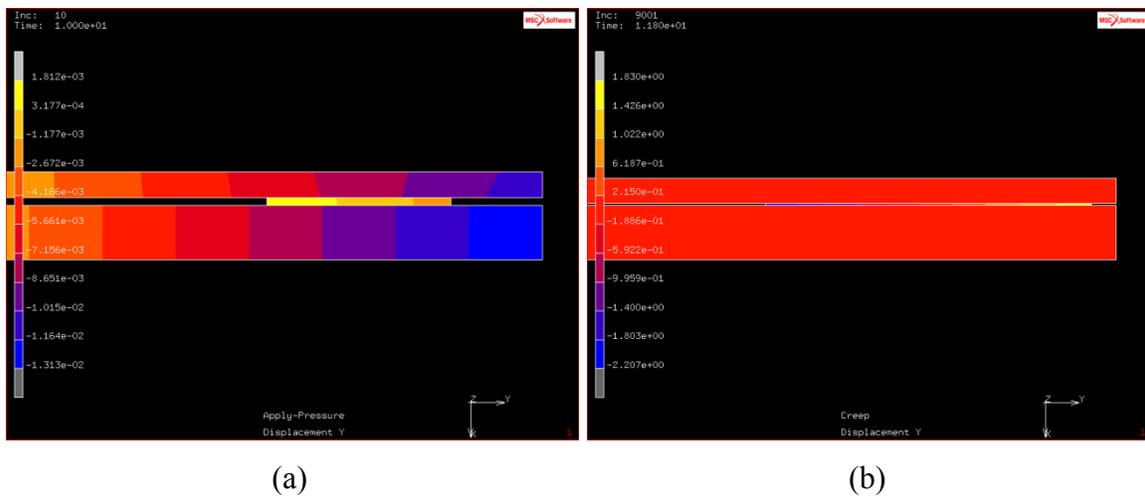


Figure 11 Deformed configurations (a) after applying top pressure; (b) creep in 2s

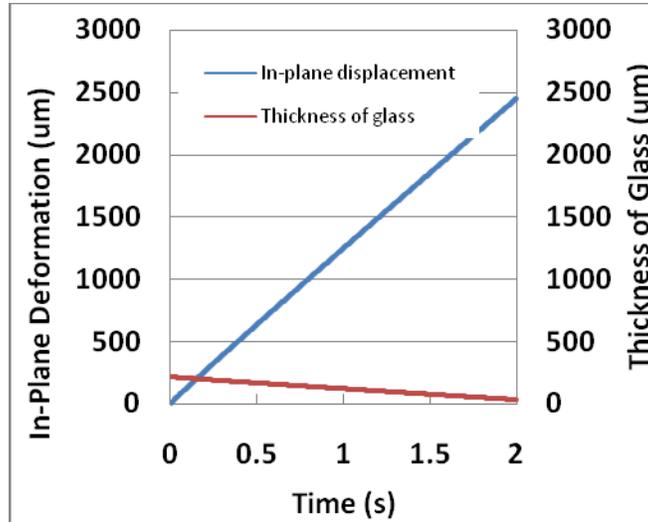
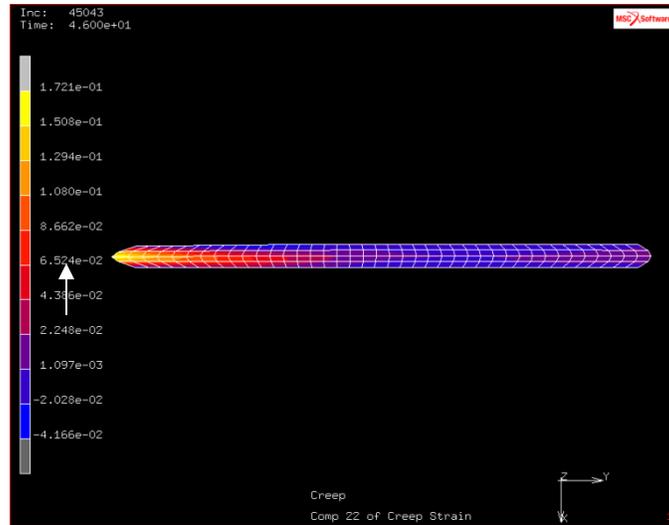
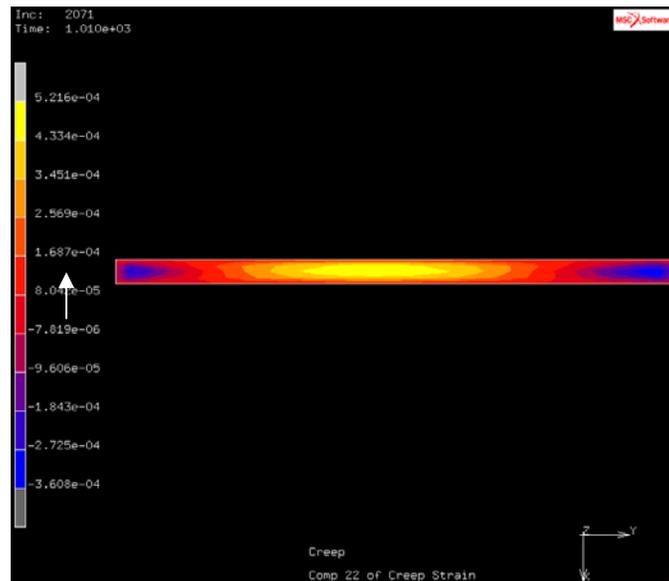


Figure 12 Deformation history of the inner side of the glass

Subsequently, we considered another case where the glass was not allowed to flow along the interfaces with PEN and IC. To compare the influence of the ceramic stopper, the results with and without the ceramic stoppers are discussed here. Figure 13 shows the deformed geometry of the glass without and with the stopper. Figure 13(a) shows the deformed configuration of glass seal after 36s of creep for the sealing system with 100% glass seal and no ceramic stopper. Figure 13(b) shows the deformed configuration of the glass seal after 1000s of creep for the sealing system with 100% glass seal but with ceramic stopper. At the constrained points along the interfaces with PEN and IC, the glass material cannot move along the interfaces, however the glass at the middle part away from the interface will still flow out. The equivalent creep strain history at the middle point of the left edge (marked by arrow in Figure 13(a) and (b)) is depicted over time in Figure 14. It is obvious that without the ceramic stopper the strain at the middle part of the glass will continue to linearly increase. Such a linear strain versus time indicates that the 100% glass seal will continue to flow over time. With the ceramic stopper, the creep deformation occurs only at the initial stage and then remains constant, because the shear stress in the glass will be relaxed out quickly and transferred into the ceramic stopper. Therefore, the geometry of the glass will be kept intact by the ceramic stopper.



(a)



(b)

Figure 13 Deformed glass with distribution of equivalent creep strain after creep. (a) 36s of creep without stopper; (b) 1000s of creep with ceramic stopper

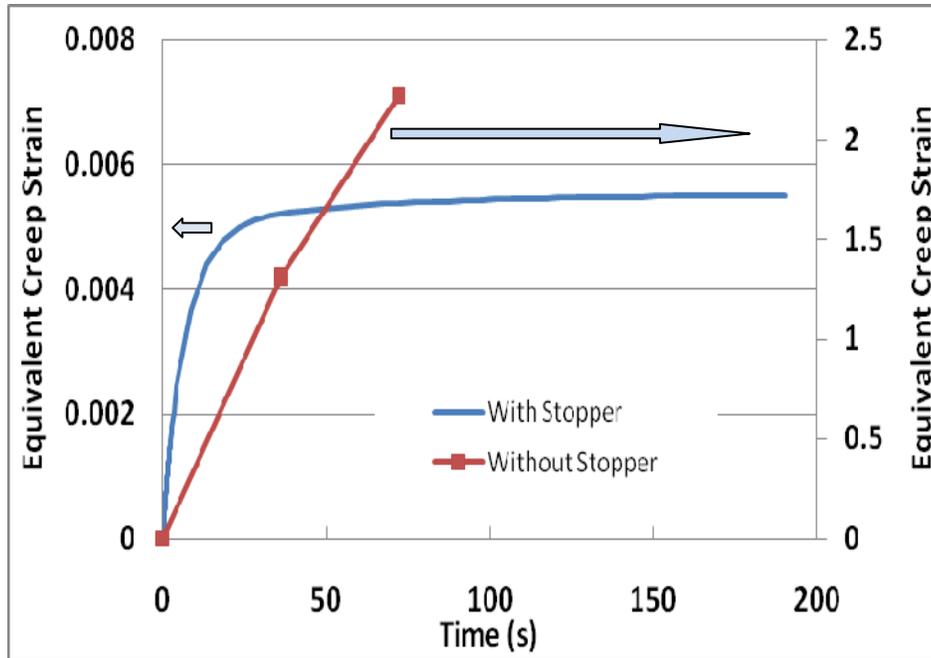


Figure 14 History of equivalent creep strain

Stresses in the Glass Seal

In order to prevent the glass seal out-flow, inner and outer ceramic stopper rings as those illustrated in Figures 2 and 3 are added to the next sealing system. Note that the glass seal and the ceramic stoppers are not bonded, and a small numerical gap of 0.063 mm is present in the model. Figure 15 and 16 show the distribution of maximum principal stress and shear stress in the glass seal at different stages of the simulation. Upon the initial temperature drop from the stress-free temperature to the SOFC operating temperature, small magnitudes of thermal stresses are induced in the glass seal. These stresses are relaxed out to even lower values during the first 3600 seconds of cell operation due to the creep behaviors of the glass seal. Upon cooling down to room temperature, relatively high thermal stress is predicted. These modeling results indicate that the final cooling step to room temperature is the most aggressive one in terms of creating high stresses in the glass seal, and is most likely to create some degree of damage in the glass seal. The exact degree of damage would depend on the room temperature bulk strength of the glass seal. However, if the glass seal is truly self-healing, then the cracks/damages will be

closed/healed up upon reheating to the stack operating temperature, and the whole process will repeat itself in Figure 15 and 16 during the next thermal cycle.

For the next sealing system design, we examine the case where the ceramic stopper and the glass seal are fully bonded. Figure 17 and 18 illustrate the modeling results of the maximum principal stress and shear stress in the glass seal at different time steps, respectively. The trend in the stress history in the seal is similar to those presented earlier with a gap. However, in this case, the magnitudes of the tensile and shear stress upon cooling to room temperature are much higher than those predicted earlier for the unbounded case with a small gap. These results indicate that leaving a small physical gap between the glass and ceramic seal will help mitigate cooling induced stress buildup in the seal, therefore reducing the degree of damage the glass seal may have upon cooling. Again, if the glass seal is truly self-healing, the magnitude of room temperature damage may not matter since all the cracks will be closed up and healed upon the next operating cycle.

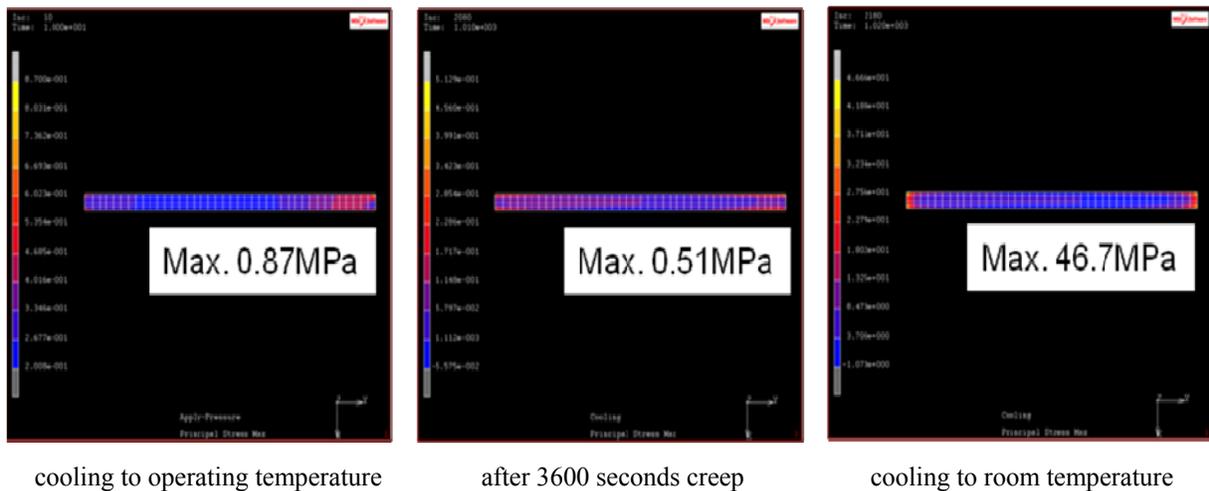
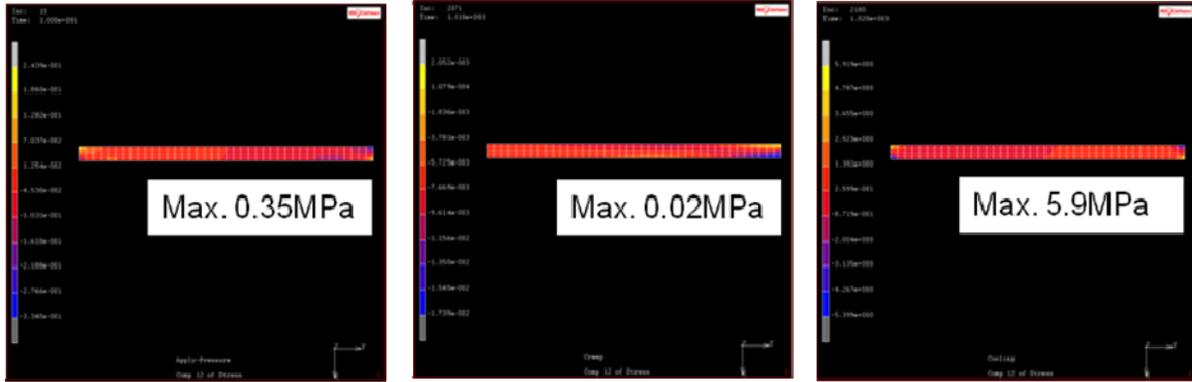


Figure 15 Maximum principal stress contour of glass seal at the different time:weak interface between stopper and glass seal

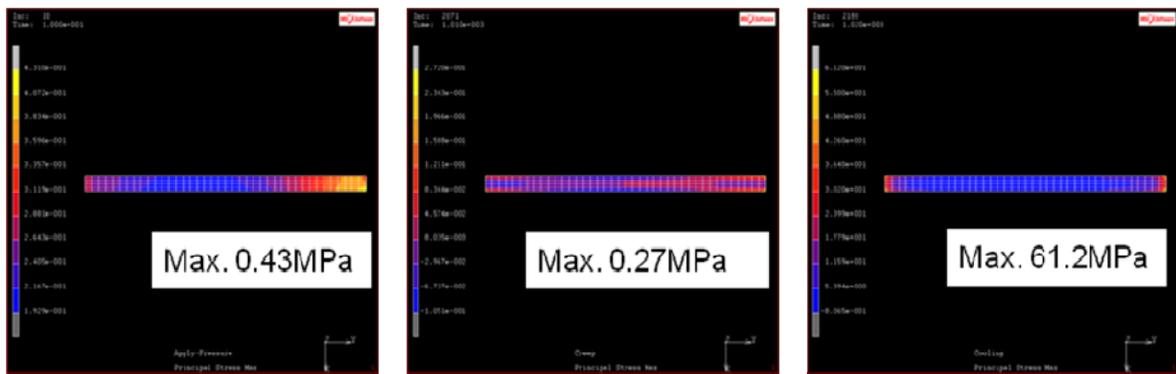


cooling to operating temperature

after 3600 seconds creep

cooling to room temperature

Figure 16 Shear stress S12 contour of glass seal at the different time: weak interface between stopper and glass seal

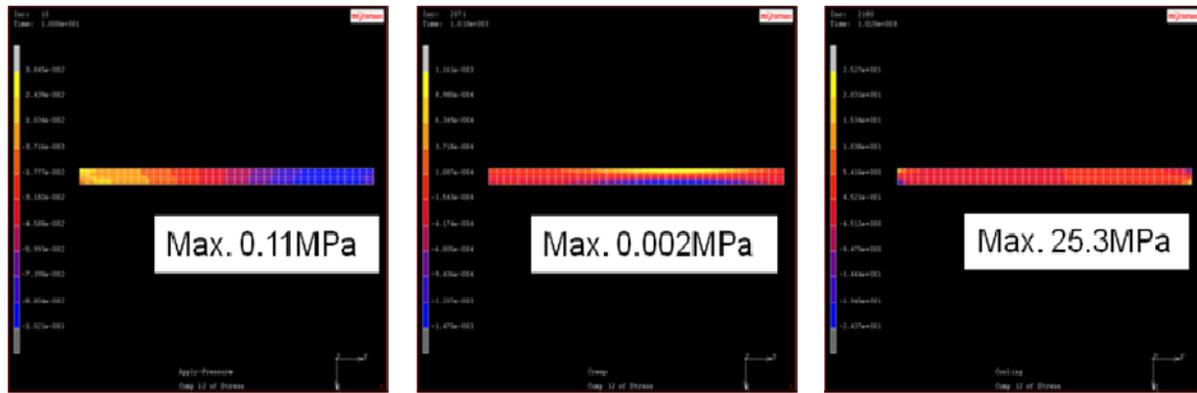


cooling to operating temperature

after 3600 seconds creep

cooling to room temperature

Figure 17 Maximum principal stress contour of glass seal at the different: strong interface of stopper and glass seal



cooling to operating temperature

after 3600 seconds creep

cooling to room temperature

Figure 18 Shear stress S12 contour of glass seal at the different time: strong interface of stopper and glass seal

Figures 19 and 20 depict the distribution of the shear stress and normal stress in the glass, respectively, during the cooling down process with a strong or weak stopper and glass interface. In Figure 19, with the weak stopper and glass interface, the stress in the glass is mainly related to the CTE mismatch between the glass, PEN, and IC. The temperature dependent CTE curves of glass and PEN intersect at the intermediate temperature of about 600K, i.e., the CTE of glass is higher than that of PEN at high temperatures, but lower than that of PEN at lower temperatures. With the strong stopper and glass interface, the stress in the glass, particularly near that interface, is influenced by the ceramic properties. After the cooling down process, the weak interface of the stopper and glass will create less interfacial shear stress and similar normal tensile stress. Whether the interface of the ceramic stopper and glass is strong or not, the highest stresses always occur at the edge of glass. In Figure 19 and 20, both of the maximum and minimum values of the stresses are given. For shear stresses, the positive and negative signs represent the directions of the shear stress, therefore only their magnitude has physical significance. For normal stresses, however, a positive value represents tensile normal stress on the interface, while a negative value represents compressive normal stress. Only a tensile normal stress on the interface will promote opening of the interface, i.e., interface delamination.

T(K)	Weak interface		Strong interface		T(K)
834					829
	6.7MPa	-8.5MPa	2.7MPa	-4MPa	
524					519
	16MPa	-19MPa	12MPa	-14MPa	
298					298
	5.9MPa	-5.4MPa	25MPa	-24MPa	

Figure 19 Shear stress distribution during the shutdown process with strong and weak interfaces of stopper and glass seal

T(K)	Weak interface		Strong interface		T(K)
834					829
	6.7MPa	-8.5MPa	2.7MPa	-4MPa	
524					519
	16MPa	-19MPa	12MPa	-14MPa	
298					298
	5.9MPa	-5.4MPa	25MPa	-24MPa	

Figure 20 Normal stress distribution during the shutdown process with strong and weak interface of stopper and glass

Effect of Interfacial Conditions of Stopper with Other Components

When applying ceramic stoppers to prevent the out-flow of the glass, the interfacial properties of the ceramic stopper with other components such as PEN, IC, and glass are closely dependent on the ceramic material and processing parameters. To evaluate the effect of the interfacial behaviors of the ceramic stopper with PEN and IC plate on the structure stability of the cell, two

extreme cases – bonded and sliding, are considered here for bounding purpose. For any boundary conditions of stopper with PEN and IC, two types of interface between the stopper and glass are always considered here: bonded and unbonded.

Figures 21 –26 show the normal and shear stresses distribution on the interfaces of glass with PEN and IC, respectively, over the distance which starts from the innermost ends of the interfaces. Regardless of the interface properties between the ceramic stopper and PEN/IC, the weak interface between the stopper and glass will always create lower interfacial shear stress on both the glass/PEN and the glass/IC interfaces. The effect of the interfacial condition of the stopper and glass on the interfacial normal stress is less than that on the interfacial shear stress. In any case, high interfacial shear stress and normal stress are predicted in the vicinity of the two edges of the glass seal. The maximum normal stress on the interface is always higher than the maximum shear stress on the interface. The high interfacial normal stresses at the edge of the glass may lead to localized failure of the glass seal at the edge after cooling down to room temperature. It may also be seen that the normal stresses on the interfaces of glass with PEN and IC remain compressive in the middle portion of the glass seal, therefore the structure integrity of the glass seals will be maintained even if some localized edge failures occur during cooling.

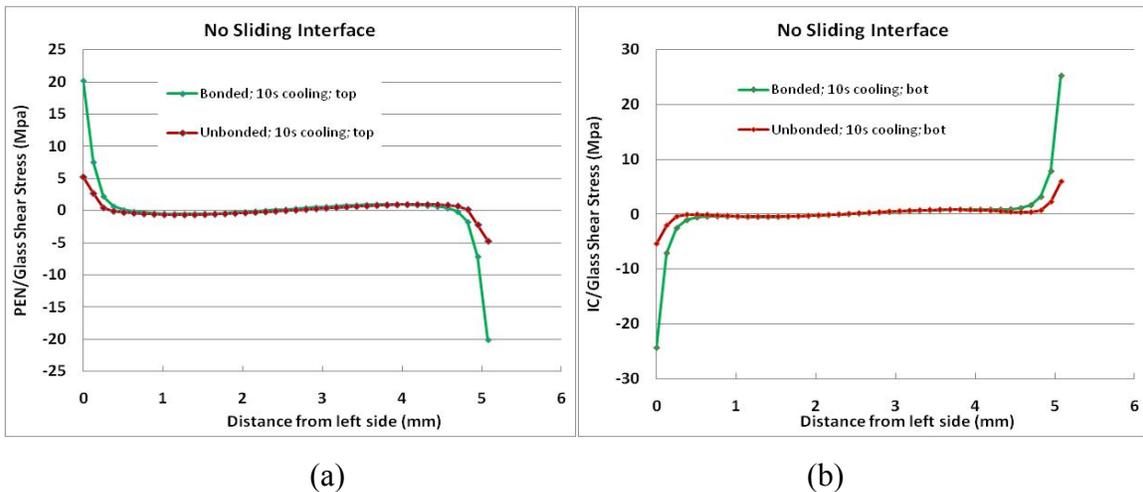


Figure 21 No sliding interface of stopper with PEN and IC: interfacial shear stress of glass (a) with PEN; (b) with IC

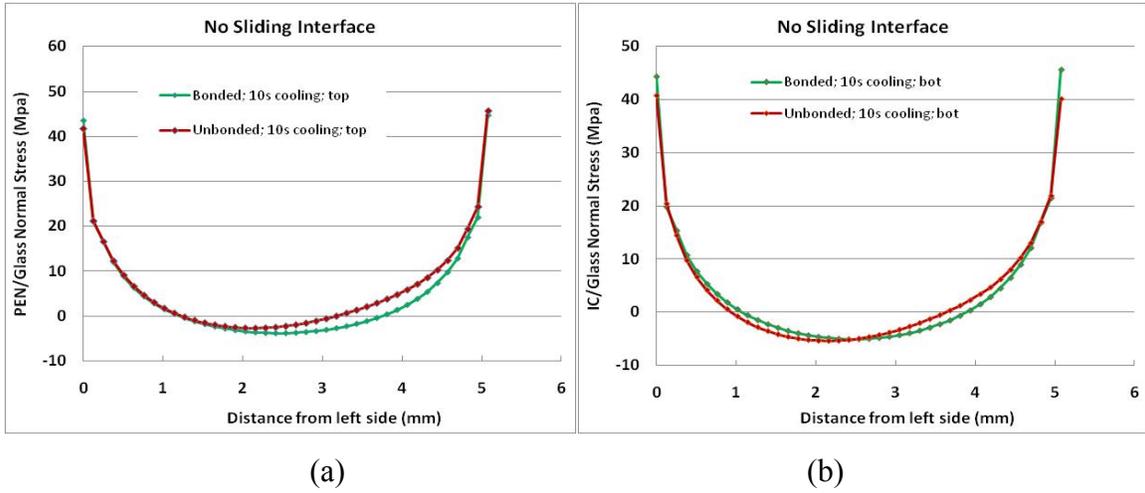


Figure 22 No sliding interface of stopper with PEN and IC: interfacial normal stress of glass with (a) PEN; (b) IC

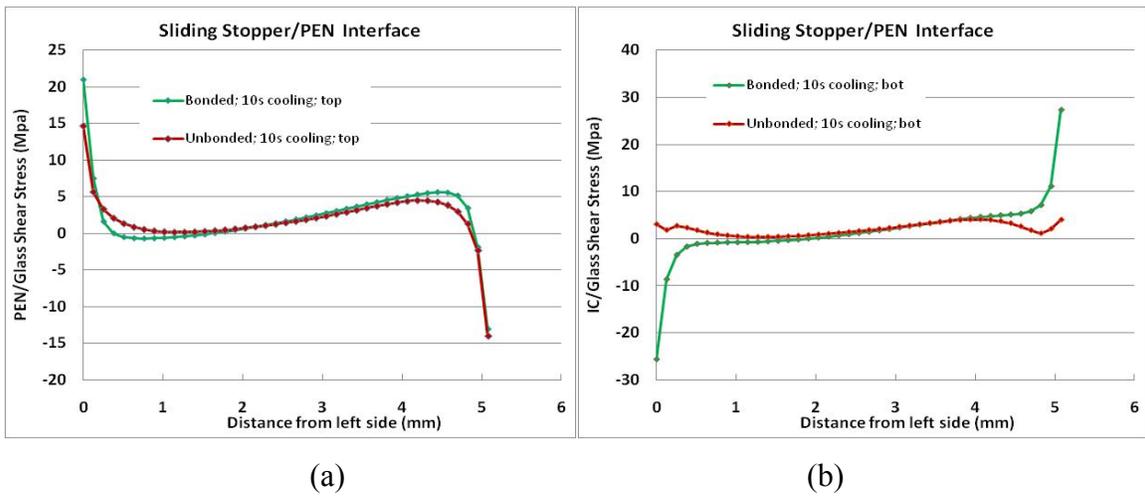


Figure 23 Sliding interface of stopper with PEN: interfacial shear stress of glass with (a) PEN; (b) IC

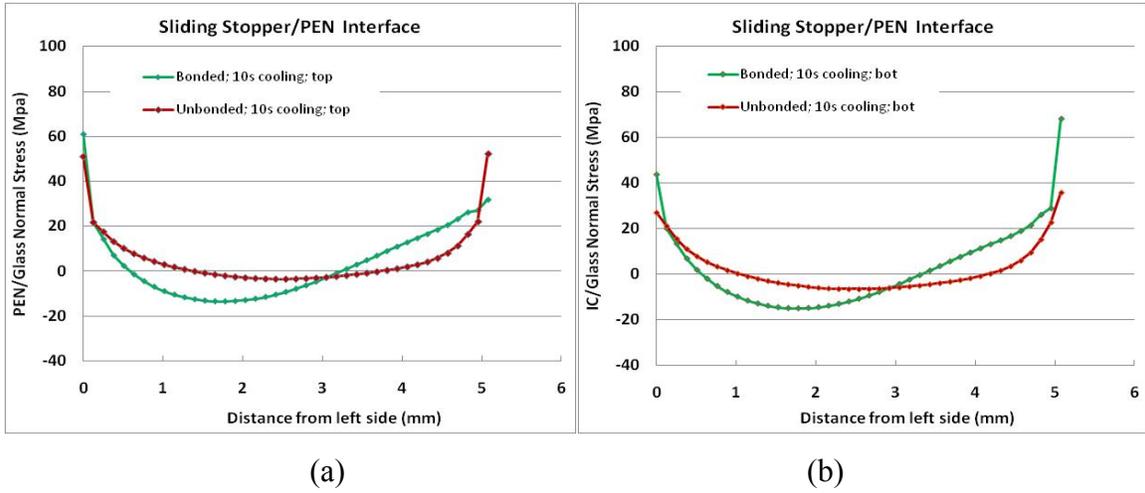


Figure 24 Sliding interface of stopper with PEN: interfacial normal stress of glass with (a) PEN; (b) IC

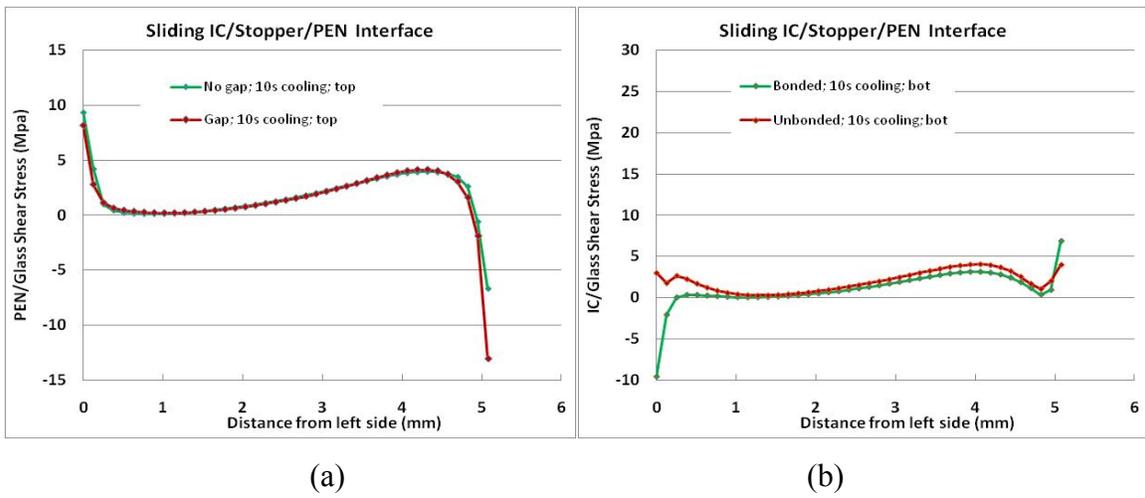


Figure 25 Sliding interfaces of stopper with PEN and IC: interfacial shear stress of glass with (a) PEN; (b) IC

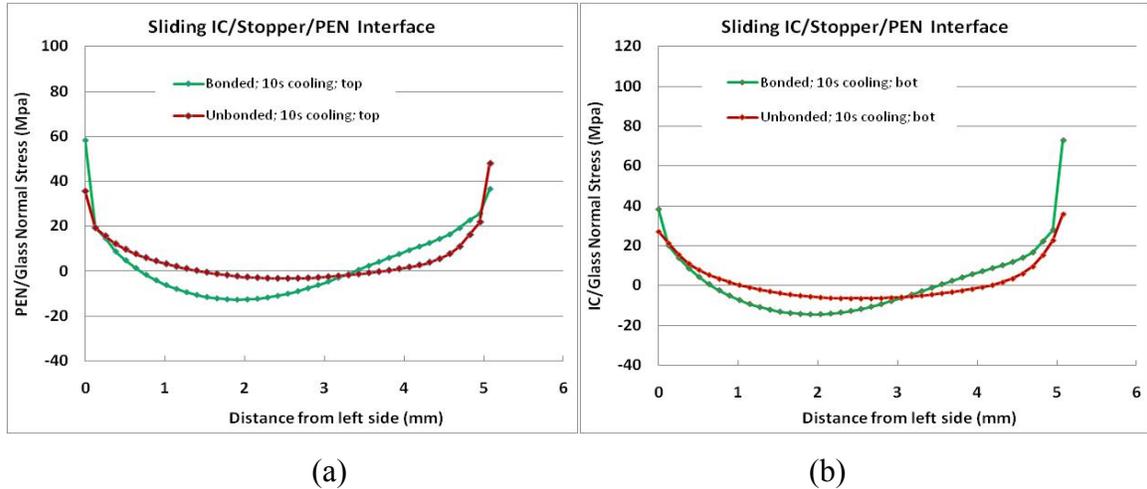


Figure 26 Sliding interfaces of stopper with PEN and IC: interfacial normal stress of glass with (a) PEN; (b) IC

4.0 Conclusions

In this report, the geometry stability of the self-healing glass and the influence of various interfacial conditions of ceramic stoppers with the PEN, IC, and glass seal on the structural integrity of the glass seal during the operating and cooling down processes are studied using finite element analyses. Two interfacial conditions of the ceramic stopper and glass seals, i.e., bonded (strong) or un-bonded (weak), are considered. The interfacial conditions of the ceramic stopper with the PEN and IC plates are assumed to be bonded or sliding.

Based on the analyses results, the following observations and conclusions can be made:

- (1) Self-healing glass alone cannot sustain its geometry in a sealing system for SOFC. The glass seal material will flow out quickly due to its low viscosity at high SOFC operating temperature.
- (2) Ceramic stoppers are necessary and helpful in maintaining the geometrical stability of the self-healing glass in a sealing system for SOFCs.

- (3) Under the operating environment of SOFCs, the stress level in the glass seal is very low. The initial stresses induced by the temperature drop from the stress-free assembly temperature to working temperature of SOFCs are relaxed quickly during the operation of SOFCs.
- (4) After the cooling down process, the CTE mismatch will result in relatively high stress in glass materials and on the interfaces of the glass seal with other components. It should be noted that the high stress occurs only locally at the edges of the glass seal, which may cause localized failure of the glass interfaces.
- (5) Despite the localized tensile normal stress at the interface between the glass seal with the PEN and IC, the compressive normal stresses in the middle portion of the glass seal will maintain the structure integrity of the glass seal system.

Our future work includes the examination of possible volume changes of the glass during the sintering process and its influence on the structural integrity of the sealing system in SOFCs. Effects of the applied glass paste volume, ceramic stoppers geometry, as well as the thermal-mechanical properties of glass on the geometry stability and structural integrity of the glass seal system will be studied. Further work in these areas is currently underway.

5.0 References

1. W.N. Liu, X. Sun, B. Koeppel, M. Khaleel, Experimental study of the aging and self-healing of the glass/ceramic sealant used in SOFCs, *International Journal of Applied Ceramic Technology*, v 7, n 1, p 22-29, January-February 2010
2. R. N. Singh, Sealing Technology for Solid Oxide Fuel Cells (SOFC), *International Journal of Applied Ceramic Technology*, 4 (2), pp. 134-144 (2007)
3. Edgar Lara-Curzio, et al., Characterization of SCN-1 Glass.
4. E. L. Bourhis, P. Gadaud, J. P. Guin, N. T ournerie, X. H. Zhang, J. Lucas, T. Rouxel, *Scripta Materilia* 45, 2001, pp. 317 – 323.
5. J. P. Andrews, *Proc. Phys. Soc.*, 36, 1924, pp. 169 – 177.
6. N. Govindaraju, et at, *Journal of Power Sources*, v 190, n 2, p 476-84, 15 May 2009
7. Rosa M. Trejo, et al, SCN-1-TMA.xls, Mar. 2010.
8. Rosa M. Trejo, et al, Viscosity and Wetting Behavior of SCN-1 Glass, Nov. 2009.