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Title: (U) Multi-Scale Friction Model in FLAG

Author(s): Kenamond, Mark A.

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# (U) Multi-Scale Friction Model in FLAG

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**M. A. Kenamond**

**X-Computational Physics**  
**Los Alamos National Laboratory**

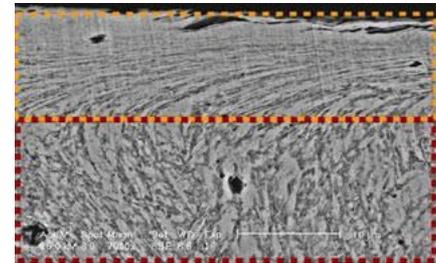
**June 9, 2014**

# Outline

- Introduction
- The model
- FLAG input syntax
- Results
- Summary, Conclusions and Future work

# Introduction

- In extreme loading regimes “friction” between metals is complex
  - Sticking: Static friction forces keep materials from sliding and materials are strong enough to withstand those forces
  - Sliding: Static friction is exceeded and Coulomb friction forces exist along with dry friction heating at the interface (rub your hands together to warm them up), materials are strong enough
  - Smearing: Sticking condition, but material strength exceeded by Coulomb forces, material plastically deforms and produces plastic work heating
  - All cases: heat conduction
- Plasticity is localized at the surface in a thin boundary layer
  - Too expensive to resolve the boundary layer
- Heat from plastic work and dry friction thermally soften the metals which can melt

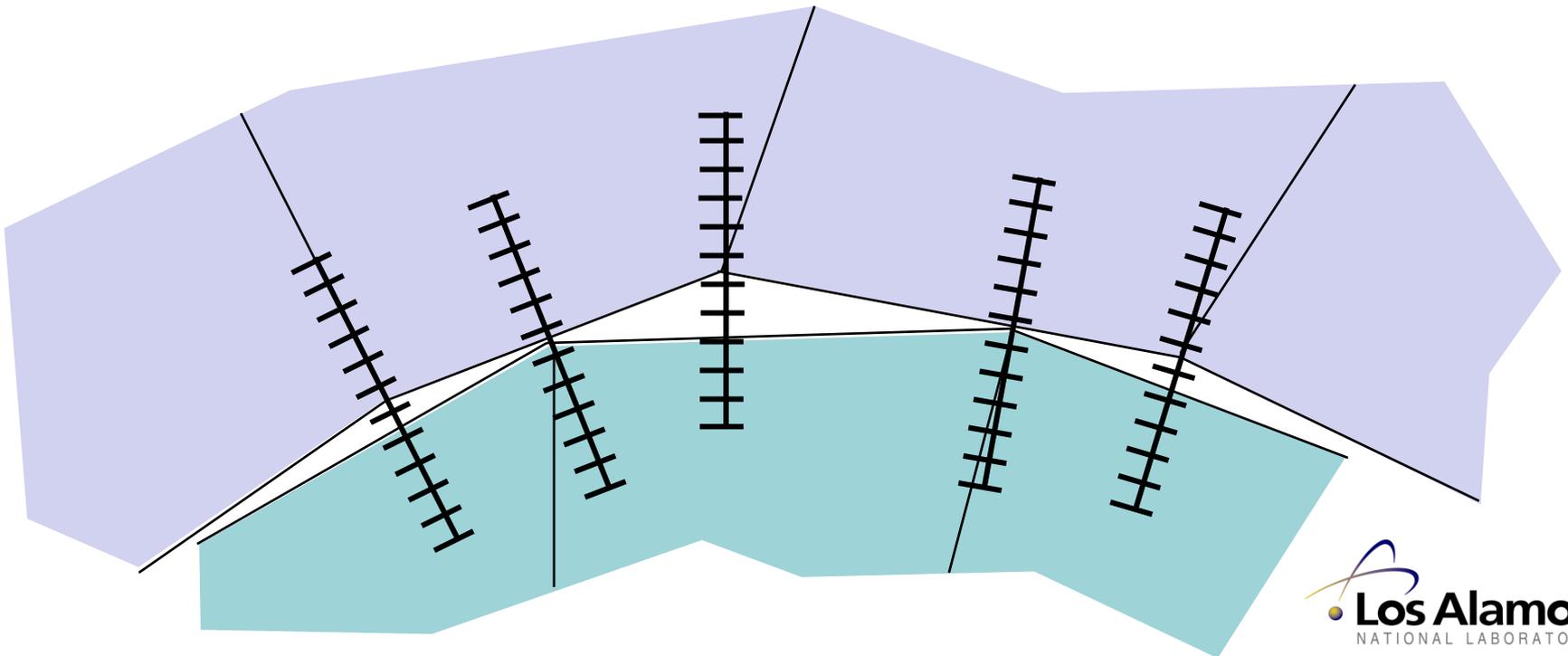


# Introduction

- Thin plastic boundary layer and high relative sliding velocity means large strain rates and huge range of strain rates
  - $O(10^8) \text{ s}^{-1}$  at the interface
- At small length scales, conduction time scales matter
- Plastic work deposits heat locally that conducts away
- Coulomb dry friction heating at the surface conducts into both materials
- Complicated.
- Want a practical model that captures much if not all of the physics

# The Model

- Coupled multi-scale friction model
  - Macro-scale = Computational mesh
  - Micro-scale = 1D boundary layer mesh at each surface point
    - Spans both contacting materials
    - Micro-scale conduction and plasticity
  - Based on similar approach by Dambakizi



# The Model: Micro-scale conduction

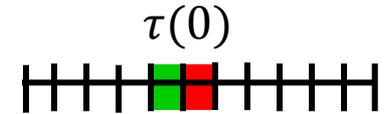
- Solves the non-steady heat equation on a 1D mesh using Crank-Nicholson time integration. Implicitly evolves temperature field by macro-scale  $\Delta t$
- Coupled to macro-scale
  - Macro-scale temperature used as Dirichlet BC at each end of the 1D mesh
  - Heat flux in/out the ends of the 1D mesh coupled to macro-scale as macro-scale internal energy source/sink
  - Macro-scale heating (PdV, radiation-electron coupling, etc.) deposited uniformly into 1D micro-scale thermal field
  - Macro-scale sliding velocity and friction force provide Coulomb dry friction heat source, deposited at the center of the 1D mesh at the contact interface
- Coupled to micro-scale plasticity
  - Micro-scale plastic heating deposited into thermal field

# The Model: Micro-scale plasticity

- Quasi-static assumption (Boo!!)
  - Quasi-uniform strain and strain rate
- Steinberg-Cochran-Guinan (SCG) strength (Boo!!)
  - Thermal softening with melt (Yay!!)
  - Not rate-dependent (Boo!!)
- Quasi-static assumption and rate independent strength enable a numerical “trick”

# The Model: Micro-scale plasticity

- Compute yield stress of each material in 1D zones connected to contact interface using SCG model
- Take minimum of the two (weaker material)
- Apply this stress to the entire 1D mesh
  - Requires quasi-static assumption
- Calculate strain increments in each 1D zone
  - Solve SCG stress-strain relation for strain increment and rate (divide increment by  $\Delta t$  to get strain rate)
  - Requires rate-independent strength model

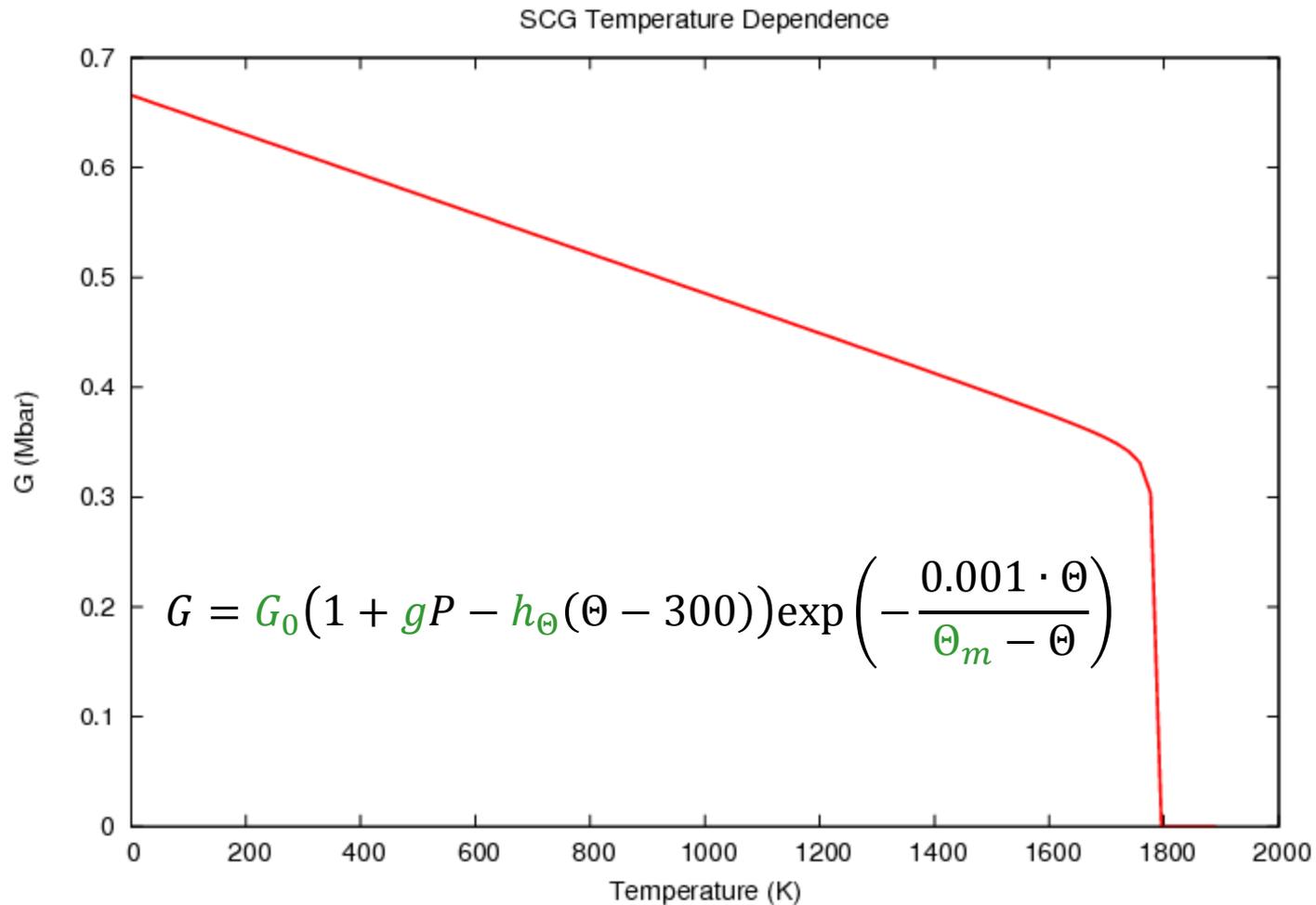


$$Y = \sqrt{3}\tau(0) = Y_0(1 + \beta\epsilon_p)^\eta \frac{G(P, \Theta)}{G_0} \quad G = G_0(1 + gP - h_\Theta(\Theta - 300)) \exp\left(-\frac{0.001 \cdot \Theta}{\Theta_m - \Theta}\right)$$

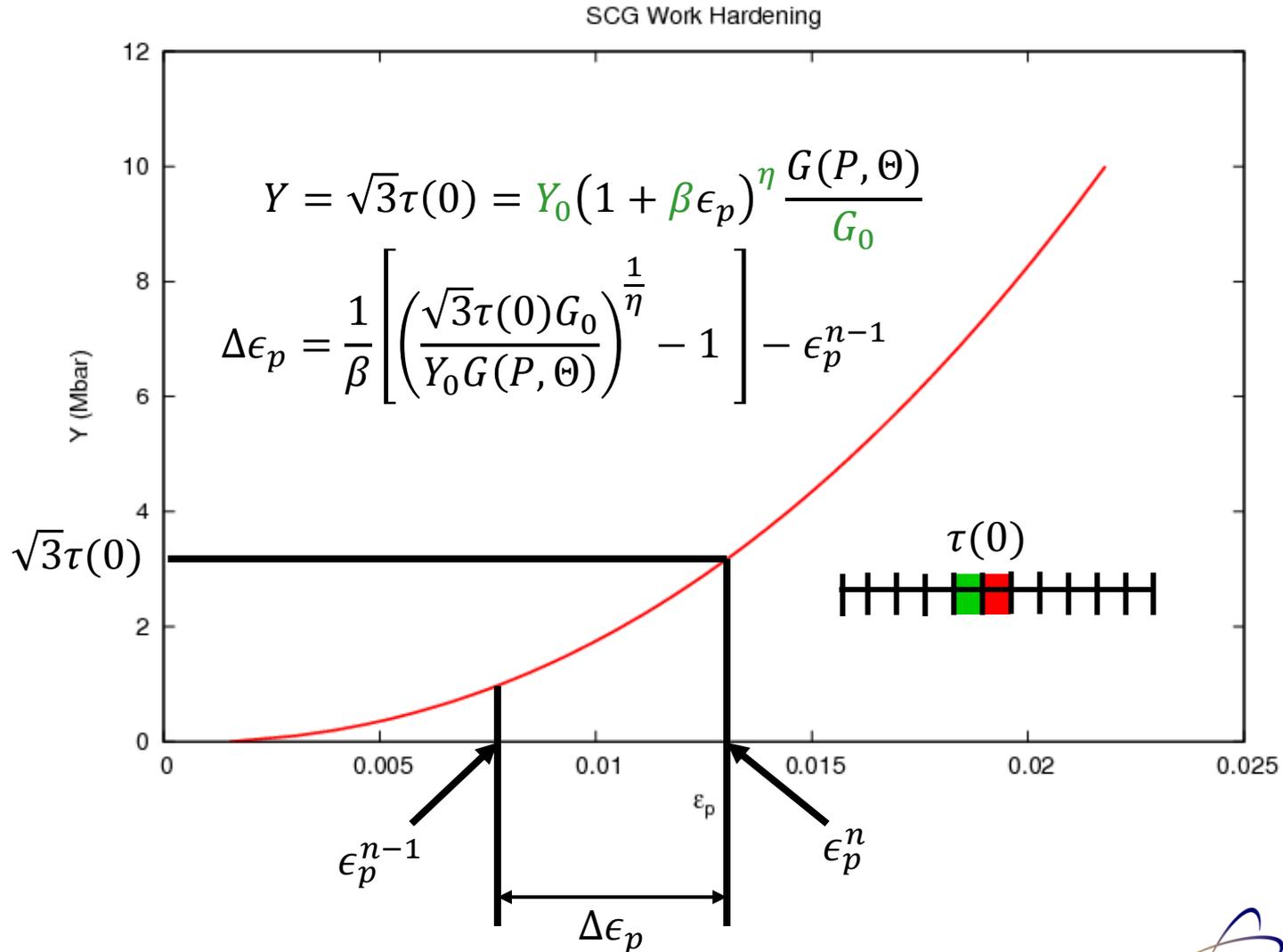
$$\epsilon_p^n = \epsilon_p^{n-1} + \Delta\epsilon_p \quad \Delta\epsilon_p = \frac{1}{\beta} \left[ \left( \frac{\sqrt{3}\tau(0)G_0}{Y_0G(P, \Theta)} \right)^{\frac{1}{\eta}} - 1 \right] - \epsilon_p^{n-1}$$

$$\dot{\epsilon}_p = \Delta\epsilon_p / \Delta t$$

# The Model: Micro-scale plasticity

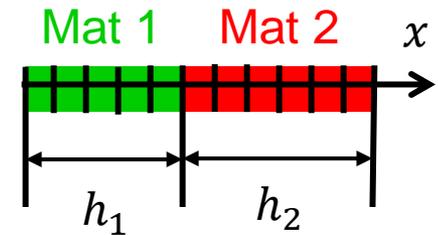


# The Model: Micro-scale plasticity



# The Model: Micro-scale plasticity

- Integrate strain rate across 1D domain
  - Gives micro-scale relative sliding velocity
  - $\Delta u_{12} = \sqrt{3} \int_{-h_1}^{h_2} \frac{\Delta \epsilon_p(x)}{\Delta t} dx$
- Get macro-scale sliding velocities  $U_1$  and  $U_2$
- Get macro-scale relative sliding velocity
  - $\Delta U_{12} = U_2 - U_1$
- Scale strain increments so integral matches macro-scale relative sliding velocity
  - $\Delta \epsilon_p^{new} = \max\left(0, \frac{|\Delta U_{12}|}{\Delta u_{12}} \Delta \epsilon_p\right)$
- Deposit heat into 1D model based on scaled strain increments
- Evolve 1D plastic strain field based on scaled strain increments
- $\tau_t = \mu_t \cdot \tau(0)$  is the Tresca (smearing) friction force



# The Model: Other stuff

- In addition to the micro-scale aspects:
- Sticking traction is computed
- Coulomb traction is computed
  - $\tau_c = \mu_c \cdot P$
- Tresca (plasticity/smearing) traction  $\tau_t$  is computed
- Minimum traction is determined
  - If Coulomb, no plasticity, but we must solve the conduction problem with a dry friction heat source
  - If Sticking, no plasticity, just conduction without a dry friction heat source
  - If Tresca, evolve plasticity, deposit plastic work, no dry friction heating
- Can change throughout a simulation

## The Model: Recap

- Handles sticking, Coulomb and Tresca friction automatically
- Coupled multi-scale model
- Micro-scale plasticity and heat conduction
- Coupled to macro-scale internal energy, velocity and stress fields
- Some big warts in the micro-scale plasticity

## FLAG Input Syntax

- Must indentify “material 1 side” vs “material 2 side”
  - Each has different SCG parameters, mesh depths, conductivities
- Must specify thermal and mechanical properties
- Only true knobs are the friction coefficients  $\mu_t$  and  $\mu_c$
- Calibrating rate-independent SCG model parameters for higher rates would probably be considered knobs

# FLAG Input Syntax

Add this to your typical slideline definition:

```
mk /global/mesh/hydro/lhydro/kbc (SlipFrict) /slide/enforcement/friction/multi_scale
alias pffrict      pffrict      $Friction forces
alias p_tau       p_tau        $Friction tractions
alias p_taustick  p_taustick   $Sticking tractions
alias p_taucoulomb p_taucoulomb $Coulomb tractions
alias p_tautresca p_tautresca  $Tresca tractions
alias p_relvel    p_relvel     $Relative sliding velocities
alias p_x0temp    p_x0temp     $Temperature at 1D interface
alias p_maxtemp   p_maxtemp    $Maximum temperature on "my" side of 1D mesh
alias p_maxeps    p_maxeps     $Maximum effective plastic strain on my side
alias p_pressure  p_pressure   $Contact pressure
alias p_heatflux  p_heatflux   $Heat flux through my end of 1D mesh
alias kp_frictype kp_frictype  $Friction type (0=none, 1=stick, 2=Tresca, 3=Coulomb)

bdy1 = "Be_Contact_Bdy"      $kbdy node name identifying side 1 (Beryllium)
map_opt = 2                  $Project onto z-axis when mapping across interface
gap_max = 0.01               $In contact if gap is less than this

rmu_c = 0.3                  $Coulomb friction coefficient
rmu_t = 0.9                  $Tresca friction coefficient
```

# FLAG Input Syntax (cont'd)

## \$ 1D mesh specs

h1 = 1.0e-3 \$1D mesh depth into material 1  
h2 = 1.0e-3 \$1D mesh depth into material 2  
nz = 40 \$1D zones in each material (80 zones total)

## \$ Be is side 1

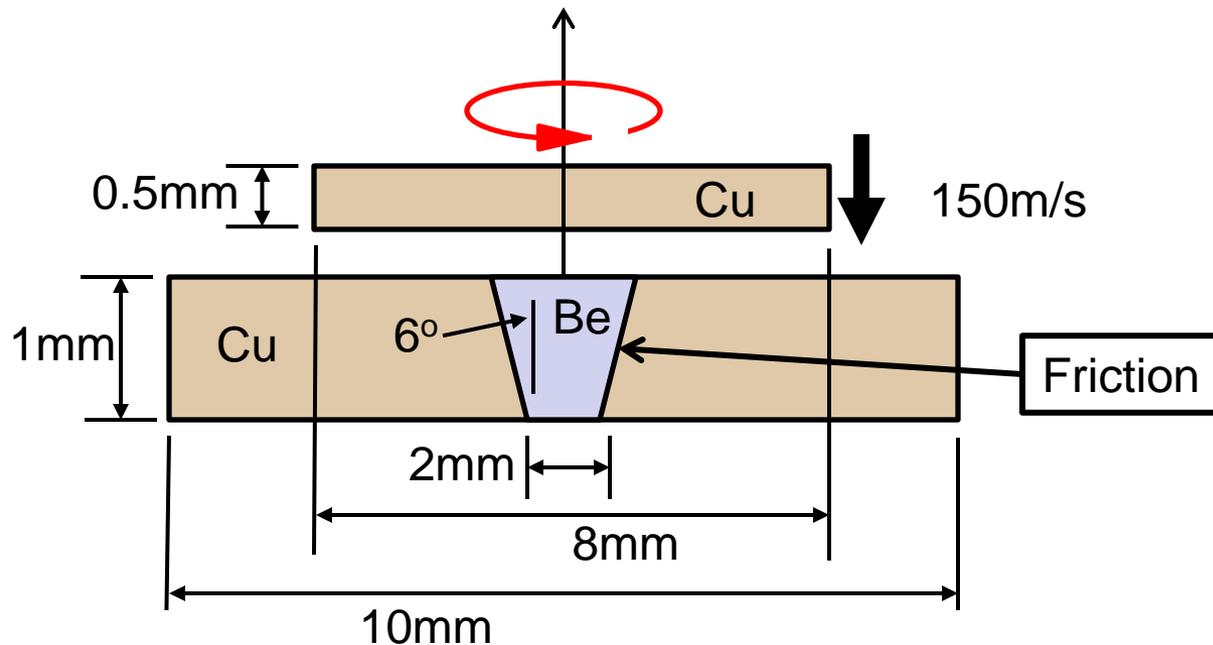
rk1 = 2e-11 \$Conductivity of material 1  
g01 = 1.51 \$G\_0 SCG initial shear modulus of material 1  
y01 = 3.3e-3 \$Y\_0 SCG initial yield stress of material 1  
ymax1 = 1.31e-2 \$Y\_max SCG maximum yield stress of material 1  
beta1 = 26.0 \$SCG work hardening coefficient of material 1  
eta1 = 0.78 \$SCG work hardening exponent of material 1  
g1 = 1.54 \$SCG P-dependent shear modulus coefficient for mat 1  
htheta1 = 2.58e-4 \$SCG T-dependent shear modulus coefficient for mat 1  
tmelt1 = 1820.0 \$Melt temperature of material 1

## \$ Cu is side 2

rk2 = 4.01e-11 \$Conductivity of material 2  
g02 = 0.477 \$G\_0 SCG initial shear modulus of material 2  
y02 = 1.2e-3 \$Y\_0 SCG initial yield stress of material 2  
ymax2 = 6.4e-3 \$Y\_max SCG maximum yield stress of material 2  
beta2 = 36.0 \$SCG work hardening coefficient of material 2  
eta2 = 0.45 \$SCG work hardening exponent of material 2  
g2 = 2.83 \$SCG P-dependent shear modulus coefficient for mat 2  
htheta2 = 3.77e-4 \$SCG T-dependent shear modulus coefficient for mat 2  
tmelt2 = 1790.0 \$Melt temperature of material 2

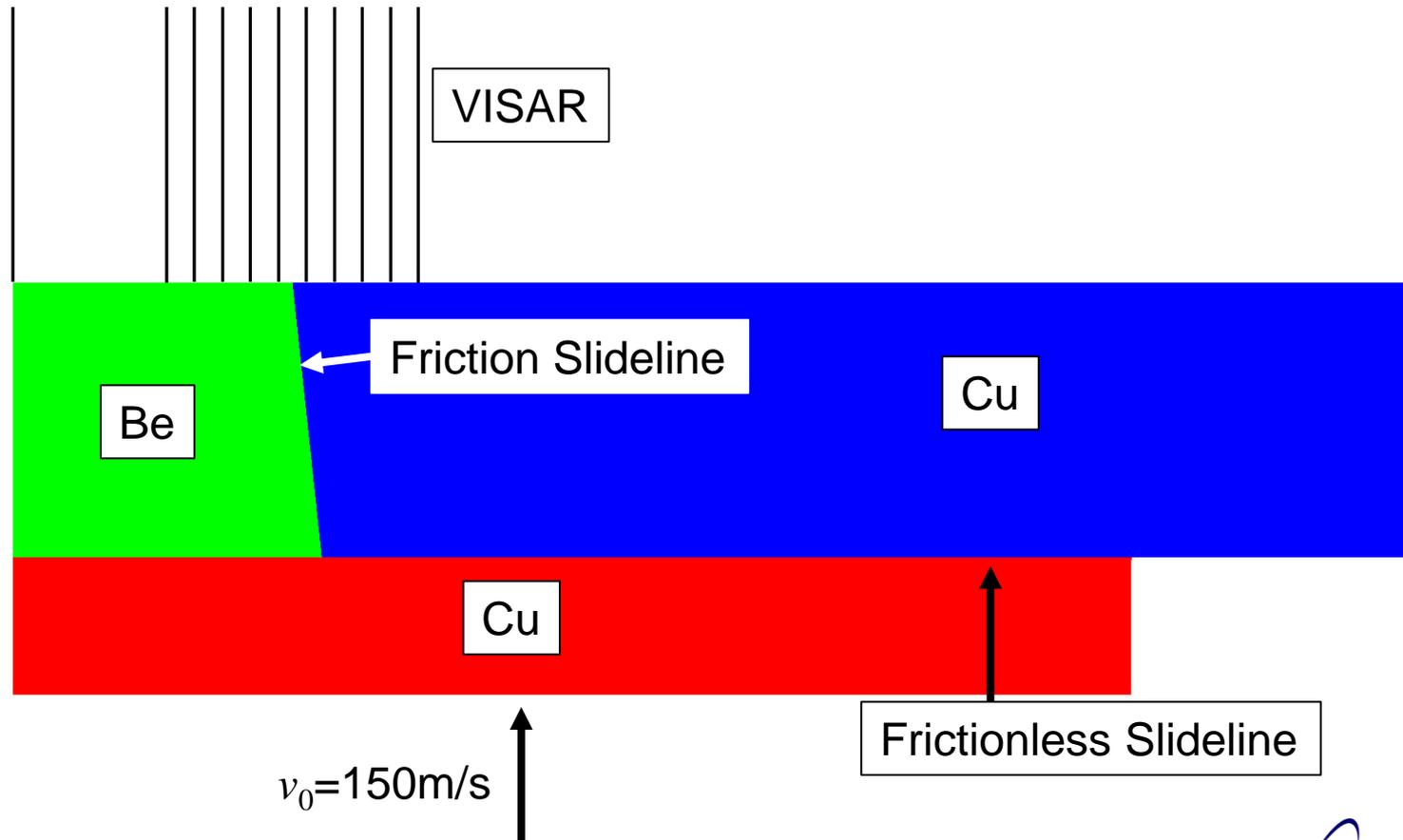
# Results

- LANL laser-driven flyer Cu/Be friction experiment
- ~150m/s Cu flyer velocity
- Composite Cu/Be target
- Point and line VISAR diagnostics and TIDI on the back surface of the target



# Results: The Model

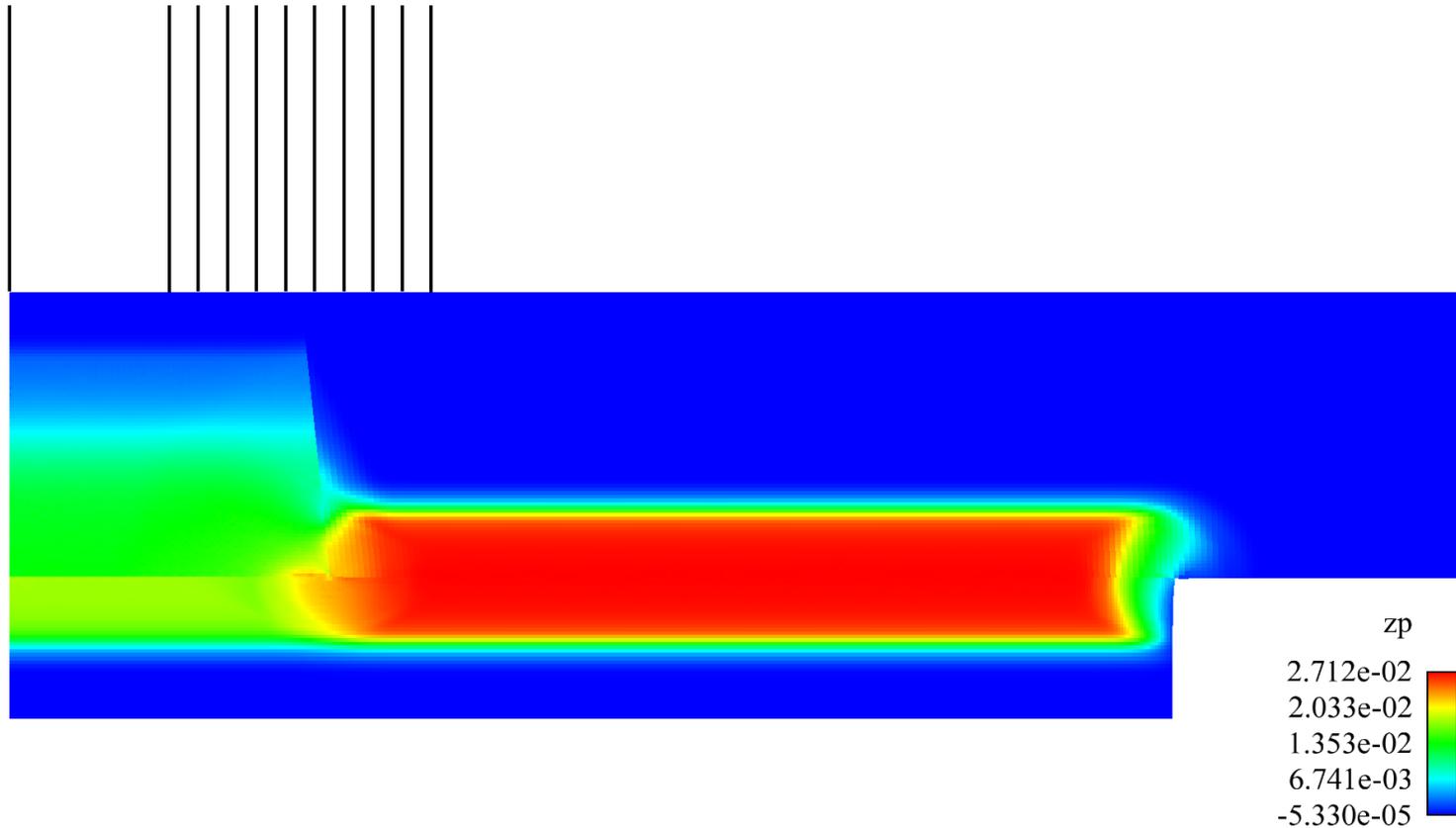
- PTW' for all materials
- 2 slidelines, one frictionless, one with friction
- Cu and Be SCG friction parameters from Steinberg compendium
- 11 VISAR probes



# Results: Pressure at 60ns

- Be sound speed higher than Cu
- Velocity difference at interface causes sliding
- Cone angle increases contact pressure

Time = 0.06021



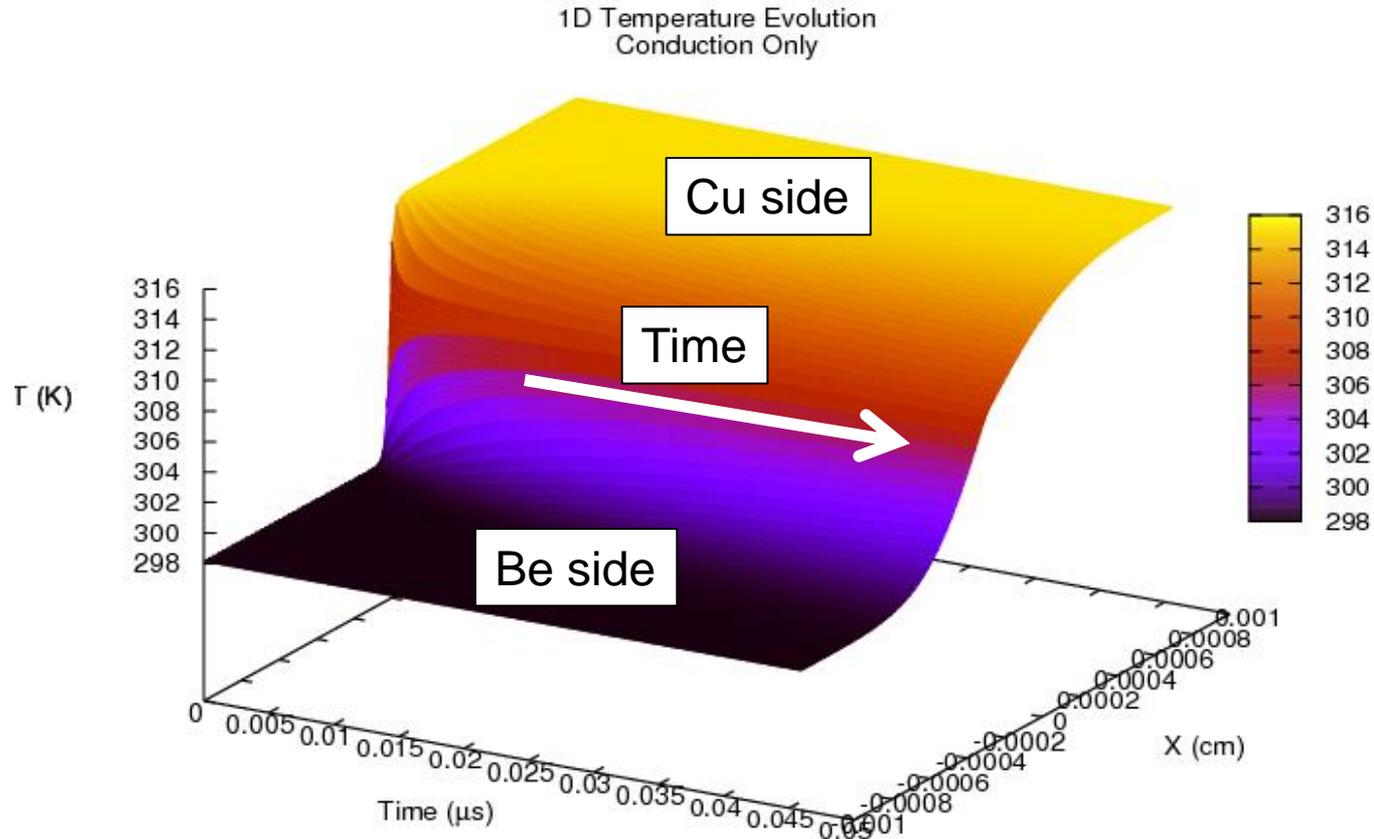
# Results: Micro-scale behavior

- Will evaluate each aspect of the model at one point on the interface
- Demonstrate increasingly coupled effects



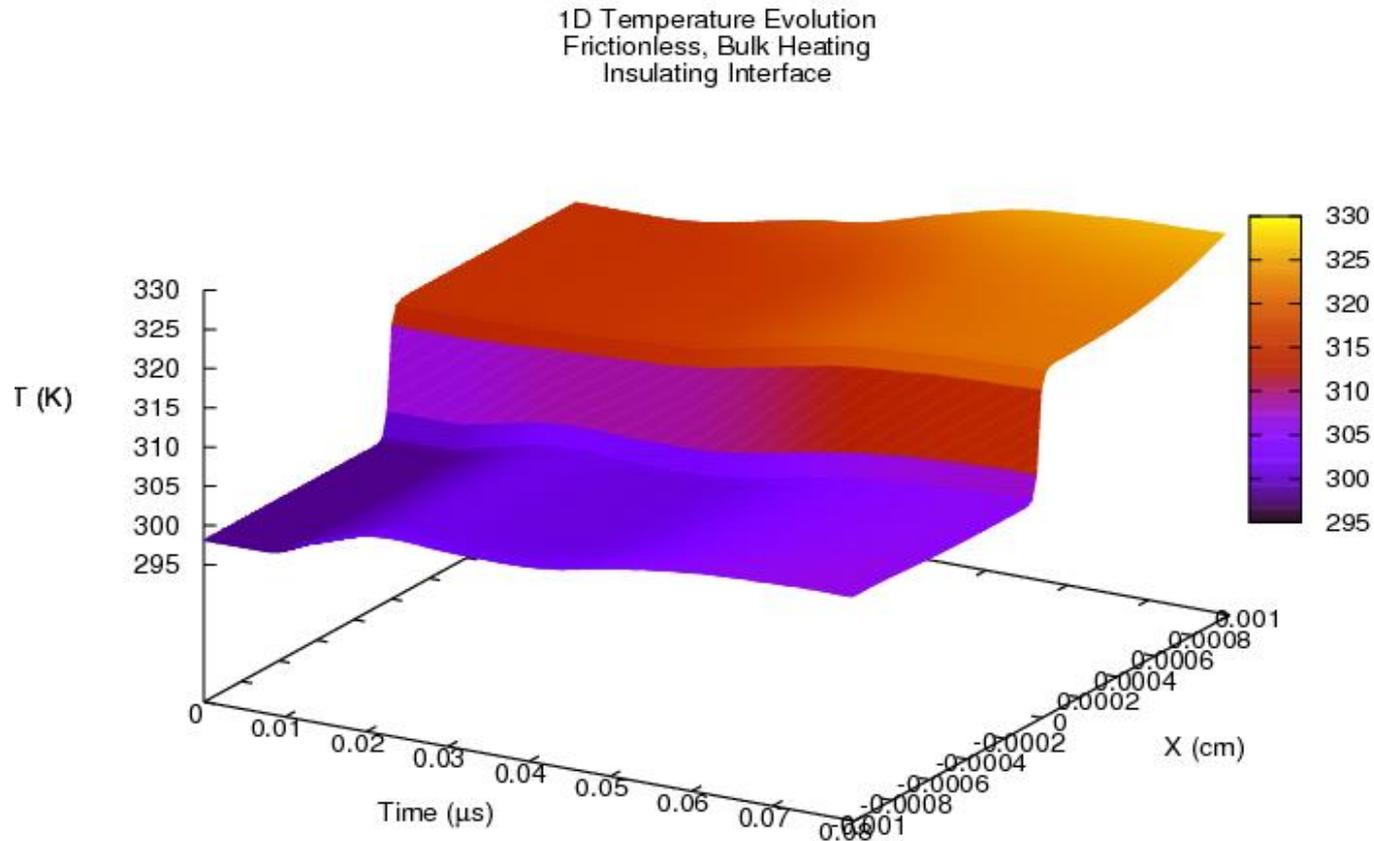
# Results: Pure Conduction

- Set impact velocity to 0
- Only conduction
- Cu and Be initialize to different temperatures



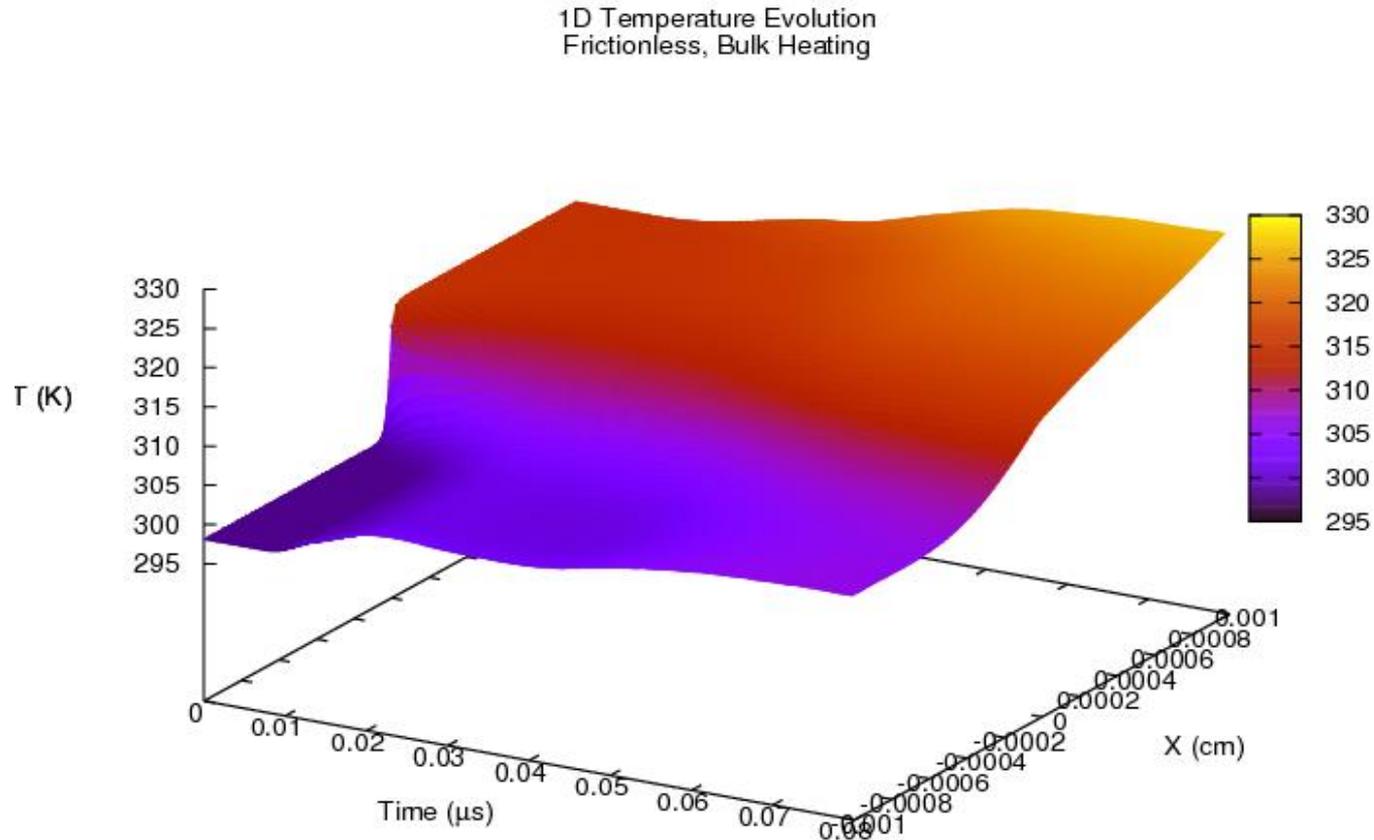
# Results: Bulk Heating

- Set impact velocity to 150m/s
- Frictionless (set Coulomb friction coefficient to zero)
- Force insulating condition at contact interface
- Cu and Be heating due to macro-scale PdV work and Dirichlet BC



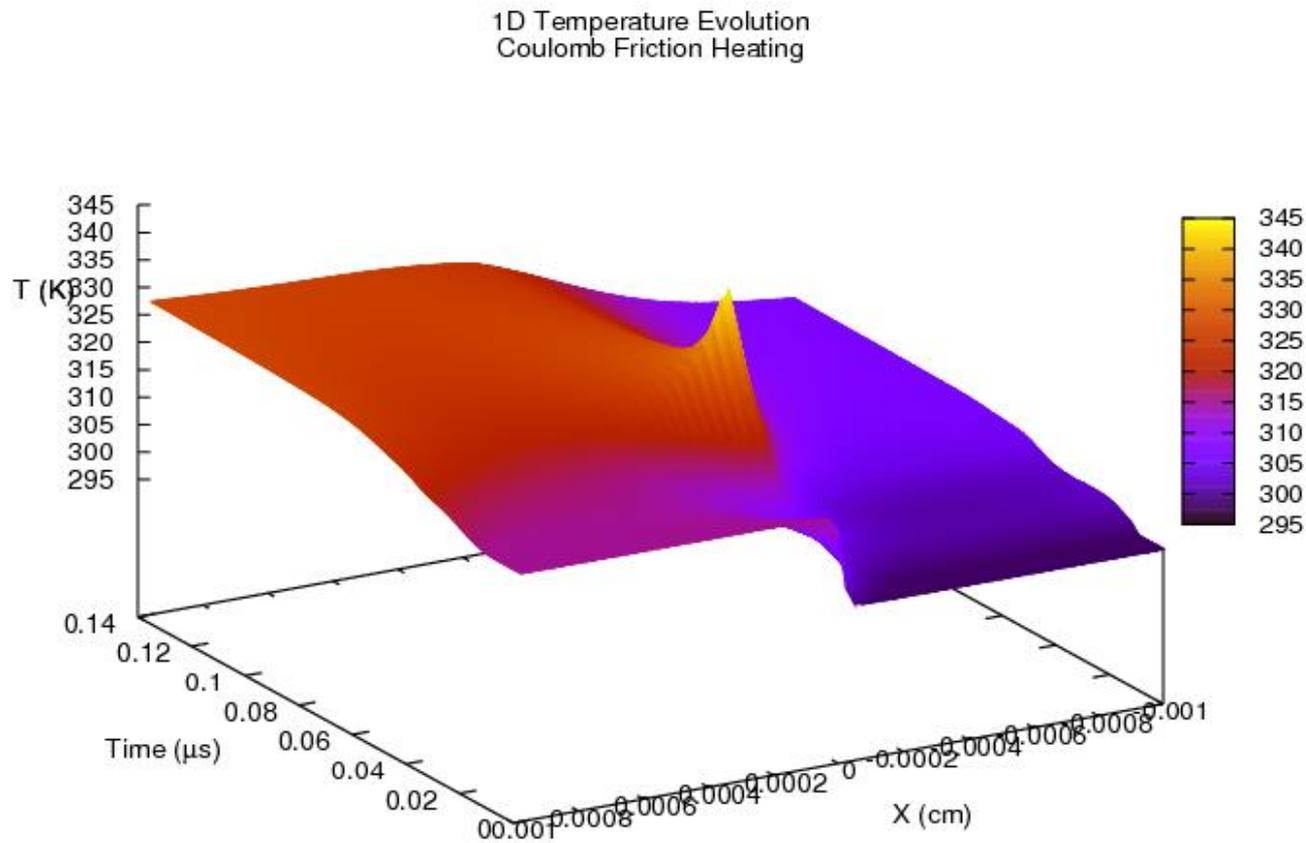
# Results: Bulk heating with conduction

- Same as before, but conduction enabled across interface



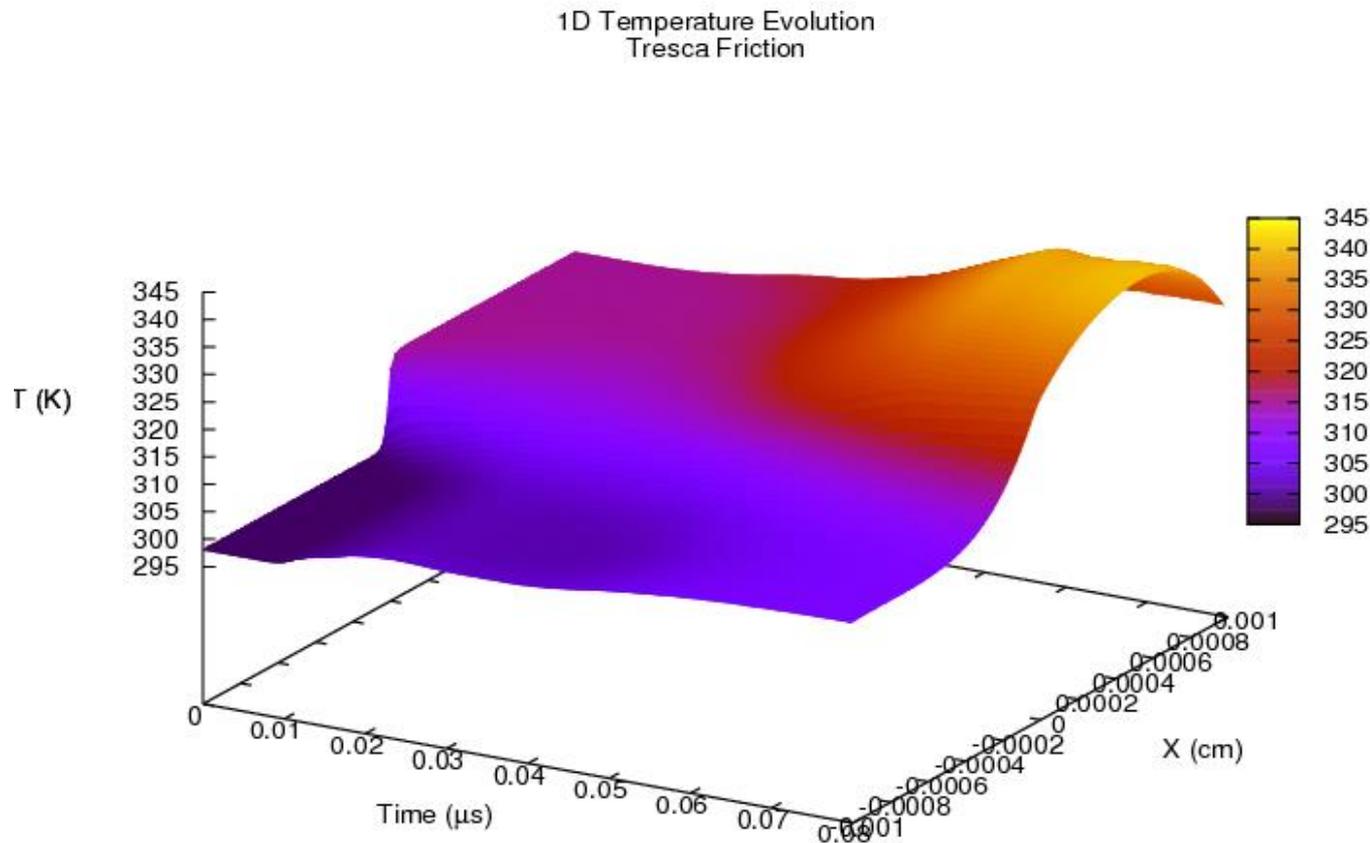
# Results: Bulk and Coulomb heating

- Turn on Coulomb friction
- Set SCG  $Y_0$  to a large number so smearing friction is disabled
- Coulomb dry friction heat source at interface until sticking condition occurs



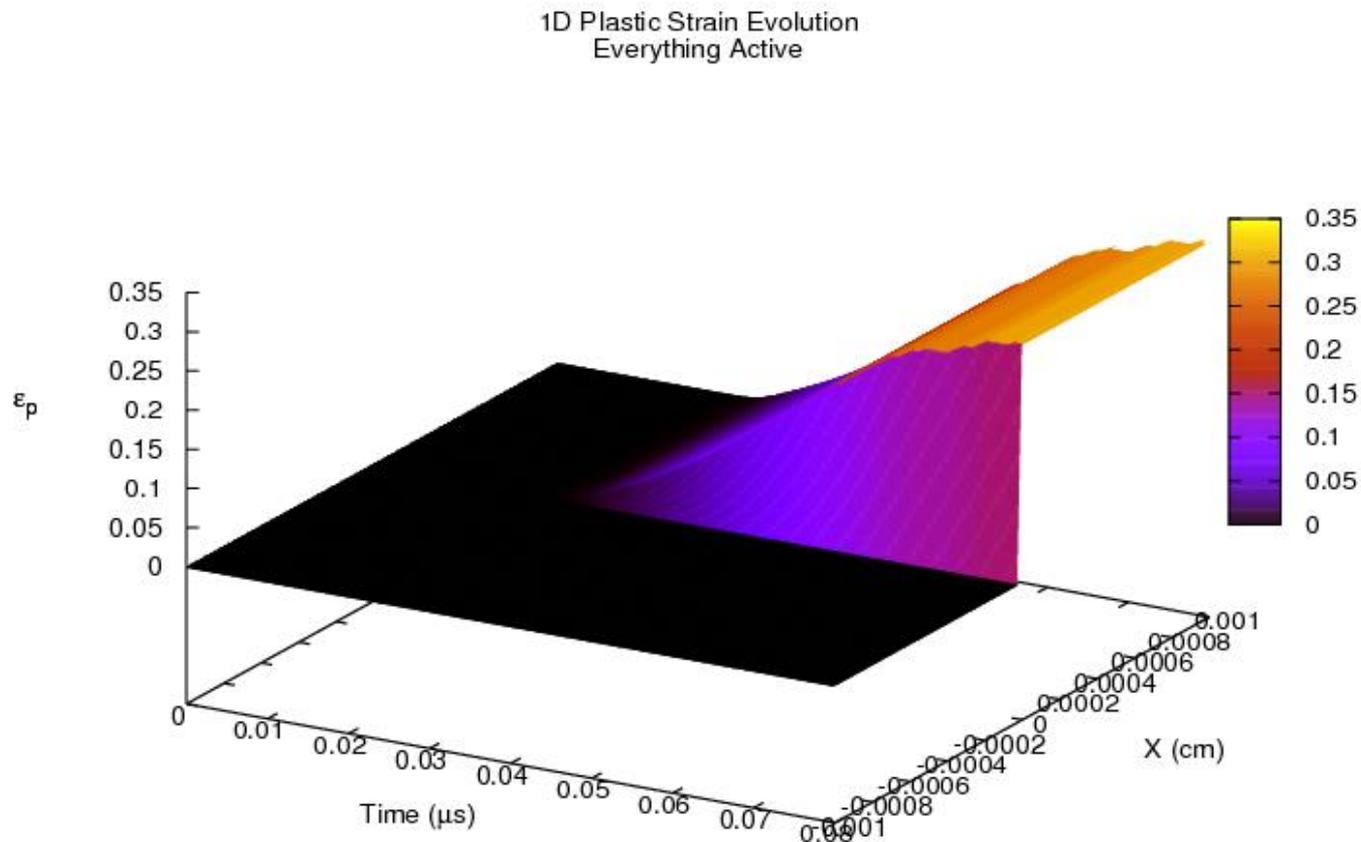
# Results: Smearing friction

- Turn off Coulomb friction (set coefficient to large value)
- Use correct SCG parameters
- Plastic work heating in Cu



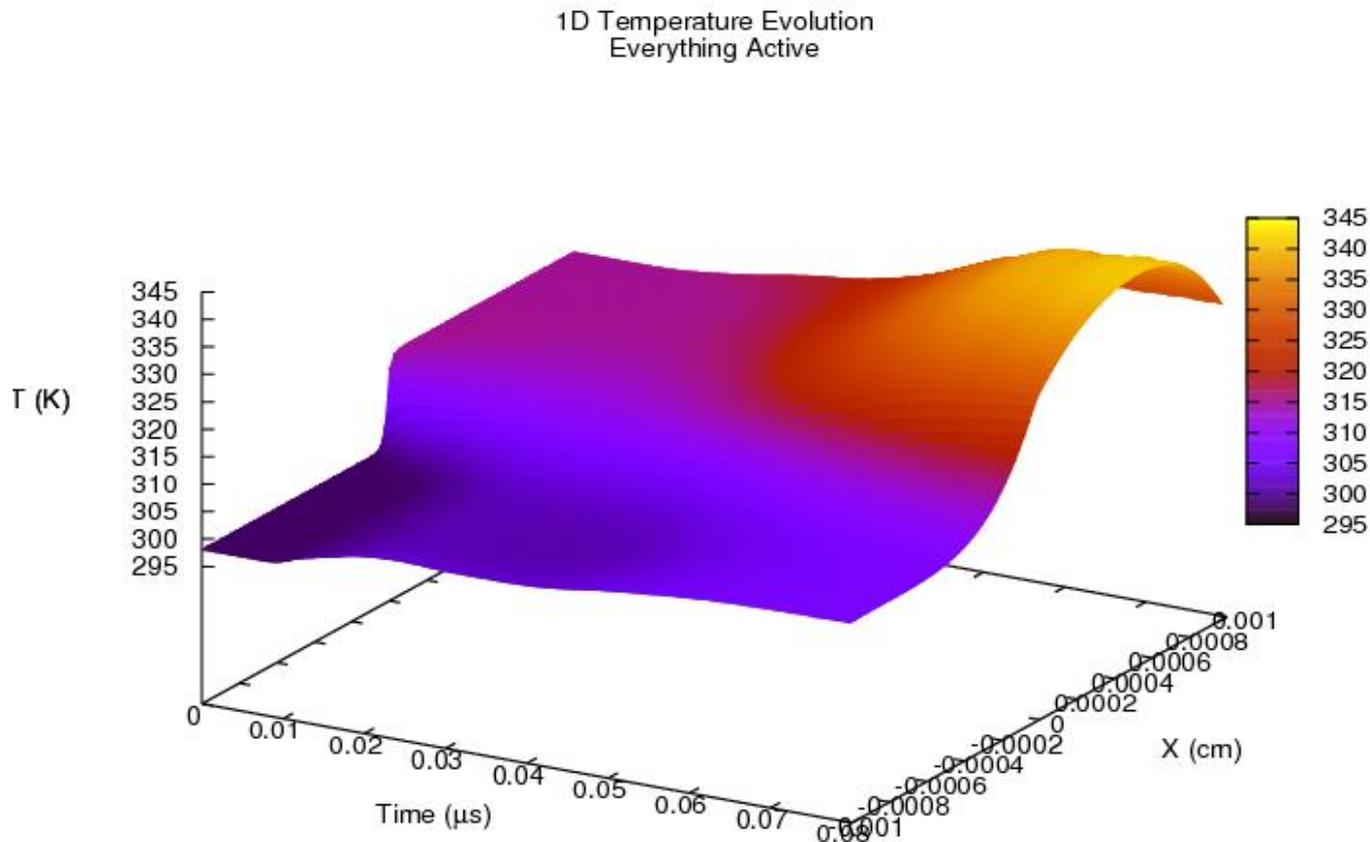
# Results: Smearing friction

- Cu plastic strain field essentially constant
- Temperatures not high enough for much thermal softening
- Not much temperature difference to see strength differences



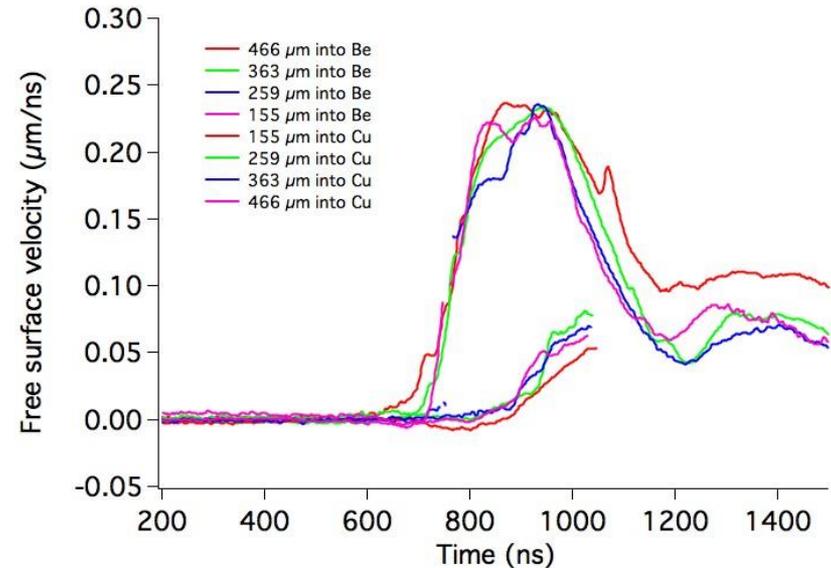
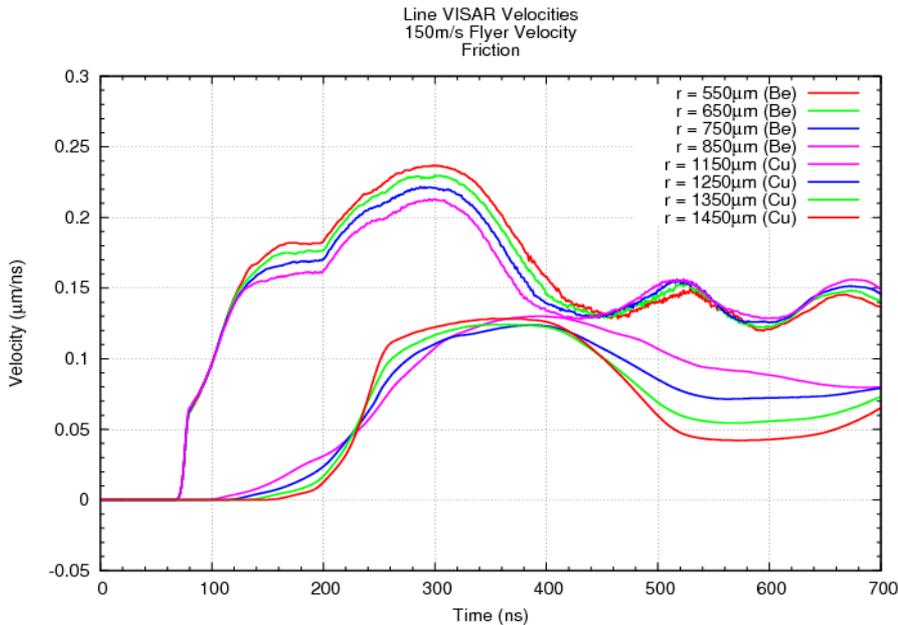
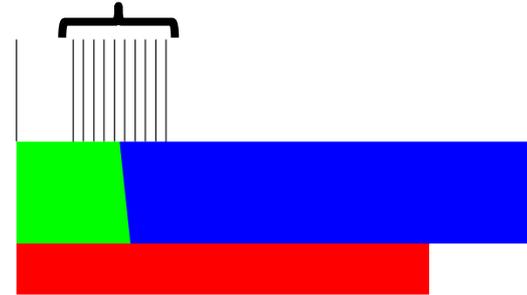
# Results: Whole tamale

- Set everything to proper values
- Essentially no Coulomb friction or sticking, purely smearing



# Results: Line VISAR

- Don't understand experimental results
- Don't follow expected trends
- Suspect experiment warrants more complicated initial conditions
- Need to simulate AWE/CEA Al/SS friction experiments



# Summary

- Coupled multi-scale friction model is in FLAG
- 1D micro-scale thermal conduction coupled to macro-scale and 1D micro-scale plasticity
- Sticking, Coulomb and Tresca conditions
- Plasticity model assumes quasi-static conditions and is rate-independent
  - Biggest concern with the current model

# Conclusions

- Much of the individual aspects of the model have been tested independently and in increasingly coupled cases, works as designed
- Plasticity assumptions are invalid, but may be “sufficiently valid”?
- Friction has an effect, but better experiments need to be simulated for validation testing

## Future Work

- Simulate AWE Al/SS friction experiments
- Test on integrated applications
- Replace plasticity model
  - 1D implicit elastic/visco-plastic hydrodynamics solver (velocity field, momentum equation, etc.)
  - 1D pure shear (xy) assumption
  - PTW' (or multiple) plasticity model(s)
    - Rate dependence, thermal softening, work hardening, etc.
- $\kappa(\rho, \Theta)$  for more accurate conduction